### Electronic Supplementary Information

# Improved sensitization efficiency in Er<sup>3+</sup> ions and SnO<sub>2</sub> nanocrystals co-doped silica thin films

Xiaowei Zhang,<sup>*a,b*</sup> Shaobing Lin,<sup>*a*</sup> Tao Lin,<sup>*a,c*</sup> Pei Zhang,<sup>*a,d*</sup> Jun Xu,<sup>\**a*</sup> Ling Xu<sup>*a*</sup> and Kunji Chen<sup>*a*</sup>

<sup>a</sup>National Laboratory of Solid State Microstructures, Department of Electronic Science and Engineering and Collaborative Innovation Centre of Advanced Microstructures, Nanjing University, Nanjing, 210093, China.

<sup>b</sup>Materials Sciences Division, Lawrence Berkeley National Laboratory, Berkeley, California, 94720, USA.

<sup>c</sup>Department of Physics, Guangxi University, Nanning, 530004, China.

<sup>d</sup>Henan Key Lab of Information-based Electrical Appliances, Department of Electrical and Information Engineering, Zhengzhou University of Light Industry, Zhengzhou, 450002, China. \*E-mail: junxu@nju.edu.cn; Fax: +86-25-83594836; Tel: +86-25-83595535.

## 1. An estimation on the average sizes of SnO<sub>2</sub> NCs with the increasing annealing temperature according to the excitation peaks' position.

As shown in Fig.4, the redshifts of the excitation from band-to-band transition of  $SnO_2$  nanocrystals (NCs) can be explained as the enlargement of the average sizes with the increasing annealing temperatures. Based on these excitation peaks, we also estimated the average size of  $SnO_2$  NCs using the effective mass theory <sup>[1]</sup>.

$$E_g(R) = E_g(R \to \infty) + \frac{h^2}{8R^2} \times \left(\frac{1}{m_e^*} + \frac{1}{m_h^*}\right) - \frac{1.8e^2}{4\pi\varepsilon\varepsilon_0 R} + smaller \ terms$$

where  $E_g(R)$  is the band gap energy of SnO<sub>2</sub> NCs, R is the average radius of SnO<sub>2</sub> NCs,  $E_g(R \to \infty)$  is the band gap energy of SnO<sub>2</sub> bulk materials and  $\varepsilon$  is the relative dielectric constant.  $m_e^*$  and  $m_h^*$  stand for the effective mass of an electron and a hole, respectively. For SnO<sub>2</sub> NCs,

$$E_g(R \to \infty) = 3.60 \ eV_{, \varepsilon} = 14, \varepsilon_0 = 8.85 \times 10^{-12} F/m_{e}$$
  
 $m_e^* = 0.35 \ m_0, \ m_h^* \gg m_e^*$ 

where  $m_0$  stands for the free electron mass. Meanwhile,  $E_g(R)$  can be calculated as follows,

$$E_g(R) = \frac{1240}{\lambda_{exc}}$$

where  $\lambda_{exc}$  stands for the excitation peak from band-to-band transition of SnO<sub>2</sub> NCs.

As shown in Table 1, it is found that the average sizes of  $SnO_2$  NCs with different annealing temperatures are consistent with the TEM observations.

Annealing temperature / °C	800	900	1000	Bulk materials
$\lambda_{exc / nm}$	293	300	322	_
Band gap / eV	4.23	4.13	3.85	3.60
Average size / nm	2.92	4.22	5.12	_

Table 1S. Band gaps and average sizes of SnO<sub>2</sub> NCs after annealing at different temperatures.

#### 2. XRD patterns of samples after annealing at 1000°C.

In order to characterize further the formation of  $SnO_2$  NCs, the aged gels were annealed at 1000 °C and then milled into powers for the X-ray diffraction (XRD, using 0.1540562 nm Cu K $\alpha$  radiation) test.

XRD pattern for corresponding sol-gel powers containing with 20% Sn after annealing at 1000°C is demonstrated in Fig. 1S(a). The pattern shows all the diffraction peaks assigned to tetragonal rutile crystalline phase of the SnO<sub>2</sub> NCs (JCPDS No. 41-1445), which is consistent with the TEM observation results. As shown in Fig. 1S(b), the related XRD spectra express a slight shifting, revealing the fact that  $Er^{3+}$  ions should be very likely incorporated in the D<sub>2h</sub> lattice site of Sn<sup>4+</sup>.



Fig.1S (a) XRD pattern of the 20% Sn doped SiO<sub>2</sub> powder samples after annealing 1000 °C. (b) Comparison of XRD spectra of pure and  $Er^{3+}$ -doped SnO<sub>2</sub> NCs silica thin films.

### Reference

[1] L. Brus, J. Phys. Chem., 1986, 90, 2555.