

## Electronic Supplementary Information (ESI)

### Band Gap Modulation and Nonlinear Optical Properties of Quaternary Tellurates $\text{Li}_2\text{GeTeO}_6$

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## Experimental section

**Synthesis.** All chemicals were used as received.  $\text{Li}_2\text{CO}_3$  ( $\geq 99.0\%$ , aladdin),  $\text{GeO}_2$  (99.99%, Hawk), and  $\text{TeO}_2$  (99.99%, Hawk) were all used as raw materials. Single crystals of  $\text{Li}_2\text{GeTeO}_6$  were obtained by using the high-temperature flux method. The mixture of  $\text{Li}_2\text{CO}_3$ ,  $\text{GeO}_2$ , and  $\text{TeO}_2$  in a molar ratio of 2.5 : 1 : 4 was fully grounded and placed in an alumina crucible. The crucible was heated to 720 °C in 3 h, and held at this temperature for 12 h. Afterward, the solution was cooled slowly to 500 °C with a rate of 2 °C  $\text{h}^{-1}$ , and then cooled the room temperature quickly. Colorless crystals of  $\text{Li}_2\text{GeTeO}_6$  were obtained from the surface of the crucible (yield: about 70% based on Ge).

The polycrystalline powder samples of  $\text{Li}_2\text{GeTeO}_6$  were prepared via the solid-state reaction method in air. A mixture of  $\text{Li}_2\text{CO}_3$ ,  $\text{TeO}_2$ , and  $\text{GeO}_2$  with a stoichiometric ratio was fully ground and loaded into a corundum crucible, heated to 700 °C at a rate of 3 °C  $\text{min}^{-1}$ , held at this temperature for 48 hours, then heat to 750 °C for 12 h with several intermittent grindings. The chemical reaction process can be expressed as follows:  $\text{Li}_2\text{CO}_3 + \text{GeO}_2 + \text{TeO}_2 + 1/2 \text{O}_2 = \text{Li}_2\text{GeTeO}_6 + \text{CO}_2$ .

**Characterization.** The single-crystal X-ray diffraction (XRD) data were collected on a Bruker D8 VENTURE dual-wavelength Mo/Cu three-circle diffractometer with a microfocus sealed X-ray tube using mirror optics as monochromator and a Bruker PHOTON III detector. The diffractometer used  $\text{MoK}\alpha$  radiation ( $\lambda = 0.71073 \text{ \AA}$ ) for data collection. All data were integrated with SAINT program<sup>1</sup> and a multi-scan absorption correction using SADABS<sup>2</sup> was applied. The structure was solved by Intrinsic Phasing methods using SHELXT<sup>3</sup> and refined by full-matrix least-squares methods against  $F^2$  by SHELXL<sup>4</sup>, embedding within the *Olex2* program.<sup>5</sup> Potential higher symmetry was checked by PLATON.<sup>6</sup> Crystal data and refinement details, atomic coordinates, and selected bond distances and angles are listed in Tables S1-S3. Powder XRD data were measured on a Dandong Haoyuan DX-27mini X-ray diffractometer with  $\text{Cu K}\alpha$  radiation ( $\lambda = 1.54056 \text{ \AA}$ ). The powder XRD pattern was scanned over the  $2\theta$  angles range of 10-70°, at a scanning step width of 0.02° and a

fixed counting time of 2 s. The thermal gravimetric (TG) analysis and differential scanning calorimetry (DSC) were studied with a NETZSCH5 instrument under air. The sample was placed in an Al<sub>2</sub>O<sub>3</sub> crucible and heated from 30 to 1000 °C with a heating rate of 10 °C·min<sup>-1</sup>. The infrared spectra were recorded on a Shimadzu IR Affinity-1 Fourier transform infrared spectrometer in the range of 400 - 4000 cm<sup>-1</sup>. The sample was grounded and mixed with KBr. The UV-Vis-NIR diffuse reflectance spectra were measured at room temperature with a Shimadzu SolidSpec-3600DUV spectrophotometer in the 200-1100 nm wavelength range. The SHG effects of the powder samples were measured by using a modified Kurtz-Perry method<sup>7</sup> with a Q-switched 1064 nm Nd: YAG laser. The samples were ground and divided into fractions according to different particle size ranges: 20-38.5, 38.5-55, 55-80, 80-125, 125-160, 160-200, and 200-250 μm. The same particle-sized KDP samples were used as the standard. The optical damage induced by the laser was tested on crystalline powder pellet samples using a Q-switched 1064 nm Nd: YAG laser (1064 nm, 7 ns, 1 Hz). The pulse energy was changed from 20 to 100 mJ and a convex lens was used to adjust the beam diameter. The damaged area can be directly observed under a microscope.

**Theoretical Calculations.** The electronic structures and optical properties for Li<sub>2</sub>GeTeO<sub>6</sub> were calculated by using the CASTEP package.<sup>8</sup> The generalized gradient-approximation (GGA) with Perdew-Burke-Ernzerhof (PBE) functional was adopted to describe the exchange-correlation potential.<sup>9</sup> The norm-conserving pseudopotentials (NCP) were adopted to model the effective interaction between the valence electrons and atom cores.<sup>10</sup> The configurations for diverse electron orbital were Li: 2s<sup>2</sup>, Ge: 4s<sup>2</sup>4p<sup>2</sup>, Te: 5s<sup>2</sup>5p<sup>4</sup>, and O: 2s<sup>2</sup>2p<sup>4</sup>, respectively. The cut off energies and the dense *K*-points in the Brillouin zone were set as 750 eV and 3 × 3 × 2. In addition, a scissors operator was used to correct the underestimation of band gap induced by PBE-GGA. On the basis of the scissor-corrected electron band structure, the second-order NLO coefficients  $d_{ij}$  were calculated using the length-gauge formalism at a zero-frequency limit.<sup>11</sup> The static second-order nonlinear susceptibilities can be ascribed to Virtual-Hole (VH) and Virtual-Electron (VE) processes. In addition, to visualize the

contribution of each atom to the SHG response, the SHG-weighted electron density method was adopted.<sup>12, 13</sup>

**Table S1.** Crystal data and structure refinements for  $\text{Li}_2\text{GeTeO}_6$ .

Compound	$\text{Li}_2\text{GeTeO}_6$
Formula weight	310.07
Temperature (K)	273(2)
Crystal system	Trigonal
Space group (number)	$R\bar{3}$ (No. 146)
$a$ (Å)	4.996(3)
$b$ (Å)	4.996(3)
$c$ (Å)	14.335(12)
Volume (Å <sup>3</sup> )	309.8(5)
$Z$	3
$\rho_{\text{calc}}$ (g·cm <sup>-3</sup> )	4.985
$\mu$ (mm <sup>-1</sup> )	14.272
$F(000)$	414
$2\Theta$ range (deg.)	8.53 to 56.61
Limiting indices	$-6 \leq h \leq 6; -6 \leq k \leq 6; -18 \leq l \leq 18$
Reflections collected / unique	1414
Independent reflections	353 ( $R_{\text{int}} = 0.0733, R_{\text{sigma}} = 0.0629$ )
Completeness to $\Theta = 25.242^\circ$	100.0 %
Data / restraints / parameters	353/7/32
Goodness-of-fit on $F^2$	1.085
$R_1, wR_2 [F_o^2 > 2\sigma(F_o^2)]^a$	$R_1 = 0.0323; wR_2 = 0.0575$
$R_1, wR_2$ (all data) <sup>a</sup>	$R_1 = 0.0424; wR_2 = 0.0598$
Largest diff. peak and hole (e·Å <sup>-3</sup> )	1.94/-1.27
Flack parameter	0.23(6)

<sup>a</sup>  $R_1 = \sum ||F_o| - |F_c|| / \sum |F_o|$  and  $wR_2 = [\sum w(F_o^2 - F_c^2)^2 / \sum w F_o^4]^{1/2}$  for  $F_o^2 > 2\sigma(F_o^2)$ .

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**Table S2.** Atomic coordinates and equivalent isotropic displacement parameters for  $\text{Li}_2\text{GeTeO}_6$ .  $U(\text{eq})$  is defined as one-third of the trace of the orthogonalized  $U_{ij}$  tensor.

<b>atom</b>	<b><i>x</i></b>	<b><i>y</i></b>	<b><i>z</i></b>	<b>U(eq)</b>	<b>BVS</b>
Li(1)	0.666667	0.333333	0.635(4)	0.02(2)	0.906
Li(2)	0.333333	0.666667	0.699(4)	0.020(15)	0.909
Ge(1)	0.000000	0.000000	0.50701(15)	0.0071(7)	3.910
Te(1)	0.333333	0.666667	0.49362(15)	0.0110(5)	6.004
O(1)	0.298(5)	0.340(5)	0.575(3)	0.013(6)	1.915
O(2)	0.678(5)	0.708(4)	0.425(2)	0.005(5)	1.995

**Table S3.** Bond lengths [Å] and angles [degree] for Li<sub>2</sub>GeTeO<sub>6</sub>.

Atom–Atom	Length [Å]	Atom–Atom–Atom	Angle [°]
Li1–O1	2.05(4)	O1–Li1–O1	103(2)
Li1–O1 <sup>#1</sup>	2.05(4)	O1–Li1–O1	103(2)
Li1–O1 <sup>#2</sup>	2.05(4)	O1–Li1–O1	103(2)
Li1–O2 <sup>#3</sup>	2.35(5)	O1–Li1–O2	156(3)
Li1–O2 <sup>#4</sup>	2.35(5)	O1–Li1–O2	89.3(8)
Li1–O2 <sup>#5</sup>	2.35(5)	O1–Li1–O2	93.4(11)
Li2–O2 <sup>#5</sup>	2.03(3)	O1–Li1–O2	89.3(8)
Li2–O2 <sup>#9</sup>	2.03(3)	O1–Li1–O2	93.4(11)
Li2–O2 <sup>#10</sup>	2.03(3)	O1–Li1–O2	156(3)
Li2–O1	2.37(5)	O2–Li1–O2	69.2(16)
Li2–O1 <sup>#11</sup>	2.37(5)	O1–Li1–O2	93.4(11)
Li2–O1 <sup>#12</sup>	2.37(5)	O1–Li1–O2	156(3)
Ge1–O1 <sup>#13</sup>	1.87(3)	O1–Li1–O2	89.3(8)
Ge1–O1 <sup>#14</sup>	1.87(3)	O2–Li1–O2	69.2(16)
Ge1–O1	1.87(3)	O2–Li1–O2	69.2(16)
Ge1–O2 <sup>#15</sup>	1.93(2)	O2–Li2–O2	103.7(19)
Ge1–O2 <sup>#11</sup>	1.93(2)	O2–Li2–O2	103.7(19)
Ge1–O2 <sup>#1</sup>	1.93(2)	O2–Li2–O2	103.7(19)
Te1–O2 <sup>#11</sup>	1.90(3)	O2–Li2–O1	93.2(10)
Te1–O2 <sup>#12</sup>	1.90(3)	O2–Li2–O1	89.0(8)
Te1–O2	1.90(3)	O2–Li2–O1	156(3)
Te1–O1 <sup>#12</sup>	1.94(2)	O2–Li2–O1	156(3)
Te1–O1 <sup>#11</sup>	1.94(2)	O2–Li2–O1	93.2(10)
Te1–O1	1.94(2)	O2–Li2–O1	89.0(8)
		O1–Li2–O1	69.2(17)
		O2–Li2–O1	89.0(8)
		O2–Li2–O1	156(3)
		O2–Li2–O1	93.2(10)
		O1–Li2–O1	69.2(17)
		O1–Li2–O1	69.2(17)
		O1–Ge1–O1	95.7(12)
		O1–Ge1–O1	95.7(12)
		O1–Ge1–O1	95.7(12)
		O1–Ge1–O2	95.5(4)
		O1–Ge1–O2	82.1(6)
		O1–Ge1–O2	168.8(9)
		O1–Ge1–O2	168.8(9)
		O1–Ge1–O2	95.5(4)

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O1–Ge1–O2	82.1(6)
O2–Ge1–O2	87.1(15)
O1–Ge1–O2	82.1(6)
O1–Ge1–O2	168.8(9)
O1–Ge1–O2	95.5(4)
O2–Ge1–O2	87.1(15)
O2–Ge1–O2	87.1(15)
O2–Te1–O2	95.8(11)
O2–Te1–O2	95.8(11)
O2–Te1–O2	95.8(11)
O2–Te1–O1	168.6(9)
O2–Te1–O1	95.5(4)
O2–Te1–O1	81.4(9)
O2–Te1–O1	95.5(4)
O2–Te1–O1	81.4(9)
O2–Te1–O1	168.6(9)
O1–Te1–O1	87.8(15)
O2–Te1–O1	81.4(9)
O2–Te1–O1	168.6(9)
O2–Te1–O1	95.5(4)
O1–Te1–O1	87.8(15)
O1–Te1–O1	87.8(15)

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Symmetry transformations used to generate equivalent atoms:

#1: 1-Y, +X-Y, +Z; #2: 1-X+Y, 1-X, +Z; #3: 1.66667-Y, 0.33333+X-Y, 0.33333+Z; #4: -0.33333+X, -0.66667+Y, 0.33333+Z; #5: 0.66667-X+Y, 1.33333-X, 0.33333+Z; #6: 0.66667+X, 0.33333+Y, 0.33333+Z; #7: 1+X, +Y, +Z; #8: +X, -1+Y, +Z; #9: 0.66667-Y, 0.33333+X-Y, 0.33333+Z; #10: -0.33333+X, 0.33333+Y, 0.33333+Z; #11: -X+Y, 1-X, +Z; #12: 1-Y, 1+X-Y, +Z; #13: -1+X, +Y, +Z; #14: +X, 1+Y, +Z; #15: -X+Y, -X, +Z; #16: -Y, +X-Y, +Z; #17: -1+X, -1+Y, +Z; #18: -0.66667+X, -0.33333+Y, -0.33333+Z; #19: 1+X, 1+Y, +Z; #20: 0.33333+X, -0.33333+Y, -0.33333+Z; #21: 0.33333+X, 0.66667+Y, -0.33333+Z;

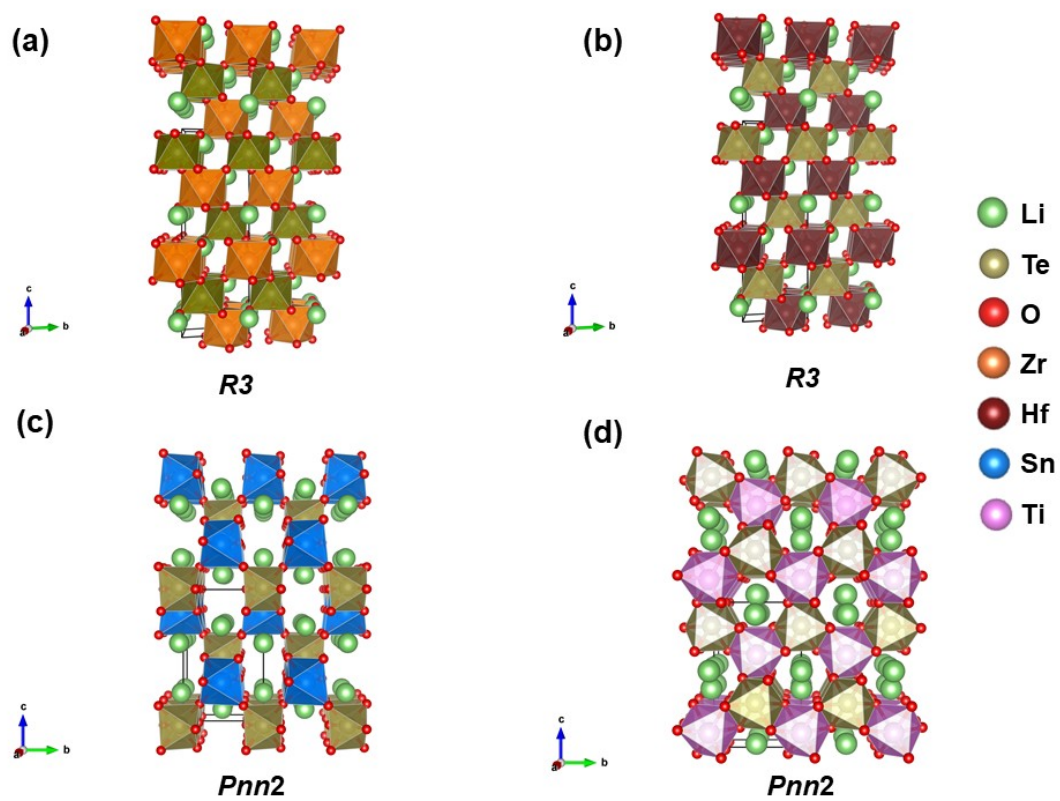


**Table S4.** Comparison of structure and optical properties of NLO active-quaternary tellurates.

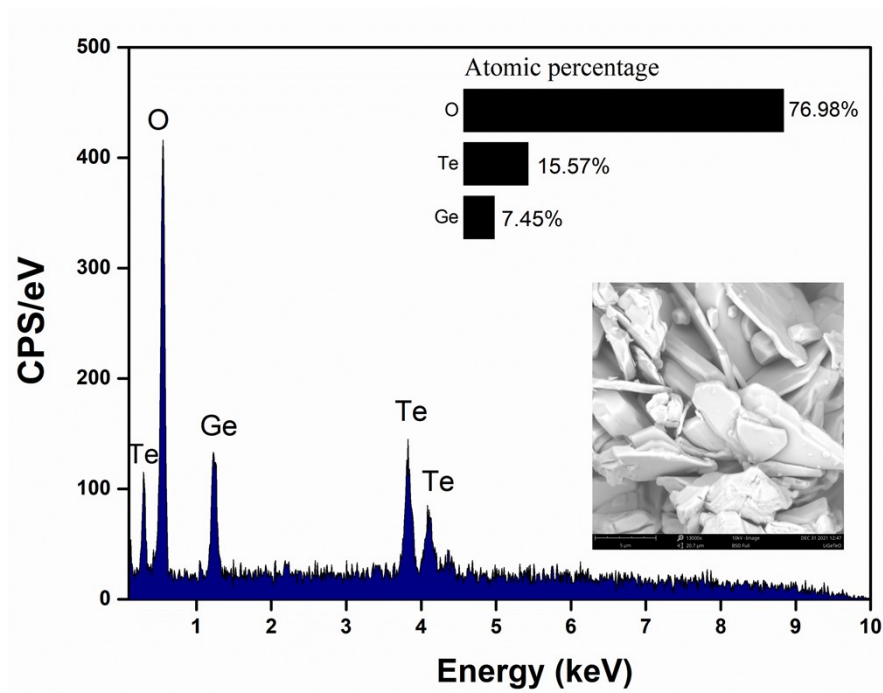
compounds	space group	octahedral distortion $\Delta d$		band gap (eV)	SHG ( $\times$ KDP)	transparent region ( $\mu\text{m}$ )	LDT ( $\text{MW cm}^{-2}$ )
		$\text{MO}_6$	$\text{TeO}_6$				
$\text{Li}_2\text{TiTeO}_6$	<i>Pnn2</i>	0.20	0.06	3.26	26	0.38-6.72	550
$\text{Li}_2\text{ZrTeO}_6$	<i>R3</i>	0.12	0.02	4.06	2.5	0.29-7.4	1300
$\alpha\text{-Li}_2\text{HfTeO}_6$	<i>R3</i>	0.19	0.06	3.98	2.2	0.27-8 <sup>#</sup>	1000
$\text{Li}_2\text{SnTeO}_6$	<i>Pnn2</i>	0.01	0.09	3.26	2.5	0.38-6.86	672
$\text{Li}_2\text{GeTeO}_6^*$	<i>R3</i>	0.16	0.13	4.65	1.3	0.24-10 <sup>#</sup>	1725

\*This work

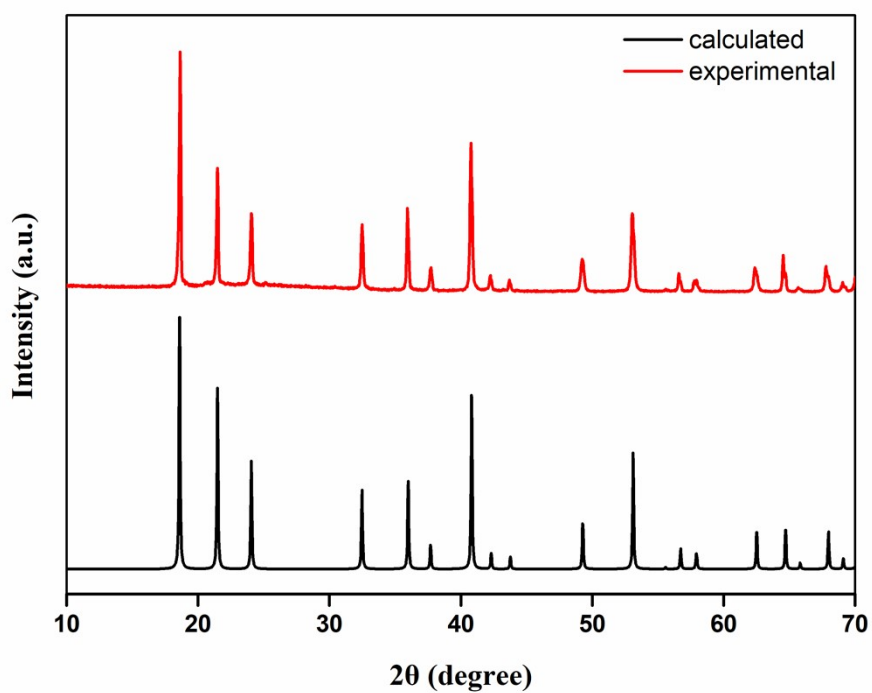
<sup>#</sup>Based on polycrystalline samples. It should be noted that the IR cut-off edges is deduced from powder IR spectrum which were not highly precise.



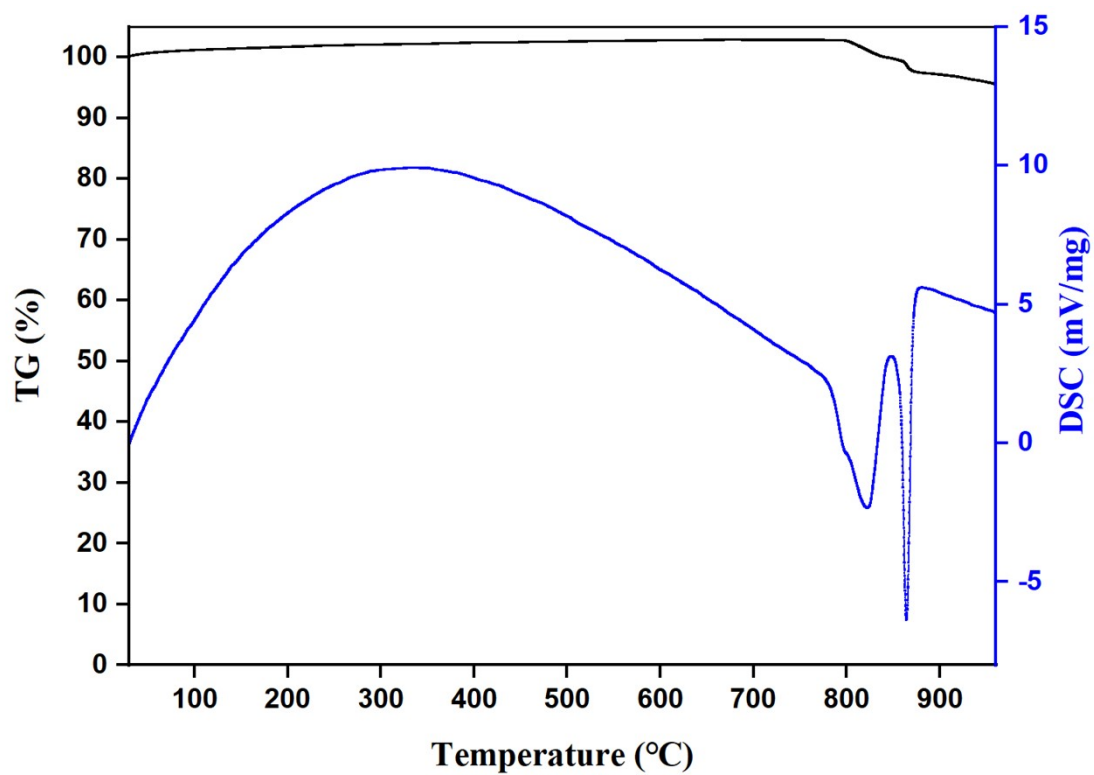
**Figure S1.** Structural comparison of (a)  $\text{Li}_2\text{ZrTeO}_6$ , (b)  $\alpha\text{-Li}_2\text{HfTeO}_6$ , (c)  $\text{Li}_2\text{SnTeO}_6$ , and (d)  $\text{Li}_2\text{TiTeO}_6$ .



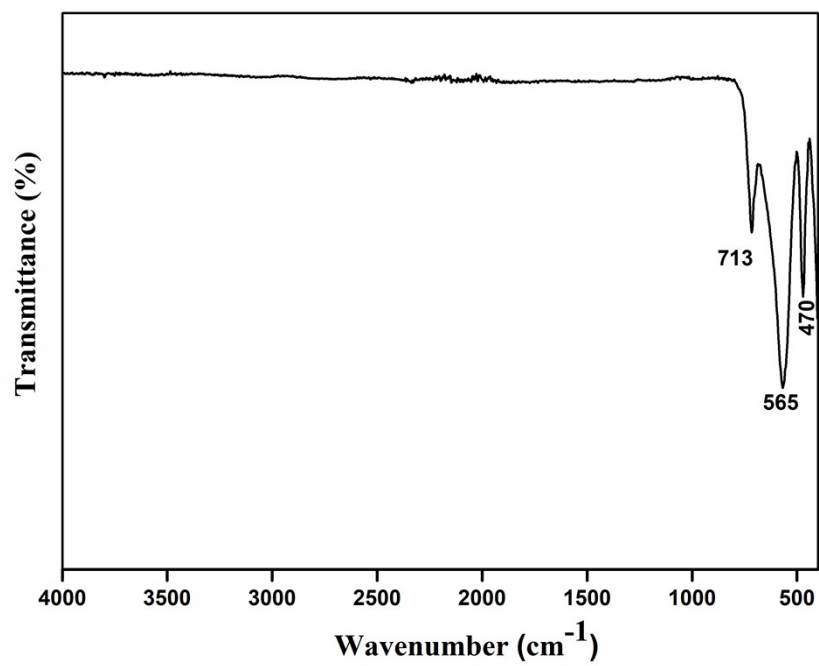
**Figure S2.** The SEM and EDX of  $\text{Li}_2\text{GeTeO}_6$ . Noted that the EDX only provides semi-quantitative results, and the Li elements are ignored due to the drawback of EDX analysis.



**Figure S3.** The calculated and experimental powder XRD patterns of  $\text{Li}_2\text{GeTeO}_6$ .



**Figure S4.** The TG and DSC curves of  $\text{Li}_2\text{GeTeO}_6$ .



**Figure S5.** The IR spectrum of  $\text{Li}_2\text{GeTeO}_6$ .

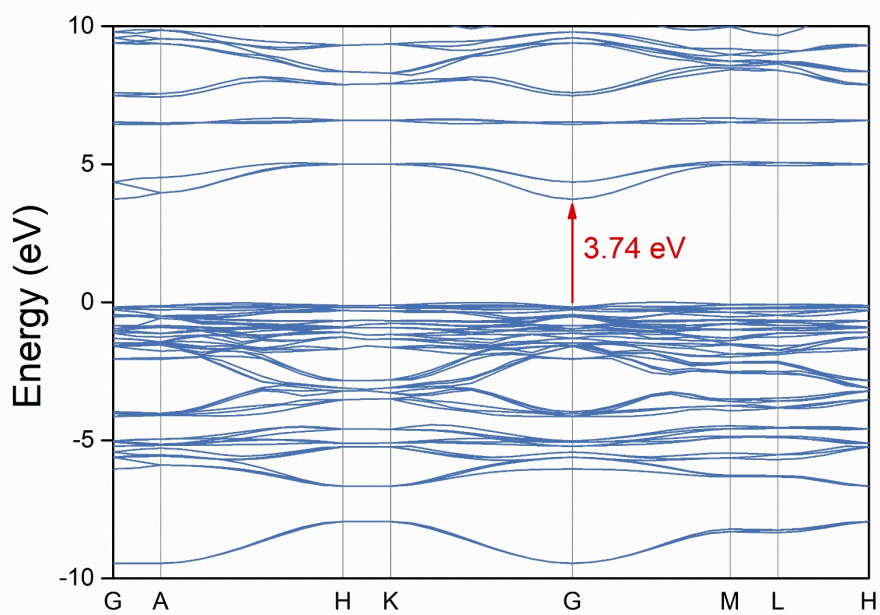


Figure S6. The band gap of  $\text{Li}_2\text{GeTeO}_6$ .

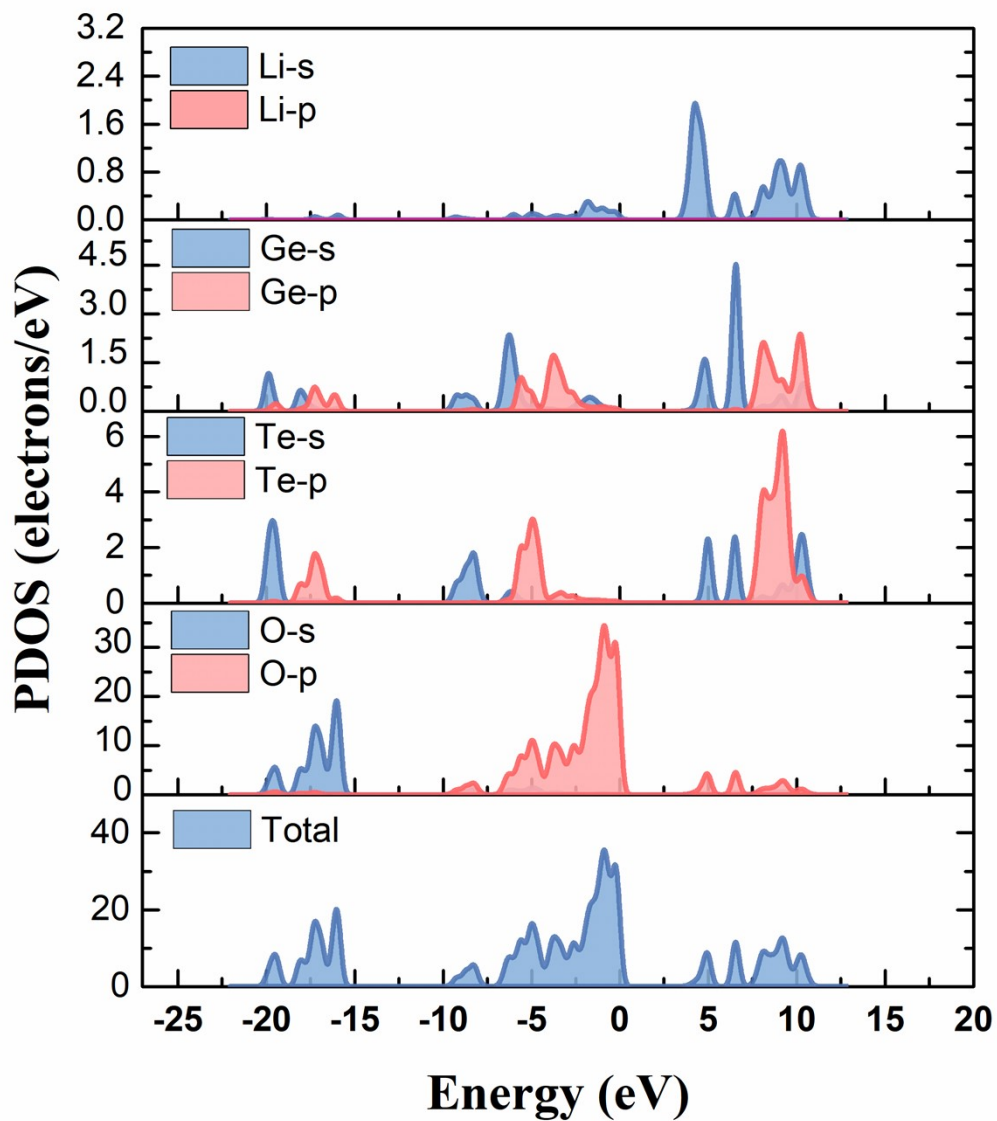
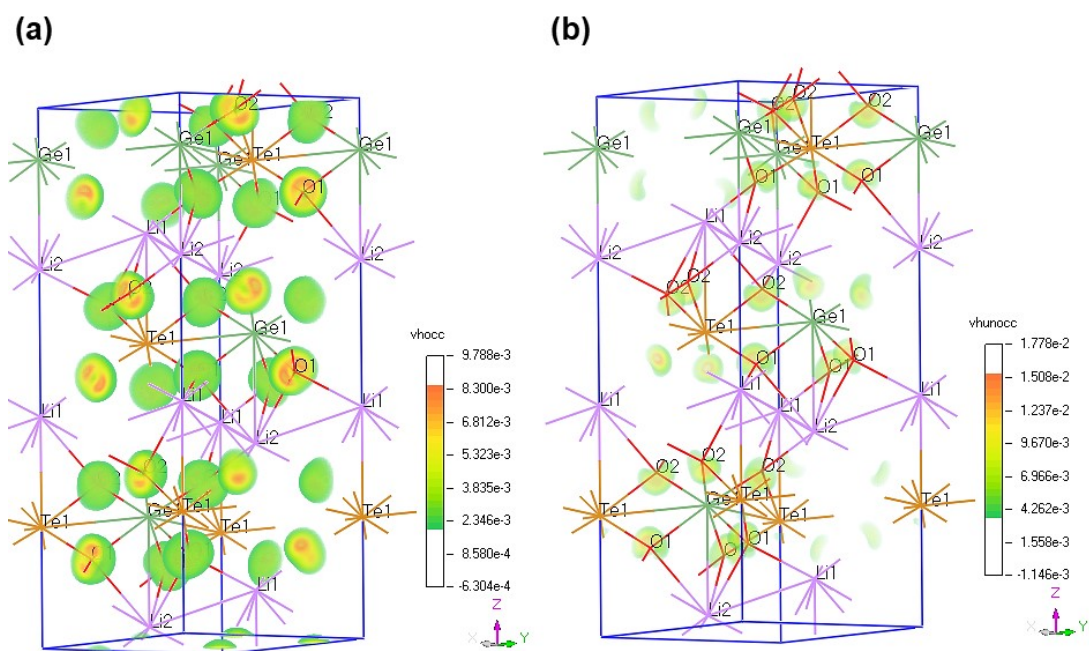


Figure S7. The DOS and PDOS of  $\text{Li}_2\text{GeTeO}_6$ .





**Figure S8.** SHG density maps of the occupied and unoccupied orbitals in the VH process of  $\text{Li}_2\text{GeTeO}_6$ . (a) VH occupied; (b) VH unoccupied.

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