

Electronic Supporting Information

High-pressure induced a stable phase of $\text{Li}_2\text{MnSiO}_4$ for an effective poly-anion cathode material from simulations

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Structural information

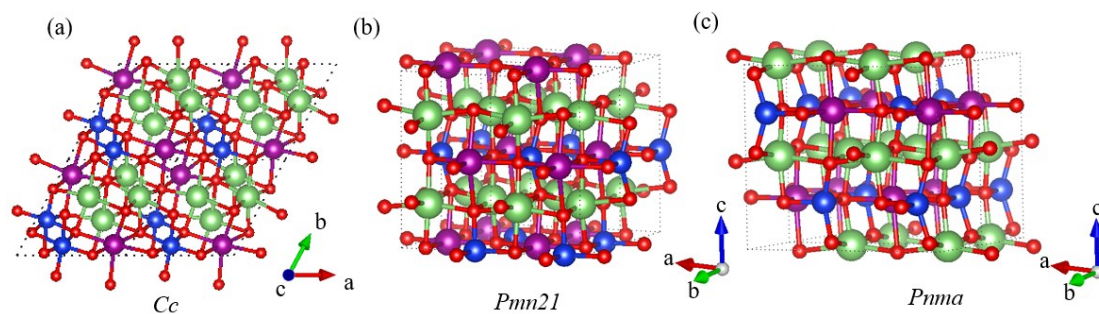


Figure S1. Geometry structures of different $\text{Li}_2\text{MnSiO}_4$ with space group of (a) Cc , (b) $Pnma$ and (c) $Pmn21$.

Structural Stability

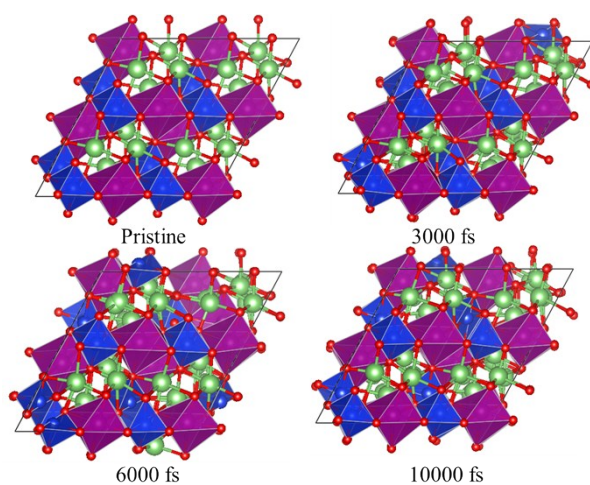


Figure S2. The snapshots of structural geometry of Cc $\text{Li}_2\text{MnSiO}_4$ extracted from AIMD simulation at 500K.

Table S1. Lattice parameters of primitive cell of predicted $\text{Li}_2\text{MnSiO}_4$ with space group of Cc after removing the pressure.

Symmetry	a=b (Å)	c (Å)	$\alpha=\beta(^{\circ})$	$\gamma(^{\circ})$
Cc	6.00	4.94	108.79	58.16
Atom	x	y	z	Sites
Li1	0.84	0.61	0.86	2
	0.61	0.84	0.36	
Li2	0.37	0.61	0.84	2
	0.61	0.37	0.34	
Mn	0.11	0.84	0.33	2
	0.84	0.11	0.83	
Si	0.37	0.10	0.87	2
	0.10	0.37	0.37	
O1	0.01	0.69	0.59	2
	0.69	0.01	0.09	
O2	0.43	0.24	0.64	2
	0.24	0.43	0.14	
O3	0.24	0.03	0.11	2
	0.03	0.24	0.61	
O4	0.46	0.79	0.65	2
	0.79	0.46	0.15	

Table S2. The charge of Mn and O atoms in *Cc* Li_{2(1-x)}MnSiO₄ ($x=0.00, 0.25, 0.5, 0.75$) based on Bader Charge analysis.

Li _{2(1-x)} MnSiO ₄	$x=0.00$	$x=0.25$	$x=0.50$	$x=0.75$
Mn1	+1.55	+1.74	+1.90	+1.95
Mn2	-1.52	+1.63	+1.90	+1.94
O1	-1.84	-1.68	-1.70	-1.52
O2	-1.84	-1.78	-1.83	-1.61
O3	-1.94	-1.79	-1.74	-1.61
O4	-1.94	-1.78	-1.61	-1.50
O5	-1.88	-1.77	-1.70	-1.51
O6	-1.88	-1.79	-1.83	-1.86
O7	-1.89	-1.85	-1.74	-1.68
O8	-1.89	-1.66	-1.61	-1.41

Table S3. Energy difference ΔE (eV) between initial and end state and migration barrier (eV) for Mn displacement to adjacent Li vacancy.

Group symmetry	<i>Pmn21</i>	<i>Pnma</i>	<i>Cc</i>
ΔE	0.53	0.46	-0.23
Migration barrier	1.10	1.06	1.48

Electronic properties

As a cathode material, the electronic conductivity is crucial to the overall battery performance, especially for charging/discharging efficiency. Usually, the poly-anion materials have large bandgap (>3eV) due to the strong bonding between polyanion groups and metal-containing polyhedrons. For instance, olivine-structure LiFePO₄ has a band gap of 4.1 eV¹ and shows low electron conductivity, which limits the application at low temperature. We calculate the band structure of *Cc* Li₂MnSiO₄ using GGA+U (U=6.0 eV) as shown in Figure S2, indicating that the *Cc* phase is a semiconductor with 3.20 eV bandgap which is comparable to those of *Pnma* (3.07 eV) and *Pmn21* (3.05 eV) phases [data from Material Project²]. To get more accurate result, the band

structure is also calculated using the HSE06 functional³, which gives the band gap of 5.0 eV. The density of states suggests that the O 2p-orbital and Mn 3d-orbital mainly contribute to the states around the Fermi level, which implies the competence of redox between Mn and O during delithiation. This is different from the situations in Li_2MnO_3 and CoO_2 .⁴

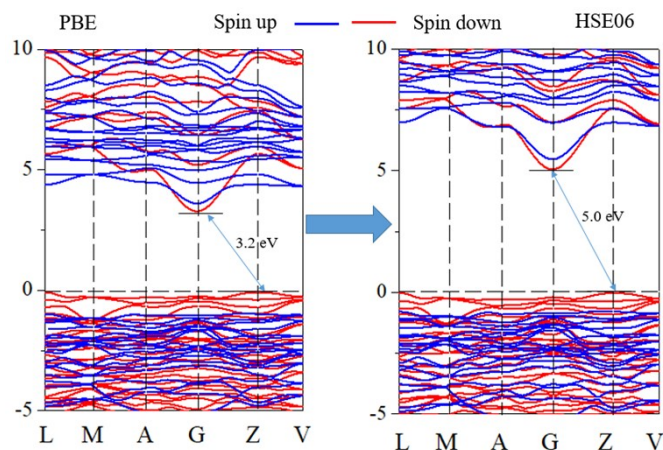


Figure S2. Band structures of Cc $\text{Li}_2\text{MnSiO}_4$ calculated using (a) PBE and (b) HSE06 functional.

Ionic Diffusion

For CI-NEB calculation, two different supercells are used: $2 \times 2 \times 3$ (256 atoms; $a=b=12.0$ Å, $c=14.8$ Å) and $2 \times 2 \times 2$ supercell (128 atoms; $a=b=12.0$ Å, $c=9.9$ Å), both give very similar results for the energy barrier as shown in Figure S3, indicating that $2 \times 2 \times 2$ supercell can be used for reducing the computational cost. To accelerate the ion-hopping process, temperature is elevated from 700K to 2500K. Thermalization over 4-ps with a time-step of 2 fs is carried out before diffusion calculation.

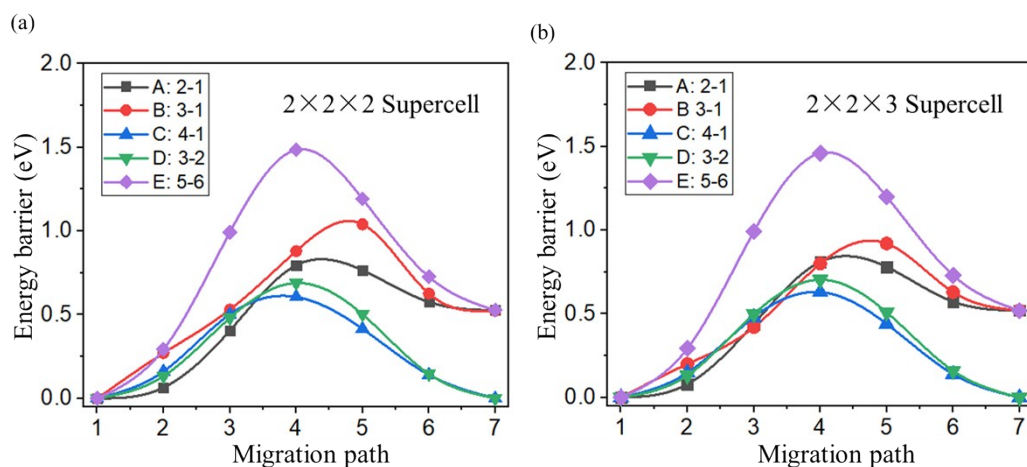


Figure S3. Energy barrier of Li diffusion in Cc $\text{Li}_2\text{MnSiO}_4$ in (a) a $2 \times 2 \times 2$ supercell and (b) a $2 \times 2 \times 3$ supercell

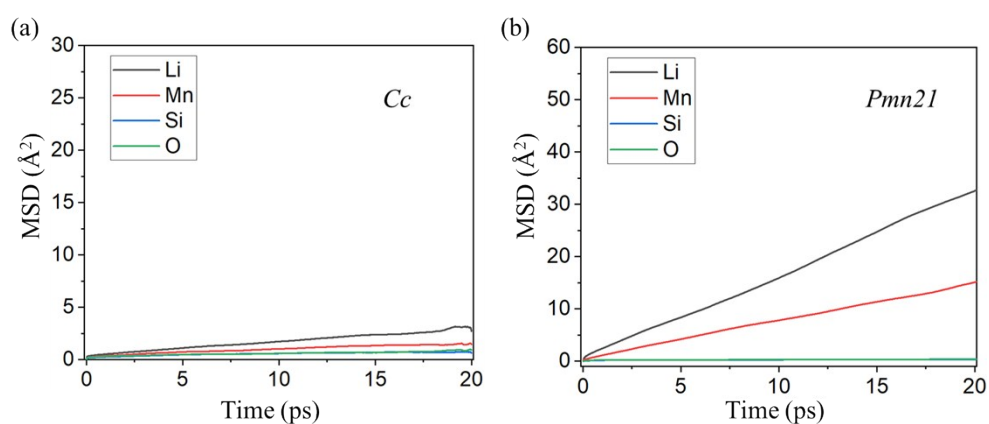


Figure S4. MSD for Li, Mn, Si, O atoms as a function of AIMD simulation time at different at 2000K for $\text{Li}_2\text{MnSiO}_4$ of (a) *Cc* phase and (b) *Pmn21* phase. Same carrier concentration is used that 8 vacancies are introduced into a $2 \times 2 \times 2$ supercell both for predict and experimental phases.

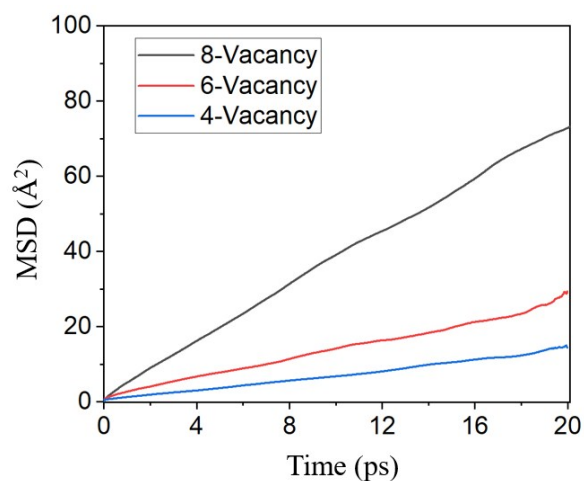


Figure S5. MSD of Li ion diffusion in a $2 \times 2 \times 2$ supercell of *Cc* $\text{Li}_2\text{MnSiO}_4$ with different carrier concentration by introducing 8, 6 and 4 vacancies.

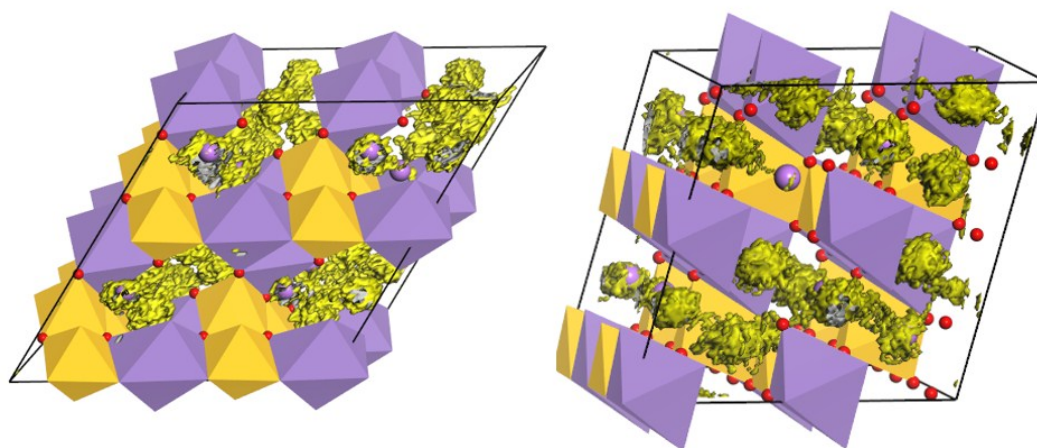


Figure S6. Projection of Li trajectories (yellow) in different views at 1500 K in a $2 \times 2 \times 2$ supercell

of *Cc* Li₂MnSiO₄.

References

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