

# Facile Template-free Preparation of Silver-coated $\text{Cu}_3\text{SbS}_4$ Hollow Spheres with Enhanced Photoelectric Properties

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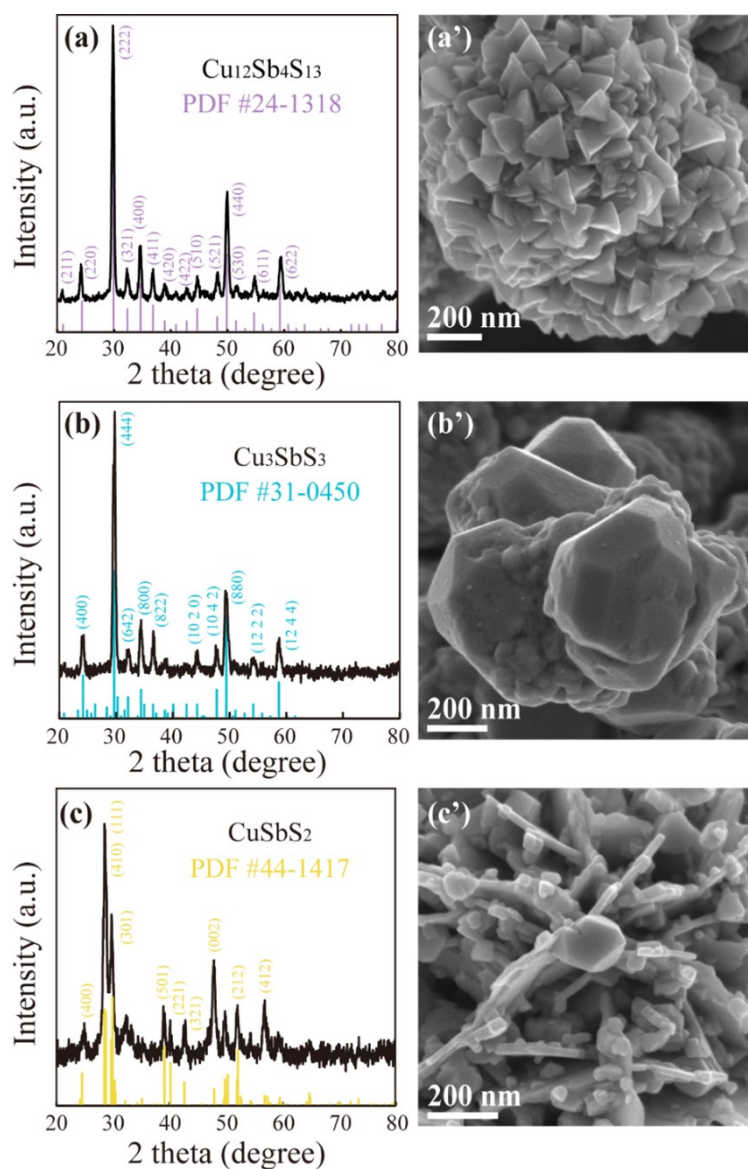


Figure S1. XRD patterns of the as-synthesized (a)  $\text{Cu}_{12}\text{Sb}_4\text{S}_{13}$ , (b)  $\text{Cu}_3\text{SbS}_3$  and (c)  $\text{CuSbS}_2$  by solvothermal method along with the standard reference pattern as #24-1318, #31-0450 and #44-1417, respectively, SEM images of (a')  $\text{Cu}_{12}\text{Sb}_4\text{S}_{13}$ , (b')  $\text{Cu}_3\text{SbS}_3$  and (c')  $\text{CuSbS}_2$ . High-purity Cu-Sb-S phases were formed and distinguishing morphology (including configuration and size) could be recognized from SEM images.

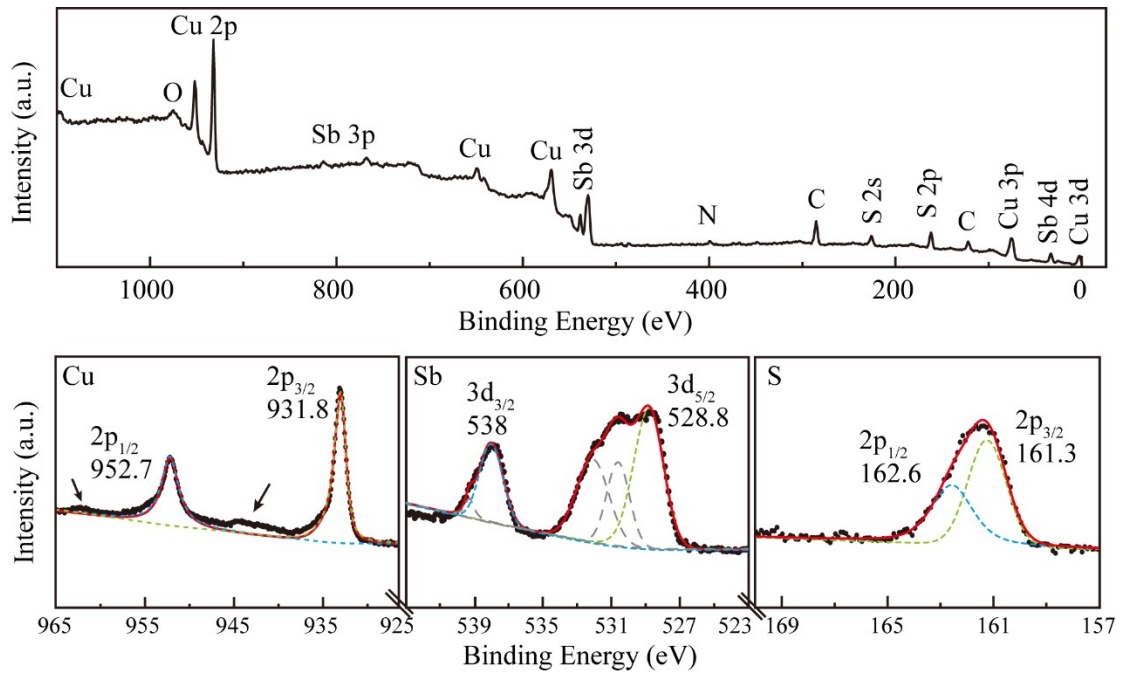


Figure S2. XPS spectra of the as-synthesized  $\text{Cu}_{12}\text{Sb}_4\text{S}_{13}$  nanocrystals including the survey spectrum along with the corresponding high-resolution XPS spectra for Cu 2p, Sb 3d and S 2p core levels.

Calculation of Ag content:

The relative content of Ag and  $\text{Cu}_3\text{SbS}_4$  in the composites can be calculated by the K value method, and the formula is described as follows<sup>1</sup>:

$$I_A/I_S = k \times W_A/W_S$$

Where I is the diffraction peak intensity, W is the weight percentage, and k is a constant value. The k value could be obtained from the standard PDF card. Both 0.5 vol% Ag and 1 vol% Ag content in the composites could be roughly estimated from XRD patterns.

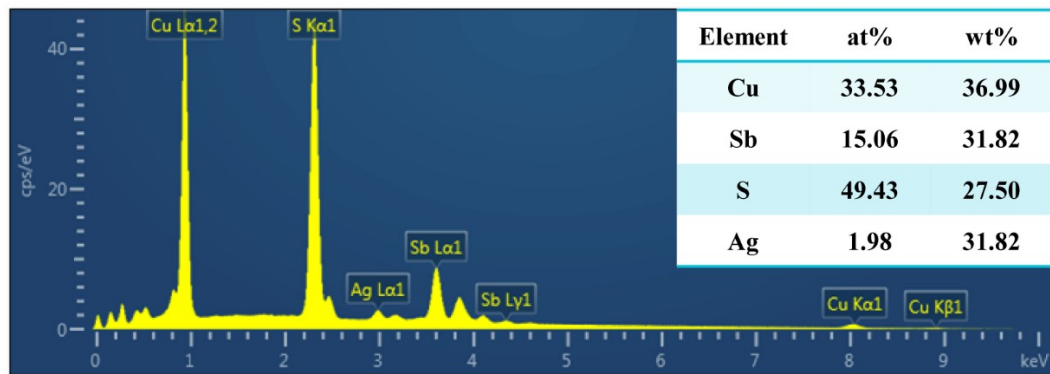


Figure S3. EDS result for the elemental mapping image along with the relative chemical composition of the as-synthesized Ag-coated  $\text{Cu}_3\text{SbS}_4$  nanocrystals hollow spheres.

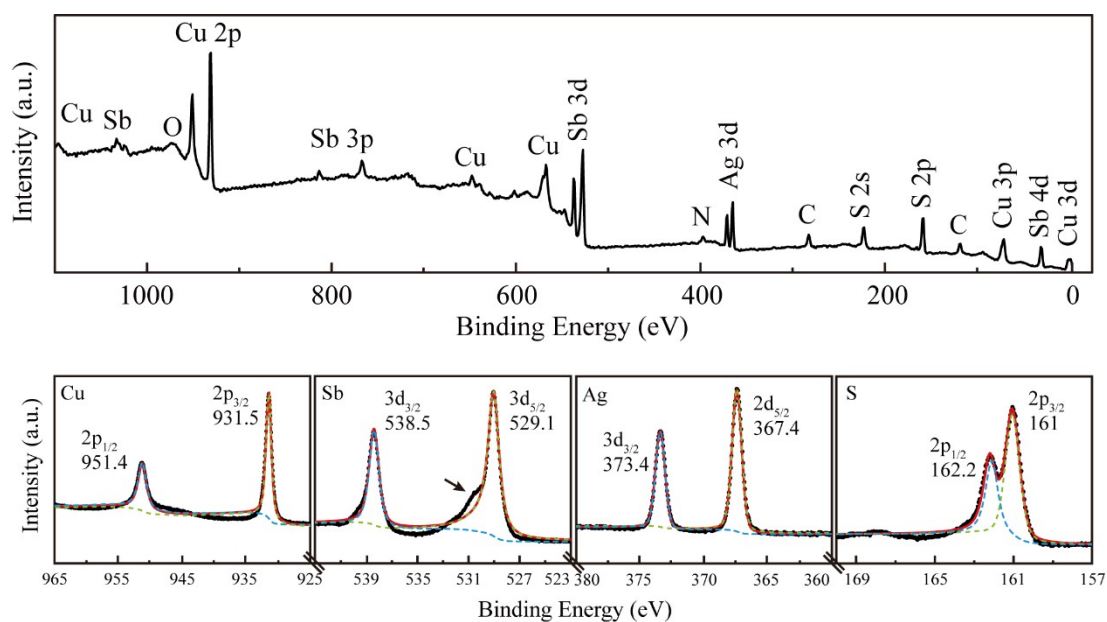


Figure S4. XPS spectra of the as-synthesized Ag-coated  $\text{Cu}_3\text{SbS}_4$  nanocrystals hollow spheres including the survey spectrum along with the corresponding high-resolution XPS spectra for Cu 2p, Sb 3d, Ag 2d and S 2p core levels. A relatively small satellite peak at 530.6 eV appeared, which means a very little reduction of  $\text{Sb}^{5+}$  into  $\text{Sb}^{3+}$  by sodium borohydride. The change has not been mentioned in the main body due to a relatively small change.

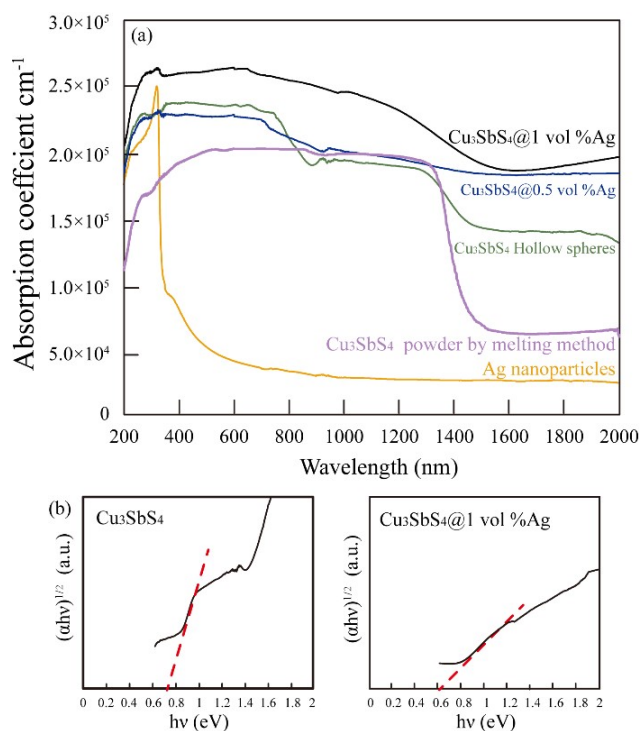


Figure S5. (a) UV-Vis-NIR spectra of Ag nanoparticles,  $\text{Cu}_3\text{SbS}_4$  powder by melting method,  $\text{Cu}_3\text{SbS}_4$  nanocrystals hollow spheres, 0.5 vol % Ag-coated  $\text{Cu}_3\text{SbS}_4$  and 1 vol % Ag-coated  $\text{Cu}_3\text{SbS}_4$ , the corresponding Tauc plot of  $(\alpha hv)^{1/2}$  versus  $(hv)$  of the as-synthesized (b)  $\text{Cu}_3\text{SbS}_4$  nanocrystals hollow spheres and (c) Ag-coated ones.

Given the absorption spectrum of Ag clusters, the Ag nanoparticles with sphere structure possessed a broad peak at 400 nm for surface plasmon resonance<sup>2</sup>. The broad peaks located around 320 nm and 900 nm attributed to the in-plane dipole plasmon and out-of-plane quadrupole resonances, respectively, which suggested the formation of Ag with triangular nanoprism structure<sup>3</sup>. This could be possibly aroused from geometry of Cu<sub>3</sub>SbS<sub>4</sub> nanoparticles precursor.

Calculation of optical band gap:

The optical bandgap of the film is determined from the absorption coefficient using

$$\alpha^m = A(h\nu - E_g) \quad (1)$$

where  $A$  is a constant,  $m$  is the transition probability,  $E_g$  is the optical band gap<sup>4</sup>. The value of  $n$  could be 1/2 or 2 for allowed indirect or direct transitions, respectively. From the experimental results and theoretical calculations for band structure,  $n = 1/2$  for Cu<sub>3</sub>SbS<sub>4</sub> with an indirect transition was chosen<sup>5-8</sup>. The bandgap of samples is determined from the plot of  $(\alpha \cdot h\nu)^2$  versus photon energy by extrapolating to  $\alpha \cdot h\nu = 0$ .

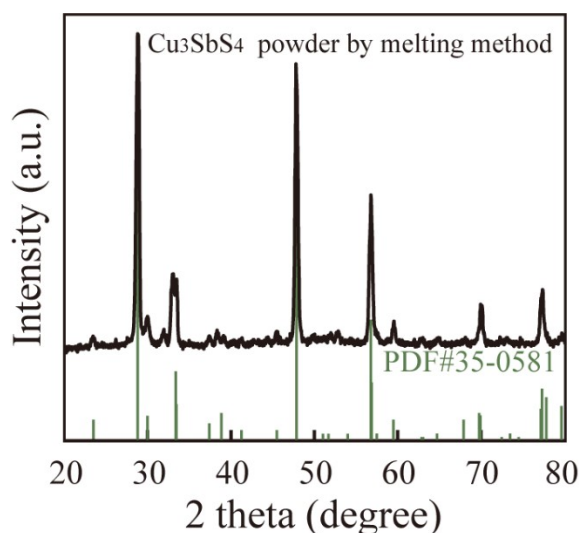


Figure S6. XRD pattern of the as-synthesized bulk Cu<sub>3</sub>SbS<sub>4</sub> by melting method, a small quantity of Cu<sub>12</sub>Sb<sub>4</sub>S<sub>13</sub> was detected. Polycrystalline sample of Cu<sub>3</sub>SbS<sub>4</sub> was synthesized by the stoichiometric amount of high purity elements (>99.99%) at 1173K for 24h, cooled down to 673K and held on for 2d. The obtained ingots were hand-ground into fine powder for hot pressing at 673 K for 20 min under a uniaxial pressure of 80 MPa.

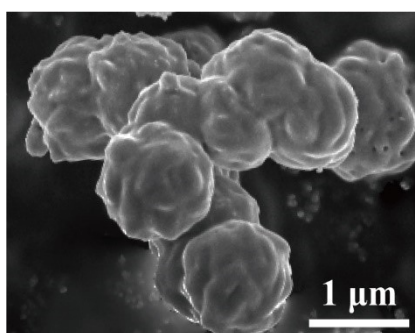


Figure S7. SEM images of the smaller spherical aggregation of crystals by only 5 min duration

time at high temperature

Table S1. The representative preparation method, structural properties and optical band gap (indirect) of  $\text{Cu}_3\text{SbS}_4$

Method	Size of crystals (nm)	Shape of crystals	Form	valence state	Optical band gap (eV)	Application	Ref.
Solvothermal method	200	-	Powder		1	-	9
Solvothermal method	26-32	sphere	Powder		0.88-1	-	10
Solvothermal method	20	-	Film (SC)		0.82	-	11
hydrothermal	40-60 in thickness	nanofibers	Powder		-	-	12
Hot-injection	10.2±1.1	spherical	Powder	1 3 2	0.9	-	6
Hot-injection	10	spherical	Powder	1 3 2	1	-	13
Hot-injection	10.5±1.7	spherical	Powder		1.72	-	14
Hot-injection	23±4	oblates	Powder		1.2	-	5
Hot-injection	61±19	irregular to tetragonal	Powder	1 3 2	1.33	-	15
Hot-injection	5.95±0.67	dots	Powder		1.5	-	7
	4.81±0.52				1.6		
	3.37±0.46				1.7		
Hot-injection	4.72±0.52	Quasi-spherical	Powder		1.4	For optoelectronic devices	16
Hot-injection	14-20	-	Film (spray deposited method using chlorobenzene)		0.89	Hole transporting material in solar cells	17
Hot-injection	21.5±8.0	Round-like	Powder		≥1	-	8
Hot-injection	19	-	Film (SC)	1 5 2	0.9	Absorber layer in TFSC	18
Microwave radiation	30-50	sphere-like	Powder	1 5 2	1.1	-	19
Mechanical alloying with spark plasma sintering	≥1000	-	Powder	1 5 2	0.85	Thermoelectric materials	20
Deep eutectic solvents synthetic approach	~7	nearly spherical	-	1 5 2	1.23	Water-splitting	21
RF magnetron sputtering with Sulfurization	≥1000	-	Film		0.89±0.01	-	22
Electron-beam evaporation with Sulfurization	≥1000	-	Film		0.88	-	23
Magnetron	50±30	-	Film		0.94-0.97	For TFSC	24

Where, TFSC is the abbreviation for thin film solar cells, SC is the abbreviation for spin-coating,  $\geq 1000$  means that the size of crystals was calculated roughly by us from SEM results in these papers and close to micrometer. The form is the state of  $\text{Cu}_3\text{SbS}_4$  when the UV-Vis measurement was conducted. Three numbers in valence state column are corresponding to valence states of element Cu, Sb and S in  $\text{Cu}_3\text{SbS}_4$ , respectively.

Reference:

- 1 C. R. Hubbard and R. L. Snyder, *Powder Diffr.*, 1988, **3**, 74–77.
- 2 H. Bi, W. Cai, L. Zhang, D. Martin and F. Träger, *Appl. Phys. Lett.*, 2002, **81**, 5222–5224.
- 3 J. Liu, X. Li and X. Zeng, *J. Alloys Compd.*, 2010, **494**, 84–87.
- 4 E. A. Davis and N. F. Mott, *Philos. Mag.*, 1970, **22**, 0903–0922.
- 5 K. Ramasamy, H. Sims, W. H. Butler and A. Gupta, *Chem. Mater.*, 2014, **26**, 2891–2899.
- 6 J. van Embden and Y. Tachibana, *J. Mater. Chem.*, 2012, **22**, 11466.
- 7 K. Chen, J. Zhou, W. Chen, Q. Chen, P. Zhou and Y. Liu, *Nanoscale*, 2016, **8**, 5146–5152.
- 8 F. Baum, T. Pretto, A. G. Brolo and M. J. L. Santos, *Cryst. Growth Des.*, 2018, **18**, 6521–6527.
- 9 L. Shi, C. Wu, J. Li and J. Ding, *J. Alloys Compd.*, 2017, **694**, 132–135.
- 10 J. Bincy, G. S. G. and L. R. A., *Mater. Res. Bull.*, 2017, **95**, 267–276.
- 11 M. Bella, C. Rivero, S. Blayac, H. Basti, M. C. Record and P. Boulet, *Mater. Res. Bull.*, 2017, **90**, 188–194.
- 12 C. An, Y. Jin, K. Tang and Y. Qian, *J. Mater. Chem.*, 2003, **13**, 301–303.
- 13 J. van Embden, K. Latham, N. W. Duffy and Y. Tachibana, *J. Am. Chem. Soc.*, 2013, **135**, 11562–11571.
- 14 S. Ikeda, S. Sogawa, Y. Tokai, W. Septina, T. Harada and M. Matsumura, *RSC Adv*, 2014, **4**, 40969–40972.
- 15 Q. Liang, K. Huang, X. Ren, W. Zhang, R. Xie and S. Feng, *CrystEngComm*, 2016, **18**, 3703–3710.
- 16 F. Zhang, K. Chen, X. Jiang, Y. Wang, Y. Ge, L. Wu, S. Xu, Q. Bao and H. Zhang, *J. Mater. Chem. C*, 2018, **6**, 8977–8983.
- 17 Q. Zeng, Y. Di, C. Huang, K. Sun, Y. Zhao, H. Xie, D. Niu, L. Jiang, X. Hao, Y. Lai and F. Liu, *J. Mater. Chem. C*, 2018, **6**, 7989–7993.
- 18 G. H. Albuquerque, K.-J. Kim, J. I. Lopez, A. Devaraj, S. Manandhar, Y.-S. Liu, J. Guo, C.-H. Chang and G. S. Herman, *J. Mater. Chem. A*, 2018, **6**, 8682–8692.
- 19 G. Chen, W. Wang, J. Zhao, W. Yang, S. Chen, Z. Huang, R. Jian and H. Ruan, *J. Alloys Compd.*, 2016, **679**, 218–224.
- 20 M. Shen, S. Lu, Z. Zhang, H. Liu, W. Shen, C. Fang, Q. Wang, L. Chen, Y. Zhang and X. Jia, *ACS Appl. Mater. Interfaces*, 2020, **12**, 8271–8279.
- 21 U. V. Ghorpade, M. P. Suryawanshi, S. W. Shin, X. Wang, E. Jo, H. Bae, K. Park, J.-S. Ha, S. S. Kolekar and J. H. Kim, *J. Mater. Chem. A*, 2018, **6**, 19798–19809.
- 22 P. A. Fernandes, A. Shongalova, A. F. da Cunha, J. P. Teixeira, J. P. Leitão, J. M. V. Cunha, S. Bose, P. M. P. Salomé and M. R. Correia, *J. Alloys Compd.*, 2019, **797**, 1359–1366.
- 23 L. Yu, R. S. Kokenyesi, D. A. Keszler and A. Zunger, *Adv. Energy Mater.*, 2013, **3**, 43–48.
- 24 N. D. Franzer, N. R. Paudel, C. Xiao and Y. Yan, in *2014 IEEE 40th Photovoltaic Specialist Conference (PVSC)*, IEEE, Denver, CO, USA, 2014, pp. 2326–2328.

