## **Supporting Information**

## Unveiling Interface Interaction Assisted Broadband Photoresponse of Epitaxial 2D Bi<sub>2</sub>O<sub>2</sub>Se on Perovskite Oxides

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## 1. Dimension of the Bi<sub>2</sub>O<sub>2</sub>Se layer and laser powers



**Fig. S1.** (a) A photo of an as-grown  $Bi_2O_2Se$  layer on the SrTiO<sub>3</sub> (100) substrate, whose dimension is 0.5mm×3mm;1 (b) laser power intensity radiated on the devices PD.I-III (P) as a function of laser wavelength ( $\lambda$ ). It is worth mentioning that the comparison of photoresponse properties among PD.I-III can only be done under the radiation of the same  $\lambda$ , as the laser power is  $\lambda$  dependent.

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# 2. Figures of merit as a function of $\boldsymbol{\lambda}$



**Fig. S2.** (a) External quantum efficiency (EQE) and (b) detectivity ( $D^*$ ) of PD.I-PD.III as a function of  $\lambda$  under 0.1 V external bias.

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### 3. I<sub>ph</sub> as a function of power intensity (P)

P of the pulsed lasers (wavelength range of 1100-1850 nm) was tuned to monitor the fluctuation of the photocurrent ( $I_{ph}$ ). As plotted in Fig. S3, all  $I_{ph}$  measured from PD.I are fitted nicely to  $I_{ph} \propto P^{\alpha}$  with  $\alpha$  values from 0.155 to 1.007, whereas the device is photoconductive or photogating predominant when  $\alpha = 1$  or  $\alpha = 0$ , respectively.<sup>2–5</sup> These data hint the co-existence of above-mentioned two effects in PD.I, and similar results were obtained in PD.II and III.



Fig. S3. (a) photocurrent ( $I_{ph}$ ) as a function of P in PD.I; the dots represent the experimental data whereas lines refer to the nice fitting of  $I^{\sim}P^{\alpha}$ ; (b) plot of  $\alpha$  values mostly in range from 0 to 1, representing the interplay of photoconductive and photogating effects.

### 4. Responsivity R as a function of P

Fig. S4 plots the R<sup>P</sup> relationships in PD.I upon the pulsed-laser radiation. These curves are mainly linear yet in some cases approach the plateau at higher P, which are typically ascribed to the co-existence of thermal-induced mechanisms.



**Fig. S4.** Plot of resbonsivity (R) versus P, showing a general decline of R with the increase of P; in some cases, for instance radiant by 1100nm, 1200nm and 1850nm, the slops of lines gradually decrease with the increase of P, indicating the enrolment of thermal-correlated effects in the photodetector

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## 5. Pyroelectric photoresponse under the visible light radiation or in the Bi<sub>2</sub>O<sub>2</sub>Se/Si photodetector

Fig.5a plots an example of the unconventionally low pyroelectric photoresponse under visible light radiation owing to the large channel area and planar electrodes' configuration. The photoresponse of the  $Bi_2O_2Se/Si$  photodetector is shown in Figure 5b, in which the measured I-t curve yields only one instant current peak right after switching on the laser. We therefore deduce that  $Bi_2O_2Se$  and  $SrTiO_3$  are both obligated to the pyroelectricity in the  $Bi_2O_2Se/SrTiO_3$  photodetectors.



Fig. S5. (a) I-t curve of PD.III illuminated by the 450 nm laser showing lower pyroelectric current than the corresponding saturated photocurrent. (b) top: a schematic of the  $Bi_2O_2Se/Si$  device; bottom: an optical microscope image of the selectively etched as-grown  $Bi_2O_2Se$  layer before transferring; an AFM image of the  $Bi_2O_2Se/Si$  photodetector; (c) I-t curve of the device under the illumination of 450 nm laser.

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#### 6. I-V curves of PD.II' and PD. III'

The shift of the I-V curves with respect to the origin is observed when illuminated at two ends. The shifting direction follows the contact polarity, in consistent with the scenario shown in Figure 5b at zero bias.

It is noteworthy that the "center" positions in I-V curves of PD I'-III' (Figure S6) have a different meaning from the one defined in the I-t curves in Figure 5b. The former refers to the position with zero internal potential, by which means, the position of zero net current under laser radiation and zero bias. The latter is defined by the geometric centre of the device. In PD.I'-III' these two "centre" positions do not overlap, elucidating slightly asymmetric work functions at two ends. This result is in consonance with the observation showing unequal absolute current values at two ends.



Fig. S6. I-V curves around the origin points measured in (a) PD. I', (b) PD. II' and (c) PD. III' where I-V curves are measured at two ends and in the channel centre.

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#### 7. Thickness dependent photoresponse under zero bias

The self-powered photoresponse is not highly seebeck coefficient (S) dependent. Based on the semiclassical Boltzmann transport theory, a trade-off between S and carrier concentration (n) can be expressed:<sup>6,7</sup>

$$S = \frac{8\pi^2 k_B^2}{3eh^2} m^* T \left(\frac{\pi}{3n}\right)^2 /_3$$
(4)

where  $k_B$  and  $m^*$  are the Boltzmann constant and the electron effective mass, respectively. Hence, owing to the positive correlation between  $\delta$  and n in our PLD-grown  $Bi_2O_2Se$  layers, S varies inversely to  $\delta$ .<sup>1</sup> Thereby, S is excluded from the predominant impact on the photoresponse.



Fig. S7. Self-powering photoresponse. Comparison of (a) R; (b) external quantum efficiency (EQE); (c) detectivity (D\*) among PD.I'-III' under zero bias

Substrate	gating	Lateral dimension (µm)	Detection band (nm)	R(A/W)	D* (Jones)	Detected wavelength (nm)	Ref.
SrTiO₃	No	500	365-1850	12.62	1.51×10 <sup>9</sup>	1100	This work
Mica	No	100	350-1600	320	-	550	8
Mica	No	10	360-1090	9.1	1.3×10 <sup>8</sup>	640	3
Mica	No	7-11	808	6.5	8.3×10 <sup>11</sup>	808	9
Mica	Yes		360-1800	108 696	8.2×10 <sup>12</sup>	360	10
Si	Yes	5		3.5×104	9×10 <sup>13</sup>	532	4

#### 8. Table S1: performance comparison of Bi<sub>2</sub>O<sub>2</sub>Se-based photodetectors under external bias of 1V

Under the optimal condition of  $Bi_2O_2Se$  transferring, a much better photoresponse can be achieved owing to the scaled channel dimension and external gating. Nevertheless, the manufacturing of photodetectors based on the as-grown  $Bi_2O_2Se$  photodetectors is facile that skips the complicated procedures of transferring  $Bi_2O_2Se$  to the Si substrate. In addition, it impedes the contamination and quality degradation of the crystal during transferring.

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