Supplementary Information

Heterostructured Ferromagnet-Topological Insulator with Dual-phase Magnetic Properties

Shu-Jui Chang ¹ , Pei-Yu Chuang² , Cheong-Wei Chong³ , Yu-Jung Chen⁴ , Jung-Chun-Andrew Huang2,3,4 , Po-Wen Chen, and Yuan-Chieh Tseng¹*

(1) Dept. Materials Science & Engineering, National Chiao Tung University, Hsinchu, Taiwan.

- (2) Dept. Physics, National Cheng Kung University, Tainan, Taiwan.
- (3) Advanced Optoelectronic Technology Center, National Cheng Kung University, Tainan, Taiwan.
- (4) Taiwan Consortium of Emergent Crystalline Materials, Ministry of Science and Technology, Taipei, Taiwan.
- (5) Division of Physics, Institute of Nuclear Energy Research, Taoyuan, Taiwan.

E-mail address[:yctseng21@mail.nctu.edu.tw](mailto:yctseng21@mail.nctu.edu.tw).

Fig. S-1 SR-PES spectra of (a) Bi-4*f* and (b) Se-3*d* where highlighted areas are used for Bi/Se ratio calculation, with the use of following equation:

$$
n=\frac{I}{S}
$$

where

$S = f \sigma \theta y \lambda A T$

where f is the Photon flux of the x-ray, σ is the photoionization cross-section of peak of element, θ is the photoelectron emission angle, y is the photoelectron yield, λ is the inelastic electron mean-free path, *A* is analysis area and *T* is a analyzer transmission function. Here, we define *S* is the sensitivity factor of core level of element.

Hence, *n* is dependent on the peak area and sensitivity factor.

$$
\frac{n_1}{n_2} = \frac{(I_1 / S_1)}{(I_2 / S_2)}\tag{1}
$$

The calculated Bi/Se ratio is 1.49 very close to the ideal stoichiometry of $Bi₂Se₃$ formula.

Fig. S-2 Cross-sectional TEM images of (a) Py/Bi₂Se₃-5nm and (b) Py/Bi₂Se₃-20 nm samples. EDX analyses are shown below the corresponding TEM images, where EDX line-scanning was performed along the yellow dashed line which starts from A and ends at B. EDX shows Ni diffusion into Bi2Se3 and it formed an Ni-Se overlapping region.

Fig. S-3 Differentiation of Ni *K*-edge XANES for pure Py (black), Py/Bi₂Se₃-5nm, Py/Bi₂Se₃-20nm and Py/Bi2Se3-40nm samples. The edge characteristics can be decomposed into feature A and B upon differentiation. Feature A is allowed by quadrupole selection rules. The weak intensity corresponds to a quadrupole 1*s*→3*d* dipole-forbidden transitions as a characteristic of chemical environment in Ni valency in which the intensity strongly depends upon the coordination symmetry. Feature A becomes less intense in Ni:Bi₂Se₃ as compared to Py, suggesting a local-structural change to centro-symmetry in coordination [1-3]. Feature B is attributed to the 1*s*→4*p* transition, which arises from a chalcogen-based charge-transfer and indicates the presence of strong Ni-Se covalent bonds. Feature B is intensified with the increase of Bi₂Se₃ thickness, implying a more stabilized phase of Ni:Bi2Se3. On the contrary, feature B is heavily suppressed in pure Py, which is a straightforward indicator to the absence of Ni:Bi2Se3.

Fig. S-4 The dependence of remanence ratios (*Mo/Msat*) on Bi2Se³ thickness, for in-plane and out-of-plane directions. *Mo/Msat* was estimated from M-H curves.

Fig. S-5 The temperature dependence of magnetization of (a) Py/Bi₂Se₃-5 nm, (b) Py/Bi₂Se₃-20 nm, (c) Py/Bi₂Se₃-40 nm, for in-plane (black) and out-of-plane (red) directions.

Fig. S-6 (left) Ni-XMCD and (right) Ni-XAS integrations applied in sum-rules analysis for the calculation of Ni's ΔESO. The orbital (*morb*) and spin (*mspin*) moments are obtained by following equations:

$$
m_{orb} = -4q(10 - n_{3d})/3r
$$
 (1)

$$
m_{spin} = -(6p - 4q)(10 - n_{3d})/r
$$
 (2)

Where

 $q = \int_{L3+L2} (\mu_{+} - \mu_{-}) d\,\varpi$ $r = \int_{L_3 + L_2} (\mu_+ + \mu_-) d\,\overline{\omega}$ $p = \int_{L3} (\mu_{+} - \mu_{-}) d\varpi$

where μ_1/μ_2 is the x-ray absorption intensity with left/right circular polarization; n_{3d} is the number of 3*d* electron per cation and the *p*, *q* and *r* are integrated values obtained from XMCD and XAS.

The calculated Ni m_{orb} are 0.02, 0.018 and 0.01 for Py/Bi₂Se₃-5 nm, Py/Bi₂Se₃-20 nm, and Py/Bi₂Se₃-40 nm, respectively, along in-plane direction, which is denoted *^morb*

Ni m_{orb} are 0.07, 0.07, 0.06 for Py/Bi₂Se₃-5 nm, Py/Bi₂Se₃-20 nm, and Py/Bi₂Se₃-40 nm, respectively, for outof-plane direction, which is denoted m_{orb}^{\perp} .

In a simple model of Bruno [4][5], a relation between m_{orb} and ΔE_{SO} is derived as:

$$
E_{SO} = -\frac{1}{2} \frac{\xi}{2\mu_B} [m_{orb}^{\perp} + (m_{orb}^{\parallel} - m_{orb}^{\perp}) \sin^2 \theta]
$$

The energy difference for the magnetization aligned along out-of-plane and in-plane directions is:

$$
\Delta E_{SO} = E_{SO}(\theta = 0^{\circ}) - E_{SO}(\theta = 90^{\circ})
$$

$$
\Delta E_{SO} = -\frac{1}{2} \frac{\xi}{2\mu_B} [m_{orb}^{\perp} - m_{orb}^{\parallel}]
$$

where ζ is spin-orbit coupling parameter for 3*d* electron and the values of m_{orb}^{\perp} and m_{orb}^{\parallel} are obtained by sum-rules [6] along the out-of- and in-plane directions, respectively.

The calculated $m_{orb}^{\parallel} - m_{orb}^{\perp}$ values are 0.02795, 0.02907 and 0.02795, for Py/Bi₂Se₃-5 nm, Py/Bi₂Se₃-20 nm, and Py/Bi₂Se₃-40 nm, respectively.

The same method was used in the calculation of Fe's ΔE_{SO} .

References

- [1] F. E. Huggins, K. C. Galbreath, K. E. Eylands, L. L. Van Loon, J. A. Olson, E. J. Zillioux, S. G. Ward, P. A. Lynch, and P. Chu, Determination of nickel species in stack emissions from eight residual oil-fired utility steam-generating units, *Environ. Sci. Technol.* **45**, 6188 (2011).
- [2] A. Kisiel, P. Zajdel, P. M. Lee, E. Burattini, and W. Giriat, XANES study of K edges of Fe, Co, Ni, and Se in transition metal selenides. Experiment and comparison with LMTO numerical calculations, *J. Alloys Compd.* **286**, 61 (1999).
- [3] S. A. Yao, V. Martin-Diaconescu, I. Infante, K. M. Lancaster, A. W. Gotz, S. DeBeer, and J. F. Berry, Electronic structure of Ni₂E complexes ($E = S$, Se, Te) and a global analysis of M₂E₂ compounds: a case for quantized E_2^{n} oxidation levels with $n = 2$, 3, or 4, *J. Am. Chem. Soc.* **137**, 4993 (2015).
- [4] D. Weller, Y. Wu, J. Stöhr, M. G. Samant, B. D. Hermsmeier, and C. Chappert, Orbital magnetic moments of Co in multilayers with perpendicular magnetic anisotropy, *Phys. Rev. B* **49**, 12888 (1994).
- [5] J. Stöhr, Exploring the microscopic origin of magnetic anisotropies with X-ray magnetic circular dichroism (XMCD) spectroscopy, *J. Magn. Magn. Mater.* **200**, 470 (1999).
- [6] P. Bruno, Tight-binding approach to the orbital magnetic moment and magnetocrystalline anisotropy of transition-metal monolayers, *Phys. Rev. B* **39**, 865 (1989).