Supporting Information

Enhanced photoluminescence stability and internal defect evolution

of the all-inorganic lead-free CsEuCl₃ perovskite nanocrystals

1. PL stability of the uncoated CsEuCl₃ nanocrystals

The gradually enhanced emissions were observed during the first 5 days. The emission decreased after 5 days, and the remnant PL reduced to about 20 % of the initial value at the 10th day. Moreover, a rapidly decomposed occurred while the pure CsEuCl₃ nanocrystal exposed to ambient atmosphere.



Figure S1 (a) The fluorescence varies with time, (d) the residual blue emission intensity, and (c) the residual red emission intensity of the uncoated CsEuCl₃ nanocrystals.

2. XRD characterization of the unmodified CsEuCl₃ perovskite nanocrystals

The CsEuCl₃ nanocrystals thin film was spin-coated on a quartz substrate at 3000 rpm for 60s in the glove box, and then the sample was sealed in a centrifuge tube. The XRD characterization and digital images of the luminescence were performed immediately when the sample removed from the glove box. Fig.S2a shows the obtained XRD pattern of the sample, the result revealing that the sample was suffer from destruction. Fig.S2b show the luminescence images of the sample exposed in environment. Apparently, the luminescence is extremely unstable and sensitive to the environment. Thus, it can be predicted that the CsEuCl₃ nanocrystals is very hard to perform the further physical characteristic analysis and practical application under nonspecial conditions.



Fig. S2 (a) The XRD pattern and (b) digital luminescence images of the unmodified sample.

3. Size distribution



Fig. S3 The size distribution of CsEuCl₃ nanocrystals.



4. Residual red emission intensity of the silica-coated CsEuCl₃ nanocrystals

Fig. S4 The residual red emission intensity of the silica-coated CsEuCl₃ nanocrystals.

5. Electronic band structure calculation models

The cutoff energy for the plane-wave basis set was set to 500 eV. The DFT+U method was applied using 2.5 eV Hubbard U value for the *d* orbital of Eu. A $3\times3\times3$ CsEuCl₃ supercell was used, and the vacancy-CsEuCl₃ was simulated by removing two CsEuCl₃ units form the same supercell. The geometry optimizations were performed until the forces on each ion was reduced below 0.01 eV/Å, and the $2\times2\times2$ K-point sampling of the Brillouin zone was used.



Fig. S5 Structural derivation of the V_{Eu} and "hollow" lattices from the proper 3D perovskite.



6. XRD characterization of the silica-coated CsEuCl₃ nanocrystals

Fig. S6 XRD pattern of silica-coated CsEuCl₃ nanocrystals after stored 80 days

7. Calculated energy band information

Spin Channel		Up	Down	Total
Model 1	Band gap	2.9677	5.2310	2.9677
(V _{Eu})	Eigenvalue of VBM	1.0971	-1.1074	1.0971
	Eigenvalue of CBM	4.0648	4.1236	4.0648
	Fermi Energy	1.3186	1.3186	1.3186
	HOMO Band	27	20	27
	LUMO Band	28	21	28
Model 2	Band gap	3.0734	5.1066	3.0734
(Hollow)	Eigenvalue of VBM	0.8767	-1.0918	0.8767
	Eigenvalue of CBM	3.9500	4.0149	3.9500
	Fermi Energy	1.1121	1.1121	1.1121
	HOMO Band	27	20	27
	LUMO Band	28	21	28

 Table S1 Calculated energy band information of the CsEuCl₃ supercell.