Supporting Information for

Role of synergic interaction in transition state formation of aldol reaction on metal oxide catalyst: A DFT investigation

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S1 Computational details

S1.1 Methods

All calculations were performed by using density-functional theory (DFT) as implemented in Vienna ab-initio simulation package (VASP).¹⁻² The interaction between ions and electrons is described using the projector augmented wave (PAW) method.³⁻⁴ The Kohn–Sham orbitals are expanded in a plane-wave basis set with a kinetic energy cutoff of 400 eV. The spin-polarized GGA-PW91 exchange-correlation functional⁵ was used with $[Zr]-4s^24p^65s^24d^2$, [Ce]- $5s^{2}5p^{6}6s^{2}4f^{1}5d^{1}$, [O]- $2s^{2}2p^{4}$, [C]- $2s^{2}2p^{2}$, and [H]-1s treated as valence electrons. The conjugate gradient algorithm was used in optimization and the convergence threshold was set to be 10^{-3} eV in total energy and 0.02 eV/Å in Hellmann-Feynman force on each atom. The calculated equilibrium lattice constant was obtained using the tetrahedron method with Blöchl corrections⁶ and a Monkhorst–Pack⁵ grid of $10 \times 10 \times 10$ k-points, with the convergence criteria of 10^{-6} eV in total energy and 10⁻⁵ eV /Å in force. For surface calculations, the Brillouin zone integration is performed using Γ point only for a $p(4\times 4)$ lateral supercell. Our test calculations show that increasing k-points has only negligible effect on the results (±0.1kcal/mol difference in binding energy by using $3 \times 3 \times 1$ k-points). All reported energies were extrapolated to $k_B T = 0$ eV. The climbing-image nudged elastic band (CI-NEB) method⁷⁻⁸ was used to locate the TS structure along the minimum-energy pathway (MEP) for aldol reaction of propanal and interconversion of propanal rotamers. Spring constant of 5.0 $eV/Å^2$ and ten images were used.

The binding energy (BE) is defined as $BE = E_{(ads/slab)} - E_{slab} - E_{(ads)}$, where $E_{(ads/slab)}$, E_{slab} , and $E_{(ads)}$ are the total energy of ads/slab, clean surface, and gas-phase adsorbate molecule.

It's been known that LDA+U and GGA+U methods⁹⁻¹¹ are commonly employed for ceria with strongly localized Ce-4f electrons, particularly for defective ceria,¹² in order to correctly address the electronic structure of ceria. In DFT+U approach, on-site electron-electron Coulomb repulsion (the empirical Hubbard U term) and exchange (parameter J) interactions are introduced to capture the correct electronic features of strongly-correlated Ce-4f electrons and their localization. We also examined such effect in our calculations. By employing the GGA+U approach by Dudarev *et al.* ¹³ with U = 7.0 eV and J = 0.7 eV,¹¹ we obtained a slightly increased BE for adsorbates on O-terminated CeO₂ (111) surface. Nevertheless, the GGA+U method does not change the overall trend of the energetics and thus the conclusions by employing GGA-DFT method in this work. The same approach has been used in previous DFT studies.¹⁴⁻¹⁵ It's worthy to note that DFT+U method is indeed critical to reduced CeO₂ (111) surface, the calculated BE is -34.2 kcal/mol (DFT+U) vs -19.6 kcal/mol (GGA-DFT), which is consistent with previous findings.¹⁶⁻¹⁷

S1.2 Slab model

The under-coordination of the surface and the ionic nature of both ZrO_2 and CeO_2 render surface metal Zr and Ce sites with Lewis acidity and O sites with Lewis basicity. It's known that zirconia undergoes phase transition from monoclinic (<1,170 °C) to tetragonal (1,170–2,370 °C) and eventually to cubic structures (>2,370 °C). At the temperature relevant to aldol condensation (<1,170 °C), monoclinic ZrO_2 ($Zr_{CN} = 7$, $O_{CN} = 3$ and 4) is the most stable phase.¹⁸ Since our study is aimed to explore the effects of acid-base pair strength on the aldol reaction, we therefore adopted cubic structure for both ZrO_2 and CeO_2 in our calculations (Figure S1 and S2), in order to isolate geometrical effect and therefore for a more direct comparison. The O-terminated surfaces are most abundant from nanoparticle catalysts.¹⁹⁻²⁰ Both O-terminated $CeO_2(111)$ surface^{10, 14-15, 21-22} and O-terminated $ZrO_2(111)$ surface²³⁻²⁴ were employed in previous DFT studies. The $ZrO_2(111)$ surface was modeled by a slab in a $p(4\times4)$ lateral supercell (14.38 Å × 14.38 Å × 19.54 Å), which contains three (O-Zr-O) repeating units, a total of nine layers (96 O atoms and 48 Zr atoms), and is sufficiently large to accommodates two propanal molecules with negligible lateral interactions with their images. The two successive slabs were separated by a 13 Å vacuum region to ensure negligible normal interactions between adsorbates and slab image. Dipole moment normal to the surface is eliminated with O-terminated surface at both top and bottom layers. In calculations, the bottom six layers (i.e., two ZrO_2 trilayers) were fixed at their equilibrium bulk phase positions with calculated lattice constant of 5.085 Å (5.070Å by experiment), while the top three layers (i.e., one ZrO_2 trilayers) and all adsorbates were allowed to relax without geometrical constraints. The top layer features (ZrO)₃ or (CeO)₃ six-member rings [Figure S1 and S2].

The O-terminated CeO₂ (111) surface was modeled in the same fashion as the ZrO₂(111) surface. The $p(4\times4)$ supercell has a dimension of 15.32 Å × 15.32 Å × 20.17 Å. The calculated lattice constant is 5.418 Å (5.411Å by experiment), which is similar to 5.419 Å obtained in previous work using GGA-PW91 functional.²⁵

We also verified the validity of methods and models used in this work by calculating surface energy of O-terminated $CeO_2(111)$ and $ZrO_2(111)$ surface. The surface energy is expressed as:²⁶

$$\sigma = \frac{1}{2A} (E_{\text{slab}} - \frac{Ns}{Nb} E_{\text{bulk}}) \tag{1}$$

where σ , A, E_{slab}, and E_{bulk} are the surface energy, surface area, total energy of the slab (containing *Ns* atoms) and the bulk (containing *Nb* atoms), respectively. Lower σ value indicates a more stable surface and less surface reactivity. We obtained σ value of 0.54 and 0.70 J/m² for relaxed O-terminated CeO₂ (111) and ZrO₂(111) surface, respectively, which are lower than those from previous work, but exhibit the same trend.^{15, 21}

S2 Projected-density of states (DOS)

Figure S5 shows the projected-DOS of labeled H, C, O atoms of TS1, MS1, TS2, MS2 in Figure 2. In Figure S5(a), one can see that the "residual" 1s-states of α -H in TS1 (blue line), i.e., H^{δ +} instead of a proton H¹⁺, is significantly broadened by 2p-states of both α -C of rotamer B and O of ZrO₂ on the surface and have a large overlap below Fermi level, an indication of direct electronic interactions among C•••H•••O in TS1. In comparison, the strongly localized 1s-states of the dissociated α -H can be found at -7.8eV with complete overlap with 2p-states of surface O [Figure S5(b)], indicating the formation of strong O-H bond in MS1. In Figure S5(c), the 1sstates of the same H in TS2 is mainly localized at -7.5eV below Fermi level and has large overlaps with 2p-states of both surface O of ZrO_2 (yellow line) and carbonyl O of rotamer A (red line), suggesting a direct electronic interactions among O•••H•••O in TS2. The peaks resemble those of MS2 with only slight shift [Figure S5(d)]. The "residual" 1s-states are populated 22% less in magnitude than those of TS1, an outcome of high electron-pulling capacity from two O in TS2.

Table S1. Characteristic bond lengths of propanal (A) and (B) along MEP for aldol reaction on (a) $ZrO_2(111)$ surface and (b) $CeO_2(111)$ surface, as shown in Figure 1 and Figure S2. Values in bold correspond to bond break and formation.

(a)											
Bond length (Å)	IS	TS1	MS1	MS2	TS2	MS2	FS				
Zr-O(B)	2.317	2.135	1.991	2.042	2.131	2.307	4.981				
$O \cdots H_{\alpha}(B)$	2.185	1.496	0.978	0.990	1.263	1.880	-				
$C_{\alpha}(B)$ - $H_{\alpha}(B)$	1.107	1.292	2.703	-	-	-	-				
C(B)-O(B)	1.236	1.272	1.344	1.348	1.258	1.233	1.217				
$C(B)-C_{\alpha}(B)$	1.486	1.404	1.345	1.345	1.425	1.490	1.515				
Zr-O(A)	-	-	-	2.389	-	-	-				
C(A)-O(A)	-	-	-	1.225	1.336	1.420	1.440				
$C(A)-C_{\alpha}(A)$	-	-	-	1.494	1.540	1.540	1.534				
$H_{\alpha}(B)$ -O(A)	-	-	-	2.675	1.190	0.994	0.974				
$C_{\alpha}(B)$ - $C(A)$	-	-	-	3.953	1.865	1.570	1.537				

(b)											
Bond length (Å)	IS	TS1	MS1	MS2	TS2	MS2	FS				
Ce-O(B)	2.631	2.425	2.213	2.229	2.499	2.611	4.500				
$O \cdots H_{\alpha}(B)$	2.586	1.367	0.979	0.977	1.203	1.769	-				
$C_{\alpha}(B)$ - $H_{\alpha}(B)$	1.107	1.397	2.772	-	-	-	-				
C(B)-O(B)	1.226	1.261	1.322	1.318	1.232	1.224	1.217				
$C(B)-C_{\alpha}(B)$	1.498	1.425	1.359	1.359	1.474	1.498	1.515				
Ce-O(A)	-	-	-	2.663	-	-	-				
C(A)-O(A)	-	-	-	1.225	1.384	1.424	1.442				
$C(A)-C_{\alpha}(A)$	-	-	-	1.494	1.555	1.539	1.534				
$H_{\alpha}(B)$ -O(A)	-	-	-	2.442	1.252	1.000	0.974				
$C_{\alpha}(B)$ - $C(A)$	-	-	-	3.978	1.675	1.562	1.534				



Figure S1. (a) Optimized structure of (a) ZrO_2 cubic unit cell; (b) Side view and (c) Top view of O-terminated $ZrO_2(111)$ surface in a supercell. Top layer features $(ZrO)_3$ six-member ring (black hexagon) with second-layer O centered underneath. Bulk: $Zr_{CN} = 8$, $O_{CN} = 4$; Surface: $Zr_{CN} = 7$, $O_{CN} = 3$. CN: coordination number.



Figure S2. (a) Optimized structure of (a) CeO_2 cubic unit cell; (b) Side view and (c) Top view of O-terminated $CeO_2(111)$ surface in a supercell. Top layer features $(CeO)_3$ six-member ring (black hexagon) with second-layer O centered underneath. Bulk: $Ce_{CN} = 8$, $O_{CN} = 4$; Surface: $Ce_{CN} = 7$, $O_{CN} = 3$.



Figure S3. Evolution of geometrical structures along MEP for aldol reaction of propanal on O-terminated CeO2(111) surface during one catalytic turnover. Imaginary frequencies: i1373.7cm⁻¹ (TS1), i729.4cm⁻¹ (TS2)



Figure S4. Comparison of adsorption of two rotamer A and B on O-terminated $ZrO_2(111)$ surface. (a) A, BE = -1.8kcal/mol; (b) B, BE = -8.7 kcal/mol.



Figure S5. Calculated PDOS for labeled H, C, and O atoms in Figure 2. (a) TS1, (b) MS1, (c) TS2, and (d) MS2. Zero-line represents Fermi level.

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