

Supplementary Information

Role of the electronically-active amorphous state in low-temperature processed In_2O_3 thin-film transistors

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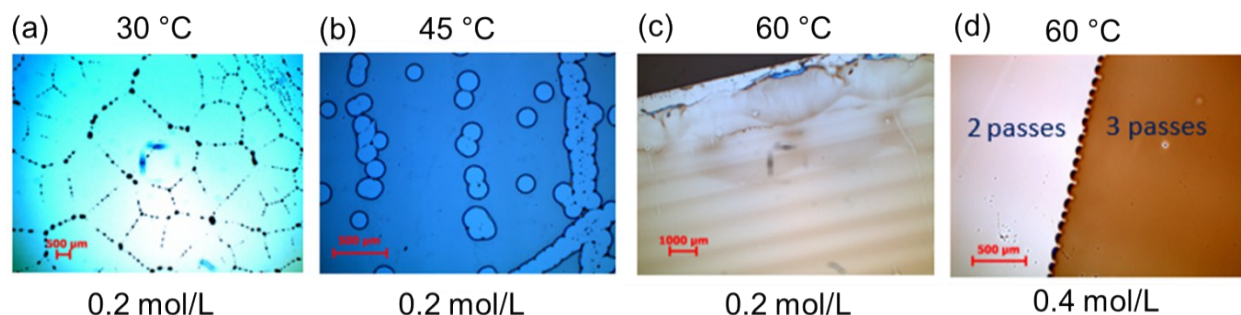


Figure S1. Optical micrographs of sg- In_2O_3 blade-coated films at various substrate temperatures. (d) corresponds to the optimized coating conditions and shows the film after 2 and 3 coats (with 250 °C annealing after each coat).

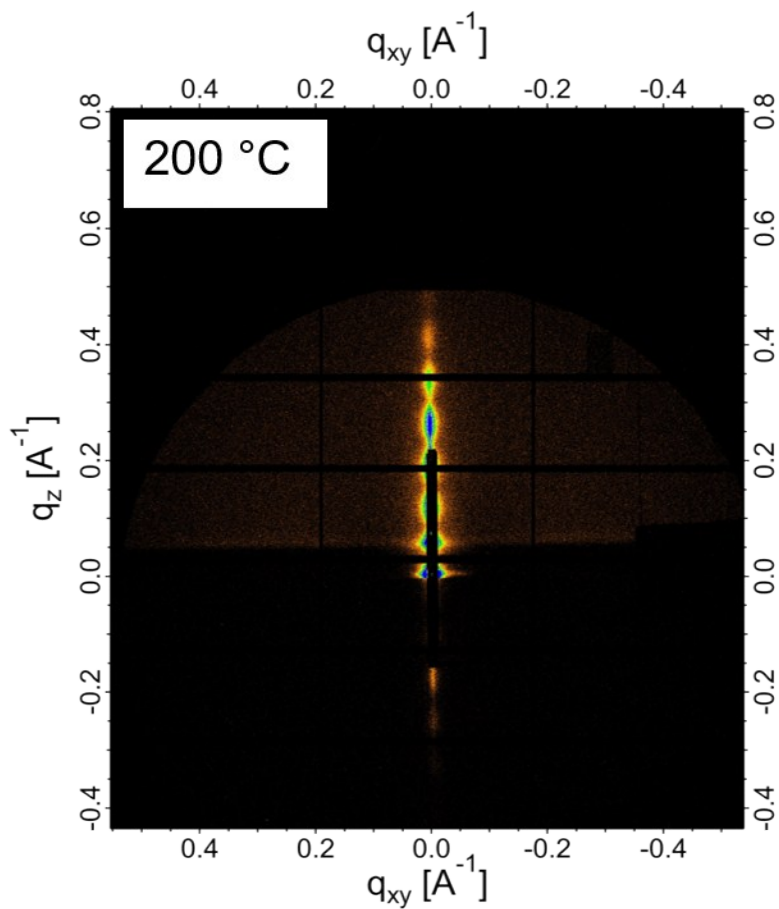


Figure S2. Grazing-incidence small-angle X-ray scattering (GISAXS) pattern of a representative 200 °C annealed bladed sg-In₂O₃ film. Periodic modulation of intensity seen along q_z (vertical) is due to the X-ray standing wave that forms inside the thin film and is evidence to the good quality and uniformity of the bladed In₂O₃ films used in this study.

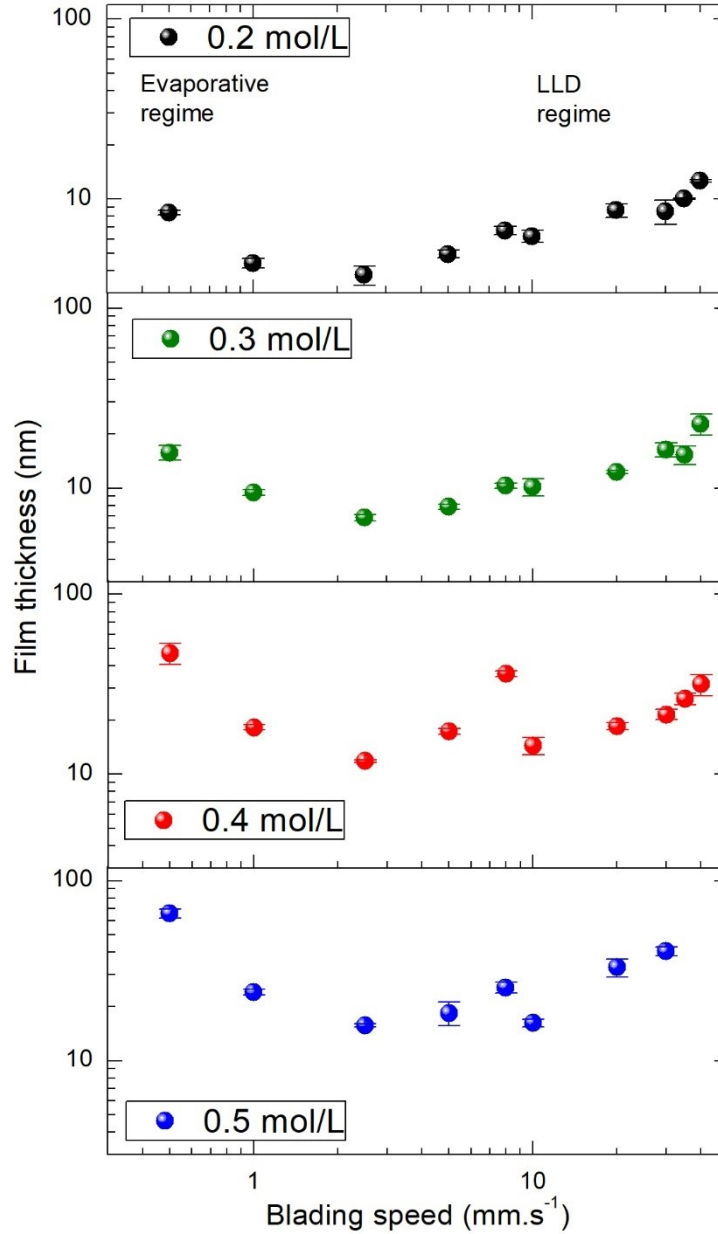


Figure S3. Thickness optimization of blade-coated sg-In₂O₃ films at 60 °C substrate temperature. Evaporative (lower speeds, negative slope) and Landau-Levich-Derjaguin (higher speeds, positive slope) coating regimes are clearly observed and suggest that the optimized films (5 mm s⁻¹) are coated in the Landau-Levich-Derjaguin regime. All uncertainties are ± one standard deviation of the mean.

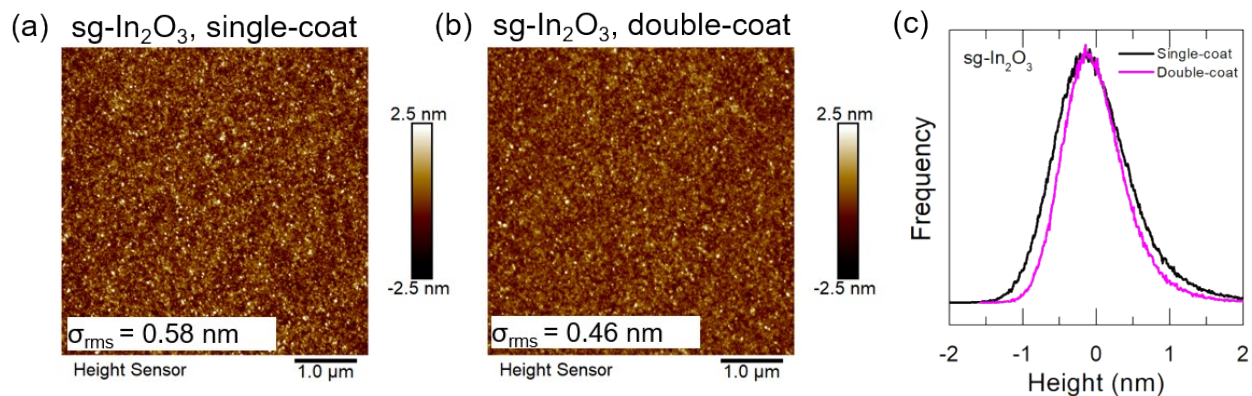


Figure S4. Atomic force microscopy (AFM) images showing the surface morphology of (a) single-coat (b) double-coat $\text{sg-In}_2\text{O}_3$ films, annealed at $325 \text{ }^\circ\text{C}$. (c) Height histogram from the two images showing a slightly narrower surface roughness distribution for the double-coat film (pink).

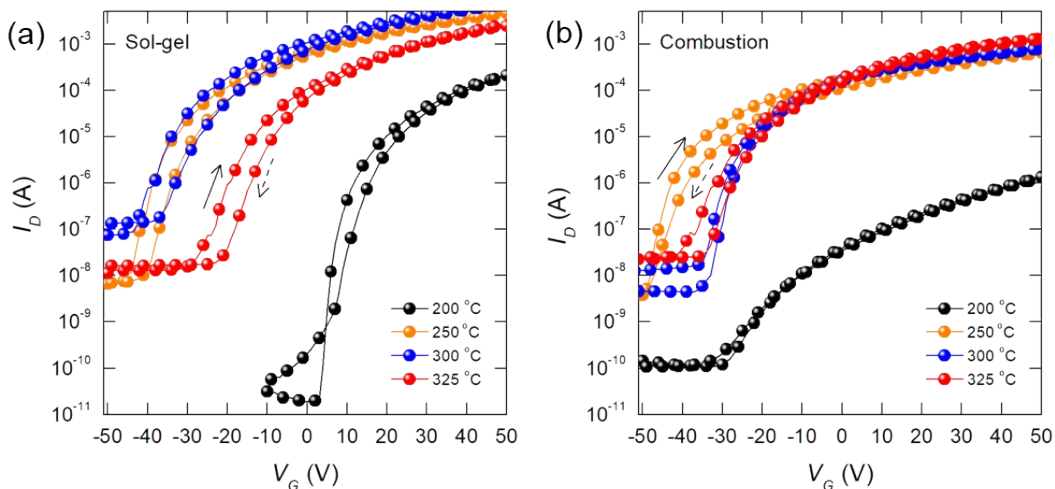


Figure S5. Forward and reverse sweeps of a few representative devices for (a) sg- and (b) $\text{c-In}_2\text{O}_3$ TFTs. Solid black arrow denotes forward sweep, while dashed black arrow denotes reverse sweep.

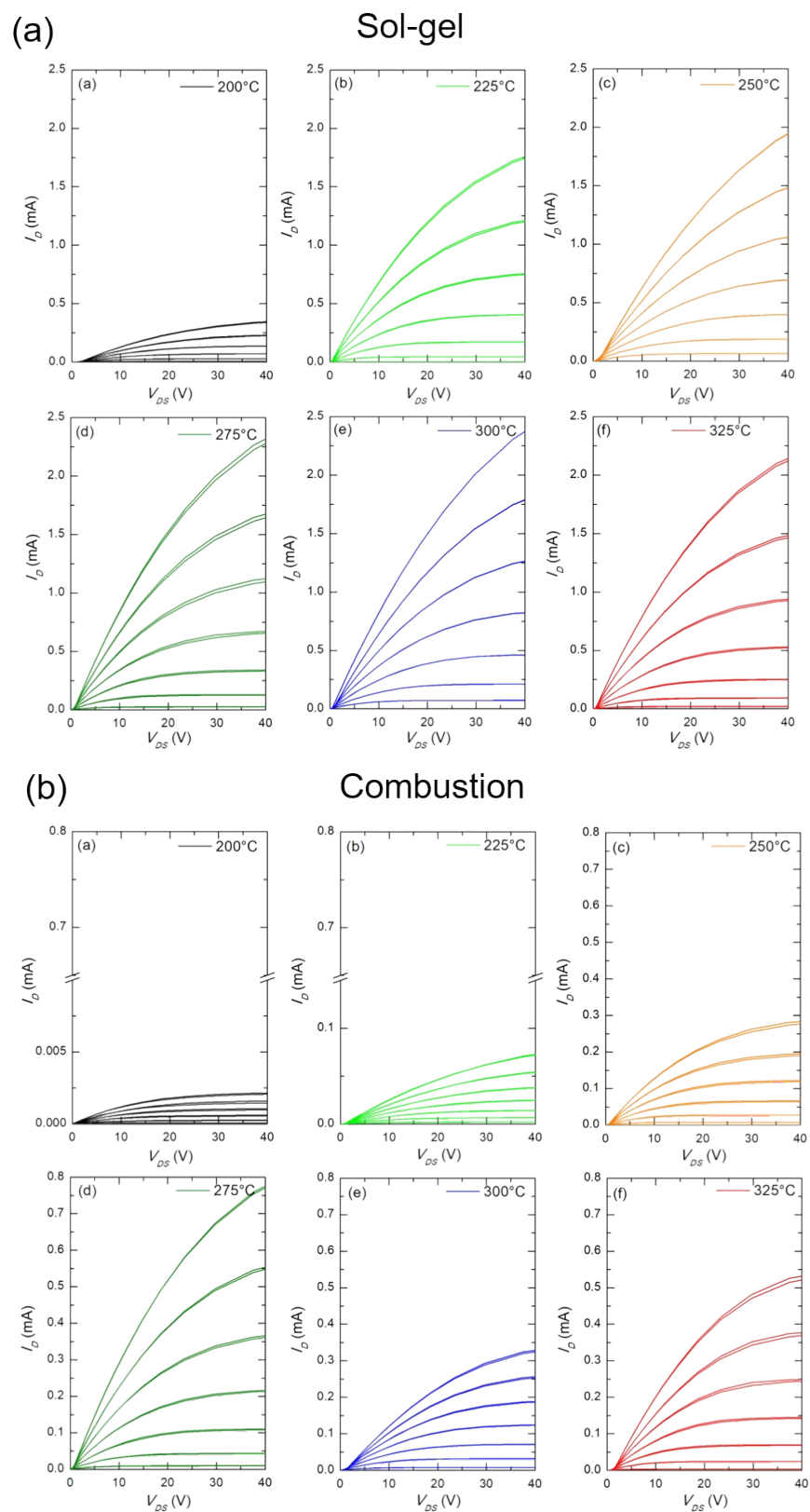


Figure S6. Representative output curves for (a). sg- and (b). c- In_2O_3 TFTs for all the annealing conditions considered in this study. $V_{GS} = 0$ V to 60 V, step = 10 V

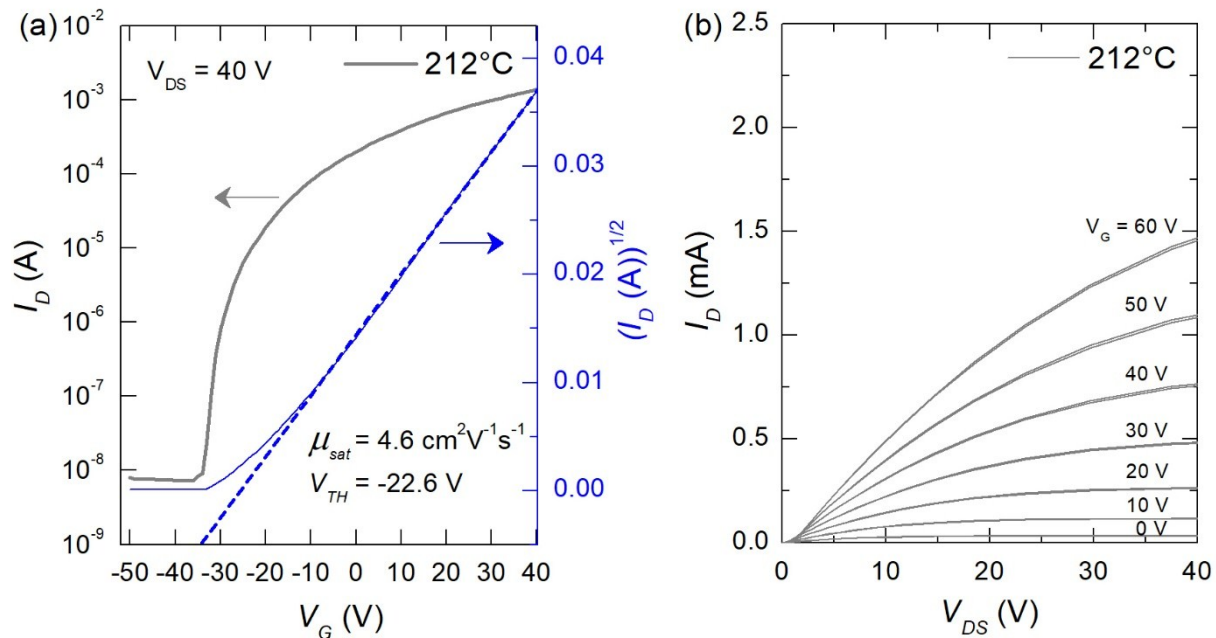


Figure S7. (a) Transfer, and (b) output curves for the best-performing 212 °C annealed sg-In₂O₃ TFT. Transfer curve was measured at a $V_{DS} = 40$ V, while output curves were measured in the range of $V_{GS} = 0$ V to 60 V with a step = 10 V.

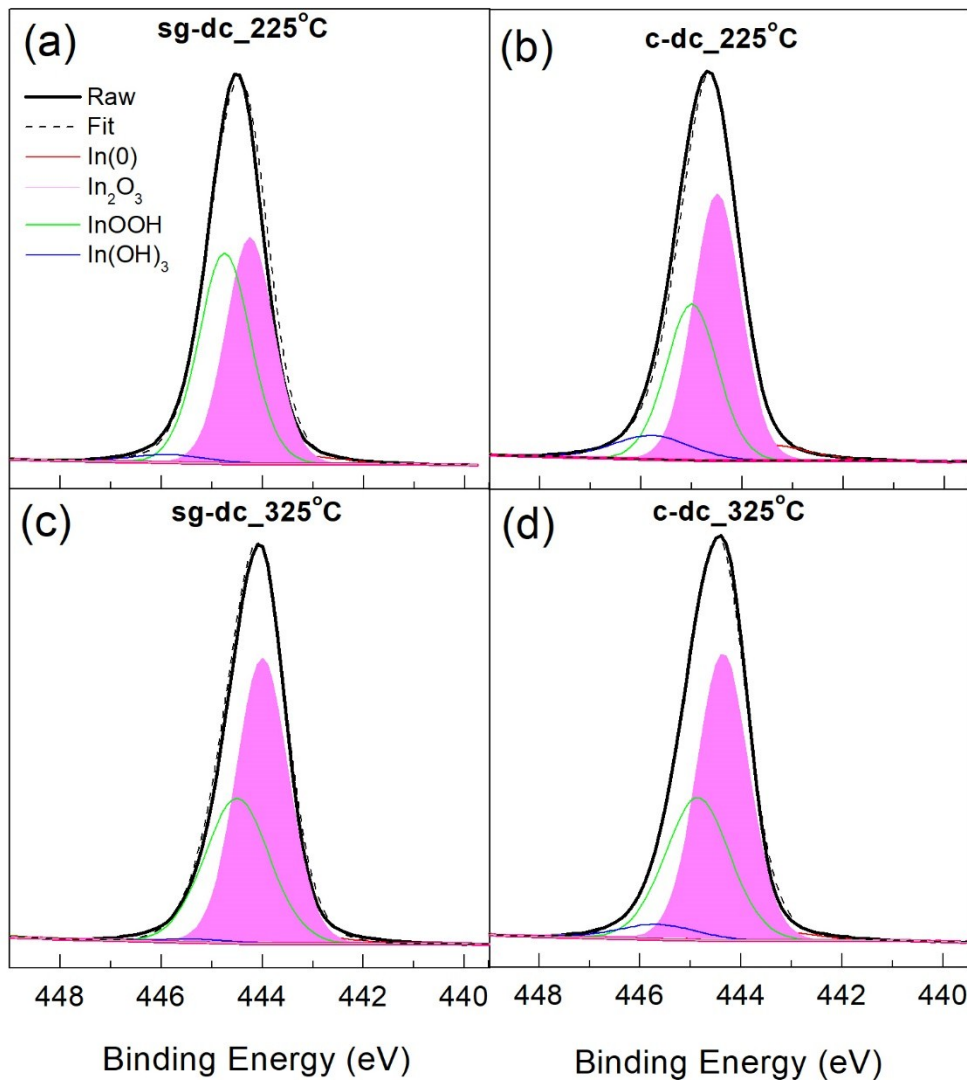


Figure S8. High-resolution X-ray photoelectron spectroscopy (XPS) spectra of the In $3d_{5/2}$ core level peaks. Pink component is the M-O-M lattice content. dc = double-coat (16 nm)

Table S1. Relative concentration of the various components of In $3d_{5/2}$ core level peak. dc = double-coat (16 nm)

| Sample | M(0) | M-O-M | M-OH | M-OR |
|-----------|------|-------|------|------|
| sg_dc_225 | 2.8 | 46.9 | 47.7 | 2.7 |
| sg_dc_325 | 2.3 | 59.1 | 37.6 | 1.0 |
| c_dc_225 | 5.5 | 52.9 | 33.9 | 7.7 |
| c_dc_325 | 3.0 | 57.0 | 35.5 | 4.5 |

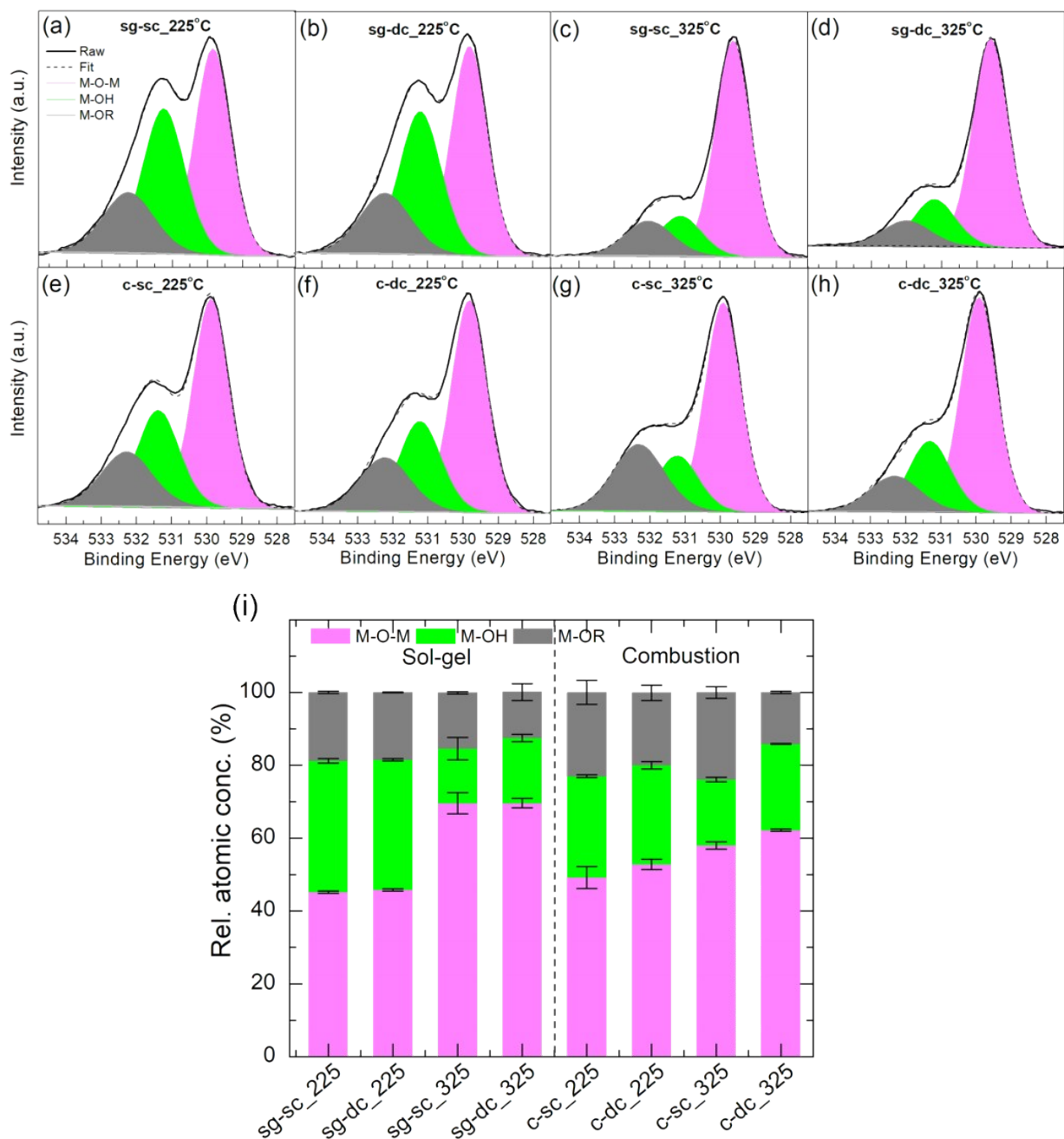


Figure S9. XPS data for O1s core level peak for the 8 nm single-coat (sc) films. For comparison, corresponding data on 16 nm double-coat (dc) is also shown. (i). Relative concentration of M-O-M, M-OH and M-OR components of the O1s peak. Error bars represent one standard deviation of the mean. sc = single-coat (8 nm), dc = double-coat (16 nm)

Table S2. M-O-M, M-OH and M-OR components (%) of the O1s core level for the different samples. sc = single-coat (8 nm), dc = double-coat (16 nm). \pm represents one population standard deviation.

| Sample | M-O-M | M-OH | M-OR |
|-----------|----------------|----------------|----------------|
| sg-sc_225 | 45.2 \pm 0.3 | 36.0 \pm 0.6 | 18.8 \pm 0.3 |
| sg-dc_225 | 45.8 \pm 0.3 | 35.7 \pm 0.3 | 18.5 \pm 0.1 |
| sg-sc_325 | 69.6 \pm 2.9 | 15.0 \pm 3.1 | 15.3 \pm 0.3 |
| sg-dc_325 | 69.6 \pm 1.3 | 17.9 \pm 1.0 | 12.6 \pm 2.3 |
| c-sc_225 | 49.2 \pm 3.0 | 27.8 \pm 0.4 | 23.0 \pm 3.3 |
| c-dc_225 | 52.8 \pm 1.4 | 27.2 \pm 1.0 | 19.9 \pm 2.1 |
| c-sc_325 | 58.0 \pm 1.0 | 18.1 \pm 0.6 | 23.9 \pm 1.6 |
| c-dc_325 | 62.2 \pm 0.3 | 23.7 \pm 0.1 | 14.1 \pm 0.3 |

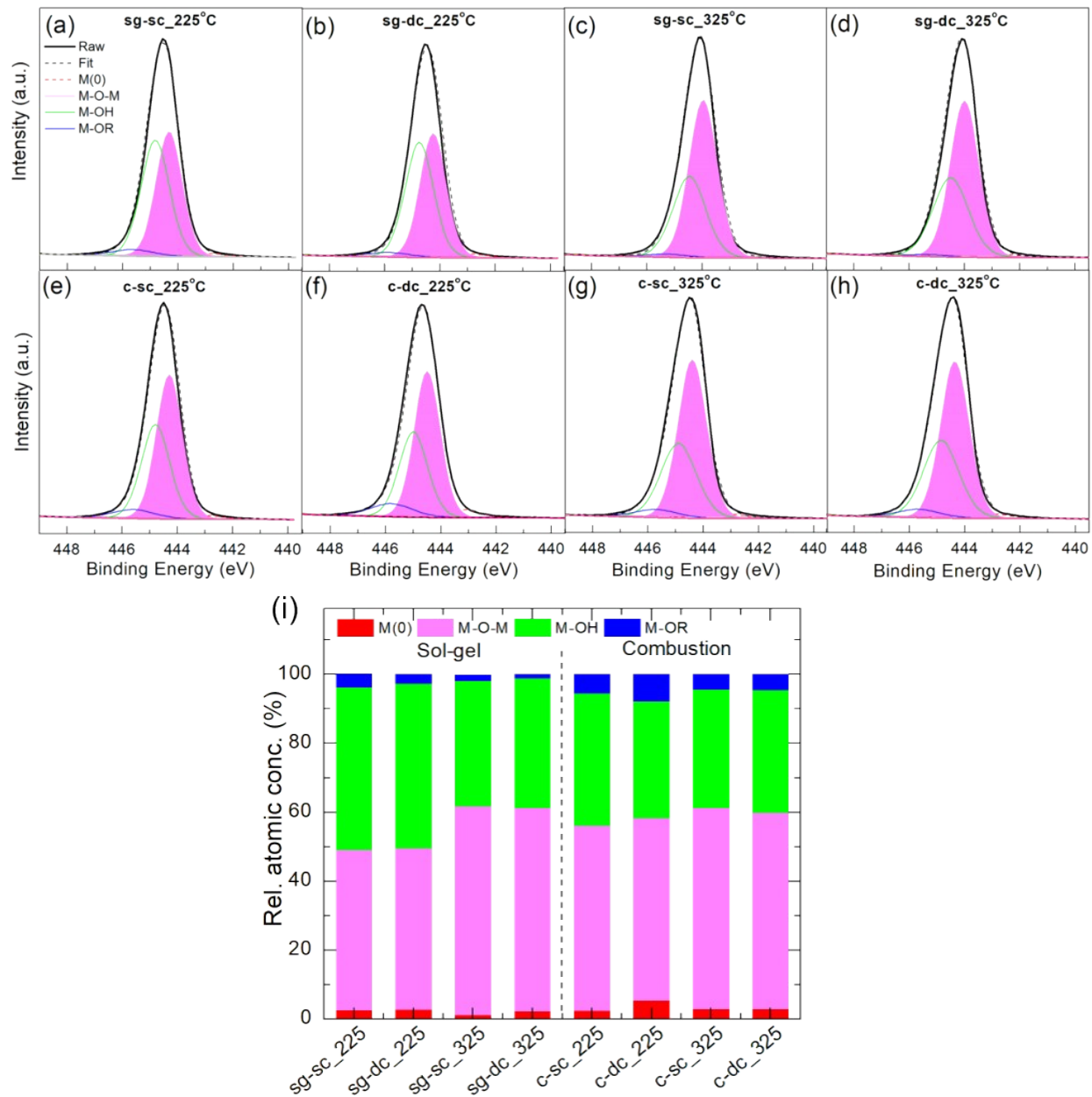


Figure S10. XPS data for In 3d_{5/2} core level peak for the 8 nm single-coat (sc) films. For comparison, corresponding data on 16 nm double-coat (dc) is also shown. (i). Relative concentration of the various components of In 3d_{5/2} peak. sc = single-coat (8 nm), dc = double-coat (16 nm)

Table S3. M(0), M-O-M, M-OH and M-OR components (%) of the In 3d_{5/2} core level for the different samples. sc = single-coat (8 nm), dc = double-coat (16 nm)

| Sample | M(0) | M-O-M | M-OH | M-OR |
|-----------|------|-------|------|------|
| sg-sc_225 | 2.7 | 46.4 | 47.2 | 3.9 |
| sg-dc_225 | 2.8 | 46.9 | 47.7 | 2.7 |
| sg-sc_325 | 1.3 | 60.5 | 36.4 | 1.7 |
| sg-dc_325 | 2.3 | 59.1 | 37.6 | 1.0 |
| c-sc_225 | 2.5 | 53.7 | 38.4 | 5.4 |
| c-dc_225 | 5.5 | 52.9 | 33.9 | 7.7 |
| c-sc_325 | 3.0 | 58.3 | 34.4 | 4.3 |
| c-dc_325 | 3.0 | 57.0 | 35.5 | 4.5 |

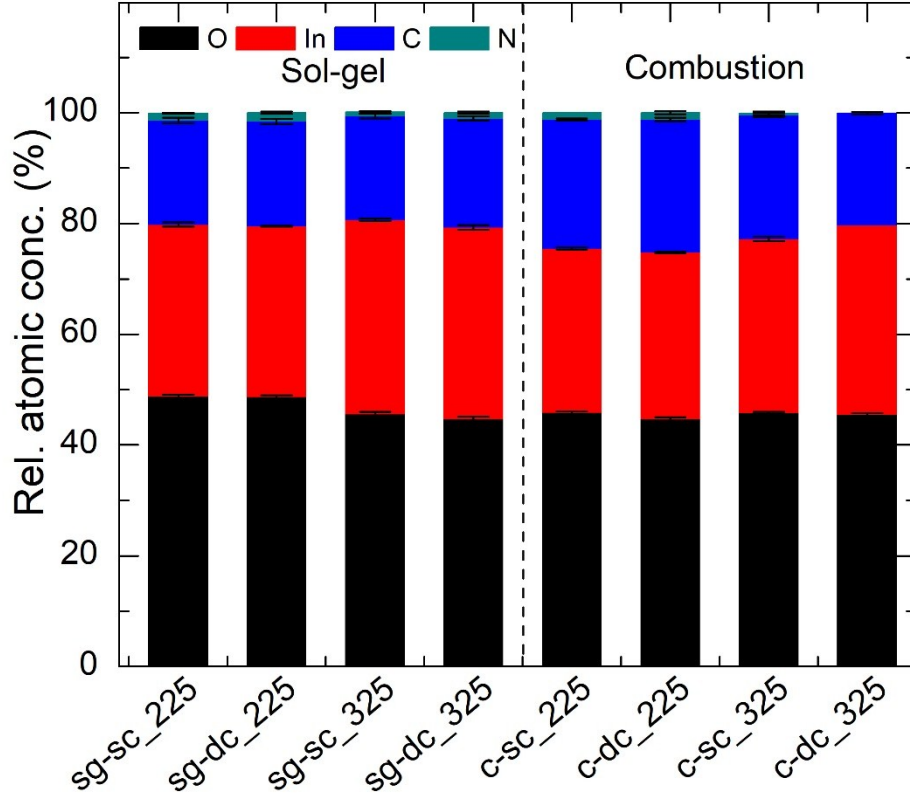


Figure S11. Relative atomic concentration of O, In, C and N for all the different In₂O₃ films considered in this study. sc = single-coat (8 nm), dc = double-coat (16 nm). Error bars represent one standard deviation of the mean.

Table S4. Relative atomic concentrations (%) for O, In, C and N for the different samples. Each value represents an average over 3 spots. sc = single-coat (8 nm), dc = double-coat (16 nm). ± represents one population standard deviation.

| Sample | O | In | C | N |
|-----------|------------|------------|------------|-----------|
| sg-sc_225 | 48.8 ± 0.3 | 31.0 ± 0.4 | 18.8 ± 0.5 | 1.3 ± 0.1 |
| sg-dc_225 | 48.7 ± 0.3 | 30.8 ± 0.1 | 18.9 ± 0.5 | 1.6 ± 0.2 |
| sg-sc_325 | 45.6 ± 0.4 | 35.1 ± 0.2 | 18.7 ± 0.4 | 0.7 ± 0.2 |
| sg-dc_325 | 44.7 ± 0.4 | 34.6 ± 0.4 | 19.7 ± 0.4 | 1.0 ± 0.2 |
| c-sc_225 | 45.9 ± 0.2 | 29.6 ± 0.2 | 23.3 ± 0.2 | 1.2 ± 0.0 |
| c-dc_225 | 44.7 ± 0.3 | 30.1 ± 0.1 | 24.0 ± 0.3 | 1.2 ± 0.3 |
| c-sc_325 | 45.8 ± 0.2 | 31.4 ± 0.4 | 22.4 ± 0.4 | 0.3 ± 0.3 |
| c-dc_325 | 45.5 ± 0.3 | 34.2 ± 0.0 | 20.2 ± 0.2 | 0.0 ± 0.0 |