

Supporting Information

Ultrastable Lithium-Sulfur Batteries with Outstanding Rate Capability Boosted by NiAs-Type Vanadium Sulfides

Chao Yue Zhang,^a Guo Wen Sun,^a Yun Fei Bai,^b Zhe Dai,^a Yi Rong Zhao,^a Xiu Ping Gao,^a Geng Zhi Sun,^{c,d} Xiao Bo Pan,^b Xiao Jun Pan*^a and Jin Yuan Zhou*^a

^a Key Laboratory for Special Function Materials & Structural Design of the Ministry of Education, and School of Physical Science & Technology, Lanzhou University, 222 South Tianshui Road, Lanzhou 730000, China

^b State Key Laboratory of Applied Organic Chemistry, College of Chemistry and Chemical Engineering, Key Laboratory of Nonferrous Metal Chemistry and Resources Utilization of Gansu Province, Lanzhou University, Lanzhou, 730000, China

^c Key Laboratory of Flexible Electronics & Institute of Adv. Mater., Nanjing Tech University, 30 South Puzhu Road, Nanjing 211816, China

^d School of Material Science and Engineering, Henan Polytechnic University, 2001 Shiji Road, Jiaozuo 454003, China

XRD patterns

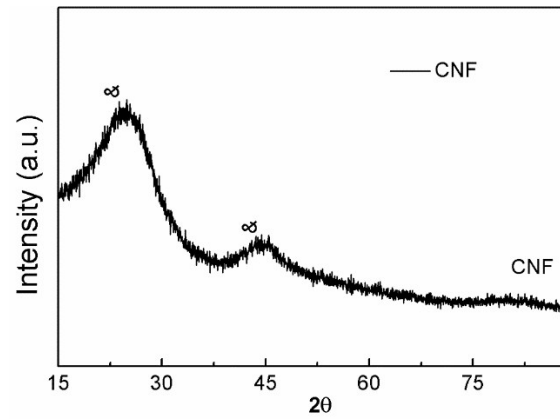


Figure S1 XRD patterns curve of the CNF.

SEM images

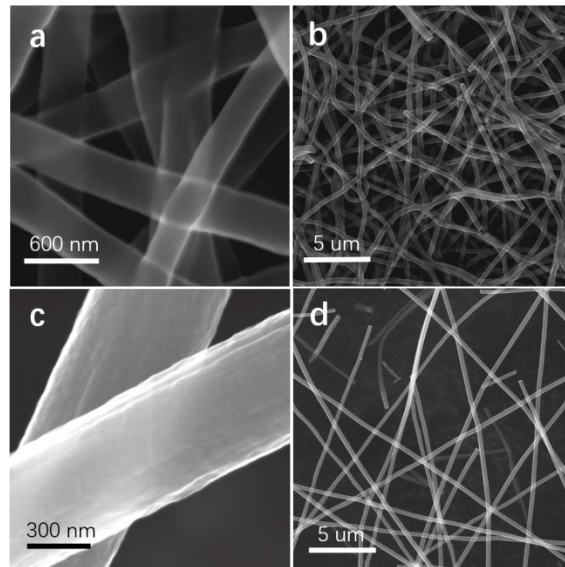


Figure S2 High and low-resolution SEM images of CNF and CNF@VO_{0.9}.

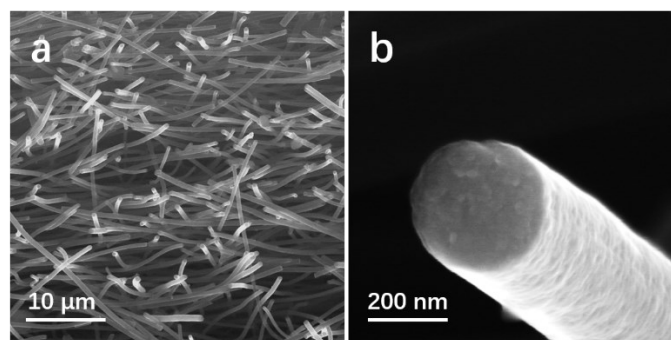


Figure S3. (a) SEM image of cut V₂S₃ nanofibers, and (b) cross-section view SEM image of a CNF@V₂S₃ composite nanofiber.

Raman pattern

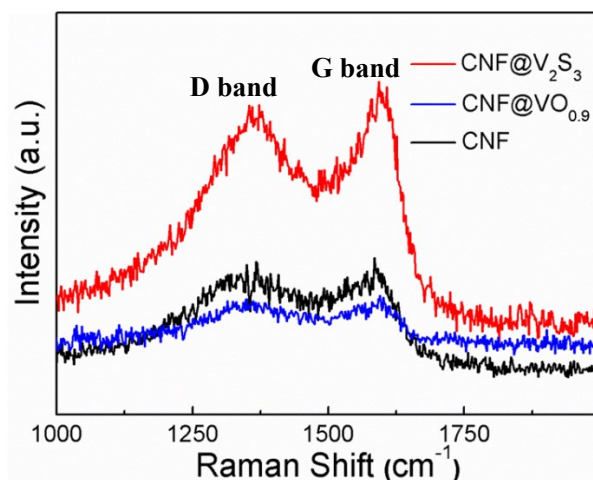


Figure S4 Raman spectrum of CNF@V₂S₃, CNF@VO_{0.9} and CNF.

XPS spectra of O 1s cores

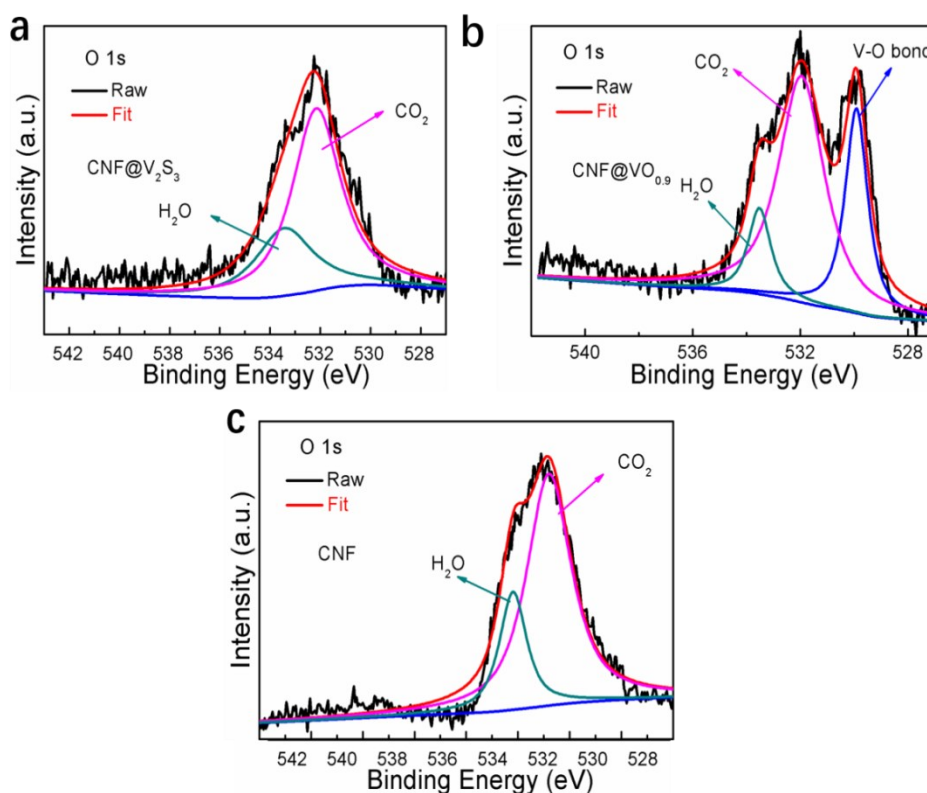


Figure S5 O elemental signal image of XPS of CNF@V₂S₃ (a) CNF@VO_{0.9} (b) and CNF (c).

To make clear the origin of oxygen, the high-resolution O 1s XPS spectra of CNF@V₂S₃, CNF@VO_{0.9} and pure CNFs have been studied together. As presented in

Figure S4, the O 1s XPS peak of CNF@V₂S₃ are similar with that of the pure CNFs, and can be decomposed into two components, located at 533.5 and 532.0 eV, respectively; while the O 1s XPS peak of CNF@VO_{0.9} can be decomposed into three components, located at 533.5, 532.0, and 529.9 eV, respectively. The peak at 529.9 eV corresponds to the V-O bonds, the peaks at 533.5 and 532.0 eV are often caused by the adsorbed H₂O and CO₂ molecules in air, respectively.¹

Optical photos

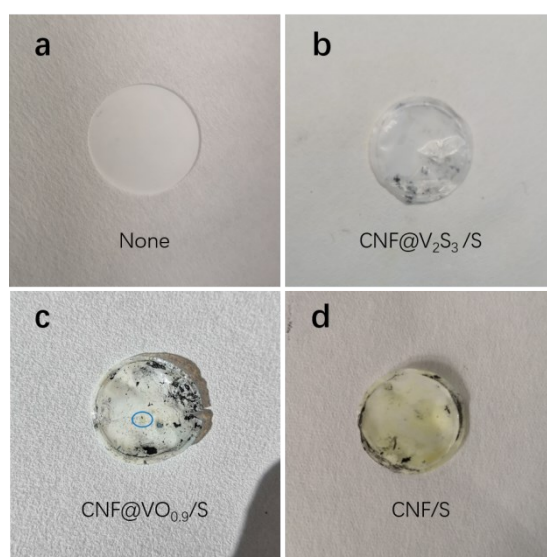


Figure S6 Photos of battery separator (a), the cycled battery separators of CNF@V₂S₃/S(b), CNF@VO_{0.9}/S(c) and CNF/S (d).

TGA data

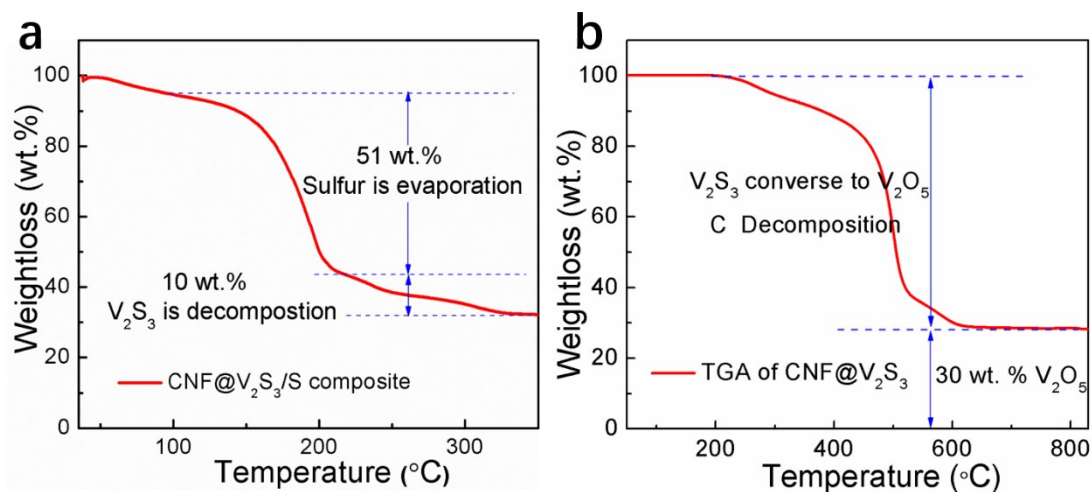


Figure S7 (a) The TGA curve of CNF@V₂S₃/S composite electrode. (b) The TGA curve of CNF@V₂S₃ electrode in air.

For the analysis of TGA curve of CNF@V₂S₃ electrode in air as shown in Figure S7b, the final phase after calcination is V₂O₅, therefore, the V₂O₅ content is about 30 wt.%. According to V element mass balance (Formula 1), the V₂S₃ content is 32.7 wt.%, and the content of the carbon is about 67.3 wt%.



Cross-section SEM image

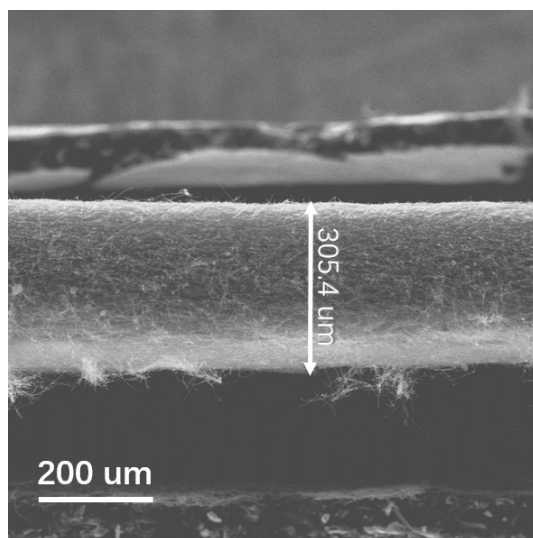


Figure S8 Thickness of CNF@V₂S₃ electrode

CV analysis

Table S1. Positions of reduction peaks in the CV curves. ²⁻⁶

	First reduction peak (V)	Second reduction peak (V)	D-value (mV)
CNF@V ₂ S ₃ /S	2.054	2.325	271
CNF@VO _{0.9} /S	2.029	2.307	278
CNF/S	2.026	2.308	282
Ref.2	2.03	2.31	280
Ref.3	2.04	2.31	270
Ref.4	2.01	2.33	320
Ref.5	2.01	2.29	280
Ref.6	1.95	2.25	300

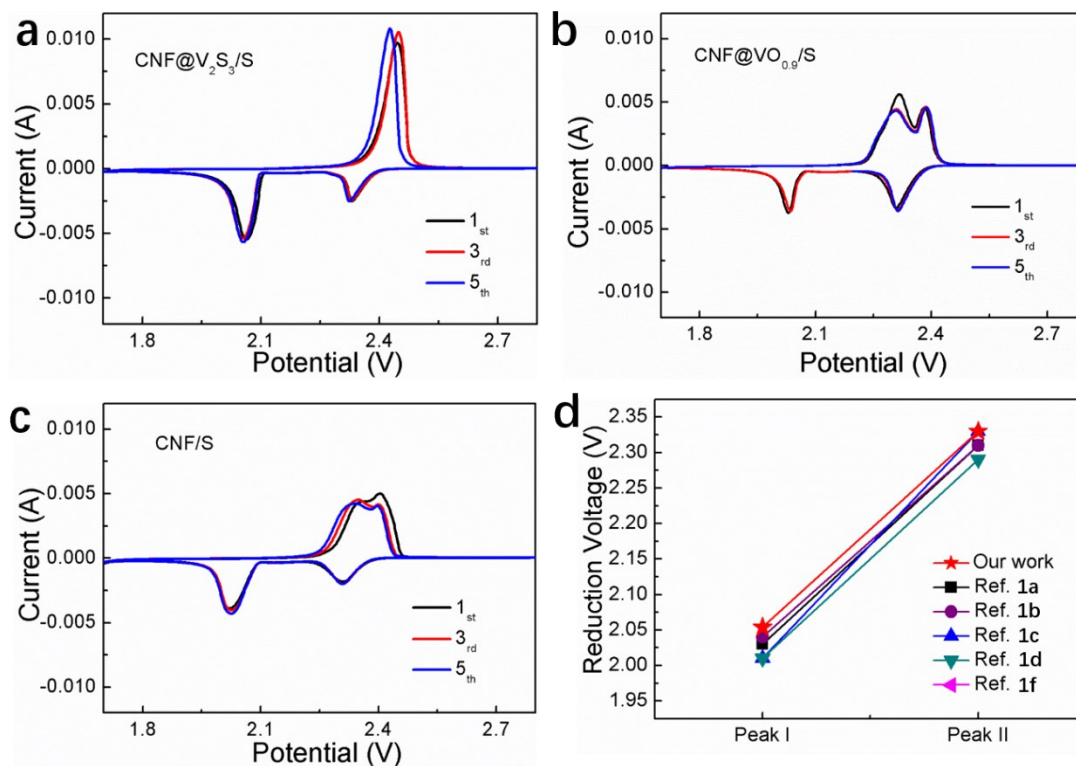


Figure S9 The initial five cycles of CV curves for (a) CNF@V₂S₃/S, (b) CNF@VO_{0.9}/S and (c) CNF/S. The Vanadium-based LSB comparison of the reduction peak position (d) in the recent literature ²⁻⁶

Electrochemical performance of host

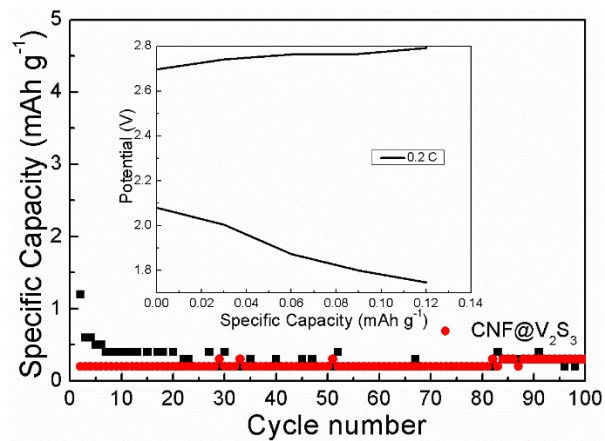


Figure S10 The electrochemical performance of CNF@V₂S₃ based compounds without sulfur.

Capacity contribution ratios

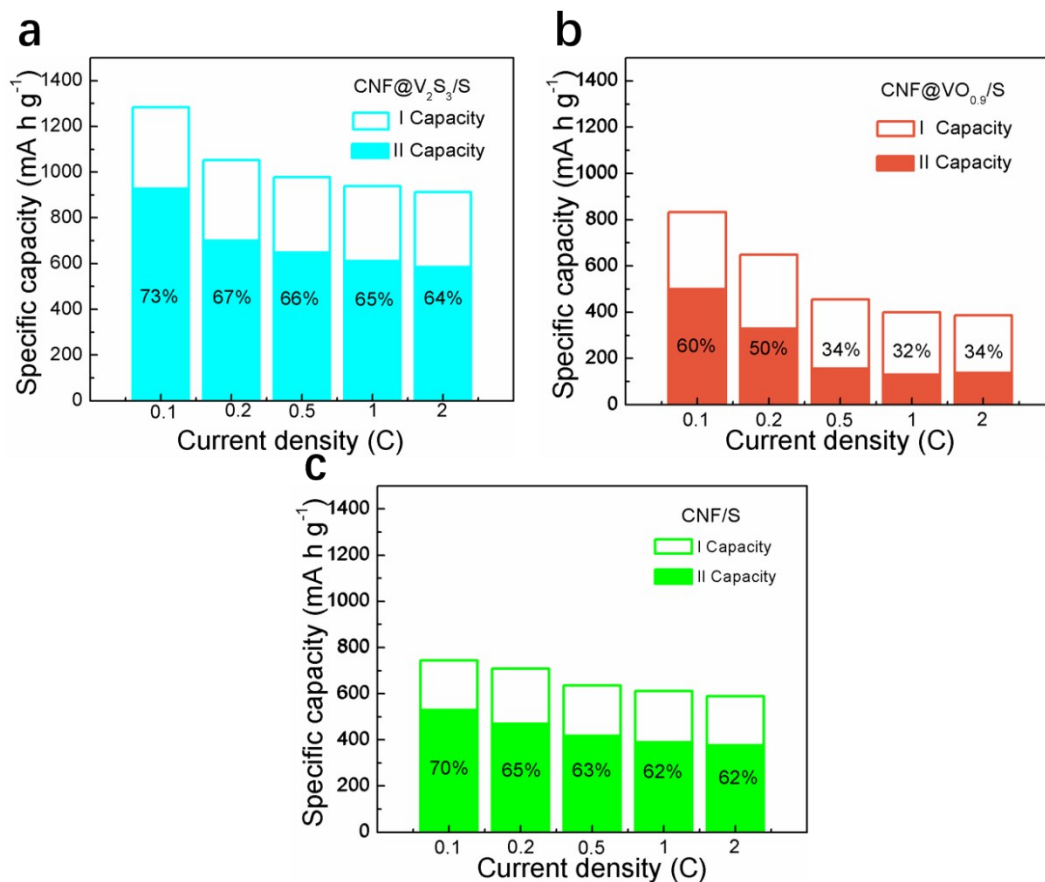


Figure S11 Capacity contributions of soluble Li_2S_4 , Li_2S_6 and Li_2S_8 conversion (I) and insoluble $\text{Li}_2\text{S}_2/\text{Li}_2\text{S}$ conversion (II) and the corresponding capacity ratios (II/I) at different rates for the (a) $\text{CNF@V}_2\text{S}_3$, (b) $\text{CNF@VO}_{0.9}/\text{S}$ and (c) CNF/S cathodes.

Due to the relatively slow reaction kinetics of intermediate LiPSs, it is difficult to convert them into the final reduction products $\text{Li}_2\text{S}/\text{Li}_2\text{S}_2$ completely during the reduction II, often resulting into an II/I ratio less than 3. Thus, the closer the ratio is to 3, the higher the reversible specific capacity is.

Calculation of reduction resistance

The calculation way of in situ reaction resistance: The difference between quasi-open-circuit potential and closed-circuit potential is used to calculate the over-potentials, and then the in-situ reaction resistance can be calculated as,⁷

$$R = \frac{\Delta U}{m \times J}, \quad [2]$$

where, ΔU is the over-potential, m is the mass loading of sulfur, and J is the charge/discharge current density.

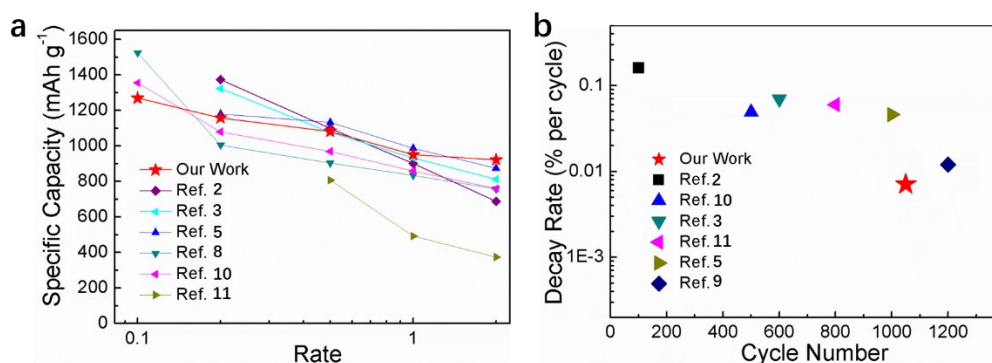


Figure S12 Long cycle comparison of the rate capabilities (a) and decay rate (b) of CNF@V₂S₃/S in the recent literature.^{2, 3, 5, 8-11}

Quasi in-situ SEM image

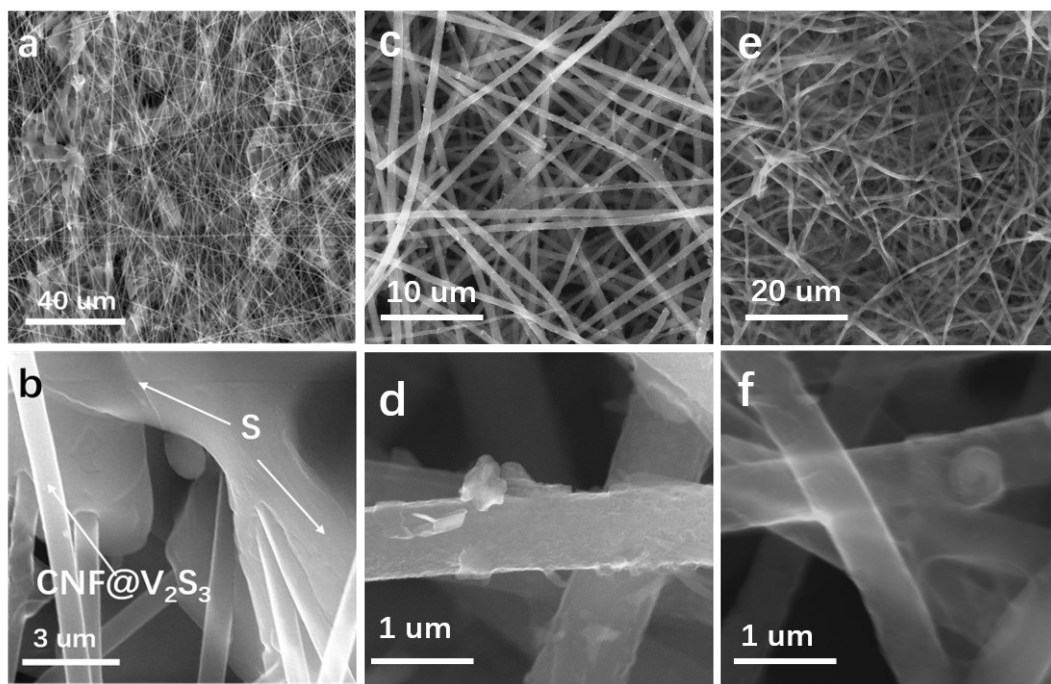


Figure S13 SEM images of the CNF@V₂S₃/S cathodes: (a-b) before cycling, (c-d) after 100 cycles at 2 C, and (e-f) after 1000 cycles at 2 C, respectively.

EIS spectra

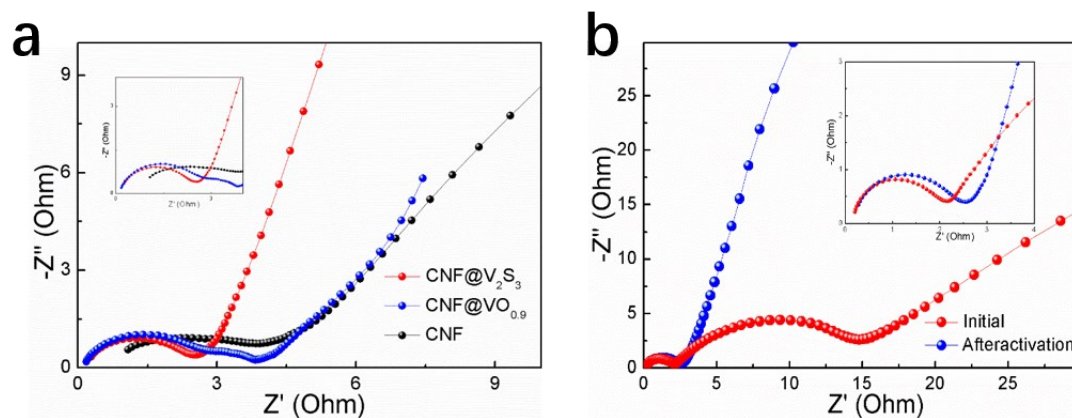


Figure S14 (a) The EIS curves of CNF@V₂S₃/S, CNF@VO_{0.9}/S and CNF/S. Inset: the magnified high frequency region. (b) Nyquist plots of the CNF@V₂S₃/S composites electrode before activation and after activation. Inset: the magnified high frequency region.

Table S2. Corresponding fitting result of R_s and R_{ct} values for the Nyquist plots.

	R_s	R_{ct}	Slope of low frequency region
CNF@V ₂ S ₃ /S	0.27	2.63	1.46
CNF@VO _{0.9} /S	0.28	3.82	0.97
CNF/S	1.10	3.97	0.62

Table S3. Corresponding CNF@V₂S₃/S result of R_s and R_{ct} values for the Nyquist plots.

	R_s	R_{ct}	Slope of low frequency region
OCV	0.28	15.2	0.83
After activation	0.27	2.63	1.46

The obtained Nyquist plots were semicircular at the high frequency section and increases sharply at the low frequency section. In the high band, the real left intersection ($-Z''$) is related to the series resistance (R_s).¹² The semi-circular region in the plots corresponds to the charge transfer resistance (R_{ct}), indicating the interface resistance of the electrolyte and the electrode surface. In the low frequency region, the greater the slope of the curve, the greater the diffusion rate of the Li ions in the electrolyte, even the proximity of 90° may ignore the diffusion resistance.¹²

EIS spectra shown in Fig. S12 indicates that compared to CNF@VO_{0.9}/S and CNF/S cathodes, the CNF@V₂S₃/S ones show a smaller series resistance (R_s) and a charge transfer resistance (R_{ct}),¹² again indicating an enhanced electron/ion kinetics.¹³ And, the curve slope in the low-frequency region of CNF@V₂S₃/S cathodes is much larger than those of CNF@VO_{0.9}/S and CNF/S ones, further implying a higher diffusivity of electrolyte ions in CNF@V₂S₃/S cathodes.

GITT analysis

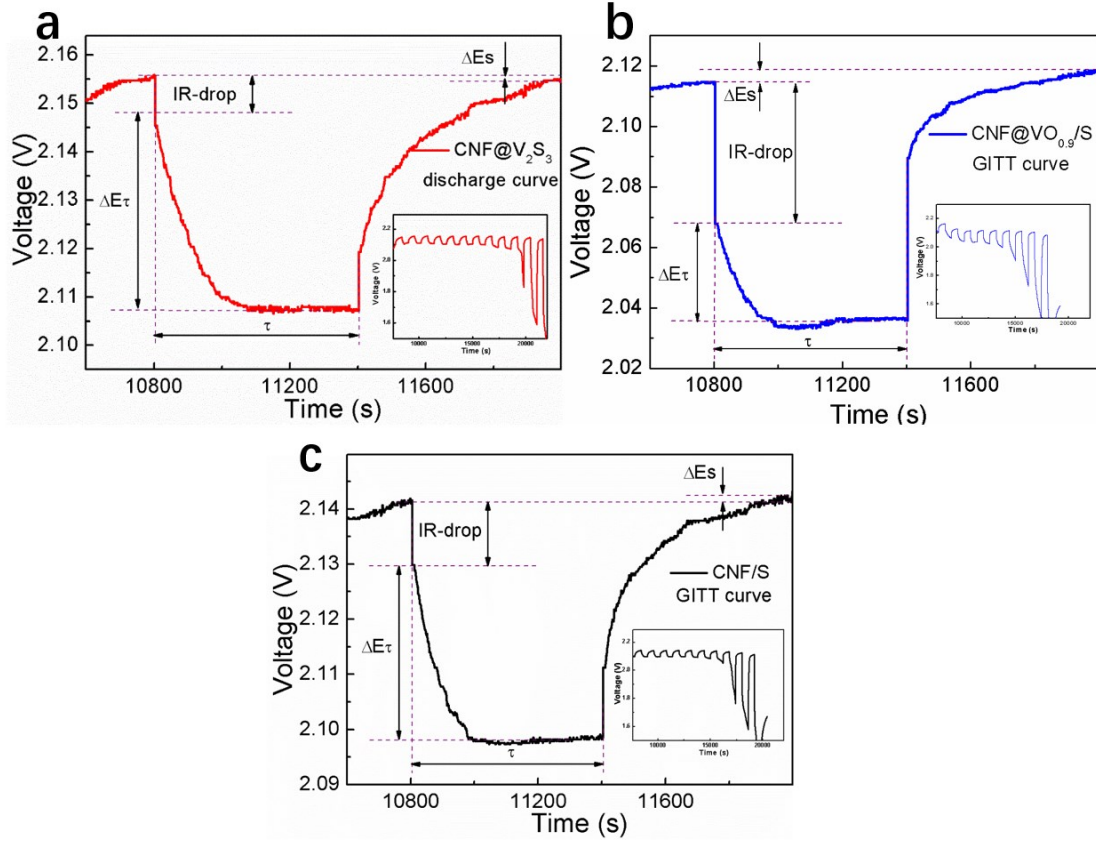


Figure S15 (a-c) GITT potential response curve with time for one typical discharge step of CNF@V₂S₃/S, CNF@VO_{0.9}/S and CNF/S. Inset: GITT profiles of the discharging process selected rest discharge-rest period.

GITT measurement was performed by a current density at 0.2 C for 10 min and rest intervals for 10 min. The Fick's second law using the equation estimated the Li⁺ diffusion coefficients in CNF@V₂S₃/S electrode:^{14, 15}

$$D_{Li^+} = \frac{4}{\pi\tau} \left(\frac{m_B V_m}{M_B S} \right)^2 \left(\frac{\Delta E_s}{\tau (dE_\tau / d\sqrt{\tau})} \right)^2, \quad [3]$$

If the potential vs. $\tau_{1/2}$ (τ is pulse current duration time) displays a linear behavior, equation diffusion can be further simplified as below:

$$D_{Li^+} = \frac{4}{\pi\tau} \left(\frac{m_B V_m}{M_B S} \right)^2 \left(\frac{\Delta E_s}{\Delta E_\tau} \right)^2, \quad [4]$$

Table S4 Li-ion diffusion rates between cathode and electrolyte

Samples	Second reduction peak ($\text{cm}^2 \text{s}^{-1}$)
CNF@V ₂ S ₃ /S	2.857×10^{-6}
CNF@VO _{0.9} /S	1.554×10^{-7}
CNF/S	1.645×10^{-7}

Optical photo



Figure S16 the Optical photo of CNF@V₂S₃/S cathode after 500 cycles.

Catalytic performance

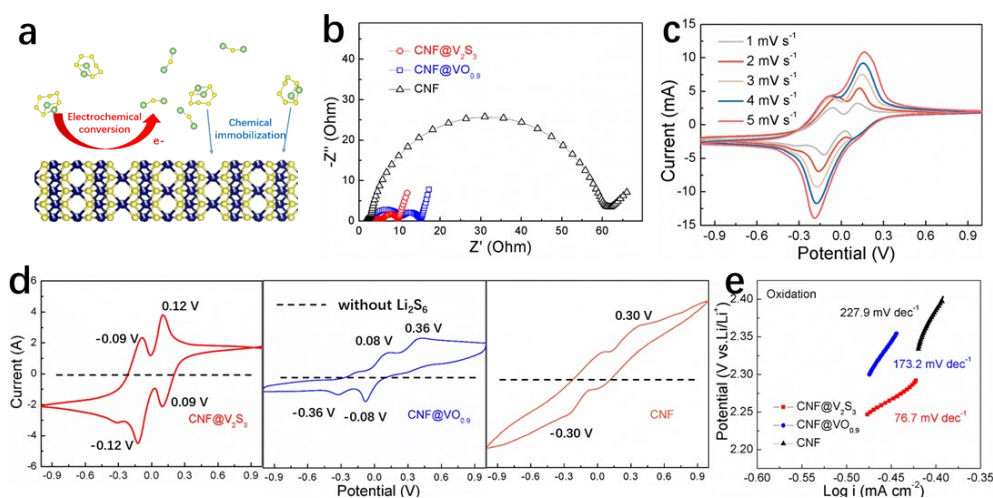


Figure S17. a) Schematic illustration of the LiPSs' conversion on the surface of V₂S₃; b) EIS spectra of CNF@V₂S₃, CNF@VO_{0.9} and CNF cells without sulfur addition. c) CV profiles of CNF@V₂S₃ cells under different scan rates. d) CV curves of CNF@V₂S₃, CNF@VO_{0.9} and CNF cells at 1 mV s⁻¹; e) Tafel plots of Li₂S oxidation on CNF@V₂S₃, CNF@VO_{0.9} and CNF electrodes.

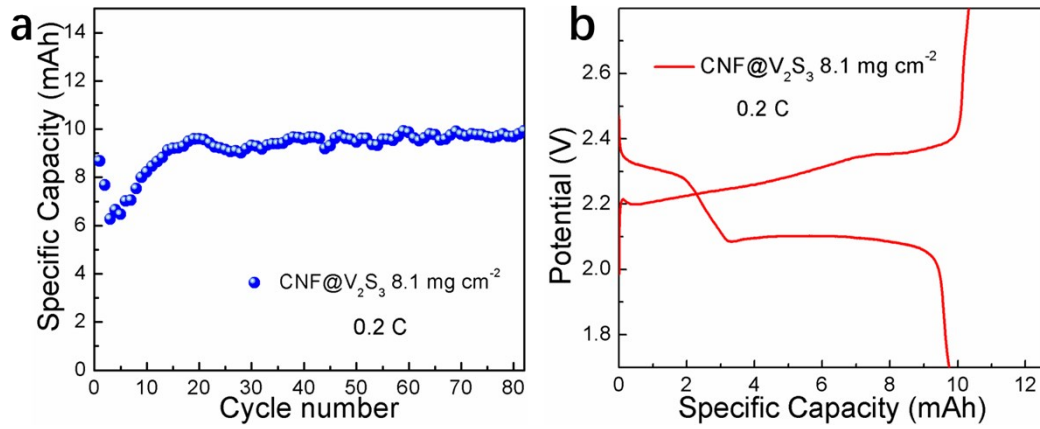


Figure S18. The sulfur high loading of CNF@V₂S₃/S electrode up to 8.1 mg cm⁻²: (a) cycle performance; (b) charge and discharge profiles.

Self-discharge formula

We define the self-discharge rate as the ratio of lost discharge capacity to initial capacity:

$$\text{Self - discharge rate} = \frac{C_{ini} - C_{fin}}{C_{ini}} \times 100\% \quad [5]$$

where C_{ini} is the initial discharge capacity and C_{fin} is the retention capacity.¹⁶

1. J. M. S. C. MORANT, L. GALAN, L. SORIANO, F. RUEDA, *Sur.Sci.*, 1989, **218**, 311-345.
2. X. Zhu, W. Zhao, Y. Song, Q. Li, F. Ding, J. Sun, L. Zhang and Z. Liu, *Adv. Energy Mater.*, 2018, **8**, 1800201.
3. Z. X. Cheng, Zhubing Pan, Hui Wang, Shiqing Wang, Ruihu, *Adv. Energy Mater.*, 2017, **8**, 1702337.
4. T. Guo, Y. Song, Z. Sun, Y. Wu, Y. Xia, Y. Li, J. Sun, K. Jiang, S. Dou and J. Sun, *J. Energy Chem.*, 2020, **42**, 34-42.
5. H. Wu, Y. Huan, D. Wang, M. Li, X. Cheng, Z. Bai, P. Wu, W. Peng, R. Zhang, Z. Ji, M. Zou and X. Yan, *J. Electrochem. Soc.*, 2019, **166**, A188-A194.
6. X. Chen, G. Du, M. Zhang, A. Kalam, S. Ding, Q. Su, B. Xu and A. G. Al-Sehemi, *Energy*

- Technol.*, 2019, **8**.
7. T. Liu, S. Sun, W. Song, X. Sun, Q. Niu, H. Liu, T. Ohsaka and J. Wu, *J. Mater. Chem. A*, 2018, **6**, 23486-23494.
 8. Y. Song, W. Zhao, N. Wei, L. Zhang, F. Ding, Z. Liu and J. Sun, *Nano Energy*, 2018, **53**, 432-439.
 9. Y. Zhang, G. Xu, Q. Kang, L. Zhan, W. Tang, Y. Yu, K. Shen, H. Wang, X. Chu, J. Wang, S. Zhao, Y. Wang, L. Ling and S. Yang, *J. Mater. Chem. A*, 2019, **7**, 16812-16820.
 10. R. Carter, L. Oakes, N. Muralidharan, A. P. Cohn, A. Douglas and C. L. Pint, *ACS Appl. Mater. Interfaces*, 2017, **9**, 7185-7192.
 11. J. Wei, H. Su, C. Qin, B. Chen, H. Zhang and J. Wang, *J. Electroanal. Chem.*, 2019, **837**, 184-190.
 12. C. Yang, X. Ou, X. Xiong, F. Zheng, R. Hu, Y. Chen, M. Liu and K. Huang, *Energy Environ. Sci.*, 2017, **10**, 107-113.
 13. X. Yang and A. L. Rogach, *Adv. Energy Mater.*, 2019, **9**, 1900747.
 14. C. Bommier, T. W. Surta, M. Dolgos and X. Ji, *Nano Lett.*, 2015, **15**, 5888-5892.
 15. Z. Jian, Z. Xing, C. Bommier, Z. Li and X. Ji, *Adv. Energy Mater.*, 2016, **6**.
 16. H. S. Ryu, H. J. Ahn, K. W. Kim, J. H. Ahn, K. K. Cho and T. H. Nam, *Electrochim. Acta*, 2006, **52**, 1563-1566.