

Supplementary Information for

Unravelling high volumetric capacity of Co₃O₄ nanograins-interconnected secondary particles for lithium-ion battery anode

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Bulk, surface and Li adsorbed models calculated by density functional theory (DFT)

1. Bulk models

To further discover why reaction pathway changes under different particle size, density-functional theory (DFT) calculations are performed. Because reaction pathways of both Co_3O_4 MPs and Co_3O_4 NPs observed by *in situ* XRD analyses on Figure 1 contain LiCo_3O_4 phase (intercalation phase), we study the conversion reaction after LiCo_3O_4 phase depending on different particle size and employed the atomic structure of LiCo_3O_4 from the Materials Project, not Co_3O_4 spinel structure. According to a previous study [36], octahedral interstitial sites of Co_3O_4 spinel should be occupied by Co ions when the Li ions inserted and the Li intercalated Co_3O_4 structure be tetragonal rocksalt (Space group: *Cmmm*) that has lattice constant, *a* and *b* are 8.362 Å and *c* is 8.746 Å.

2. Surface models

Because this structure is composed of Li^+ and Co^{3+} layer and Co^{2+} layer which is anisotropic along c-axis, the cleaved surface plane can be 5 types of planes: (001), (100), (110), (101) and (111). We determined the representative surface model as the most stable (001) plane, Li-O-Co termination and thickness of nine layers converged by surface energy as function of the number of layers (Figure S15). The surface energy, γ was calculated by following equation,

$$\gamma = \frac{1}{2A} \left[E_{\text{Li}_x\text{Co}_y\text{O}_z}^{\text{slab}} - \frac{N_{\text{Co}}}{3} E_{\text{LiCo}_3\text{O}_4}^{\text{bulk}} - \left(N_{\text{O}} - \frac{4}{3} N_{\text{Co}} \right) \mu_{\text{O}} - \left(N_{\text{Li}} - \frac{1}{3} N_{\text{Co}} \right) \mu_{\text{Li}} \right]$$

where $E_{\text{Li}_x\text{Co}_y\text{O}_z}^{\text{slab}}$ and $E_{\text{LiCo}_3\text{O}_4}^{\text{bulk}}$ are the total energy of slab and bulk; *A* is the surface area; N_i and μ_i are the number of atoms and the chemical potential of *i* species respectively. The surface model consists of 36 atoms and 16.895 Å and vacuum was inserted more than 26 Å to avoid

overlap between periodic images. The structures was relaxed by fixing center layer to consider bulk region. We then studied the structural and electrochemical properties of LiCo_3O_4 upon the lithiation process after intercalation step, starting from one Li atom adsorped on LiCo_3O_4 (001) surface. We empolyed the surface model including 80 atoms which was enlarged from 17.483 \AA^2 to 69.930 \AA^2 surface area to avoid too high concentration of adsorbed ion density and removed four layers in bottom to reduce the computational cost.

3. Li-adsorbed models

Then we found 8 kinds of Li adsorption sites in the supercell that can be classified by top (T1, T2, T3), bridge (B1, B2) and hollow (H1, H2, H3) sites (Figure S16). The adsorption structures was relaxed by using DFT energy and the adsorption energy was calculated by following equation (Table S2).As a result, Li ion located on bridge sites (B1 and B2) moves to hollow site (H2 and H1) respectively resulting that the bridge site is unstable and Li ion can not stay in there. On the other hands, the adsorption of Li ion on LiCo_3O_4 surface is favorable because adsorption energy of top and hollow sites are negative. Especially, the hollow sites are most stable and should be occupied first when Li ion adsorbed.

To describe high concentration of Li ion adsorption, we reduced cell size to 36 atoms allocated in 17.48 \AA^2 surface area and the adsorbed structure was relaxed by fixing bottom layer to be considered as bulk resulting in lithiated Co_3O_4 (001) structure (Fig. 3d)

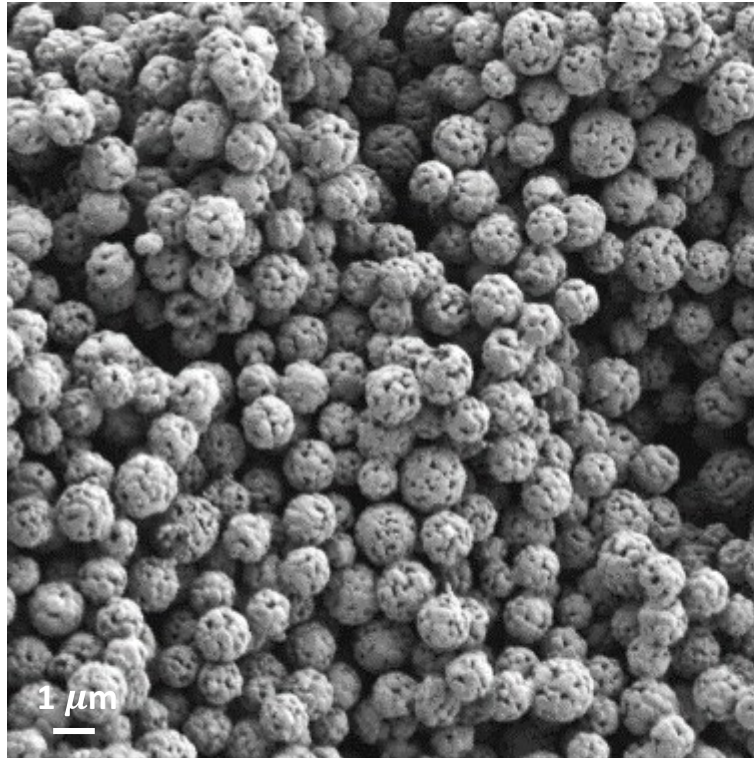


Fig.S1 Low magnification SEM image of Co_3O_4 NISP.

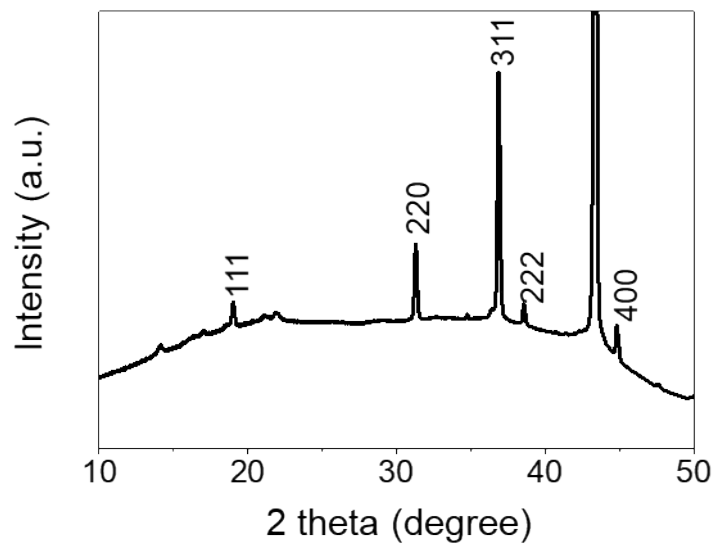


Fig. S2. XRD pattern of Co_3O_4 NISP, acquired from first pattern of *in situ* XRD experiment.

Note that strong peak in 43.3 degree is from Cu foil.

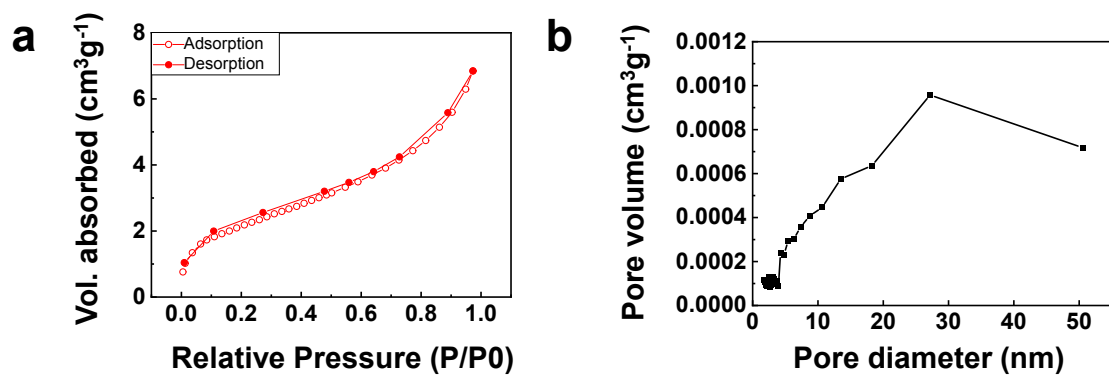


Fig. S3 (a) Nitrogen adsorption-desorption isotherms and (b) pore size distribution graph of Co₃O₄ NISP.

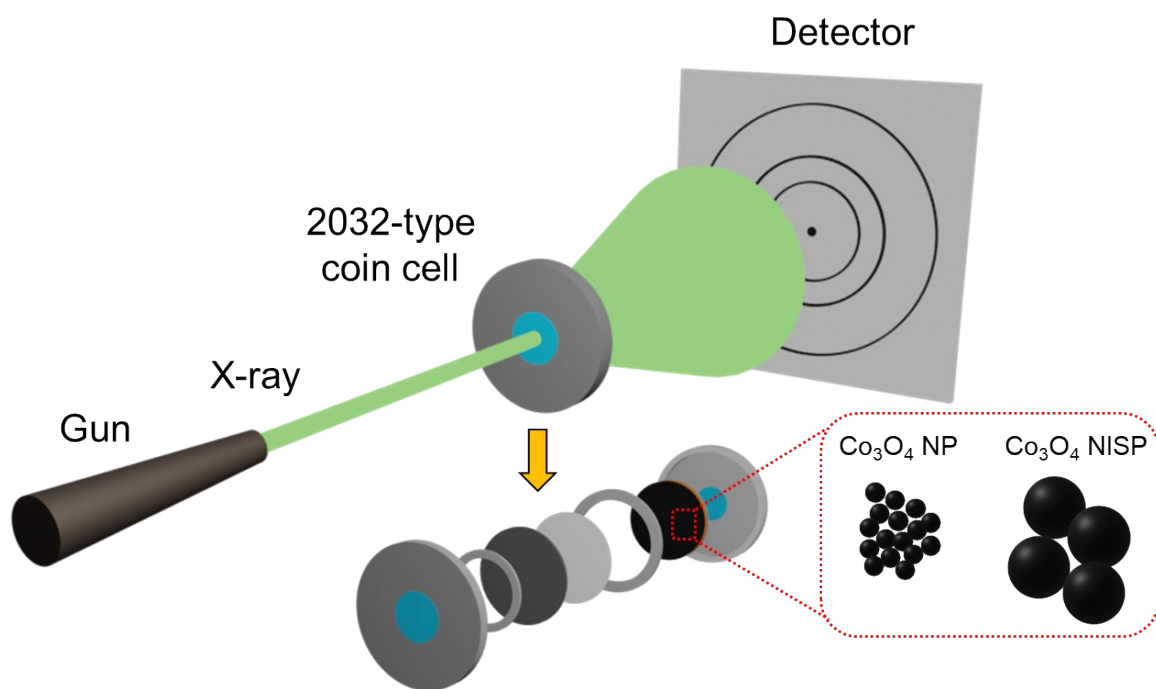


Fig. S4. Schematic illustration of *in situ* XRD experiment.

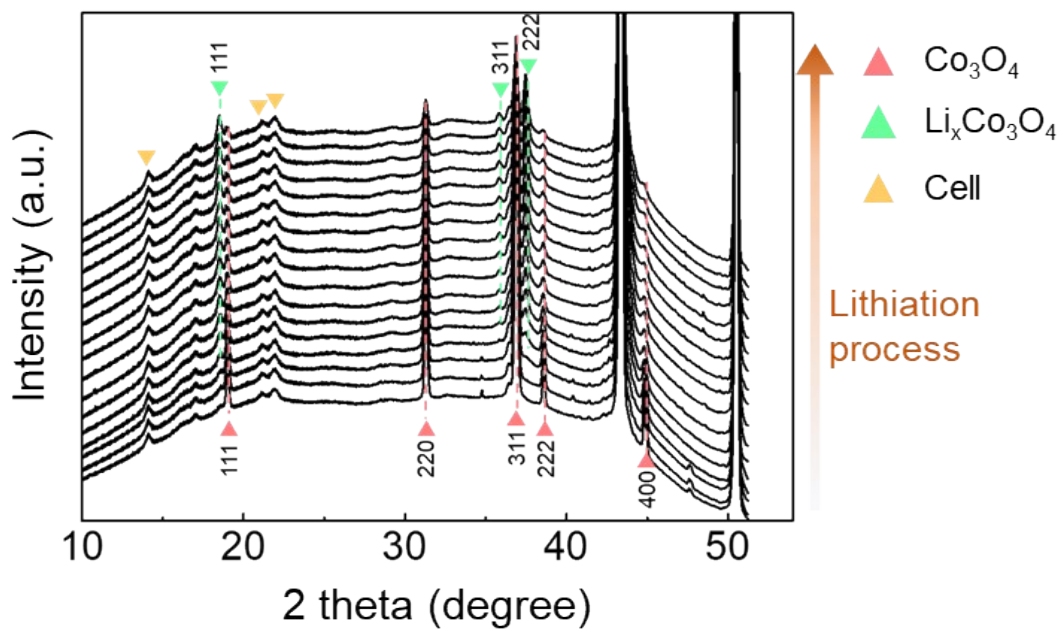


Fig. S5. *In situ* XRD patterns Co_3O_4 NISP in range of 10 to 55 degree.

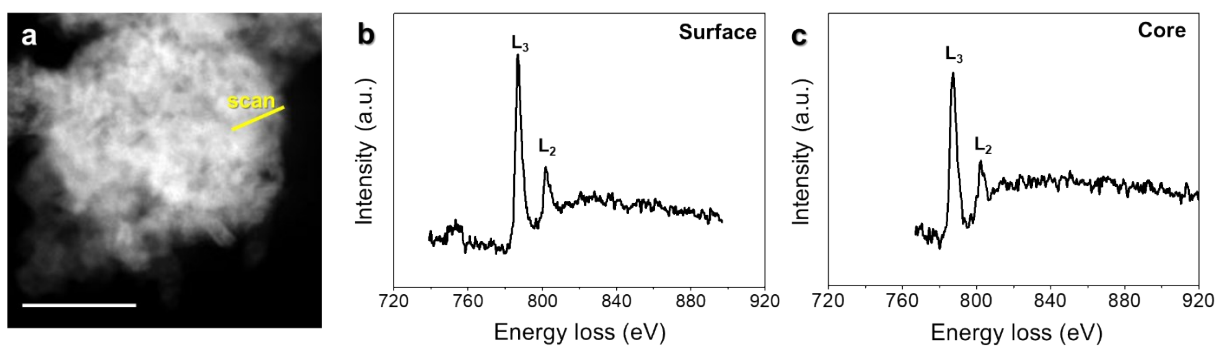


Fig. S6. (a) HAADF-STEM image of Co_3O_4 NISP at 100% of SOC. (b) EELS spectrum from the surface of Co_3O_4 NISP. The ratio between L_3 and L_2 was 3.36, indicating Co^{0+} . (c) EELS spectrum from inner part of Co_3O_4 NISP. The ratio was 2.54, indicating $1/3 \text{Co}^{2+}$ and $2/3 \text{Co}^{3+}$. The scale bar in (a) is 500 nm.

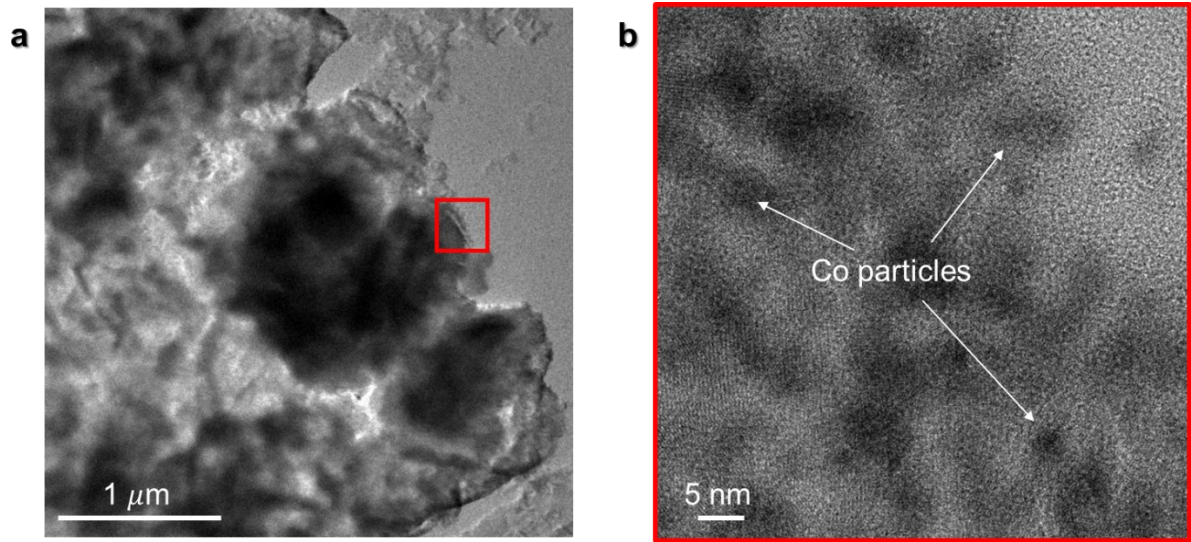


Fig. S7. (a) TEM image of Co_3O_4 NISP at 100% of SOC. (b) Magnified TEM image of (a), showing Co nanoparticles are present in nanograins.

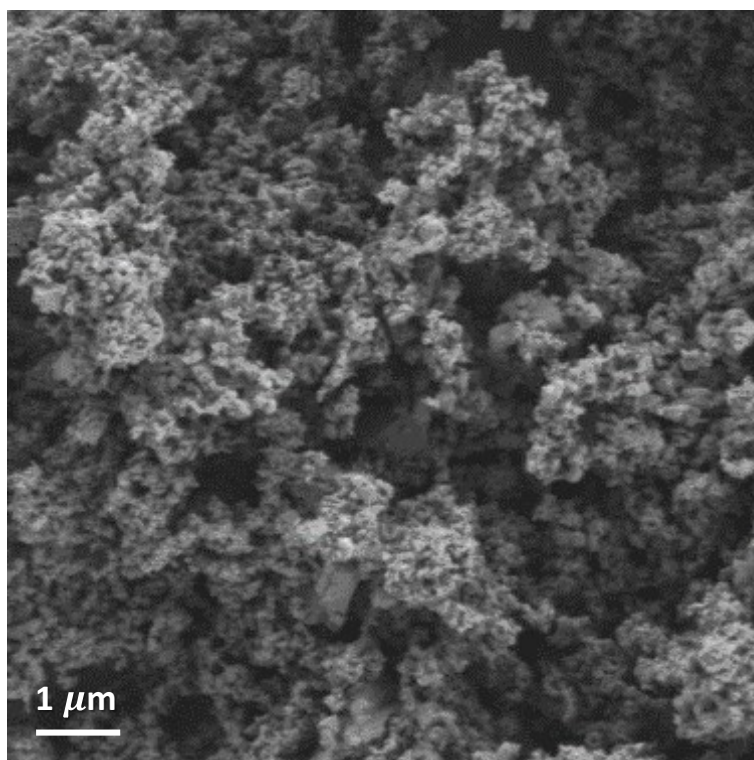


Fig. S8. Low magnification SEM image of Co₃O₄ NP.

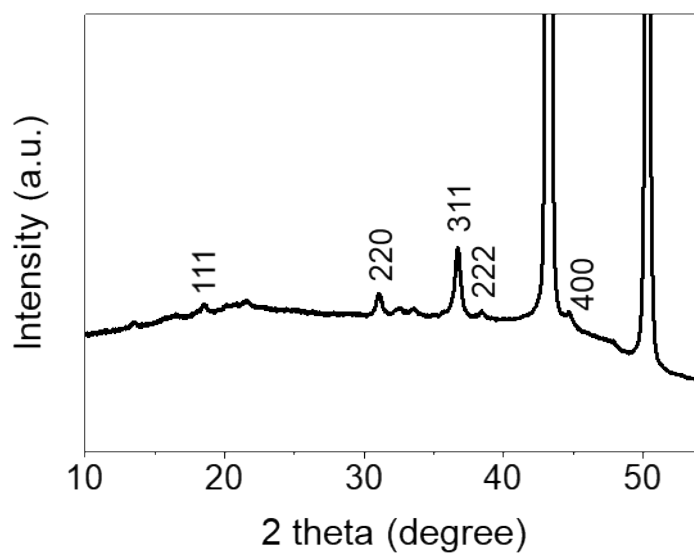


Fig. S9. XRD pattern of Co_3O_4 NP acquire from first pattern of *in situ* XRD experiment. Note that strong peak in 43.3 and 50.3 degree are from Cu foil.

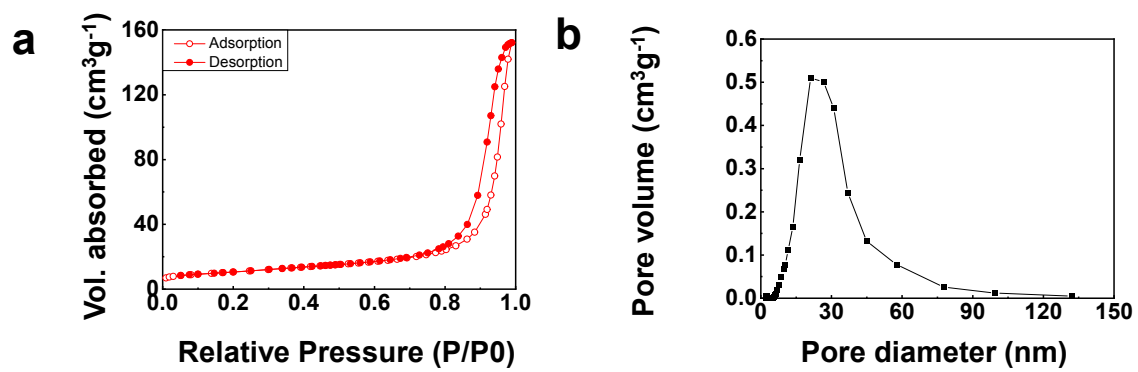


Fig. S10. (a) Nitrogen adsorption-desorption isotherms and (b) pore size distribution graph of Co_3O_4 NP.

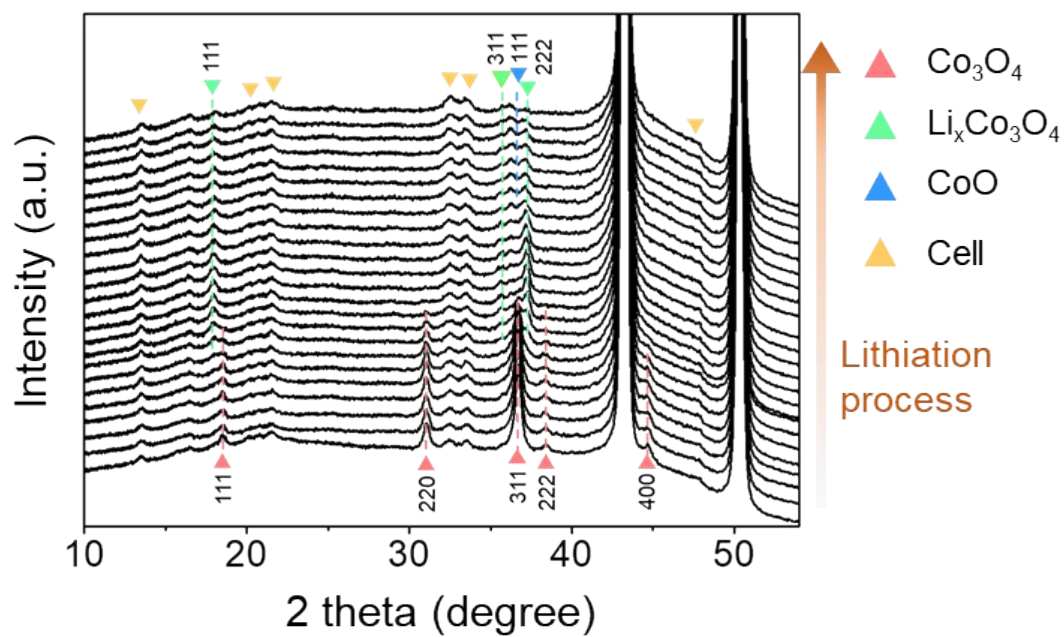


Fig. S11. *In situ* XRD patterns Co_3O_4 NP in range of 10 to 55 degree.

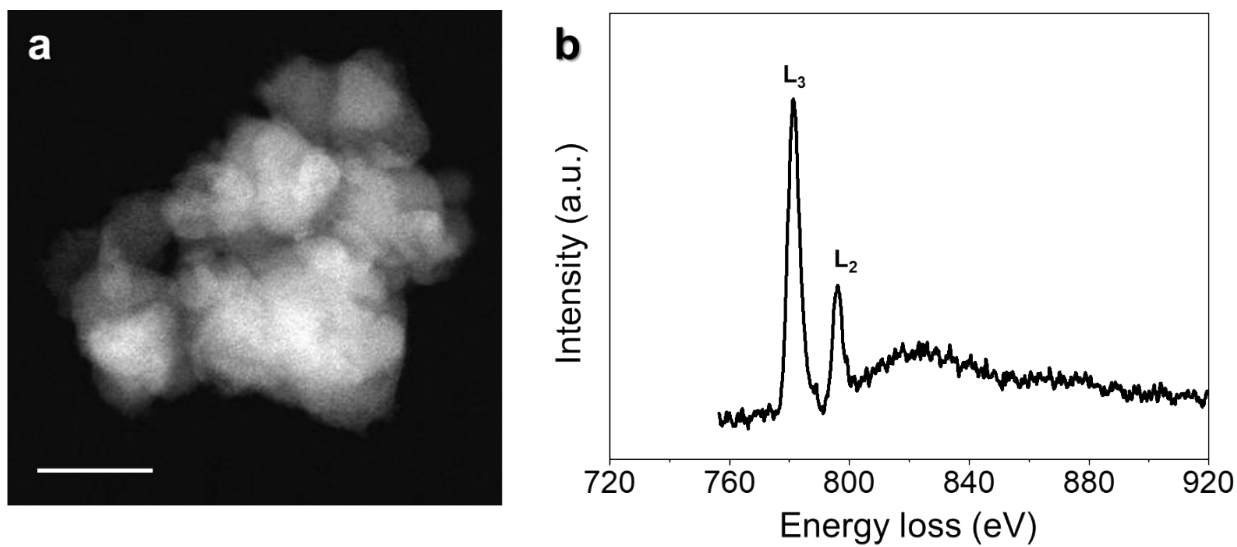


Fig. S12. (a) HAADF-STEM image and (b) EELS spectrum of Co_3O_4 NP after 1st cycle of lithiation. The ratio between L_3 and L_2 peak is 3.21, indicating electronic state of Co^{0+} . The scale bar in (a) is 100 nm.

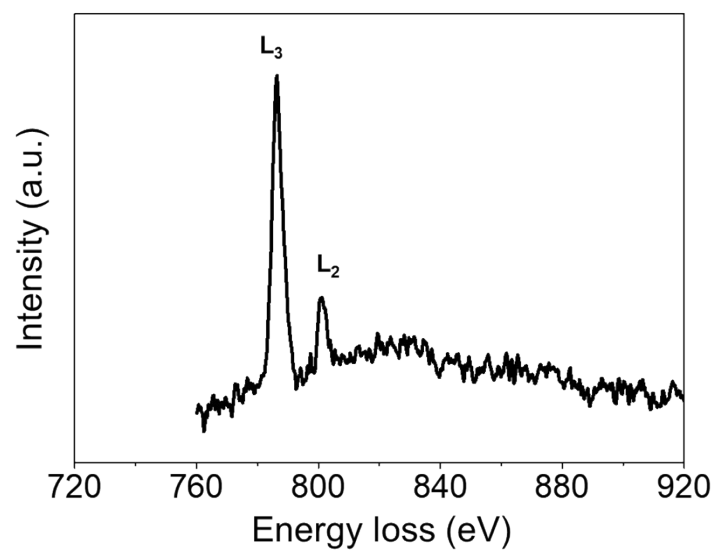


Fig. S13. EELS spectrum confirming the presence of CoO on the surface of Co₃O₄ NISP, having the peak ratio of 4.55.

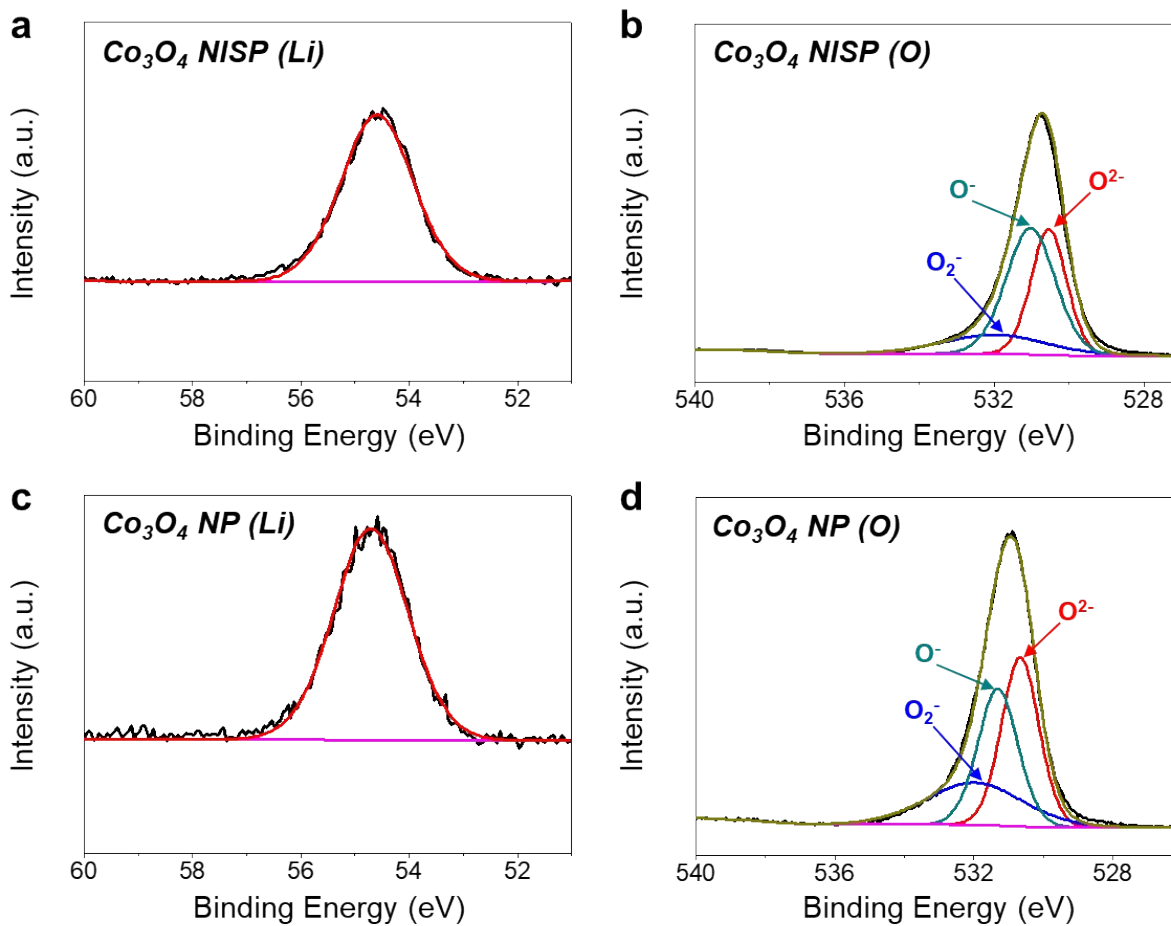


Fig. S14. *Ex situ* XPS analysis of (a) Li and (b) O for Co_3O_4 NISP after 1st cycle of lithiation.

Ex situ XPS analysis of (c) Li and (d) O for Co_3O_4 NP after 1st cycle of lithiation.

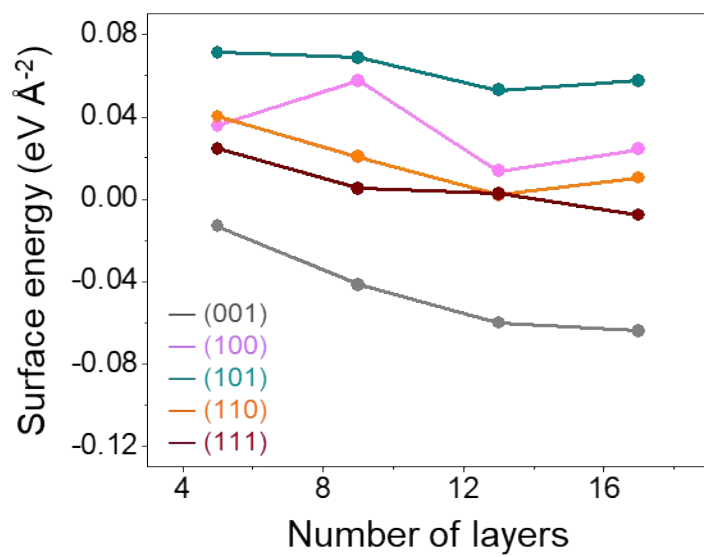


Fig. S15. Surface energies of the LiCo₃O₄ (001), (100), (101), (110) and (111) surfaces as a function of the number of layers.

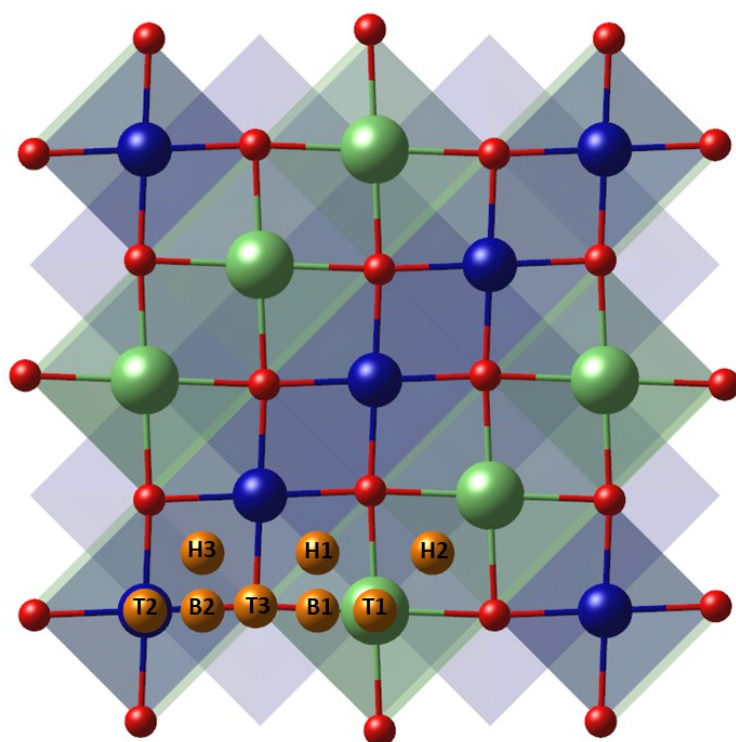


Fig. S16. Top view of eight possible Li adsorption sites on the (001) surface: Top sites (T1, T2, T3), bridge sites (B1, B2), and hollow sites (H1, H2, H3).

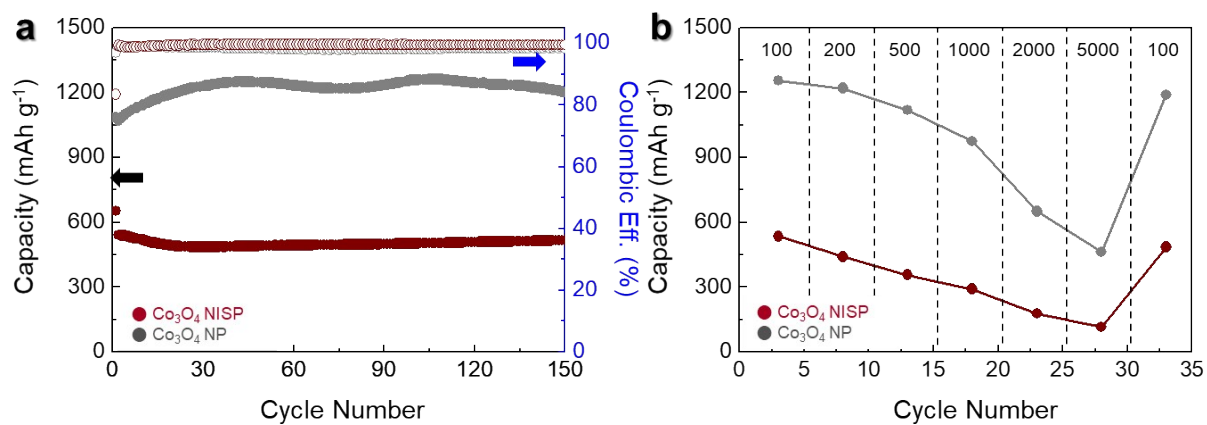


Fig. S17. (a) Cycle retention characteristics (at a current density of 500 mA g^{-1}) and (b) rate capabilities (at current densities expressed in mA g^{-1}) of Co_3O_4 NISP and Co_3O_4 NP. All units are expressed in terms of gravimetric capacity.

Table S1. L_3/L_2 ratio of metallic Co and Co oxide [1].

	L_3/L_2 ratio
Co_3O_4	2.42
CoO	4.51
Metallic Co	3.15

Table S2. Adsorption energies with respect to adsorption sites on (001) surface and each site before and after relaxation

Index	Site (before relaxation)	Site (after relaxation)	E_{ads} (eV)
T1	Top (Li)	Top (Li)	-0.36
T2	Top (Co)	Top (Co)	-1.08
T3	Top (O)	Hollow (Li-O-Li-O)	-1.95
B1	Bridge (Li-O)	Hollow (Li-O-Li-O)	-2.43
B2	Bridge (Co-O)	Hollow (Li-O-Co-O)	-1.02
H1	Hollow (Li-O-Co-O)	Hollow (Li-O-Co-O)	-2.38
H2	Hollow (Li-O-Li-O)	Hollow (Li-O-Li-O)	-0.29
H3	Hollow (Co-O-Co-O)	Hollow (Co-O-Co-O)	-1.92

Table S3. Tap density calculation of Co₃O₄ NP and NISP.

Particle size	Tap density (g/cm ³)
Co ₃ O ₄ NP	0.5
Co ₃ O ₄ NISP	4.195

Table S4. Comparison of electrochemical performance and tap density with other previous literatures.

Spinel type oxides	Tap density (g cm ⁻³)	Cycle retention	Rate capabilities	Ref
Li ₄ Ti ₅ O ₁₂ -C	1.1	~96 mAh cm ⁻³ at 7.0 A g ⁻¹ for 1000 cycles	103 mAh cm ⁻³ at 14.0 A g ⁻¹	[2]
Li ₄ Ti ₅ O ₁₂	1.4	235.2 mAh cm ⁻³ at 1.0 C For 400 cycles*	169.4 mAh cm ⁻³ at 5.0 C*	[3]
ZnCo ₂ O ₄	1.48	1480 mAh cm ⁻³ at 0.45 A g ⁻¹ for 70 cycles*	1406 mAh cm ⁻³ at 0.9 A g ⁻¹ *	[4]
Fe ₃ O ₄ @CNT	1.22	1540.0 mAh cm ⁻³ at 0.1 A g ⁻¹ for 125 cycles*	54.9 mAh cm ⁻³ at 3.2 A g ⁻¹ *	[5]
CoMn ₂ O ₄ -S	1.73	1366.7 mAh cm ⁻³ at 0.4 A g ⁻¹ for 100 cycles*	907.1 mAh cm ⁻³ at 1.6 A g ⁻¹	[6]
NiFe ₂ O ₄ -S	1.33	~900 mAh cm ⁻³ at 0.1 C for 200 cycles	674.6 mAh cm ⁻³ at 2.0 C*	[7]
ZnFe ₂ O ₄	0.85	460.7 mAh cm ⁻³ at 1.0 A g ⁻¹ for 488 cycles*	277.1 mAh cm ⁻³ at 1.5 A g ⁻¹ *	[8]
NiCo ₂ O ₄ -S	1.66	808.4 mAh cm ⁻³ at 1.0 C for 1500 cycles	664 mAh cm ⁻³ at 5.0 C	[9]
Co ₃ O ₄ NISP	4.195	2167.3 mAh cm ⁻³ at 0.5 A g ⁻¹ for 150 cycles	611.7 mAh cm ⁻³ at 5.0 A g ⁻¹	This Work

References

- [1] Y. Zhao, T.E. Feltes, J.R. Regalbuto, R.J. Meyer, R.F. Klie, *In situ* electron energy loss spectroscopy study of metallic Co and Co oxides, *J. Appl. Phys.* 108 (2010) 0-7.
- [2] Y. Xiang, W. Zhang, B. Chen, Z. Jin, H. Zhang, P. Zhao, G. Cao, Q. Meng, Nano- $\text{Li}_4\text{Ti}_5\text{O}_{12}$ particles in-situ deposited on compact holey-graphene framework for high volumetric power capability of lithium ion battery anode, *J. Power Sources* 447 (2020) 227372.
- [3] J. Ren, H. Ming, Z. Jia, Y. Zhang, J. Ming, Q. Zhou, J. Zheng, High Tap Density $\text{Li}_4\text{Ti}_5\text{O}_{12}$ Microspheres: Synthetic Conditions and Advanced Electrochemical Performance, *Energy Technol.* 5 (2017) 1680-1686.
- [4] R.A. Adams, V.G. Pol, A. Varma, Tailored Solution Combustion Synthesis of High Performance ZnCo_2O_4 Anode Materials for Lithium-Ion Batteries, *Ind. Eng. Chem. Res.* 56 (2017) 7173-7183.
- [5] H. Ming, H. Zhou, X. Zhu, S. Zhang, P. Zhao, M. Li, L. Wang, J. Ming, Advanced Metal Oxide@Carbon Nanotube for High-Energy Lithium-Ion Full Batteries, *Energy Technol.* 6 (2018) 766-772.
- [6] S. Chen, Y. Ming, B. Tan, S. Chen, Carbon-free sulfur-based composite cathode for advanced Lithium-Sulfur batteries: A case study of hierarchical structured CoMn_2O_4 hollow microspheres as sulfur immobilizer, *Electrochim. Acta* 329 (2020) 135128.
- [7] Z. Zhang, D.-H. Wu, Z. Zhou, G.-R. Li, S. Liu, X.-P. Gao, Sulfur/nickel ferrite composite as cathode with high-volumetric-capacity for lithium-sulfur battery, *Sci. China Mater.* 62 (2019) 74-86.

- [8] L. Hou, H. Hua, L. Lian, H. Cao, S. Zhu, C. Yuan, Green Template-Free Synthesis of Hierarchical Shuttle-Shaped Mesoporous ZnFe_2O_4 Microrods with Enhanced Lithium Storage for Advanced Li-Ion Batteries, *Chem. Eur. J.* 21 (2015) 13012-13019.
- [9] Y.-T. Liu, D.-D. Han, L. Wang, G.-R. Li, S. Liu, X.-P. Gao, NiCo_2O_4 Nanofibers as Carbon-Free Sulfur Immobilizer to Fabricate Sulfur-Based Composite with High Volumetric Capacity for Lithium-Sulfur Battery, *Adv. Energy Mater.* 9 (2019) 1803477.