# Electronic Supplementary Information (ESI) 

# High-entropy amorphous oxycyanide as an efficient pre-catalyst for oxygen evolution reaction 

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## 1. Experimental section

### 1.1 Synthesis of the Prussian blue analogue nanocube precursors

For the synthesis of the quinary $\boldsymbol{H E} \boldsymbol{E} \boldsymbol{P B A}$ nanocubes, namely, CoFeNiCuMn PBA, equivalent amounts of cationic metal salts (namely, $2 \mathrm{mmol} \mathrm{NiCl} 2 \cdot 6 \mathrm{H}_{2} \mathrm{O}, 2 \mathrm{mmol}$ $\mathrm{CuCl}_{2} \cdot 2 \mathrm{H}_{2} \mathrm{O}$ and $2 \mathrm{mmol} \mathrm{MnCl}_{2} \cdot 4 \mathrm{H}_{2} \mathrm{O}$ ) and 9 mmol trisodium citrate dehydrate were dissolved in 100 mL of distilled water by vigorous stirring. Meanwhile, equivalent
 $\left.\mathrm{K}_{3}\left[\mathrm{Fe}(\mathrm{CN})_{6}\right]\right)$ were dissolved in 100 mL of distilled water under stirring. After that, the abovementioned solutions were mixed and stirred vigorously until precipitates can be obtained after reacting for 30 min . The powdery product was then rinsed with distilled water and absolute ethanol for several times, and finally dried under vacuum. For the synthesis of the quaternary, ternary, binary PBAs and unary PB nanocube precursors, the metal salts were modified as following.

CoFeCuMn PBA: The cationic metal salts were modified as $3 \mathrm{mmol} \mathrm{CuCl}_{2} \cdot 2 \mathrm{H}_{2} \mathrm{O}$ and $3 \mathrm{mmol} \mathrm{MnCl} 2 \cdot 4 \mathrm{H}_{2} \mathrm{O}$, and the other parameters remain unchanged.

CoFeCu PBA: The cationic metal salts were modified as $6 \mathrm{mmol} \mathrm{CuCl}_{2} \cdot 2 \mathrm{H}_{2} \mathrm{O}$, and the other parameters remain unchanged.

CuFe PBA: The cationic metal salts were modified as $6 \mathrm{mmol} \mathrm{CuCl} 2 \cdot 2 \mathrm{H}_{2} \mathrm{O}$, and the anionic metal salts were replaced by $4 \mathrm{mmol} \mathrm{K}_{3}\left[\mathrm{Fe}(\mathrm{CN})_{6}\right]$. The other experimental parameters remain unchanged.

FeFe PB: The cationic metal salts were modified as $6 \mathrm{mmol} \mathrm{FeCl} 2 \cdot 4 \mathrm{H}_{2} \mathrm{O}$, and the anionic metal salts were replaced by $4 \mathrm{mmol}_{3}\left[\mathrm{Fe}(\mathrm{CN})_{6}\right]$. The other parameters remain unchanged.

### 1.2 Synthesis of the amorphous oxycyanide pre-catalysts

For the synthesis of high-entropy amorphous HE-CNO, 100 mg HE-PBA precursor was placed in a porcelain boat and heated to $200^{\circ} \mathrm{C}$ in air in a tube furnace at a heating rate of $10{ }^{\circ} \mathrm{C} \mathrm{min}^{-1}$, and maintained for 2 h . After that, the porcelain boat was cooled, and the powdery product was collected. The synthesis procedures of the
counterparts, namely, $\mathrm{CoFeCuMn}-\mathrm{CNO}, \mathrm{CoFeCu}-\mathrm{CNO}, \mathrm{CuFe}-\mathrm{CNO}$ and $\mathrm{Fe}-\mathrm{CNO}$, were the same as the fabrication of HE-CNO, where the precursors were replaced by the quaternary, ternary, binary PBAs and unary PB nanocube precursors, respectively.

### 1.3 Structural characterizations

The X-ray diffraction (XRD) was performed on a Philips X'Pert Pro Super diffractometer with $\mathrm{Cu} \mathrm{K} \alpha$ radiation $(\lambda=1.54178 \AA$ ). The Fourier transform infrared (FT-IR) spectra were measured on MAGNA-IR 750 (Nicolet Instrument). The scanning electron microscopy (SEM) images were taken on a JEOL JSM-6700F SEM. The transmission electron microscopy (TEM) was carried out on a JEM-2100F field emission electron microscope at an acceleration voltage of 200 kV . The highresolution TEM (HRTEM), high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) and corresponding elemental mapping analyses were performed on a Thermo Fischer Talos F200X TEM. The nitrogen adsorption-desorption isotherms were carried out by using a Micromeritics ASAP 2460 system, and all the gas adsorption experiments were performed at liquidnitrogen temperature $(77 \mathrm{~K})$ after degassed at $300^{\circ} \mathrm{C}$ for 6 h . The atomic ratio of the metals was determined by inductively coupled plasma optical emission spectrum (ICP-OES) on a Perkin Elmer Optima 7300DV ICP emission spectroscope. The X-ray photoelectron spectroscopy (XPS) analyses were performed on a VGESCALAB MKII X-ray photoelectron spectrometer with an excitation source of $\mathrm{Mg} \mathrm{K} \alpha=1253.6$ eV , and the resolution level was lower than 1 atom\%.

### 1.4 Electrocatalytic study

All the electrochemical measurements were performed in a three-electrode system linked with an electrochemical workstation (Ivium Vertex. C. EIS). All potentials were calibrated to a reversible hydrogen electrode (RHE) according to the Nernst equation and the data were presented without $i \mathrm{R}$ correction. Typically, 4 mg of catalyst and $50 \mu \mathrm{~L}$ Nafion solution (Sigma Aldrich, $5 \mathrm{wt} \%$ ) were dispersed in 1 mL water-isopropanol mixed solution (volume ratio of $3: 1$ ) by sonicating for at least 30
min to form a homogeneous ink. Then $5 \mu \mathrm{~L}$ of the dispersion (containing $20 \mu \mathrm{~g}$ of catalyst) was loaded onto a glassy carbon electrode with 3 mm diameter, resulting in a catalyst loading of $0.285 \mathrm{mg} \mathrm{cm}^{-2}$. The as-prepared catalyst film was allowed to be dried at room temperature. The cyclic voltammetry (CV) and linear sweep voltammetry (LSV) with a scan rate of $2 \mathrm{mV} \mathrm{s}^{-1}$ were conducted in $\mathrm{O}_{2}$-purged 1 M KOH solution. $\mathrm{A} \mathrm{Hg} / \mathrm{HgO}$ electrode was used as the reference electrode, a platinum gauze electrode ( $2 \mathrm{~cm} \times 2 \mathrm{~cm}, 60 \mathrm{mesh}$ ) was used as the counter electrode, and the glassy carbon electrodes loaded with various catalysts were served as the working electrodes. The Faradaic efficiency (FE) was measured at a constant anodic current of 20 mA . The electrochemical impedance spectroscopy (EIS) measurements were operated in the same configuration at 1.6 V vs. RHE from $10^{-2}-10^{5} \mathrm{~Hz}$. The EIS data were fitted according to the following equivalent circuit, where $R_{s}, R_{c t}$ and $R_{\text {int }}$ are the series resistance, charge-transfer resistance, and solid-electrolyte interface resistance, respectively; $\mathrm{CPE}_{1}$ and $\mathrm{CPE}_{2}$ correspond to the capacitances derived from the solidelectrolyte interface process and the Faradic process.


## 2. Additional physical and electrochemical characterizations



Fig. S1 (A) SEM and (B) TEM images of HE-PBA nanocube precursor.


Fig. S2 XPS spectra of the HE-PBA nanocube precursor. (A) Survey spectra. (B-I) XPS spectra of $\mathrm{Co}, \mathrm{Fe}, \mathrm{Ni}, \mathrm{Cu}, \mathrm{Mn}, \mathrm{C}, \mathrm{N}$ and O , respectively.


Fig. S3 FT-IR spectra of HE-PBA precursor and HE-CNO.

The structural characteristics of HE-PBA and HE-CNO were investigated by Fourier transform infrared spectroscopy (FT-IR). As shown in Fig. S3, the peaks centered at $1600 \mathrm{~cm}^{-1}$ and $3407 \mathrm{~cm}^{-1}$ can be attributed to the tensile and in-plane deformation of -OH , indicating the presence of water molecules. ${ }^{1,2}$ In the case of the HE-PBA, the absorption peaks of -OH can be assigned to the tensile vibration of water molecules bound to the metal ions, corresponding to the coordinated water in PBA lattice. ${ }^{3,4}$ The high-intensity band at $2093 \mathrm{~cm}^{-1}$ can be assigned to the $\mathrm{C} \equiv \mathrm{N}$ stretching vibration, while such signal significantly declined for the HE-CNO, indicating that the oxidation process could effectively break the $\mathrm{C} \equiv \mathrm{N}$ bonds. ${ }^{1,2}$ Besides, the peaks at $470 \mathrm{~cm}^{-1}$ corresponding to C-M bonds also show obvious decrement after the oxidation process, while in contrast, the peaks located at $597 \mathrm{~cm}^{-1}$ and $1349 \mathrm{~cm}^{-1}$ corresponding to $\mathrm{O}-\mathrm{M}$ bonds and C-O tensile deformation increase emerge along with the oxidation, suggesting the formation of oxide species during the thermal oxidation progress. ${ }^{5-8}$


Fig. S4 (A) SEM and (B) TEM images of the quaternary CoFeCuMn-CNO.


Fig. S5 (A) SEM and (B) TEM images of the ternary CoFeCu-CNO.


Fig. S6 (A) SEM and (B) TEM images of the binary CuFe-CNO.


Fig. S7 (A) SEM and (B) TEM images of the unary Fe-CNO.


Fig. S8 (A) Nitrogen adsorption-desorption isotherm and the calculated BET specific surface area of HE-CNO. (B) Pore size distribution.

Table S1 Comparison of the OER performance. All the data were obtained in 1 M KOH electrolyte, and the catalysts were in powdery form without loading on porous and conductive supports.

|  | $\begin{gathered} \boldsymbol{\eta}_{10} \\ {[\mathrm{mV}]} \end{gathered}$ | $\begin{gathered} \mathbf{j g e o} @ \eta_{500 \mathrm{mv}} \mathrm{v} \\ {\left[\mathrm{~mA} \mathrm{~cm}^{-2}\right]} \end{gathered}$ | $\mathbf{j}_{\mathbf{C d}} @ \mathbf{\eta}_{400 \mathrm{mV}}$ $\left[\mathbf{A} \mathbf{F}^{-1}\right]$ | Stability | Ref. |
| :---: | :---: | :---: | :---: | :---: | :---: |
| HE-PBA | - | 2.1 | 18.2 | - | this work |
| HE-CNO | 328 | 137.5 | 283.6 | 1220\%@72 h | this work |
| HE-CNO-pc | 278 | 216.8 | 556.8 | - | this work |
| CoFeCuMn-CNO | 366 | 63.7 | 145.0 | - | this work |
| $\mathrm{CoFeCu}-\mathrm{CNO}$ | 388 | 28.6 | 114.2 | - | this work |
| $\mathrm{CuFe}-\mathrm{CNO}$ | 503 | 9.8 | 17.8 | - | this work |
| Fe-CNO | - | 5.5 | 21.5 | - | this work |
| $\left(\mathrm{Ni}_{0.62} \mathrm{Fe}_{0.38}\right)_{2} \mathrm{P}$ hollow nanocube derived from PBA | 290 | - | - | 80.5\%@10 h | 9 |
| Ni-Co oxide nanocage derived from PBA | 380 | - | $\sim 0.3$ | 100\%@10 h | 10 |
| FeCoNi-P derived from PBA | 265 | - | $\sim 14.8$ | - | 11 |
| Dodecagon N-doped Pd-$\mathrm{CoNi}-\mathrm{C}$ nanosheet derived from PBA | 309 | 104.0 | $\sim 0.3$ | - | 12 |
| $\mathrm{CoFe}_{2} \mathrm{O}_{4}$ nanosheet derived from PBA | 275 | 88.9 | 1.3 | 87.5\%@10 h | 13 |
| CoNiFe PBA ( $\mathrm{Co} / \mathrm{Ni}=1: 3$ ) | 320 | - | 21.7 | - | 14 |
| CoFeZn PBA | 343 | 92.6 | 0.5 | 82.6\%@12 h | 15 |
| Defective Co PBA | 400 | 70.0 | 0.08 | 72\%@10000 s | 16 |
| $\mathrm{Co}_{3} \mathrm{~S}_{4} @ \mathrm{MoS}_{2}$ hollow heterostructure derived from PBA | 280 | - | - | $\begin{aligned} & 87.5 \% @ 2000 \\ & \text { cycles } \\ & / 90.2 \% @ 10 \mathrm{~h} \end{aligned}$ | 17 |
| $\mathrm{Fe}_{0.5} \mathrm{Co}_{0.5} \mathrm{MoO}_{4-\mathrm{x}} \mathrm{S}_{\mathrm{x}}$ nanoflower derived from PBA | 263 | - | - | $\begin{aligned} & \text { 97.5\%@1000 } \\ & \text { cycles } \\ & \text { / 87\%@72 h } \end{aligned}$ | 18 |
| $\mathrm{Co} / \mathrm{CoTe}$ microcube derived from PBA | 337 | 231.2 | 2.0 | $\begin{aligned} & 94.8 \% 1000 \\ & \text { cycles } \end{aligned}$ | 19 |
| Pt-decorated NiFe PBA nanocube | 333 | 80.5 | 11.0 | $\begin{aligned} & \text { 99.6\%@1000 } \\ & \text { cycles } \\ & / 85.6 \% @ 12 \text { h } \end{aligned}$ | 20 |
| CoCo PBA nanocube | 329 | 126.6 | 270.0 | 499.1\%@60 h | 21 |
| CoFe PBA nanocube | 463 | 19.6 | 22.0 | 205\%@60 h | 21 |


| Fe PB nanocube | 528 | 7.8 | 19.0 | 151\%@60 h | 21 |
| :---: | :---: | :---: | :---: | :---: | :---: |
| NiFe PBA nanocube | 587 | 3.2 | 8.0 | 114\%@60 h | 21 |
| Ni-P porous nanoplate/C derived from PBA | 300 | - | - | $\begin{gathered} 74.1 \% @ 1000 \\ \text { cycles } \\ / \sim 102.1 \% @ 10 \\ \text { h } \end{gathered}$ | 22 |
| $\mathrm{CoSe}_{2}$ nanobox derived from PBA | 335 | - | 2.0 | $\begin{aligned} & \text { 94.4\%@1000 } \\ & \text { cycles } \\ & / \sim 91 \% @ 4 \text { h } \end{aligned}$ | 23 |
| Fe-doped CoP core-shell nanocage derived from PBA | 300 | - | - | $\begin{gathered} 101.9 \% @ 1000 \\ \text { cycles } \\ / \sim 37.8 \% @ 12 \text { h } \end{gathered}$ | 24 |
| $\mathrm{CoFe} / \mathrm{CoFeO}_{\mathrm{x}}$ derived from PBA embedded in $\mathrm{N}-\mathrm{C}$ | 330 | - | 4.5 | $\begin{gathered} 92.5 \% @ 2000 \\ \text { cycles } \\ \text { / 59.3\%@20 h } \\ 97.2 \% @ 5000 \end{gathered}$ | 25 |
| CoFe PBA hollow cage | 330 | - | 0.2 | cycles /~82.7\%@12h | 26 |
| Partially phosphated Fedoped CoP derived from PBA | 262 | 79.4 | 1.8 | $\begin{aligned} & 95.0 \% @ 3000 \\ & \text { cycles } \\ & / 96.0 \% @ 18 \mathrm{~h} \end{aligned}$ | 27 |
| CoFe oxide nanoframe derived from PBA | 290 | - | - | $\begin{gathered} 94.8 \% @ 500 \\ \text { cycles } \end{gathered}$ | 28 |
| MnFe PBA@ $\mathrm{H}_{3} \mathrm{PMo}_{12} \mathrm{O}_{40}$ hollow nanocube | 440 | - | 0.3 | - | 29 |
| $\mathrm{RuO}_{2}$ | 340 | 52.0 | - | - | 30 |
| $\mathrm{IrO}_{2}$ | 320 | 50.0 | 2.9 | $\begin{aligned} & 72.9 \% @ 1000 \\ & \text { cycles } \end{aligned}$ | 31 |

Table S2 The fitted $\mathrm{R}_{\mathrm{s}}$ and $\mathrm{R}_{\mathrm{ct}}$ values of the catalysts.

|  | $\mathbf{R}_{\mathbf{s}}[\mathbf{\Omega}]$ | $\mathbf{R}_{\mathbf{c t}}[\mathbf{\Omega}]$ |
| :--- | :---: | :---: |
| HE-PBA | 7.8 | 1190.0 |
| HE-CNO | 0.7 | 24.8 |
| CoFeCuMn-CNO | 2.1 | 41.1 |
| CoFeCu-CNO | 0.5 | 277.9 |
| CuFe-CNO | 6.9 | 826.1 |
| Fe-CNO | 0.8 | 1110.0 |



Fig. S9 CV curves in the non-redox region for the estimation of electrochemical double-layer capacitance $\left(\mathrm{C}_{\mathrm{dl}}\right)$.

The estimation of the effective active surface area of the samples was carried out according to literature. ${ }^{32,33}$ Cyclic voltammetry (CV) was performed in $\mathrm{O}_{2}$-degassed 1 M KOH solution at various scan rates $\left(20,40,60,80\right.$ and $\left.100 \mathrm{mV} \mathrm{s}^{-1}\right)$ in the range of $0.8-0.9 \mathrm{~V}$ vs. RHE (Fig. S9). The $\mathrm{C}_{\mathrm{dl}}$ values can be determined by plotting the $\Delta \mathrm{j}$ ( $\mathrm{janode} \mathrm{j}_{\text {cathode }}$ ) at 0.85 V vs. RHE against the scan rate (Fig. 3D), where the slope is twice $\mathrm{C}_{\mathrm{d} 1}$. The $\mathrm{C}_{\mathrm{dl}}$ values are expected to be linearly proportional to the effective surface area.


Fig. S10 The calculated Faradaic efficiency of HE-CNO and HE-CNO-pc.


Fig. S11 Comparison of the electrochemical parameters of HE-CNO before and after stability test. (A) LSV curves. (B) $\mathrm{C}_{\mathrm{d} 1}$ values. (C) $\mathrm{C}_{\mathrm{d} 1}$-normalized activity.


Fig. S12 FT-IR spectra of HE-CNO before and after stability test.

The FT-IR spectra were conducted to study the structural characteristics of HECNO before and after the stability test. As shown in Fig. S12, the peaks centered at $1600 \mathrm{~cm}^{-1}$ and $3401 \mathrm{~cm}^{-1}$ can be attributed to the tensile and in-plane deformation of OH. ${ }^{1,2}$ In the case of HE-CNO, the band at $2086 \mathrm{~cm}^{-1}$ can be assigned to the $\mathrm{C} \equiv \mathrm{N}$ stretching, while the peak vanished after the stability test, indicating the complete oxidation of the cyan group during the pre-oxidation process. ${ }^{3,4}$ Besides, the intensity of peak at $576 \mathrm{~cm}^{-1}$ increases after the stability test, indicating the enhanced O-M bonding owing to the formation of the catalytically active metal (oxy)hydroxides. ${ }^{6}$ In addition, the emerging peak at $1230 \mathrm{~cm}^{-1}$ can be assigned to the C-C bonds owing to the addition of Nafion solution in the electrochemical test. ${ }^{34}$

activated catalyst
Fig. S13 Possible OER mechanism on the surface of the activated HE-CNO. The activated HE-CNO could offer more catalytically active high-valence sites for OER owing to the efficient pre-oxidation process, which is responsible for the substantially increased OER activity during the long-term operation.

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