# Electronic Supplementary Information

## Directly Synthesis of 1T-phase MoS<sub>2</sub> Nanosheets with Abundance

### Sulfur-Vacancies through $(CH_3)_4N^+$ Cations-Intercalation for

### **Hydrogen Evolution Reaction**

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Fig.S1 SEM images of (a-b) bulk  $MoS_2$  and (c-d) TMA-MoS<sub>2</sub>.



**Fig.S2** (a-b) TEM and (c) the section profile along the blue line in (b) of bulk  $MoS_2$ . (d-e) HRTEM images and (f) HAADF-TEM with corresponding EDS elemental mapping images of bulk  $MoS_2$ . The average interlayer spacing is 6.13 Å consistent with XRD result, and the structure is mainly pure 2H-phase.



**Fig.S3** (a-b) TEM and (c) the section profile along the blue line in (b) of TMA-MoS<sub>2</sub>. (d-e) HRTEM images and (f) HAADF-TEM with corresponding EDS elemental mapping images of TMA-MoS<sub>2</sub>. The average interlayer spacing of (002) plane is 9.9 Å, and the dominant structure is 1T-phase.



**Fig.S4** Fourier transform infrared (FT-IR) spectra of TMA, TMA-MoS<sub>2</sub>, and ST-MoS<sub>2</sub>, suggesting the presence of TMA in the TMA-MoS<sub>2</sub> and absence of TMA in the ST-MoS<sub>2</sub> (the main range marked in yellow).



**Fig.S5** (a) Optical images of C-MoS<sub>2</sub>, MH-MoS<sub>2</sub>, and ST-MoS<sub>2</sub> exfoliated from C-MoS<sub>2</sub>, bulk MoS<sub>2</sub>, and TMA-MoS<sub>2</sub> before and after centrifugation. The concentration of all samples is 1 mg/mL. (b) The colloidal photographs of C-MoS<sub>2</sub>, MH-MoS<sub>2</sub>, and ST-MoS<sub>2</sub> dispersed in 4:1 v/v IPA/H<sub>2</sub>O solution and after 4 months. It clearly presents that the precipitation is not observed in the colloidal ST-MoS<sub>2</sub> solution even after 4 months. However, it is found in the C-MoS<sub>2</sub> and MH-MoS<sub>2</sub>, respectively, suggesting that as-synthesized colloidal ST-MoS<sub>2</sub> nanosheets possess the best stability among the samples. This will be examined in the following zeta-potential measurement (**Fig. S9**).



**Fig.S6** Raman spectrum of C-MoS<sub>2</sub>. Only two peaks at 380 and 406 cm<sup>-1</sup> are observed, which corresponds to the in-plane vibration  $(E^{12}_{g})$  and out-of-plane mode  $(A_{1g})$ , respectively, suggesting the 2H-phase structure.



**Fig.S7** (a) High-resolution XPS spectra of Mo 3d core-level peaks of C-MoS<sub>2</sub> and (b) S 2p core-level peaks of ST-MoS<sub>2</sub>, MH-MoS<sub>2</sub> and C-MoS<sub>2</sub>, respectively.



**Fig.S8** High-resolution XPS spectra of Mo 3d core-level peaks of bulk  $MoS_2$  and TMA- $MoS_2$ . Both the Mo  $3d_{3/2}$  and  $3d_{5/2}$  peaks of TMA- $MoS_2$  were shifted to lower binding energies by about 1 eV with respect to those of bulk  $MoS_2$  peaks, suggesting the 1T-pahse was formed during hydrothermal process rather than ultrasonic exfoliation.



**Fig.S9** Zeta-potential measurement of (a) C-MoS<sub>2</sub>, (b) MH-MoS<sub>2</sub> and (c) ST-MoS<sub>2</sub> suspensions in 4:1 v/v IPA/H<sub>2</sub>O solution with a concentration of 0.01 mg/mL. All samples are tested three times in parallel. Owing to the electrical double layers on the nanomaterial surface reflected to the nanomaterial solution stability, zeta-potential measurements were carried out. In general, good dispersed nanomaterials possess the zeta-potential values of less than -30 mV. <sup>[S16]</sup> Clearly, we can observe that the zeta-potential of MH-MoS<sub>2</sub> suspensions is -32.1 mV, indicating the as-synthesized MH-MoS<sub>2</sub> nanosheets possess a relative stability and negative charge surface. Notably, the zeta-potential of ST-MoS<sub>2</sub> (-53.9 mV) is more negative than that of C-MoS<sub>2</sub> and MH-MoS<sub>2</sub>, suggesting the outstanding dispersed stability, which can be expected to the more exposed edges and basal planes of ST-MoS<sub>2</sub>.



Fig.S10 The Mo K-edge X-ray absorption spectroscopy (XAS) spectra of MH-MoS $_2$  and ST-MoS $_2$ .



**Fig.S11** (a) Mo K-edge XAS spectra and (b) derivative XAS spectra of ST-MoS<sub>2</sub> and TMA-MoS<sub>2</sub>. (c) Mo K-edge EXAFS oscillations and (d) Fourier transform of the  $k^2$ -weighted Mo K-edge of the EXAFS spectra of ST-MoS<sub>2</sub> and TMA-MoS<sub>2</sub>.



Fig.S12 The EXAFS fitting curves in R space of MH-MoS<sub>2</sub>.



Fig.S13 The EXAFS fitting curves in R space of ST-MoS<sub>2</sub>.



Fig.S14 SEM images of (a-b) pure CdS and (c-d) CdS/ST-MoS<sub>2</sub>.



**Fig.S15** (a) TEM, (b) HRTEM of pure CdS. (c) The corresponding FFT pattern marked with red square and (d) the section profile along the blue line in (b). (e) HAADF-TEM image with corresponding EDS elemental mapping images of pure CdS.



**Fig.S16** (a) TEM, (b) HRTEM and (c) HAADF-TEM image with corresponding EDS elemental mapping images of CdS/ST-MoS<sub>2</sub>.



**Fig.S17** XRD patterns of CdS and  $MoS_2/ST-MoS_2$  nanocomposites with varying  $MoS_2$  loading. All the samples have similar diffraction peaks, which can be indexed to the hexagonal phase of CdS (PDF#41-1049). With the increase of the loading amount of ST-MoS<sub>2</sub>, it can be observed that the weak diffraction peaks of ST-MoS<sub>2</sub>, e.g., (100), (101), and (103) planes, gradually appear; this can be ascribed to the small amount or the low crystallization of ST-MoS<sub>2</sub>.



Fig.S18 Calibration curve collected with linear fit showing strong linear dependence of signal intensity to  $H_2$  volume.



**Fig.S19** The  $H_2$  evolved rate changes along with the mass of CdS/ST-MoS<sub>2</sub>. It is noted that the  $H_2$  evolved rates don't linearly increase with respect to the photocatalyst mass due to the light-shielding effect. ref: 10.1016/j.joule.2021.01.001



**Fig.S20** UV/Vis diffuse reflectance spectra of CdS/ST-MoS<sub>2</sub> with various ST-MoS<sub>2</sub> loading. The pure CdS and CdS/ST-MoS<sub>2</sub>-x samples appear a significant absorption at wavelength shorter than 520 nm, which is associated with the intrinsic bandgap absorption of the CdS semiconductor. Besides, it can be observed that the absorbance continuously increase, in visible light region from 520 to 800 nm, as the increase of the loading amount of UT-MoS<sub>2</sub>. Although the adsorption of CdS/ST-MoS<sub>2</sub> is enhanced greatly when the loading amount of cocatalyst exceeds 7wt%, the HER performances exhibit a downward trend, which is ascribed to the "shielding effect".



Fig.S21 The  $H_2$  evolved rate changes along with the reaction time over various CdS/ST-MoS<sub>2</sub> photocatalysts.



Fig.S22 (a) XPS, (b) Raman, (c) TEM, and (d) HRTEM images of the spent CdS/ST- $MoS_2$ .



**Fig.S23** Tauc plots  $((\alpha h \upsilon)^2 \text{ vs. } h \upsilon)$  converted form UV/Vis diffuse reflectance spectra of St-MoS<sub>2</sub>, MH-MoS<sub>2</sub>, and CdS.



Fig.S24 Mott-Schottky plots of CdS collected at various frequencies.

Table S1. Elemental Analyses of ST-MoS<sub>2</sub>.

Methods	Mo atom (%)	S atom (%)	S/Mo	S vacancies (%)
XPS	36.4	63.6	1.75	12.5
EDS	35.7	64.3	1.80	10.0

Table S2. The AQE values of various wavelength over CdS/ST-MoS<sub>2</sub>.

		0			
Wavelength (nm)	420	450	500	550	
Light intensity (mW)	125	137	162	148	
Evolved H <sub>2</sub> (umol/h)	579	621	724	26.6	
AQE (%)	73.4	67.0	59.4	2.2	
For example, here we calculate the AQE value at 420 nm:					

$40F - \frac{2 \times N_{H_2}}{100\%} \times 100\% -$	$2 \times 6.02 \times 10^{23} \times 579 \times 10^{-6}$	× 100% - 73 4%
$AQL = \frac{1}{I \times \lambda \times t} \times 100 \% =$	$\overline{125 \times 10^{-3} \times 420 \times 10^{-9} \times 3600}$	~ 100 /0 = 7 3.4 /0
h  imes c	$6.61 \times 10^{-34} \times 3 \times 10^{8}$	

Table S3. EXAFS fitting parameters at the Mo K-edge for various samples

	01			0	1	
Sample	Shell	N <sup>a</sup>	R (Å) <sup>b</sup>	σ <sup>2</sup> (Å <sup>2</sup> ·10 <sup>-3</sup> ) <sup>c</sup>	ΔE0 (eV) <sup>d</sup>	R factor (%)
MH-MoS <sub>2</sub>	Mo-S	5.8	2.41	3.2	3.4	0.1
	Mo-S-Mo	4.6	3.16	4.6	3.1	
ST-MoS <sub>2</sub>	Mo-S	4.6	2.41	7.9	0.4	0.8
	Mo-Mo	1.2	2.76	7.5	1.5	
Standard MoS <sub>2</sub>	Mo-S	6	2.41	2.9	2.1	0.3
	Mo-S-Mo	6	3.16	3.5	1.0	

<sup>*a*</sup> *N*: coordination numbers; <sup>*b*</sup> *R*: bond distance; <sup>*c*</sup>  $\sigma^2$ : Debye-Waller factors; <sup>*d*</sup>  $\Delta E_0$ : the inner potential correction. *R* factor: goodness of fit. *S*02 were set as 0.965/0.965 for Mo-S/Mo-Mo, which was obtained from the experimental EXAFS fit of reference MoS<sub>2</sub> by fixing CN as the known crystallographic value and was fixed to all the samples.

Samples	η(mV) at	Tafel slope	References
	$10 \text{ mA/cm}^2$	(mV/dec)	
ST-MoS <sub>2</sub>	178	59	This work
MoS <sub>2</sub> -8A	300	85	ACS Nano 2019, <b>13</b> , 6824
H <sub>2</sub> O <sub>2</sub> treated MoS <sub>2</sub>	131	48	J. Am. Chem. Soc. 2020, 142, 4298
Co-1T MoS <sub>2</sub>	195	46	ACS Energy Lett. 2018, 3, 7
Ni-1T MoS <sub>2</sub>	191	47	
1T- MoS <sub>2</sub>	219	106	Nat. Commun. 2019, 10, 982
FeO@1T- MoS <sub>2</sub>	187	108	
$CoO@1T-MoS_2$	117	84	
Monolayer MoS <sub>2</sub>	256	93	J. Phys. Chem. Lett. 2019, 10, 4763
Ni-MoS2	>200	89	J. Am. Chem. Soc. 2017, 139, 15479
Metallic 1T MoS <sub>2</sub>	175	41	Nat. Commun. 2016, 7, 10672
1T- MoS <sub>2</sub>	203	48	J. Am. Chem. Soc. 2016, <b>138</b> , 7965
MoS <sub>2</sub> /Ni <sub>3</sub> S <sub>2</sub>	110	83	Angew. Chem. 2016, 55, 6702
1T/2H- MoS <sub>2</sub>	320	61	ACS Appl. Mater. Inter. 2017, 9, 25291
T- MoS <sub>2</sub>	290	78	Nat. Energy 2017, 2, 17127

Table S4. Comparison of HER performance in  $0.5 \text{ M H}_2\text{SO}_4$  solution for 1T phase MoS<sub>2</sub>-based electrocatalysts.

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