Supporting Information

Novel A-π-D-π-A-type BODIPY dyads as small-molecule donor

for solution-processed organic solar cells

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1. Theoretical calculation

The design concept of BODIPY molecules is presented in **Fig. S1**. Density functional theory (DFT) and time-dependent density functional theory (TD-DFT) calculations were conducted by Gaussian 09 software at the B3LYP/6-31G(d) level.¹ The optimized geometries, frontier orbital energy levels, electronic density distributions and the predicted UV-vis absorption spectra of **CBDP**, **DCBDP**, **CTBDP** and **DCTBDP** are shown in **Fig. S2**. The calculated data of electronic transitions for these molecules are also summarized in **Table S1**.



Fig. S1 The design concept of BODIPY molecules with DFT calculation method.



Fig. S2 (a) The optimized geometries; (b) energy-level and electronic density distributions; (c) predicted UV-vis

absorption spectra of CBDP, DCBDP, CTBDP and DCTBDP respectively.

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Molecule	HOMO ^{DFT} (eV)	LUMO ^{DFT} (eV)	EDFT g (eV)	Dipole moment (Debye)	Total energy (×10 ⁶ kJ mol ⁻¹)		
CBDP	-5.31	-2.97	2.34	14.3	-6.8		
DCBDP	-4.88	-2.86	2.02	0.0	-7.7		
CTBDP	-5.05	-3.01	2.04	15.3	-9.7		
DCTBDP	-4.80	-2.95	1.85	7.7	-10.6		

 Table S1 The energy level, bandgap, dipole moment and total energy of CBDP, DCBDP, CTBDP and DCTBDP

 based on ground-state optimized configurations with DFT calculation.

Table S2 The electronic transitions for CBDP, DCBDP, CTBDP and DCTBDP with TD-DFT calculation.

Molecule	State	E(eV)	λ (nm)	f	Composition
CBDP	S 1	2.04	609	0.4902	HOMO \rightarrow LUMO (70%)
	S 6	2.82	440	0.2448	HOMO-2 \rightarrow LUMO+1 (67%)
	S 8	3.14	394	0.3891	HOMO-4 \rightarrow LUMO (52%)
	S9	3.18	390	0.6872	HOMO-3 \rightarrow LUMO (66%)
	S11	3.41	363	0.2388	HOMO-4 \rightarrow LUMO+1 (53%)
	S13	3.48	356	0.4499	HOMO-6 \rightarrow LUMO+1 (56%)
	S20	3.93	315	0.3234	HOMO \rightarrow LUMO+2 (57%)
DCBDP	S 1	1.72	721	0.2265	HOMO \rightarrow LUMO (70%)
	S5	2.36	526	0.2077	HOMO-1 \rightarrow LUMO+1 (69%)
	S 7	2.96	419	0.2441	HOMO-3 \rightarrow LUMO (66%)
	S9	3.15	394	0.3378	HOMO-5 \rightarrow LUMO (48%)
	S11	3.21	386	0.6903	HOMO-4 \rightarrow LUMO+2 (66%)
	S14	3.43	362	0.3469	HOMO-7 \rightarrow LUMO (44%)
	S1	1.78	695	0.4888	HOMO \rightarrow LUMO (69%)
	S 7	2.90	428	1.6122	HOMO \rightarrow LUMO+2 (47%)
CTBDP	S9	3.04	408	0.3071	HOMO-3 \rightarrow LUMO (40%)
	S10	3.09	401	0.4202	HOMO-5 \rightarrow LUMO+1 (47%)
	S11	3.12	398	0.5389	HOMO-5 \rightarrow LUMO (35%)
DCTBDP	S 1	1.63	763	0.2614	HOMO \rightarrow LUMO (66%)
	S9	3.01	412	0.8450	HOMO-4 \rightarrow LUMO (52%)
	S13	3.11	398	0.2499	HOMO \rightarrow LUMO+2 (44%)
	S14	3.19	389	1.2934	HOMO \rightarrow LUMO+3 (43%)
	S27	3.72	334	0.5948	HOMO-1 \rightarrow LUMO+4 (43%)

2. ¹H NMR and ¹³C NMR spectra































3. Photovoltaic device fabrication

BHJ-OSCs were fabricated with a conventional device The structure of ITO/PEDOT:PSS/SMDs:PC71BM/Al. The indium tin oxide (ITO) coated glass substrates (10 Ω sq⁻¹) were cleaned with deionized water, methanol, acetone, toluene and iso-propyl alcohol, successively. Then poly(3,4-ethylenedioxythiophene):poly(styrene sulfonate) (PEDOT:PSS) as hole-transporting layer was spin-coated on the pre-cleaned ITO glass substrates at 4000 rpm for 60 s. Subsequently, the substrates were baked at 140 °C for 20 min in air and transferred to a N₂-filled glove box. The solutions of CTBDP or DCTBDP:PC71BM (1:2, w/w) at total concentration 12 mg mL⁻¹ in chlorobenzene were stirred overnight at room temperature in N₂-filled glove box, and then D/A blend solutions were spin-casted on the top of PEDOT:PSS layer at 800 rpm for 60 s. Next, the blend films were dried at 100 °C for 5 min so as to remove residual organic solvent, and the prepared samples were exposed or not to CH₂Cl₂ vapor atmosphere at different duration. Finally, an aluminum (Al) cathode was deposited by vacuum thermal evaporation (ca. 10^{-4} Pa) through a shadow mask, yielding eight individual OSCs with 5.0 mm² effective area. Holeonly and electron-only OSCs were fabricated with the same process, apart from gold (Au) as cathode and zinc oxide (ZnO) as electron buffer layer, respectively. As for ZnO layer fabrication, a dispersive solution of ZnO nanoparticles in iso-propyl alcohol (Aladdin reagent) was spun-cast on the top of ITO substrates at 3000 rpm for 60 s, and then were annealing at 165 °C for 1 h in air.

Donor	$J_{ m SC}$ (mA cm ⁻²)	V _{oc} (V)	FF	PCE _{max} (PCE [#] _{ave.}) (%)
CTBDP	11.33	0.91	0.29	3.02(2.59) ^a
	15.69	0.81	0.41	5.26(5.02) ^b
	17.70	0.81	0.41	5.85(5.28)°
	14.90	0.81	0.43	5.22(4.31) ^d
	12.88	0.80	0.31	3.24(2.80) ^a
DCTBDP	14.59	0.75	0.35	3.85(3.57) ^b
	13.43	0.74	0.37	3.66(3.24)°
	11.74	0.55	0.37	2.39(2.02) ^d

4. Photovoltaic data

 Table S3 PV data of the OSCs based on CTBDP and DCTBDP/PC71BM films under different conditions.

[#] The average PCE is obtained from eight devices and all active-layer films are spin-casted from chlorobenzene solution (D/A=1:2, 12 mg/mL). For active layers: a----as-casted; b-d----CH₂Cl₂ vapor treatment for 30, 45 and 60 s respectively.

5. UV-vis absorption spectra of blend films



Fig. S11 UV-vis absorption spectra of **CTBDP** and **DCTBDP**/PC₇₁BM blend films (w/w, 1:2) under different conditions.

6. Study of charge recombination

The charge recombination in **CTBDP** and **DCTBDP**-based devices were further investigated by the dependence of J_{sc} and V_{oc} with light intensity (P_{in}) (see **Fig. S12**). In general, the relationship between J_{sc} and P_{in} can be described as the power law: ²

$$J_{\rm sc} \propto (P_{\rm in})^{\alpha}$$

in which α is the power exponent and represents the degree of bimolecular recombination in OSCs. The large *a* value from $\lg J_{sc}$ versus $\lg P_{in}$ curve means that bimolecular recombination process in OSCs at short circuit condition can be strongly suppressed. As shown in **Fig. S12(a)**, less bimolecular recombination was observed for the optimized **CTBDP** and **DCTBDP**-devices with slightly large α values of 0.923 and 0.921, respectively in comparison with the as-casted counterparts.

Under the condition of only bimolecular combination, the V_{oc} of a solar cell is primarily determined by the following equation: ³

$$V_{\rm oc} = \frac{E_{\rm gap}}{q} - \frac{kT}{q} \ln(\frac{(1-P_D)\gamma N_c^2}{P_D G})$$

where E_{gap} is the energy downshift between LUMO of the acceptor material and HOMO of the donor material; *T* is Kelvin temperature; *k* is Boltzmann constant; *G* refers to the generation rate of electron-hole pairs; P_D is the dissociation probability of electron-hole pairs; γ is the Langevin recombination constant and N_c is defined as effective density of states. Among them, P_D and N_c are independent of *P* while *G* is positively related to it. In principle, if bimolecular recombination is unique loss mechanism, the slope (S) of V_{oc} versus $\ln P_{in}$ curve should be equal to unity kT/q. However, when Shockley-Read-Hall (SRH) recombination (namely trap-assisted recombination) is involved, the S value remains the range of $1kT/q \sim 2kT/q$. If S value is very close to 2kT/q, the recombination mechanism would be dominated by monomolecular recombination. As shown in **Fig. S12(b)**, the optimized OSCs especially for **CTBDP**-device exhibit the significantly suppressed trap-assisted recombination with smaller S value of 1.17kT/qamong all OSCs, and therefore displaying better photovoltaic performance.



Fig. S12 Variation of (a) J_{sc} and (b) V_{oc} with the light intensity (5.6-94 mW cm⁻²) for OSCs based on CTBDP and DCTBDP/PC₇₁BM blend films under different conditions.

7. References

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