SUPPLEMENTARY INFORMATION

- 2 Electrochemical Reduction of CO₂ to Ethylene on Cu/Cu_xO.GO composites in Aqueous
- 3 Solution

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7 The role of GO in the electrolyte on the synthesis of composites bath was accounted for

8 by taking EIS of the electrolyte bath with and without GO and with Pt or CFP as working

9 electrode.

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11 Figure SI-1: EIS of Pt and CFP in a 0.2 M TSC solution with or without GO. Inset showing the

12 change in the solution resistance. The graphs were fitted with $[R_s([RQ]Q) W]$ circuit.

13 Material Characterization:

The XRD patterns of pCu/Cu_xO and pCu/Cu_xO-GO (figure SI-2 (b)) depicts major 14 diffraction peaks positioned at 43° and 51° corresponding to (111) and (200) planes of fcc 15 structure of metallic copper (JCPDS No. 85-1326). Few minor peaks located at 36° and 42° 16 related to diffraction from (111) and (200) planes of fcc Cu₂O (JCPDS No. 05-0667) can also 17 be noticed in these XRD patterns. Interestingly, upon addition of GO in case of pCu/Cu_xO-GO, 18 comparatively more oxide phase of copper was observed than pCu/Cu_xO was noticed. The 19 XRD pattern of pCu/Cu_xO-GO depicts diffraction peaks positioned at 36° and 42° related to 20 reflections from (111) and (200) planes of Cu₂O phase. Besides, few peaks located at 35°, 38°, 21 and 49° corresponding to diffraction from (111), (111), and (200) planes of monoclinic CuO 22 (JCPDS file No. 48-1548), few significant peaks from metallic Cu are also noted. Thus, the 23 presence of GO in the electrophoretic setup seems to favour the oxide phase than the metallic 24 Cu in pCu/Cu_xO-GO which is a manifestation of oxygen group transfer from GO to copper 25 26 occurring when copper interacts with the available -C-O bonds of GO. This oxide phase preference can be realised from XRD peak ratio of Cu_xO/Cu as given in table SI-1; composites 27 have more intense peaks for Cu_xO than Cu. 28 The XRD patterns for fCu/Cu_xO, fCu/Cu_xO-GO, and fCu/Cu_xO@GO (figure SI-2 (c)) 29

30 show the combination of different copper phases and again addition of GO leads to more oxide 31 phases as inferred from pCu/Cu_xO and pCu/Cu_xO -GO (table SI-1). However, the change in

cathode material from Pt to CFP does not appreciably change the crystallographic phases 32 obtained through electrochemical deposition though morphological changes are observed from 33 FESEM images. These observations from the XRD project the influence of GO on phase 34 preference of copper/copper oxide nano-structures which we attribute to the interaction 35 between GO and copper, where the former one is reduced and the later adheres to GO as oxide. 36 This variation in crystalline phases would directly influence the electrocatalytic activity of 37

- these copper-based electro-catalyst materials. 38
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SI-2. XRD patterns of Cu/Cu_xO and its composites with GO. (a) overlay of Cu/Cu_xO -GO 42 composites with intense CFP peaks; (b) (a trace) pCu/Cu_xO , and (b trace) pCu/Cu_xO-GO ; (c); 43 (c trace) fCu/Cu_xO , (d trace) fCu/Cu_xO-GO , (e) $fCu/Cu_xO(a)GO$. The bar diagram in both the 44 panels corresponds to JCPDS data from file numbers 48-1548 of fcc-CuO (green bars), 05-45 0667 of fcc-Cu₂O (red bars), and 85-1326 of fcc-metallic Cu (black bars). 46

Further to support the formation of a composite and deduce an idea about the extent of 47 reduction of GO into rGO (reduced graphene oxide because of composite formation) (which 48 also reflects the extent of defects in GO sheets manifested in I_D/I_G ratio) hence a presentation 49 of extent of metal - GO interactions. Raman spectra were recorded for all the composite 50 samples containing GO, as shown in the SI-3. The characteristic peaks were observed for 51 pristine GO sample i.e. peaks positioned at 1350 cm⁻¹ corresponding to D band (related to 52 degree of order/disorder due to a breathing k-point phonon of A1g symmetry) and at 1590 cm⁻ 53 ¹ corresponding to G band (related to the E_2g phonon of sp^2 hybrid carbon atoms, which is an 54

- indicator of the stacking structure) [1]. All the composite samples demonstrate an enhancement 55 in intensity of D band and lowering in intensity of G band as compared to pristine GO 56 indicating the formation of a composite between Cu/Cu_xO and GO. Furthermore, ratio of 57 intensities of D band to that of G band (i.e. I_D/I_G ratios) estimated from the Raman spectra (table 58 SI-1) suggest a trend which is in accordance with the observations made from XRD and EDX 59 analysis. Particularly, electrodeposited sample using Pt cathode i.e. pCu/Cu_xO-GO shows 60 highest I_D/I_G intensity ratio of 1.23, followed by fCu/Cu_xO@GO with $I_D/I_G = 1.19$, fCu/Cu_xO-61 GO with $I_D/I_G = 1.05$, pCu/Cu_xO@GO with $I_D/I_G = 0.92$, and pristine GO showing the lowest 62 I_D/I_G ratio of 0.88. Pertinent to mention that higher I_D/I_G ratio reflects a greater extent of charge 63
- 64 shared between metal and the GO support.



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66 **Figure SI-3.** Raman spectra of the composites with D and G bands at 1350 cm⁻¹ and 1590 67 cm⁻, (a) $fCu/Cu_xO@GO$, (b) GO, (c) pCu/Cu_xO-GO , (d) fCu/Cu_xO-GO and (e)

68 $pCu/Cu_xO@GO$.

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70 *Table SI-1: Summary of the physico-chemical parameters obtained from XRD and Raman* 71 *spectral analysis.*

Samples	GO	Cathode	I_D/I_G ratio	Cu _x O/Cu before	Cu _x O/Cu after
	support	(synthesis)		electrolysis	electrolysis
pCu/Cu _x O		Pt		0.890	0.593
pCu/Cu _x O-GO	GO	Pt	1.31	2.60	0.82
fCu/Cu _x O		CFP		1.17	0.977
fCu/Cu _x O-GO	GO	CFP	1.01	2.02	1.263
fCu/Cu _x O@GO	GO	CFP	0.93	0.851	0.536
pCu/Cu _x O@GO	GO				
GO	GO		0.92		

73 SI-4: Elemental mapping of different samples.

All samples were found devoid of any impurity, and distribution of constituent elements washomogenous.

76 (a) pCu/Cu_xO



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83 (c) fCu/Cu_xO



(e) fCu/Cu_xO@GO



Figure SI-4: Detailed EDX elemental scans for the materials.



Figure SI-5: Detailed XPS scan for C 1s in the materials prepared, showing peaks for -C-C,

-C=O and -COOH groups. Materials without GO show presence of all three functional groups

102 while materials where GO was added show that -COOH group is no longer present suggesting

- 103 locking of the group by interacting with metal centre. (a) pCu/Cu_xO , (b) pCu/Cu_xO -GO, (c)
- fCu/Cu_xO , (d) fCu/Cu_xO -GO, and (e) fCu/Cu_xO @GO.



Figure SI-6: Detailed XPS scan for O 1s in all of the materials showing peaks for oxide phase 107 and -OH phase. (a) pCu/Cu_xO , (b) pCu/Cu_xO -GO, (c) fCu/Cu_xO , (d) fCu/Cu_xO -GO, and (e)

 $fCu/Cu_xO@GO$.

110 Indie SI-2 . Al S peak positions atomy with ratio of $Ca = 10 Ca$	110	Table SI-2: XPS	peak positions	along with	ratio of Cu^{2+}	to Cu^+ .
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S.No.	Sample	Cu 2p	O 1s	C 1s	Cu ⁺⁺ /Cu ⁺ (area)
1	pCu/Cu _x O	932.6	530.7	284.6	1.64
		933.8	533.7	286.6	
				290.0	
2	pCu/Cu _x O-GO	932.5	530.0	284.6	3.99
		933.9	531.6	286.5	
3	fCu/Cu _x O	931.8	531.1	284.4	1.56
		933.7	521.8	286.5	
				288.7	
4	fCu/Cu _x O-GO	931.8	530.9	284.4	4.81
		933.5	531.6	286.5	
5	fCu/Cu _x O@GO	932.8	531.1	284.7	1.25
		934.0	533.5	285.4	
				288.3	



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113 *Figure SI-7:* (a) LSV traces of $fCu/Cu_xO@GO$ and $pCu/Cu_xO@GO$ and GO in 0.2 M KHCO₃ 114 saturated with CO_2 . (b) Current as a function of scan rate to calculate capacitance of the 115 materials, $\blacktriangle pCu/Cu_xO$, $\triangleright pCu/Cu_xO$ -GO, $\blacktriangledown fCu/Cu_xO$, $\diamond fCu/Cu_xO$ -GO, $\bullet fCu/Cu_xO@GO$, 116 $\square pCu/Cu_xO@GO$.

117 The only detected liquid product was formic acid (NMR) at 8.4 ppm and its accumulative

118 concentration for one hour was highest recorded at fCu/Cu_xO with 8%, and it was recorded at

119 3.9% with pCu/Cu_xO-GO and not detected at electrodes modified with fCu/Cu_xO@GO and

120 pCu/Cu_xO@GO.



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122 *Figure SI-8*: The time-dependent current density and F.E. curves for the production of ethylene

123 on pCu/CuxO-GO at -0.98 V vs RHE over the time span of 100 minutes.

Electrochemical Impedance analysis (EIS): To probe the charge transfer dynamics, EIS studies (SI-9) at different potentials and in presence and absence of CO_2 were recorded. The equivalent circuits for the Nyquist plots are given in Figure SI-9. At open circuit potential in presence of argon, charge transfer resistances are more than in presence of CO_2 suggesting that reduction of CO_2 may be easy on these surfaces than HER, occurring in absence of CO_2 , and pCu/Cu_xO-GO composites show better electrochemical activity. At the onset potential, the reaction starts and charge transfer resistances in argon saturated are lowered and suggesting

- 131 HER occurring at the surfaces. In presence of CO₂, pCu/Cu_xO-GO shows minimum resistance
- 132 hence should be efficient towards the ERC; also, pCu/Cu_xO -GO is not very good for HER even



133 in absence of CO_2 .

136 *Figure SI-9*: Nyquist plots recorded in presence and absence of CO_2 at OCPs (upper panel),

137 and onset potentials (bottom panel). (a) $fCu/Cu_xO@GO$, (b) fCu/Cu_xO-GO , (c) pCu/Cu_xO , (d) 138 fCu/Cu_xO (c) pCu/Cu_xO , (d)

138 fCu/Cu_xO , (e) pCu/Cu_xO -GO and (f) pCu/Cu_xO @GO.



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140 Figure SI-10 (a): Equivalent circuit in Argon saturated solution.



142 *Figure SI-10 (b):* Equivalent circuit in CO₂ saturated solution.

143 Stability of catalysts:

144 ERC at higher negative potentials is associated with reduction of the copper oxide and this reduction leads to the degradation of catalyst activity and breakdown of morphology. The 145 nano-morphological stability of the materials after prolonged electrolysis was checked by 146 analyzing the spent (used materials) with XRD and FSEM. Figure SI-10 shows the phase 147 148 transformation of the catalysts after electrolysis of 7 hours at -0.985 V vs RHE. Peaks because of metallic copper are more intense than the copper oxide peaks which is due to the 149 electrochemical reduction of CuxO during ERC. The morphology of the catalysts after bulk 150 electrolysis was found to be intact as seen in FESEM images, SI-11, wherein pCu/CuxO, 151 conical structures are not ubiquitous. The pCu/CuxO-GO shows agglomeration of particles but 152 153 looks similar to the unspent assembly of particles on GO. fCu/CuxO and fCu/CuxO-GO show no pronounced effect of prolonged electrolysis, while fCu/CuxO@GO shows agglomeration 154 of the particles over GO sheets. This morphological stability even after 7 hours of continuous 155 electrolysis highlights the stability and durability of these synthesized catalysts. 156



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158 *Figure SI-11:* XRD graphs of the catalyst surfaces post-electrolysis at -0.985V, (left panel) (a 159 trace) pCu/Cu_xO , (b trace) pCu/Cu_xO -GO, and (right panel) (c trace) fCu/Cu_xO , (d trace) 160 fCu/Cu_xO -GO, (e trace) fCu/Cu_xO (a)GO.



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- 163 Figure SI-12: morphological transformation of the spent materials. The images show that the
- 164 prolonged electrolysis does not affect the morphology of composites and hence the catalysts
- 165 are stable for longer run times for ERC, (a) pCu/Cu_xO , (b) pCu/Cu_xO -GO, (c) fCu/Cu_xO , (d)
- 166 pCu/Cu_xO -GO and (f) fCu/Cu_xO @GO.

167 **References:**

- 168 1. R. Saito, M. Hofmann, G. Dresselhaus, A. Jorio, M. S. Dresselhaus, Advances in Physics, 2011,
- 169 **60.**