# **Electronic Supplementary Information**

## **Cluster-derived TiO<sup>2</sup> nanocrystals with multiple carbon coupling for**

## **interfacial pseudo-capacitive lithium storage**

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- **Figure S10** TEM images, XRD patterns, N<sub>2</sub> adsorption-desorption isotherms, Raman spectra, and TG curves of s-TiO<sub>2</sub>-C-rGO, TiO<sub>2</sub>-C-rGO and l-TiO<sub>2</sub>-C-rGO.
- **Figure S11** SEM images and Nyquist plots of s-TiO<sub>2</sub>-C-rGO, TiO<sub>2</sub>-C-rGO and l-TiO<sub>2</sub>-C-rGO after cycling for more than 300 cycles at the current rate of  $0.1 \text{ A g}^{-1}$ .
- Table S1 Comparisons of lithium storage performances with reported TiO<sub>2</sub>-based nanostructures.



**Figure S1** XRD patterns of the as-synthesized Ti8Ph cluster and the simulated ones.



**Figure S2** XRD patterns (a) and Raman spectra (b) of the three typical samples obtained with 2 ml of DMF and 8 ml of  $H_2O$  using 0.8 mmol, 1.2 mmol and 1.4 mmol of NaF using Ti<sub>8</sub>Ph as the precursor.



**Figure S3** TEM (a,d), HRTEM (b, e) images and SAED patterns (c, f) of the samples obtained with 2 ml of DMF and 8 ml of  $H_2O$  using 0.8 mmol (a-c) and 1.4 mmol (d-f) of NaF using Ti<sub>8</sub>Ph as the precursor.



Figure S4 SEM images of products obtained with 10 ml H<sub>2</sub>O and different amounts of NaF. (a) 0.4 mmol, (b) 0.8 mmol, (c)1.2 mmol, (d)1.6 mmol, (e) 2.0 mmol.



**Figure S5** SEM images of products obtained with *x* ml of DMF, *y* ml of H2O and *z* mmol of NaF. (a, e, I, m, q) *x*=2, *y*=8, (b, f, j, n, r) *x*=4, *y*=6, (c, g, k, o, s) *x*=6, *y*=4, (d, h, l, p, t) *x*=8, *y*=2; (a-d) *z*=0.4, (e-h) *z*=0.8, (i-l) *z*=1.2, (m-p) *z*=1.6, (q-t) *z*=2.0.

To scrutinize the influences of  $H_2O$ , DMF and NaF on the products, a series of control experiments were conducted. With exclusive DMF, only sol-like species could be obtained. With H2O as the sole solvent, only small irregular nanoparticles could be observed (**Figure S4**). With mixed solvent of DMF and  $H<sub>2</sub>O$ , well-defined truncated tetragonal bipyramids gradually showed up and grew bigger and sharper with increasing dosage of NaF (**Figure S5**). As the ratio of  $H_2O$  increased, less NaF was needed to convert Ti<sub>8</sub>Ph into faceted TiO<sub>2</sub> nanocrystals. However, in the presence of insufficient and excessive NaF, small irregular nanoparticles and aggregated species other than  $TiO<sub>2</sub>$  were produced, respectively, regardless of the solvent composition. Hence, appropriate amount of DMF, H2O and NaF was essential to form uniform and well-faceted  $TiO<sub>2</sub>$  nanocrystals. Since DMF could well dissolve  $Ti<sub>8</sub>Ph$ , the cluster probably existed in the form of molecules in the reaction solution. It could serve either as building blocks to directly participate in the construction of nanocrystals, or as reservoirs that were able to continuously and steadily release Ti ions during the reaction. Such bottom-up synthetic processes may be accounted for the homogeneous nucleation and the resulting uniform generation of  $TiO<sub>2</sub>$ . H<sub>2</sub>O, another solvent, acted as both the source and medium for the subsequent hydrolysis and condensation of the Ti-O cluster. As is well-established, fluoride ions and HF are effective in promoting the exposure of  $(001)$  facets of anatase TiO<sub>2</sub> by tightly adsorbing on the (001) facets. Herein, NaF also played vital roles in facilitating the oriented growth of faceted  $TiO<sub>2</sub>$  nanocrystals. Yet to our surprise, the percentage of (001) facets gradually decreased with the increase of the amount of NaF. This unusual phenomenon might be rationalized by the large electronegativity of fluorine, which enabled fluoride ions to form hydrogen bonds with both  $H_2O$  and the aldehyde group of DMF. Besides, because of the electrostatic interaction, sodium ions would be attracted to the carboxylic acids surrounding the core of  $Ti_8Ph$ , and meanwhile the majority of fluoride ions would stay close to the sodium ions. As a result, the number of fluoride ions capable of adsorbing on the surfaces of  $TiO<sub>2</sub>$ , particularly (001) facets, would be greatly declined, leading to less inhibited growth of the nanocrystals and gradual elimination of (001) facets. Therefore, it was reasonable to conclude that NaF was of paramount importance to the shape and size control of the  $TiO<sub>2</sub>$  nanocrystals. Its cooperative work with the precursor and solvents was also responsible for the oriented growth of the well-defined truncated tetragonal bipyramids of anatase  $TiO<sub>2</sub>$ .



**Figure S6** SEM and TEM images of pristine TiO<sub>2</sub> nanocrystals with the middle size.



**Figure S7** SEM and TEM images of TiO<sub>2</sub>-rGO, in which TiO<sub>2</sub> nanocrystals were of the middle size.



**Figure S8** Ti 2p (a, d, g, j), O 1s (b, e, h, k) and C 1s (c, f, i, l) XPS of the four typical samples of TiO<sub>2</sub> (a-c), TiO<sub>2</sub>-C (d-f), TiO<sub>2</sub>-rGO (g-i), TiO<sub>2</sub>-C-GO (j-l), in which TiO<sub>2</sub> nanocrystals were of the same middle size.



**Figure S9** Characterizations of the active materials corresponding to TiO<sub>2</sub>-C-rGO after cycling for more than 300 cycles at the current rate of 0.1 A  $g^{-1}$ . (a-b) TEM images (peeled off the copper foils), (c) XRD patterns (loaded on the copper foils), (d) Nyquist plots recorded at the open circuit voltage in the frequency range from 100 kHz to 100 mHz with an amplitude of 5 mV. For comparison, the corresponding Nyquist plots of TiO<sub>2</sub>-C, TiO<sub>2</sub>- rGO, and TiO<sub>2</sub> were also provided.



**Figure S10** TEM images (a, s-TiO<sub>2</sub>-C-rGO; b, l-TiO<sub>2</sub>-C-rGO), XRD patterns (c), N<sub>2</sub> adsorptiondesorption isotherms (d), Raman spectra (e), and TG curves (f) of s-TiO<sub>2</sub>-C-rGO and l-TiO<sub>2</sub>-C-rGO. For comparison, the latter four characterizations of TiO<sub>2</sub>-C-rGO were also provided.



**Figure S11** SEM images (a-c) and Nyquist plots (d) of the three composites of s-TiO<sub>2</sub>-C-rGO (a),  $TiO<sub>2</sub>-C-rGO$  (b) and l-TiO<sub>2</sub>-C-rGO (c) after cycling for more than 300 cycles at the current rate of  $0.1 A g^{-1}$ . The Nyquist plots were recorded at the open circuit voltage in the frequency range from 100 kHz to 100 mHz with an amplitude of 5 mV





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