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1 Appendix A. Supplementary data

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sample				condi	tion				lable
	NaW	NaWO ₄ ·H ₂ O(g)		$Pb(AC)_2 \cdot 3H_20(g)$		(ml)	Ethylene glycol(ml)		
1		0.53	0		3		1		WO ₃
2		0.53	0.0758		3		1		WP-1
3		0.53	0.1516		3		1		WP-2
4		0.53	0.2274		3		1		WP-3
5		0.53	0.3032		3		1		WP-4
6		0.53	0.3790		3		1		WP-5
7		0.53	0.4548		3		1		WP-6
8		0.53	0.62		3		1		PbWO ₄
		Т	able S2. 2	PW, PWI	H-5, and	W			
-		O 1s	W 4f	Pb 4f	VB	0	W	Pb	_
		(eV)	(eV)	(eV)	(eV)	(%)	(%)	(%)	
-	WO ₃	529 ^[1] .7	34.7	-	2.68	55.31	21.07	-	_
			36.8						
	WP-4	529.7	34.7	138.1	-	49.31	15.14	4.73	
			36.8	143.0					
	PbWO ₄	529.6	34.4	137.6	1.66	35.30	9.16	8.96	
			36.5	142.4					

Table S1. The short name of synthesized samples with different synthesis condition

Table S3. BET surface areas, pore volumes, and pore size of the samples

Samples	BET surface areas (m ² g ⁻¹)	Pore volume (cm ² g ⁻¹)	Pore size (nm)	
WO ₃	5.3826	0.022076	15.9904	
WP-1	37.5730	0.235167	25.0357	
WP-2	34.0102	0.185593	21.8279	
WP-3	9.4986	0.093349	39.3107	
WP-4	12.6849	0.087182	27.4916	
WP-5	1.8648	0.011727	25.1536	
WP-6	1.9938	0.026214	52.5989	
PbWO ₄	1.8770	0.016863	35.9362	

6 Table S4.The corresponding report of WO_3 , $PbWO_4$ and composite photocatalysts for RhB

7	degradation
'	acgradation

Photocatalysts	RhB (mg/L)	Time (min)	Dosage	Light Source	Removeal efficiency (%)	Refs.
WO ₃ /Ag/Ag ₃ PO ₄	20	120	100ml/100mg	UV filter 500 W Xe-lamp	92%	[2]
WO ₃ /MoS ₂	10	120	100ml/50mg	500 W tungsten halogen lamp	98.2%	[3]
WO ₃	4	150	100ml/30mg	UV filter 300 W Xe-lamp	99%	[4]
WO ₃ /Bi ₂ WO ₆	5	150	10ml/10mg	UV filter 500 W Xe-lamp	78.1%	[5]
PbWO ₄ :9%Eu ³⁺	5	240	50ml/50mg	UV light 500 W mercury lamp	90.85%	[6]
PbWO ₄	2.4	105	50ml/50mg	UV filter 500 W Xe-lamp	100%	[7]
WP-4	5	30	100ml/50mg	300 W Xe lamp	98.6%	This work

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10 Fig. S1 SEM images of WP-4 at various reaction stages (a) 1h, (b) 3h, (c) 5h, (d) 8h, (e) 12h.

It can be seen from the Fig.S1 that with the extension of reaction time, the rod structure material shows a growth process, from the initial 2-3 um to 5-6 um. The rod-shaped sample is lead tungstate (it has been proved in the paper), and the nanoparticles on the surface of $PbWO_4$ are WO_3 . In the process of reaction, the nanoparticle tungsten trioxide gradually compounds on the lead tungstate surface, forming a close contact phase interface.



Fig. S2. (a) The band gap of WO₃ and PbWO₄. (b) The internal electric field and band edge
bending at the interface of WO₃/PbWO₄. (c) The Z-scheme charge transfer mechanism between WO₃
and PbWO₄ under light irradiation



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Fig. S3 The internal electric field

22 When semiconductors with different work function combined into a composite, electrons are migrated from one semiconductor to another, forming internal electric field (IEF) around the interface, 23 which becomes the driving force of photogenerated carrier migration^[8]. Recently, IEF engineering has 24 emerged as an efficient method to improve photogenerated charge separation^[9-11]. Scholars have found 25 that interface interaction can enhance IEF^[12-14]. Many composite photocatalysts have been 26 27 demonstrated to construct IEF at heterogeneous interface. Such as, g-C₃N₄@Bi₂WO₆^[15], CdS/BiOI/WO3^[16], WO3/g-C3N4^[17], CdS/WO3^[1], Bi2WO6/BiOI^[18]. Therefore, when WO3 and PbWO4 28 are combined, IEF will be built due to different work function. WO₃ and PbWO₄ are both n-type 29 30 semiconductor, the Fermi level of them is located near the CB of them. Based on the CB position 31 results of this work, it suggests the work function of WO₃ is higher than $PbWO_4^{[19, 20]}$. The 32 establishment process of the built-in electric field is described in Fig. S2. When $PbWO_4$ contacts with 33 WO₃, electrons of $PbWO_4$ could transfer to WO₃ until reaching equilibrium, then band bending at grain 34 boundary is established, and an IEF with $PbWO_4$ pointing to WO₃ is formed (Fig. S2b). Under light 35 irradiation, IEF drives the photogenerated holes of $PbWO_4$ to move toward WO₃, while the 36 photogenerated electrons of WO₃ move in the opposite direction, thus photogenerated carriers of WO₃ 37 and $PbWO_4$ are separated according to the Z-type system (Fig. S2c).





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Fig. S4 (a) Steady-state PL spectra, (b) time-resolved transient PL decay

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of WO₃, PbWO₄ and WP-4

The steady-state photoluminescence (PL) spectrum reveals the carrier separation and transfers 41 42 efficiency of the composites, as shown in Fig. S4a. It displays that the PL peak intensity of WP-4 is 43 lower than WO₃ and PbWO₄, indicating that the recombination of photogeneration electron-hole pairs in WP-4 is forcefully suppressed due to the rapid separation of charge carrier across grain boundary. 44 Moreover, the time-resolved photoluminescence spectroscopy (TRPL) is employed to elaborate on the 45 specific charge carrier dynamics of WO₃, PbWO₄ and WP-4 (Fig.S4b). The fluorescent lifetime of PW-46 47 4 (10.3 ns) is larger than WO_3 (7.7 ns) and $PbWO_4$ (8.8 ns), indicating the photogenerated carriers of PW-4 need more times to decay than WO3 and PbWO4 after stopping illumination, which increases the 48 possibility of photocarriers to participate in photo-oxidation reaction^[21]. 49





Fig. S5 UV-Vis absorption spectrum of PW-4 (a), UM (b), and PC (c)





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Fig. S6 (a) EDX spectra and (b) TEM with definite phase interface of the PC

55 According to the mass ratio of the WP-4, PbWO₄/WO₃ composite with interface is obtained according to He, Hongcai's^[22] research. The EDX spectra and high-resolution TEM image of the 56 physically contacted WO₃/PbWO₄ composite (PC) are measured to further prove that the combination 57 of two similar crystal units can form a high-quality contact interface (Fig. S6 (a, b)). From Fig. S6b, a 58 59 clear interface is observed, which is different from the TEM result of grain boundary with continuous 60 phase transition. Due to PW-4 is connected by the same crystal structure unit, strong interaction exists between WO₃ and PbWO₄, and no gap between these two parts. As result, it is easier for photocarriers 61 to transmit and separate in two parts. 62



63 64

Fig. S7 UV-Vis absorption spectrum of PW-4

According to the UV-vis absorption spectra of RhB at different degradation time and the blank experiment (same condition but without RhB), as shown in Fig. S7. The slight blue shift of RhB maximum absorption wavelength is attributed to the N-deethylation of RhB. The main absorption peak disappears after 40 min, indicating the RhB molecule was thoroughly cracked. Carefully, the peaks below 400 nm disappear after 20 min, suggesting small aromatic ring compounds gradually disappear. The blank experiment result (the range of 300 nm - 200 nm), indicating the baseline will be raised

- 71 when using photocatalyst, which may be due to the small particles contained in centrifugal supernatant.
- 72 The same phenomenon is also observed in the photocatalytic degradation experiment of RhB over PW-
- 73 4. But the baseline is a little higher than a blank experiment, which may be due to some kinds of small
- 74 molecules exist^[23-26]. However, most RhB will be mineralized over PW-4 under illumination.
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