Supporting information

Non linear magnetic behaviour around zero field of an assembly of superparamagnetic nanoparticles

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Multiparametric Testing

In an attempt for multiparametric feasibility, two SPM probes : lab-made γ -Fe₂O₃ nanoparticles and commercial Estapor beads (1020/50, Merck), are mixed at different ratios. The iron concentration of the two ferrofluids was adjusted in order to obtain MIAplex[®] signatures at isointensity. Fig. S1A shows the MIAplex $\mathcal B$ signatures of the two kinds of particles.

Fig. S1B-D shows the experimental MIAplex \otimes signatures measured by mixing the two kinds of particles in various proportions and the corresponding calculated curves. A rather good agreement is obtained between experimental and calculated curves. This confirms the high sensitivity of measuring $d^2B(H)/dH^2$ for characterization and separation purpose.

Figure S1: A : MIAplex signature for nanoparticles (dashed line) and Estapor beads (solid line). Experimental (open circles) and calculated (solid line) curves for 75%-25% (B), 50%-50% (C), and 25%-75% (D) nanoparticles-Estapor mixtures.

Synthesis and characterization of $\gamma \text{Fe}_2\text{O}_3$ **nanoparticles**

Sodium n-dodecyl sulfate (99%, Alfa Aesar), dimethylamine solution (40%, Fluka), Iron (II) Chloride Tetrahydrate (Sigma Aldrich) were used as received. Water is purified with a Millipore system (resistivity 18.2 M Ω .cm).

Concerning lab-made nanoparticles, they are synthesized according to a procedure already described. Briefly, non coated γFe_2O_3 particles were synthesized by reaction of ferrous dodecyl sulfate with dimethylamine in water for two hours at $28^{\circ}C^{1}$. The nanocrystal surface is then functionalized with 5hydroxy-5,5-bis(phosphono)pentanoic acid, which allows a good dispersion in water of the nanoparticles. The nanoparticles surface is characterized via infrared spectroscopy (Fig. S1).

Figure S2: Infrared spectra of free 5-hydroxy-5,5-bis(phosphono)pentanoic acid (HMBP) (dotted line) and 10nm Fe2O3@HMBP nanoparticles (solid line).

The spectrum of free HMBP molecule contains peaks at 1706cm^{-1} (C=O), 1637cm^{-1} (carboxylic C-O), 1309 and 1213cm⁻¹ (δ CH₂), 1146 and 1059cm⁻¹ (P-O). The spectrum of coated nanoparticles presents a very intense band at 577cm⁻¹ attributed to Fe and a large intense vibration band between 1150 and 950cm⁻¹ which corresponds to the P-O absorption band. Compared to free molecule, this modified band indicates that the molecule is bonded at the surface of the particle via the hydroxymethylenebisphosphonate terminal function. The coating rate has been quantified via the P-O area band¹ and is evaluated as about 1500 HMBP molecules per nanoparticle.

Nanoparticles crystal size and structure were characterized by X-ray diffraction (XRD) and Transmission Electron Microscope (TEM). XRD pattern (Fig. S2 and S3) was measured with an X-ray powder diffractometer model X'Pert PRO, MPD, PANalytical, Almelo, the Netherlands, CoKα beam in Bragg Brentano geometry (y/y) was used combined with a fast detector based on real time multiple strip technology (X'Celerator). Crystal XRD size was determined with the Scherrer formula. TEM images were obtained using a FEI CM10 Microscope and samples were prepared by depositing a drop of nanoparticles suspension on carbon coated copper grids placed on a filter paper.

Figure S3: XRD pattern 10nm Fe2O3@HMBP nanoparticles.

Figure S4: XRD pattern of Ferrotec Nanoparticles

The stars Figure S3 indicate the presence of NaCl residue. Phases identified are maghemite γ Fe₂O₃ and magnetite Fe₃O₄ (EVA software (version 13, Bruker-AXS, Karlsruhe, Germany, 1996–2007) and JCPDS-International Centre for Diffraction Data Powder Diffraction File (PDF-2, JCPDS-ICDD, Newtown Square, PA)). Crystal size is determined via the Scherrer equation and is 9.9nm.

The median diameter d_0 and standard deviation w are deduced from TEM data measurements, simulating the diameter d distribution with a lognormal function g(d) described in equation (1).

$$
g(d) = \frac{1}{\sigma d \sqrt{2\pi}} \exp\left(-\frac{\left(\ln \frac{d}{d_0}\right)^2}{2\sigma^2}\right)
$$
(1)

The parameter σ is related to the standard deviation w by the relation (2).

$$
\sigma = \sqrt{\ln\left(1 + \left(\sqrt{1 + 4\left(\frac{w}{d_{med}}\right)^2} - \frac{1}{2}\right)\right)} \approx \sqrt{\ln\left(1 + \left(\frac{w}{d_{med}}\right)^2\right)}
$$
(2)

Magnetic properties of γ **Fe₂O₃ nanoparticles**

Magnetic properties were measured using a superconducting quantum interference device (Quantum Design SQUID Magnetometer MPMS-5T) at 300 K and the MIAplex[®] technology. Samples measured with SQUID are liquid suspensions in water, 0.5wt%, conditioned in silicone capsules supplied by Plastem S.A. Concerning the MIAplex®, samples are also liquid suspensions in water 0.5wt% and 2wt%, conditioned in 500µL Eppendorf.

The saturation magnetization is obtained by the extrapolation of the plot of magnetization vs the inverse of the field at the origin² (Fig. S4).

Figure S5: Magnetization M as function of 1/H.

The red line is the asymptote when 1/H tends to zero. The intersection between this asymptote and yaxis is the experimental saturation magnetization M_{sat} .

Considering a core-shell system, the spin canting thickness e and the corresponding magnetic size $(d_0$ -2e) are deduced from the equation $(2)^3$.

$$
M_{sat} = M_{satbulk} \times \left(\frac{d_0 - 2e}{d_0}\right)^3
$$
 (2)

For the lab-made nanoparticles with $d_0 = 10.5$ nm as TEM diameter and $M_{sat} = 61$ emu/g, we calculate a magnetic size of $d_m = 9.6$ nm, corresponding to a non magnetic layer of e ≈ 0.45 nm. We estimate the contribution of the shell in the total volume of the particles from the equation (3).

$$
\% (V_{shell}) = 100 \times \frac{V_{shell}}{V_{criticalline}} = 100 \times \frac{V_{cristalline} - V_{magnetic}}{V_{cristalline}} = 100 \times \frac{d_0^3 - d_m^3}{d_0^3}
$$
(3)

We so obtained a shell contribution of 24% of the particle volume.

The Langevin model of superparamagnetism⁴ can be used to describe the non-linear magnetization

curve of the nanoparticles, assuming that particles do not interact, by the equation (4).
\n
$$
\frac{M}{M_{sat}} = \coth\left(\frac{v\rho M_{sat}\mu_0 H}{k_b T}\right) - \frac{k_b T}{v\rho M_{sat}\mu_0 H}
$$
\n(4)

v is the volume of the magnetic core, ρ is the particle density (4900kg.m⁻³ for maghemite), M_{sat} $(emu.g⁻¹)$ is the saturation magnetization of the particle, T is the sample temperature in Kelvin (taken to be 300 K), μ_0 is the permeability of vacuum ($4\pi \times 10^{-7}$ H.m⁻¹), H is the applied field (in A.m⁻¹) and k_b is the Boltzmann constant, 1.38×10^{-23} J.K⁻¹. As to model the experimental magnetic data, we have used the Langevin equation weighted by a lognormal function (equation 1).

The experimental curves at high and medium magnetic field ranges are fitted with 1 Langevin equation weighted by log-normal distribution.

At low magnetic field range, the experimental curve are fitted with the sum of 2 Langevin equations weighted by log-normal distribution and corresponding to separate contributions.

The second derivative of the magnetization fit at high and medium magnetic field ranges is compared with the MIAplex \otimes signature (Figure S5).

Figure S6: Second derivative of the magnetization fit at high and medium magnetic field **range** (dotted line) superposed with the MIAplex[®] signal (solid line).

The second derivative of the high and medium magnetic field magnetization fit presents a peak-topeak line width in field $\Delta H_{pp} = 14.4 \pm 0.2 \text{ kA.m}^{-1}$ which is about 3.7 times bigger than the MIAplex® signature ($\Delta H_{pp} = 3.9 \pm 0.1 \text{ kA.m}^{-1}$).

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