

## Supplementary Information

### Selective area grown AlInGaN nanowire arrays with core-shell structures for photovoltaics on silicon

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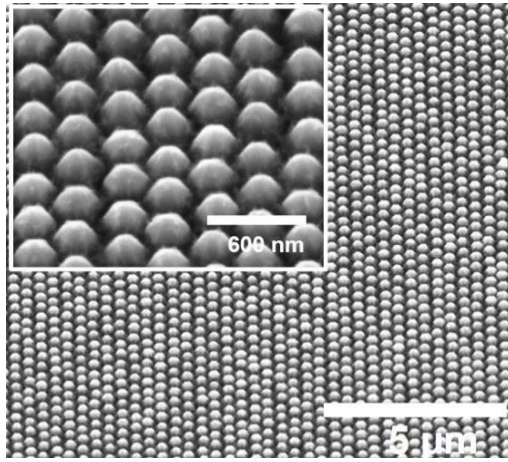
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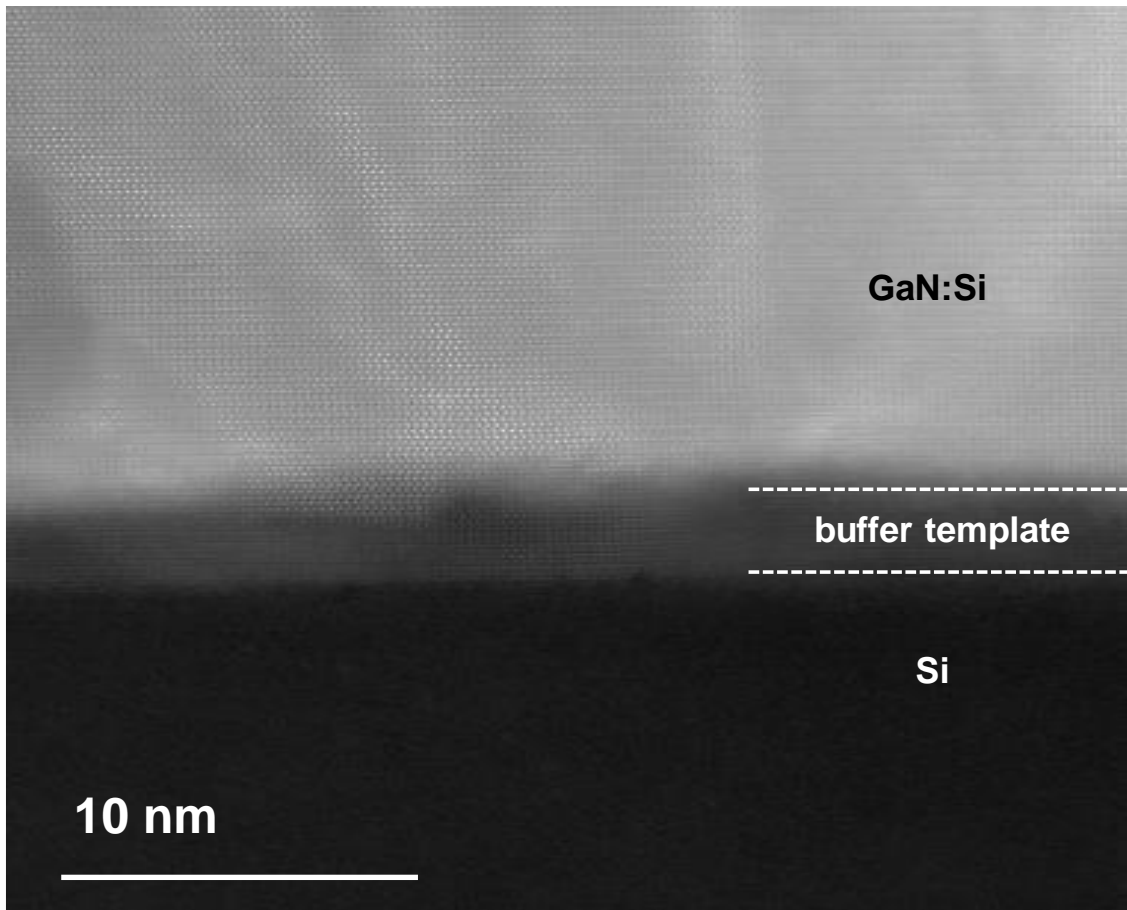
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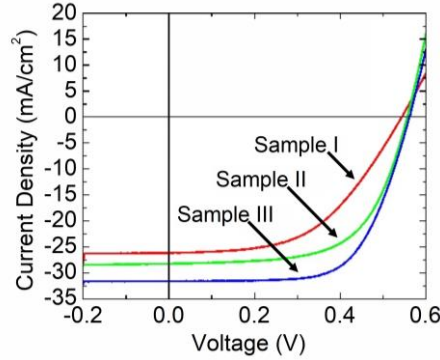
**Fig. S1** SEM image of exposed GaN:Mg segments of NW array coated with polyimide resist. The inset is magnified image of exposed GaN:Mg array. Facility for Electron Microscopy Research, McGill University.



**Fig. S2** The high-resolution HAADF-STEM image showing the interface between GaN:Si segment and Si substrate. In the thin AlN/GaN:Ge buffer layer, the presence of slightly different contrast may be due to the variations of composition and crystal orientations of buffer template. Canadian Centre for Electron Microscopy, McMaster University.

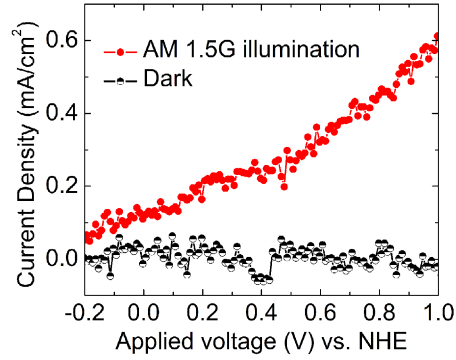
**Table S1.** MBE parameters of AlN layers on Si tunnel-junction SCs and the resulting PV characteristics.

Sample	Al BEP [ $\times 10^{-8}$ Torr]	$T_{\text{Substrate}}$ [ $^{\circ}\text{C}$ ]	$J_{\text{sc}}$ [ $\text{mA}/\text{cm}^2$ ]	$V_{\text{oc}}$ [V]	$\eta$ [%]	$FF$ [%]
I	No AlN	No AlN	26.12	0.54	7.28	51.2
II	5.46	750	28.24	0.56	9.77	62.1
III	3.18	810	31.61	0.56	11.59	65.2

**Fig. S3** Current density-voltage characteristics of Si SCs with a Si TJ. Sample I is as-fabricated Si TJ SCs, Sample II and Sample III were Si TJ SCs with thin AlN layer.

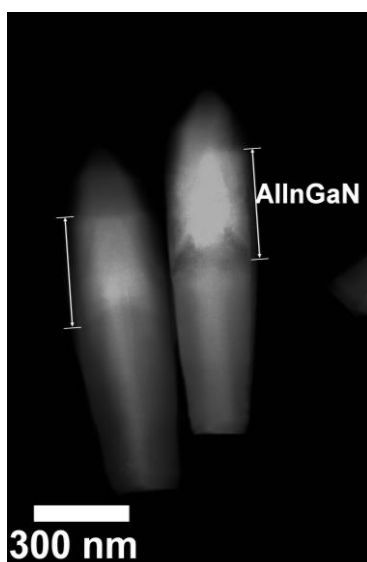
The effects of MBE-grown AlN layers on the PV characteristics of Si SCs with a  $p^+$ -Si/ $n$ -Si TJ were investigated. The preparation of Si SC wafer included spin coating of liquid boron (B) dopant precursor and liquid phosphorous (P) dopant precursor on the two surfaces of double-side-polished  $n$ -Si, and thermal annealing of the coated wafers for the conversions to  $p^+$ -Si and  $n^+$ -Si layers via thermal-diffusion doping process. Subsequently, liquid P-dopant precursor was spin coated on the surface of  $p^+$ -Si and underwent rapid thermal annealing to form few-nm-thick  $n$ -Si layer at the top of  $p^+$ -Si layer. The as-synthesized sample is denoted as Sample I (reference sample). The Si TJ-SC samples were loaded into RF PA MBE (Veeco, GEN II) system for the growth of thin AlN layer on the Si TJ ( $p^+$ -Si/ $n$ -Si) using various growth parameters shown in Table S1. To avoid  $\text{SiN}_x$  formation between Si and AlN layer, Al shutter were open prior to the excitation of  $\text{N}_2$  plasma. Sample I, Sample II and Sample III were fabricated using the following procedures. Ti (20 nm)/Au (100 nm) metal was deposited on the  $n^+$ -Si layer as bottom metal contact by using e-beam evaporation. Ti (20 nm)/Au (100 nm) metal grid patterns were then deposited on TJ ( $p^+$ -Si/ $n$ -Si) as top metal contact by using standard photolithography and e-beam evaporation, followed by a rapid thermal annealing (RTA) at  $550^{\circ}\text{C}$  in  $\text{N}_2$  gas ambient for 1 min.

PV characteristics of Sample I, Sample II and Sample III under  $100 \text{ mW}/\text{cm}^2$  illumination from an AM 1.5G solar simulator are summarized in Fig. S3. and in Table S1. Sample II and Sample III exhibit higher short-circuit current density ( $J_{\text{sc}}$ ), conversion efficiency ( $\eta$ ) and fill factor ( $FF$ ) than Sample I. The increase in open-circuit voltage ( $V_{\text{oc}}$ ) due to the addition of AlN layer is negligible, which is in the range of  $\sim 0.02 \text{ V}$ . The improved  $J_{\text{sc}}$  and  $\eta$  are possibly due to AlN antireflection effect<sup>1-3</sup> instead of surface passivation effect. This study demonstrates the good electrical conductivity of AlN layer, and paves the way to integrate SAG III-nitride NWs with Si SCs for multijunction (Al)InGa $\text{N}/\text{Si}$  integrated SCs and double-band (Al)InGa $\text{N}/\text{Si}$  photoelectrochemical cells (PECs).



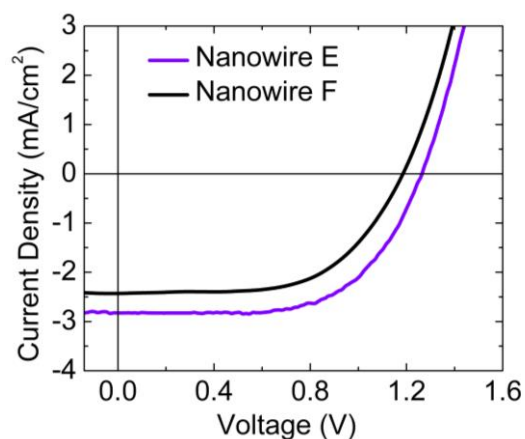
**Fig. S4** Photoelectrochemical (PEC) characterization of *n*-GaN NW-array photoanodes selective-area grown on Si substrate. The variations of current density with applied voltage versus normal hydrogen electrode (NHE) for 0.014 cm<sup>2</sup> NW array in 1 M HBr solution under AM 1.5G illumination of 100 mW/cm<sup>2</sup> (red and solid symbol) and in dark environment (black and semi-hollow symbol).

The device preparation procedures and PEC characterization configurations can be found in literatures<sup>4-7</sup>. Both a thermopile (818P-100-55, Newport) and a photodiode sensor (818-ST2-UV/DB, Newport) with attenuator were used to calibrate the illumination intensity. Under AM 1.5G illumination of 100 mW/cm<sup>2</sup>, the photo-current density ( $J_{ph}$ ) reached 0.61 mA/cm<sup>2</sup> at 1.0 V versus normal hydrogen electrode (NHE).



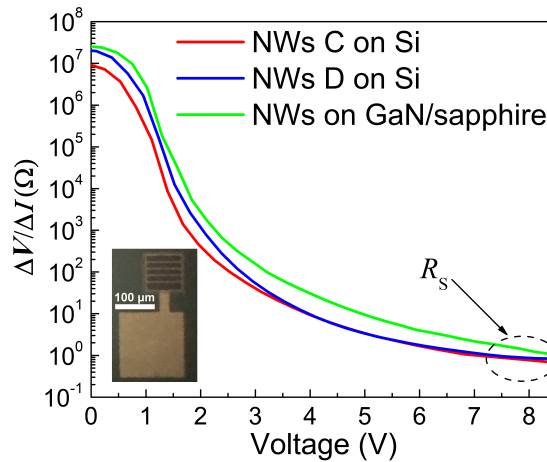
**Fig. S5** An HAADF-STEM image of SAG GaN:Si/AlInGaN/GaN:Mg NWs. The labelled regions are AlInGaN segments showing In-rich AlInGaN core and AlGaN shell based on atomic-number contrast. Facility for Electron Microscopy Research, McGill University.

Figure S5 shows HAADF-STEM image of SAG AlInGaN NWs based on atomic-number contrast. Light element (Al) distribution presents dark contrast while heavy element (In) distribution presents bright contrast. It is presented clearly from bottom to top that *n*-GaN, AlInGaN and *p*-GaN segments exhibit tapered and faceted hexagonal structure with Ga polarity. AlInGaN segments presents brighter intensity at the center and darker intensity at the sidewall, indicating the formation of core-shell structure in SAG AlInGaN NWs. A core-shell structure has been reported from spontaneously grown AlInGaN NWs.<sup>8</sup>



**Fig. S6** Current density-voltage ( $J$ - $V$ ) output characteristics of two SAG AlInGaN NW SCs on Si substrates with high Al incorporation than that of NWs A, B, C and D. The samples differ in the MBE parameters of AlInGaN active segments.

The AlInGaN segments of NWs E and F have higher Al incorporation than those of NWs A, B, C and D. Compared to the small blueshift in peak wavelength of NWs A, B, C and D from 527.9 to 515.8 nm, the increased Al incorporation in NWs E and F lead to remarkable blueshift in PL emission from green emissions of NWs A, B, C and D to blue emission (465 nm) of NW E and deep-blue emission (432 nm) of NW F. As shown in Fig. S6,  $J$ - $V$  output characteristics of NWs E and F exhibit lower  $J_{sc}$ , lower FF in the range of 0.594-0.615 and lower  $\eta$  in the range of 1.71-2.19%, compared to those of NW D SC.



**Fig. S7**  $\Delta V/\Delta I$  of  $n$ -GaN/AlInGaN/ $p$ -GaN NW arrays (NWs C and NWs D) selective-area grown on  $n$ -Si substrate and  $n$ -GaN/InGaN quantum dots/ $p$ -GaN selective-area grown on GaN/sapphire substrate.  $\Delta V/\Delta I$  is calculated based on the  $I$ - $V$  characteristics of  $\sim 100 \times 100 \mu\text{m}^2$  SAG NW array under forward bias in the dark environment. The series resistances ( $R_s$ ) are estimated by using similar methods in literatures<sup>9-11</sup>. The estimation of  $R_s$  is based on Equation 8. The inset is the image of  $\sim 100 \times 100 \mu\text{m}^2$  SAG NW array device (NWs C) on  $n$ -Si.

## Reference

1. P. M. Kaminski, K. Bass and G. Claudio, *Phys. Status Solidi C*, 2011, **8**, 1311-1314.
2. G. Krugel, A. Sharma, W. Wolke, J. Rentsch and R. Preu, *Phys. Status Solidi RRL*, 2013, **7**, 457-460.
3. C. Wang, W. Cheng, P. Ma, R. Xia and X. Ling, *J. Mater. Chem. A*, 2017, **5**, 2852-2860.
4. S. Fan, B. AlOtaibi, S. Y. Woo, Y. Wang, G. A. Botton and Z. Mi, *Nano Lett.*, 2015, **15**, 2721-2726.
5. S. Fan, I. Shih and Z. Mi, *Adv. Energy Mater.*, 2017, **7**, 1600952.
6. S. Vanka, Y. Wang, P. Ghamari, S. Chu, A. Pandey, P. Bhattacharya, I. Shih and Z. Mi, *Solar RRL*, 2018, **2**.
7. S. Fan, S. Y. Woo, S. Vanka, G. A. Botton and Z. Mi, *APL Materials*, 2016, **4**.
8. R. Wang, X. Liu, I. Shih and Z. Mi, *Appl. Phys. Lett.*, 2015, **106**, 261104.
9. J.-K. Sheu, K.-H. Chang, S.-J. Tu, M.-L. Lee, C.-C. Yang, C.-K. Hsu and W.-C. Lai, *Opt. Express*, 2010, **18**, A562-A567.
10. M. Wolf and H. Rauschenbach, *Advanced Energy Conversion*, 1963, **3**, 455-479.
11. Y.-j. Liu, C.-c. Huang, T.-y. Chen, C.-s. Hsu, S.-y. Cheng, K.-w. Lin, J.-k. Liou and W.-c. Liu, *Progress in Natural Science: Materials International*, 2010, **20**, 70-75.