

## Electronic supplementary information

### Nanoscale mapping of edge-state conductivity and charge-trap activity in topological insulators

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#### I. Current and noise mapping by a conducting-AFM integrated with scanning noise set-up

For the electrical current and noise measurements, a platinum (Pt) tip (25-Pt300B, Park-Systems) installed on a conducting AFM (XE-70, Park-Systems) made a contact with the surface of the Bi<sub>2</sub>Se<sub>3</sub>. While the tip was scanning the surface of the Bi<sub>2</sub>Se<sub>3</sub> film, a small DC bias (20 mV) was applied between the Au substrate and the grounded Pt tip by a function generator (DS345, Stanford Research Systems). Importantly, the adhesion of the Bi<sub>2</sub>Se<sub>3</sub> multilayer film on the Au substrate was strong, allowing us to reliably measure an image in contact mode with a 300 nN contact force. We typically scanned an area of 5×5 μm<sup>2</sup> with a scan speed of 0.2 Hz. The electrical currents through the Pt tip were measured and converted to amplified voltage signals by a low-noise preamplifier (SR570, Stanford Research Systems). The amplified signals were filtered by a band-pass filter (6 dB) of the SR570 preamplifier to obtain the electrical noise signal which was the fluctuating component of the current signals. The root mean square power of the noise signal was obtained by the homemade custom-designed spectrum analyzer and the RMS-to-DC converter based on an AD737 chip. Note that, the RMS power was measured over the entire frequency range of the pass-band of the filter. Hence, we obtained the absolute noise PSD value at the central frequency of the band, by dividing the total RMS power to the bandwidth of the filter. Using this system, we could obtain the maps of the topography, the current and the noise PSD with a specific frequency on the Bi<sub>2</sub>Se<sub>3</sub> multilayer film, simultaneously. Finally, the measured maps were analyzed using a vertical transport

model to obtain the maps of conductivities and charge-trap densities in the Bi<sub>2</sub>Se<sub>3</sub> multilayer film.

#### II. Preparation of Bi<sub>2</sub>Se<sub>3</sub> multilayer sample

Bi<sub>2</sub>Se<sub>3</sub> flakes (purity 99.999%) were purchased from Alfa Aesar (Product Id 13126, 1-6 mm in sizes) and preserved inside a vacuum desiccator in the dark condition. First, a Bi<sub>2</sub>Se<sub>3</sub> flake was placed on a Cu tape, and a few top layers of the Bi<sub>2</sub>Se<sub>3</sub> were cleaved by a scotch tape to get the clean surfaces of Bi<sub>2</sub>Se<sub>3</sub> on the Cu tape. This step also made top layers brittle and detached slightly from the bulk piece. Then, the exfoliated Bi<sub>2</sub>Se<sub>3</sub> multilayer was transferred on an Au substrate by a dry transfer method with a polydimethylsiloxane (PDMS) stamp, providing a clean surface free from organic impurities.

#### III. Calculation of conductivity and charge-trap density

Since the conductivity of the Bi<sub>2</sub>Se<sub>3</sub> film was rather low, we could assume that charge carriers mainly flowed in a short vertical current path from an underlying Au substrate to a Pt probe through the film layer in a sandwich structure. We also assumed that parasitic currents in a lateral direction inside the layer were limited. Assuming a vertical charge transport, a conductivity  $\sigma$  can be calculated like  $\sigma = Id(VA)^{-1}$ , where  $I$ ,  $d$ ,  $V$ , and  $A$  represent a *measured current*, the *thickness of the film*, an *applied voltage*, and the *contact area of a conducting probe*, respectively. In our experiment, the thickness of the film was ~25 nm. The contact area  $A$  of our conducting probe was estimated as ~2000 nm<sup>2</sup> from the effective contact radius.<sup>1</sup> The measured conductivity values were comparable to the previously-reported values of the conductivity of Bi<sub>2</sub>Se<sub>3</sub> film, showing the reliability of our method. Then, the noise PSD values of the mean-square fluctuation in the number of occupied charge-traps in the small segment of the Bi<sub>2</sub>Se<sub>3</sub> film within the contact area  $A$  of the tip can be written as<sup>2,3</sup>

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$$S_{N_T}(f, x, y) = A \cdot \int_{-\infty}^{\infty} \int \frac{4\tau(E, x, y, z)}{1 + [2\pi f \cdot \tau(E, x, y, z)]^2} \cdot f_t(1 - f_t) \cdot N_T(E, x, y, z) \cdot dz \cdot dE \quad (1)$$

where the  $N_T$ ,  $\tau$ , and  $f$  are the *density of charge-traps over the space and energy*, a *trapping time constant*, and a *frequency*, respectively. The integral over  $z$  ranges from 0 to the film thickness  $d$ . The trap occupancy function can be written as  $f_t(E) = [1 + \exp\{(E - E_f)/kT\}]^{-1}$  where  $E_f$  is Fermi level. At a rather low temperature including a room temperature,  $f_t(1 - f_t)$  behaves like a delta function around the Fermi level  $E_f$ , and the eqn (1) after the integral over  $E$  can be simplified as<sup>2,3</sup>

$$S_{N_T}(f, x, y) = A \cdot kT \cdot \int \frac{4\tau(E_f, x, y, z)}{1 + [2\pi f \cdot \tau(E_f, x, y, z)]^2} \cdot N_T(E_f, x, y, z) \cdot dz \quad (2)$$

Assuming that charge-traps are distributed uniformly over the  $z$  direction, the eqn (2) can be approximated as

$$S_{N_T}(f, x, y) = A \cdot d \cdot kT \cdot N_T(x, y) \frac{4\tau(x, y)}{1 + [2\pi f \cdot \tau(x, y)]^2} \quad (3)$$

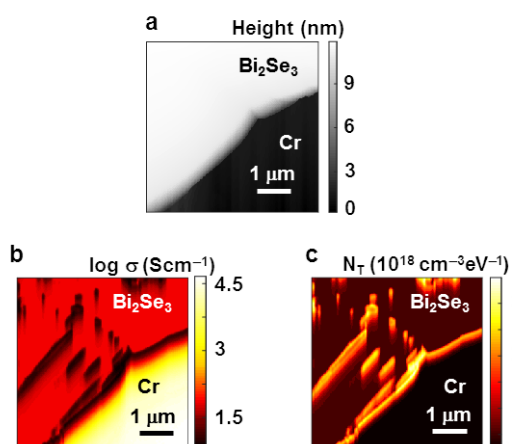
The noise PSD  $\Delta S_I$  can be written as

$$\Delta S_I(f, x, y) = \frac{(I)^2}{(\Delta C)^2} S_{N_T}(f, x, y) \quad (4)$$

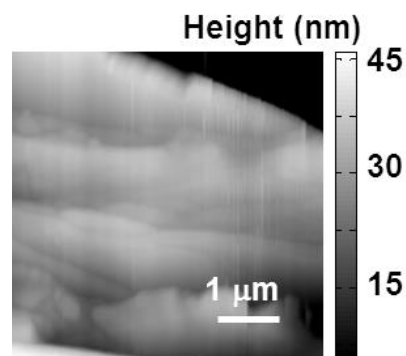
where  $\Delta C$  is the carrier number in the segment of the  $\text{Bi}_2\text{Se}_3$  film. Then, the charge-trap density  $N_T$  can be written like,

$$N_T(x, y) = \Delta S_I(f, x, y) \frac{(\Delta C)^2}{(I)^2} \cdot \frac{1}{A \cdot d \cdot kT} \cdot \frac{1 + [2\pi f \cdot \tau(x, y)]^2}{4\tau(x, y)} \quad (5)$$

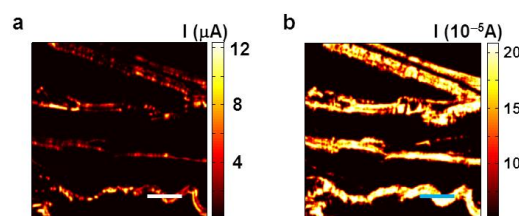
The average *trapping time*  $\tau$  and the *carrier density* in the  $\text{Bi}_2\text{Se}_3$  were reported as  $\sim 10^{-5}$  seconds and  $\sim 10^{21} \text{ cm}^{-3}$ , respectively.<sup>4</sup>



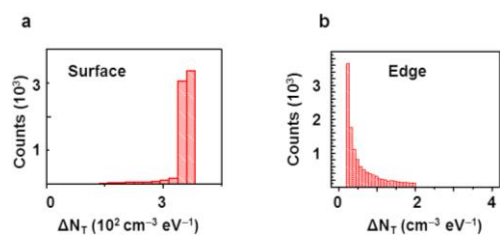
**Fig. S1** Scanning noise microscopy mapping of  $\text{Bi}_2\text{Se}_3$  on magnetic Cr samples. (a) Topography image showing the height of  $\text{Bi}_2\text{Se}_3$  on Cr substrate. (b) Conductivity map of  $\text{Bi}_2\text{Se}_3$  on Cr substrate. Edge regions showed lower conductivity than that of surface regions. (c) Charge trap density map of  $\text{Bi}_2\text{Se}_3$  on Cr substrate. Edge regions showed higher trap density in accordance with non-topological edges.



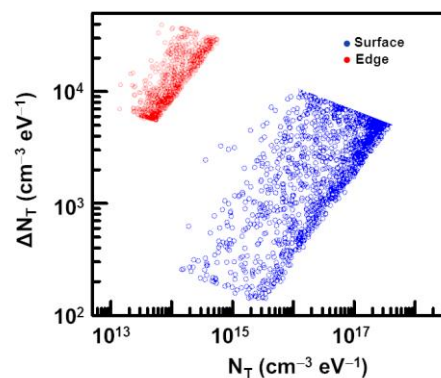
**Fig. S2** Topography image of  $\text{Bi}_2\text{Se}_3$  multilayer film. Contact mode AFM was utilized for imaging. The height of the  $\text{Bi}_2\text{Se}_3$  film was  $\sim 40$  nm corresponding to 40 van der Waal planes.



**Fig. S3** Current maps of  $\text{Bi}_2\text{Se}_3$  multilayer film at various bias voltages. (a) Current map of  $\text{Bi}_2\text{Se}_3$  at the low bias of 0.02 V. (b) Current map of  $\text{Bi}_2\text{Se}_3$  at the high bias of 0.2 V. At the high bias, edges showed rapid increase in the current. Also, the edge-currents became broader due to stronger spin orbit coupling. Scale bars are 1  $\mu\text{m}$ .



**Fig. S4** Change in trap density ( $\Delta N_T$ ) with increasing bias voltage from 0.02 V to 0.2 V. (a) Histogram showing distribution of  $\Delta N_T$  in flat surface when bias voltage was increased from 0.02 V to 0.2 V. (b) Histogram showing distribution of  $\Delta N_T$  in edges when bias voltage was increased from 0.02 V to 0.2 V.  $\Delta N_T$  was nearly homogeneous in surface or edge.



**Fig. S5** Effect of white light illumination on the trap density ( $N_T$ ) of  $\text{Bi}_2\text{Se}_3$ . Scatter plot showing  $\Delta N_T$  dependence on  $N_T$  when the  $\text{Bi}_2\text{Se}_3$  was illuminated with white light.  $\Delta N_T$  showed linear dependences on  $N_T$  in both edges and the surface.

## Notes and references

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