

# Self-assembly of *bis-β*-diketone-based [M<sub>2</sub>L<sub>2</sub>] dinuclear platforms into 2-dimensional coordination polymers.

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## Electronic Supplementary Information (ESI)

**S1.** SCXRD details

**S2.** Topology diagram for {[ZnL<sup>1</sup>]<sub>n</sub>} and {[CuL<sup>1</sup>]<sub>n</sub>}

**S3.** References

## S1. SCXRD refinement tables and details

	<b>H<sub>2</sub>L<sup>1</sup></b>	<b>[Pd<sub>2</sub>L<sup>1</sup><sub>2</sub>]</b>
<b>Empirical formula</b>	C <sub>19</sub> H <sub>25</sub> NO <sub>4</sub>	C <sub>38</sub> H <sub>46</sub> N <sub>2</sub> O <sub>8</sub> Pd <sub>2</sub>
<b>Formula weight</b>	331.40	871.57
<b>Temperature/K</b>	190	100(2)
<b>Crystal system</b>	monoclinic	triclinic
<b>Space group</b>	<i>P2<sub>1</sub>/n</i>	<i>P-1</i>
<b>a/Å</b>	6.2281(5)	5.7610(12)
<b>b/Å</b>	19.4672(19)	9.921(2)
<b>c/Å</b>	15.2664(13)	16.902(3)
<b>α/°</b>	90	75.20(3)
<b>β/°</b>	95.993(8)	88.40(3)
<b>γ/°</b>	90	74.52(3)
<b>Volume/Å<sup>3</sup></b>	1840.8(3)	899.2(4)
<b>Z</b>	4	1
<b>ρ<sub>calc</sub>/cm<sup>3</sup></b>	1.196	1.609
<b>μ/mm<sup>-1</sup></b>	0.677	1.055
<b>F(000)</b>	640	444
<b>Crystal size/mm<sup>3</sup></b>	0.5 × 0.2 × 0.2	0.05 × 0.2 × 0.2
<b>Radiation</b>	CuKα (λ = 1.54184)	synchrotron (λ = 0.7108)
<b>2θ range for data collection/°</b>	7.384 to 130.312	4.41 - 52.746
<b>Index ranges</b>	-7 ≤ h ≤ 6, -22 ≤ k ≤ 18, -17 ≤ l ≤ 17	-7 ≤ h ≤ 7, -11 ≤ k ≤ 12, -10 ≤ l ≤ 21
<b>Reflections collected</b>	5919	3298
<b>Independent reflections</b>	3073 [R <sub>int</sub> = 0.0255, R <sub>sigma</sub> = 0.0426]	3298 [R <sub>int</sub> = 0.0316]
<b>Data/restraints/parameters</b>	3073/74/294	3298/213/252
<b>Goodness-of-fit on F<sup>2</sup></b>	1.051	1.214
<b>Final R indexes [I ≥ 2σ (I)]</b>	R <sub>1</sub> = 0.0533, wR <sub>2</sub> = 0.1489	R <sub>1</sub> = 0.1386, wR <sub>2</sub> = 0.4017
<b>Final R indexes [all data]</b>	R <sub>1</sub> = 0.0729, wR <sub>2</sub> = 0.1692	R <sub>1</sub> = 0.1391, wR <sub>2</sub> = 0.4018
<b>Largest diff. peak/hole / e Å<sup>-3</sup></b>	0.34/-0.20	3.53/-4.36

	$\{\{\text{Zn}_2\text{L}_2\}_n \cdot \text{dmf}_n\}$	$\{\{\text{Cu}_2\text{L}_2\}_n \cdot \text{dmf}_n\}$
<b>Empirical formula</b>	$\text{C}_{38}\text{H}_{44}\text{N}_2\text{O}_8\text{Zn}_2$	$\text{C}_{38}\text{H}_{46}\text{Cu}_2\text{N}_2\text{O}_8$
<b>Formula weight</b>	787.36	785.85
<b>Temperature/K</b>	100(2)	100(2)
<b>Crystal system</b>	monoclinic	monoclinic
<b>Space group</b>	$P2_1/c$	$P2_1/c$
<b>a/Å</b>	16.584(3)	16.234(3)
<b>b/Å</b>	11.077(2)	11.130(2)
<b>c/Å</b>	12.668(3)	12.499(3)
<b><math>\alpha/^\circ</math></b>	90	90
<b><math>\beta/^\circ</math></b>	107.26(3)	105.22(3)
<b><math>\gamma/^\circ</math></b>	90	90
<b>Volume/Å<sup>3</sup></b>	2222.3(8)	2179.1(8)
<b>Z</b>	4	2
<b><math>\rho_{\text{calc}}/\text{g/cm}^3</math></b>	1.177	1.198
<b><math>\mu/\text{mm}^{-1}</math></b>	1.123	1.021
<b>F(000)</b>	820	820
<b>Crystal size/mm<sup>3</sup></b>	$0.5 \times 0.5 \times 0.5$	$0.1 \times 0.05 \times 0.02$
<b>Radiation</b>	synchrotron ( $\lambda = 0.7108$ )	synchrotron ( $\lambda = 0.7108$ )
<b>2<math>\theta</math> range for data collection/<math>^\circ</math></b>	4.488 to 52.744	4.49 to 54.208
<b>Index ranges</b>	$-20 \leq h \leq 20, -13 \leq k \leq 13, -15 \leq l \leq 15$	$-20 \leq h \leq 20, -14 \leq k \leq 14, -16 \leq l \leq 16$
<b>Reflections collected</b>	30449	31832
<b>Independent reflections</b>	4349 [ $R_{\text{int}} = 0.1110, R_{\text{sigma}} = 0.0516$ ]	4471 [ $R_{\text{int}} = 0.0372, R_{\text{sigma}} = 0.0209$ ]
<b>Data/restraints/parameters</b>	4349/0/284	4471/0/264
<b>Goodness-of-fit on <math>F^2</math></b>	1.075	1.055
<b>Final R indexes [<math>I \geq 2\sigma(I)</math>]</b>	$R_1 = 0.0851, wR_2 = 0.1892$	$R_1 = 0.0318, wR_2 = 0.0870$
<b>Final R indexes [all data]</b>	$R_1 = 0.1072, wR_2 = 0.2078$	$R_1 = 0.0325, wR_2 = 0.0876$
<b>Largest diff. peak/hole / e Å<sup>-3</sup></b>	0.86/-0.75	0.41/-0.42

## Specific Refinement Details

### *[Pd<sub>2</sub>L<sup>1</sup><sub>2</sub>]*

The crystals obtained were very small plates which were difficult to manually separate. Synchrotron radiation was required to obtain suitable diffraction data and because of the instrument design we were limited to a single 360 ° phi scan. The diffraction pattern obtained, however, was still significantly broadened indicative of some degree of twinning and the structure was refined as a two component twin (180 ° rotation about 0 0 1) with approximate twin fractions of 0.57 and 0.43. After refinement, however, there still remains some evidence of further unresolved twinning indicated by large residual peaks in chemically unreasonable positions, which result in higher than ideal statistics. A number of rigid bodies and restraints were required to facilitate realistic modelling particularly in the pyridyl ring and the disordered tertiary butyl group. The connectivity of the structure is unambiguous.

### *{[Zn<sub>2</sub>L<sup>1</sup><sub>2</sub>]<sub>n</sub>·dmf<sub>n</sub>}* and *{[Cu<sub>2</sub>L<sup>1</sup><sub>2</sub>]<sub>n</sub>·dmf<sub>n</sub>}*

During the refinement of the crystal structures of *{[Zn<sub>2</sub>L<sup>1</sup><sub>2</sub>]<sub>n</sub>·dmf<sub>n</sub>}* and *{[Cu<sub>2</sub>L<sup>1</sup><sub>2</sub>]<sub>n</sub>·dmf<sub>n</sub>}* a region of disordered electron density was observed across a point of inversion. Despite many attempts to refine this as a disordered DMF solvent molecule no reasonable model was obtained. The region of electron density was therefore masked using Olex2.<sup>1</sup> For *{[Zn<sub>2</sub>L<sup>1</sup><sub>2</sub>]<sub>n</sub>·dmf<sub>n</sub>}*, the electron density correlated to 73.2 electrons and for *{[Cu<sub>2</sub>L<sup>1</sup><sub>2</sub>]<sub>n</sub>·dmf<sub>n</sub>}*, the electron density correlated to 80.0 electrons. This is in good agreement with the 73 electrons associated with a single full-occupancy DMF molecule.

## S2. Topology<sup>2</sup>

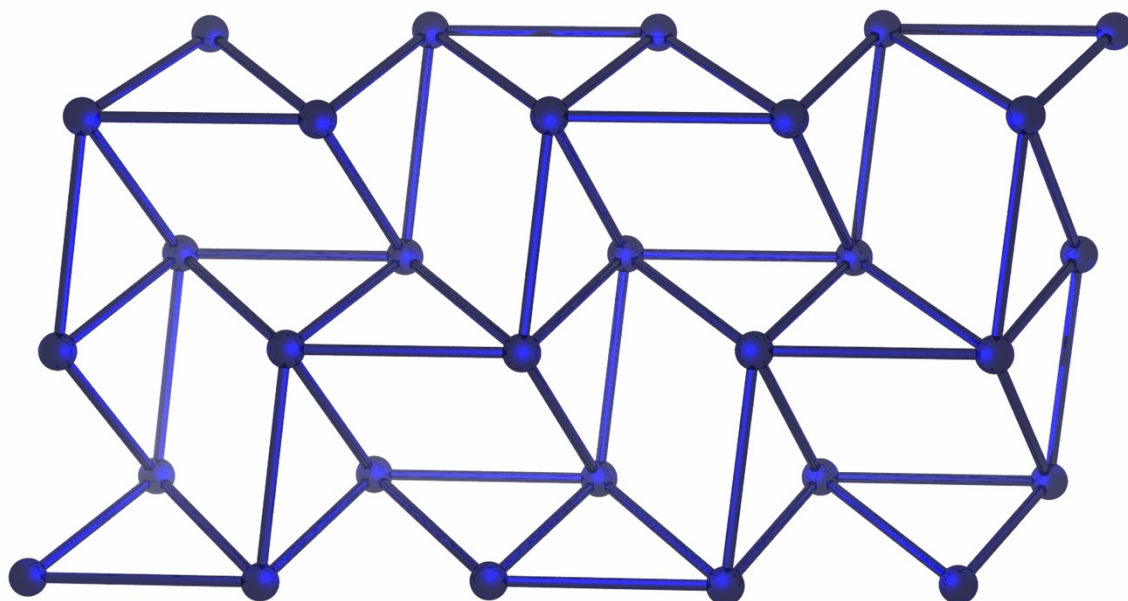


Figure S1. Topological diagram derived from crystal structure of  $[\text{Zn}_2\text{L}^1]_n$  and  $[\text{Cu}_2\text{L}^1]_n$ , showing **tts** topology. Metal atoms are shown as blue spheres.

## S3. References

1. O. V. Dolomanov, L. J. Bourhis, R. J. Gildea, J. A. K. Howard and H. Puschmann, *J. Appl. Cryst.*, 2009, **42**, 339-341.
2. V. A. Blatov, A. P. Shevchenko and D. M. Proserpio, *Cryst. Growth Des.*, 2014, **14**, 3576.