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Supplementary Information

## Evidence for Intrinsic Nature of Band-Gap States Electrochemically Observed on

## Atomically Flat TiO<sub>2</sub>(110) Surfaces

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# Supplementary S1: A set of AFM images and RHEED patterns before and after UV illumination.



Figure S1 shows a set of AFM images and RHEED patterns for a different, but similar sample before and after UV illumination. After UV illumination, a spotty RHEED pattern that appears different from spots along a Laue circle characteristic to the  $TiO_2(110)$ -(1×1) surface before UV illumination, resulted from transmission of the primary beam through asperities rising above the plane of a rough surface, was observed and consequently the surface morphology, which had been atomically flat before UV illumination, became rough, but still can be seen the original steps-and-terraces structure. This result is a direct evidence for the photoetching that has been inferred to occur during UV illumination, but it is very limited to near the surface region.

Supplementary S2: Estimation of the defect density by a simple analytical model



The defect density was estimated using the analytical model proposed by Tomkiewicz in 1979(1). In this model, the following equations can be derived:

$$\frac{dU}{d(1/C^2)} = \frac{\varepsilon \varepsilon_0 e N_D}{2} + \frac{\varepsilon \varepsilon_0 e^2 N_D}{2C_H} \left(\frac{dN_{SS}}{d\Delta \phi_s}^+\right)$$
(1)

$$\Delta \phi_S = U - U_{fb} \tag{2}$$

$$\frac{dN_{ss}^{+}}{d\Delta\phi_{s}} = \frac{eN_{t}}{\sqrt{4\pi\lambda kT}} \exp\left(-\frac{e^{2}(U-U_{t})^{2}}{4\lambda kT}\right)$$
(3)

U: the electrode potential

 $U_{\rm fb}$ : the flat band potential

 $U_t$ : the most probable potential

 $N_{\rm D}$ : the donor density

 $N_{\rm SS}^{+}$  the area concentration of ionized surface states

 $N_{\rm t}$ : the defect density of the surface states

 $\epsilon_0$ : the vacuum permittivity

 $\epsilon$ : the relative permittivity of TiO<sub>2</sub> (=89) (2)

 $C_{\rm H}$ : the capacitance of the Helmholtz layer  $\lambda$ : the reorganization energy k: the Boltzmann constant

Figure S2 is a plot of the values taken from the derivative of the nonlinear Mott-Schottky plot as a function of U. It was fitted by Gaussian line shape to compare with the following theoretical function obtained by combining Eqs. (1)~(3).

$$\frac{dU}{d(1/C^2)} = y_0 + A \exp\left(-\frac{(U-x_0)^2}{width}\right)$$
(4)

Then each quantity can be obtained using the value of the Helmholtz layer capacitance  $C_{\rm H}$  63µC/cm<sup>2</sup> (1) as follows:

$$N_{D} = \frac{2y_{0}}{\varepsilon \varepsilon_{0} e}$$

$$U_{t} = x_{0}$$

$$N_{t} = \frac{C_{H} A \sqrt{width}}{e y_{0}} \sqrt{\pi}$$

$$\lambda = width / 4kT$$

Note that the values of reorganization energy ranged from 0.48 to 1.55eV, which is within a typical range of  $\lambda$  for semiconductors (3).

# **Supplementary S3:** Behavior of the donor density $N_D$ near the surface region



Figure S3 is a plot of the values of  $N_D$  as a function of UV illumination time ( $\bullet$ ), together with those after the subsequent reduction ( $\bullet$ ) and growing the TiO<sub>2</sub> film ( $\bullet$ ). With increasing the UV illumination time, the donor density increases, approaching to the nominal value of the Nb dopant concentration of  $1.4 \times 10^{20}$  cm<sup>-3</sup> (i.e. 0.5wt%). This indicates that the as-supplied Nb-doped TiO<sub>2</sub>(110) is likely to have a kind of dead layer near the surface region in which some of Nb dopants may become inactive. The photoetching process removes the dead layer, resulting in a fresh TiO<sub>2</sub> surface exposed. The donor density changed little or sometimes rather decreased even after the reduction, while the grown TiO<sub>2</sub> film had a small donor density of  $1.9 \times 10^{19}$  cm<sup>-3</sup> near the surface region. The origin of the donors is bulk oxygen vacancies and they can be easily re-oxidized near the film surface region during cooling the sample in O<sub>2</sub> (4).

#### References

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