

## Electronic Supporting Information

### Utilizing MOF Precursors toward One-Step, Calcination-Free Synthesis of MnO<sub>2</sub> Superstructure for Superior Lithium Storage

*Yang Fan,<sup>a</sup> Deli Luo,<sup>a</sup> Yan Wu,<sup>a</sup> Tianlang Peng,<sup>a</sup> Qi Qi,<sup>a</sup> Xubing Han,<sup>a</sup> Jinxin Zhou,<sup>a</sup> Yanling Wang,<sup>c</sup>*

*Bao Lin,<sup>a</sup> Qinqin Xiong,<sup>a</sup> Yongjun Yuan,<sup>a</sup> Haiying Qin,<sup>a</sup> and Xiaoshi Hu<sup>\*a,b</sup>*

<sup>a</sup>Key Laboratory of Novel Materials for Sensor of Zhejiang Province and New Energy Materials Research Center, College of Materials and Environmental Engineering, Hangzhou Dianzi University, Hangzhou, 310018, PR China.

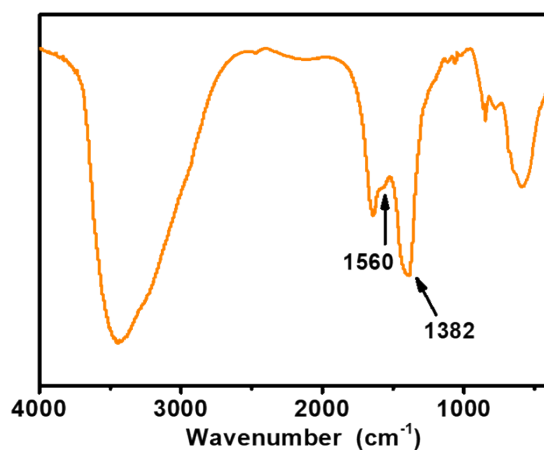
<sup>b</sup>State Key Laboratory of Silicon Materials, School of Materials Science & Engineering, Zhejiang University, Hangzhou, 310027, PR China

<sup>c</sup>College of Information Engineering & Art Design, Zhejiang University of Water Resources and Electric Power, Hangzhou, 310018, PR China

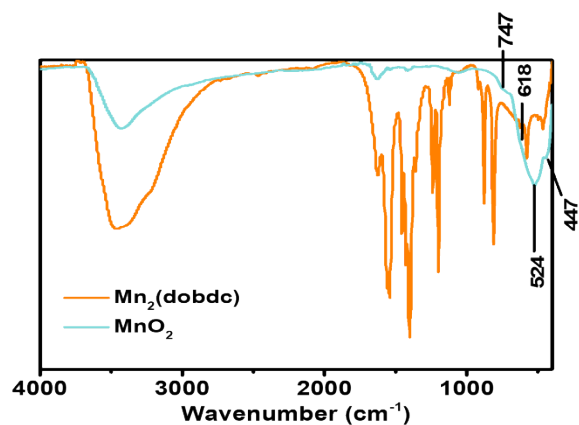
\* Corresponding author.

*E-mail addresses: xshu@hdu.edu.cn (X. S. Hu); huxiaoshi.happy@163.com (X. S. Hu)*

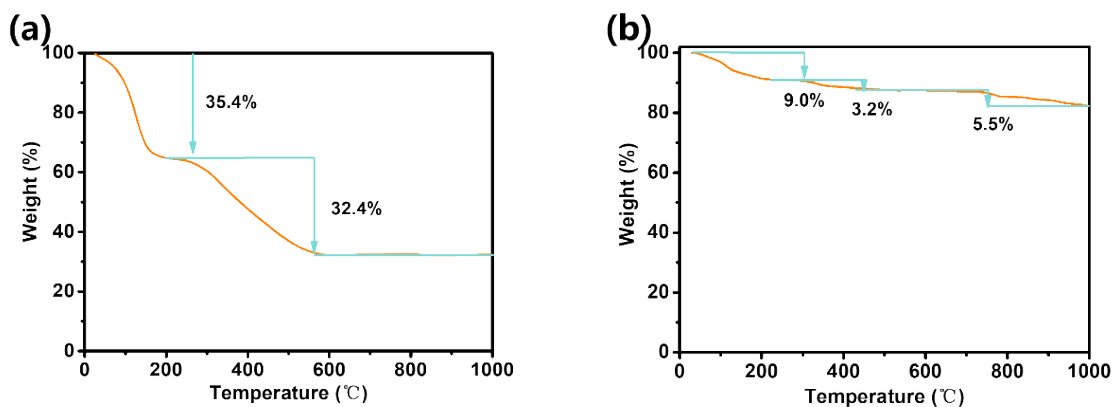
## Supporting Figures and Table



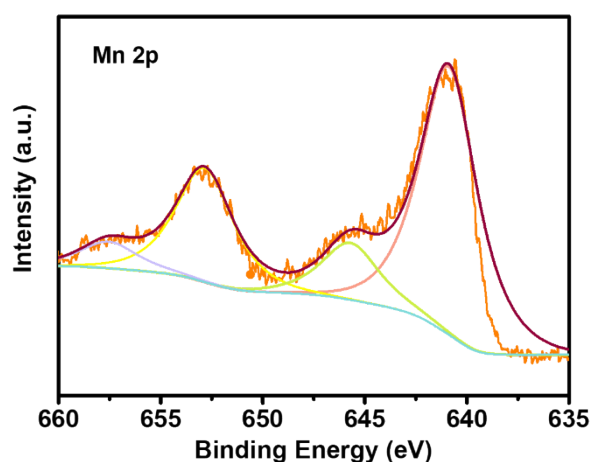
**Fig. S1** FTIR spectrum of the powder sample obtained by drying the filter liquor after reaction. It can be seen that obvious symmetric and asymmetric stretching vibrations of the carboxylate group situated at 1560 and 1382 cm<sup>-1</sup> assigned to released  $\text{dobdc}^{4-}$  linker from  $\text{Mn}_2(\text{dobdc})$  were present in the spectrum.<sup>[1]</sup>



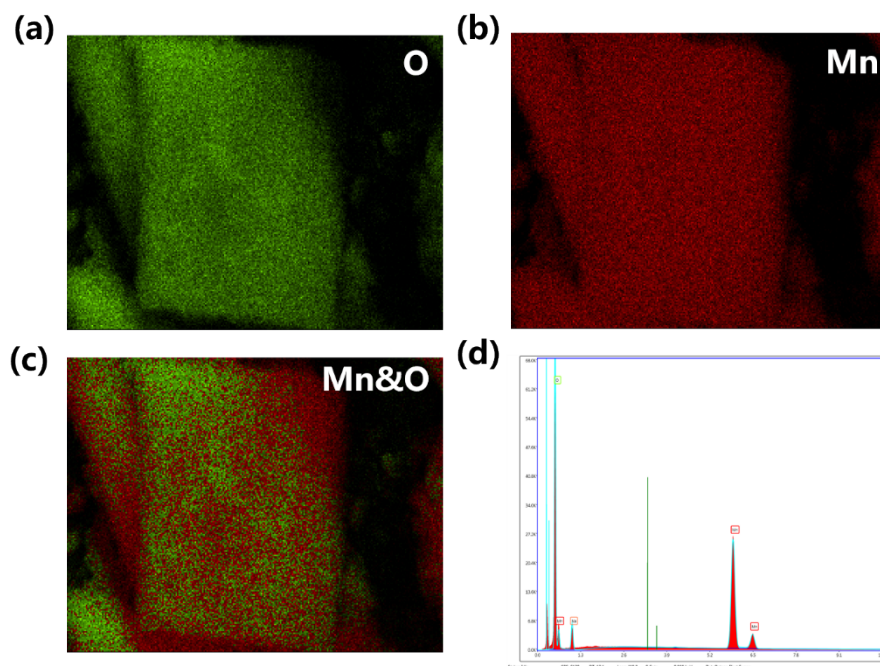
**Fig. S2** Comparison of FTIR spectrum of  $\text{Mn}_2(\text{dobdc})$  precursor and synthesized  $\text{MnO}_2$  materials. For  $\text{MnO}_2$  material, the peaks at about 3425 cm<sup>-1</sup> and other appreciable weak bands above 800 cm<sup>-1</sup> can be assigned to H–O–H stretching vibration and bending vibration of physically adsorbed water molecules by the  $\text{MnO}_2$  products. The  $\text{MnO}_2$  shows the absorptions at around 747, 618, 524 and 447 cm<sup>-1</sup> peaks which are due to the Mn–O–Mn asymmetric stretching vibration and bending vibration mode. It can be seen that all the vibrations assigned to inorganic  $\text{dobdc}^{4-}$  linker of  $\text{Mn}_2(\text{dobdc})$  in the 400–1590 cm<sup>-1</sup> region almost disappeared, suggesting that the organic portion of the framework had been completely consumed.<sup>[1, 2]</sup>



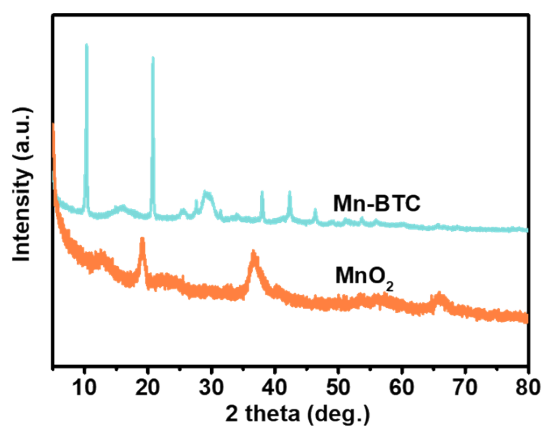
**Fig. S3** TGA curves of (A) Mn<sub>2</sub>(dobdc) and (B) MnO<sub>2</sub> samples in air. For the Mn<sub>2</sub>(dobdc) sample, the mass loss of 35.4% below 240 °C is related to the loss of coordinated solvent molecules and adsorbed water, while the weight loss of 32.4% in range of 300-600 °C stands for the decomposition of the dobdc<sup>4-</sup> ligands. No obvious weight loss is detected over 600 °C, suggesting the complete decomposition of the framework to manganese oxides. For the MnO<sub>2</sub> samples, a weight loss of 9.0 wt.% below 240 °C is attributed to the loss of surface and interlayer water. The weight loss of 3.2 wt.% in the range of 320-600 °C is due to the change from MnO<sub>2</sub> to Mn<sub>2</sub>O<sub>3</sub>. The weight loss of 5.5 wt.% occurs at over 600 °C is mainly due to the change from Mn<sub>2</sub>O<sub>3</sub> to Mn<sub>3</sub>O<sub>4</sub>.<sup>[3, 4]</sup>



**Fig. S4** High resolution XPS spectra for Mn 2p region measured for Mn<sub>2</sub>(dobdc). The high-resolution XPS spectra of Mn 2p exhibit two peaks at 640.9 and 652.8 eV, which are corresponded to the Mn 2p<sub>3/2</sub> and Mn 2p<sub>1/2</sub> spin-orbit. The spin orbital splitting ΔE (Mn 2p<sub>3/2</sub> - Mn 2p<sub>1/2</sub>) of 11.9 eV can be assigned to the Mn<sup>2+</sup>.<sup>[5]</sup>



**Fig. S5** EDX analysis of hierarchical mesoporous  $\text{MnO}_2$  polyhedron particle: (A) O, (B) Mn and (C) Mn&O elemental mapping images of single particle, and (D) EDX spectra.



**Fig. S6** XRD of rod-shaped Mn-BTC MOF precursors synthesized by following a modified literature method<sup>[6]</sup> (the amount of all raw materials were magnified five times in our modified synthesis with other conditions kept constant) and its derived  $\text{MnO}_2$  product.

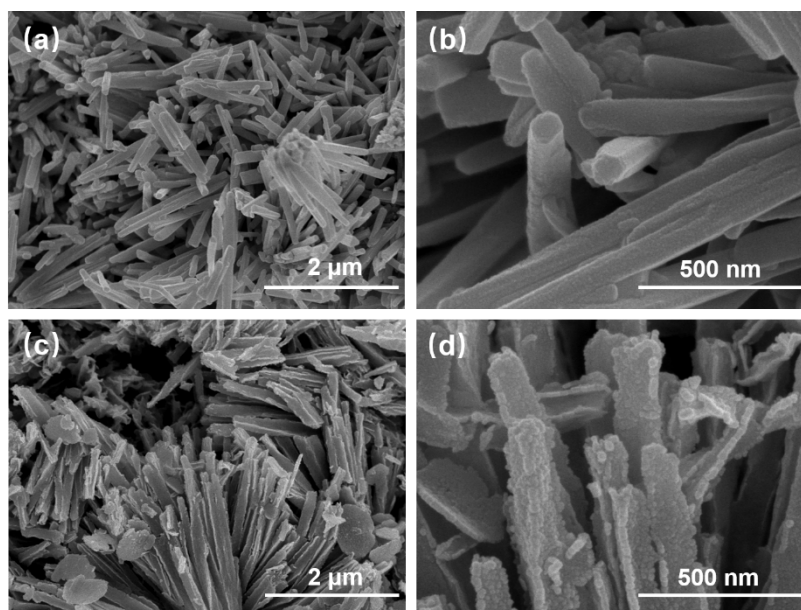


Fig. S7 SEM images of (a, b) rod-shaped Mn-BTC MOF precursors and (c, d) its derived MnO<sub>2</sub> product.

**Table S1.** Comparison of cycling performance of this work with other reported MnO<sub>2</sub>-related anode materials.

Sample	Rate(mA g <sup>-1</sup> )	Retention capacity (mA h g <sup>-1</sup> )	Cycle number	Ref.
Hollow MnO <sub>2</sub> nanospheres	100	840	60	[7]
α-MnO <sub>2</sub> nanotubes	200	618	300	[8]
3D hierarchical MnO <sub>2</sub> microspheres	300	715	200	[9]
α-MnO <sub>2</sub> hollow urchins	270	481	40	[10]
MnO <sub>2</sub> Nanowires	85	710	50	[11]
Three dimensional hierarchically porous crystalline MnO <sub>2</sub>	400	778	200	[12]
Core-shell porous MnO <sub>2</sub> /Carbon nanosphere composite	300	604	200	[13]
MnO <sub>2</sub> nanoflakes on vertically aligned carbon nanotubes	250	803	150	[14]
MnO <sub>2</sub> nanowire/graphdiyne	200	660	120	[15]
MnO <sub>2</sub> /3D porous graphene-like composites	100	836	200	[16]
MnO <sub>2</sub> -coated carbon nanohorns	100	705	60	[17]
Core-shell MnO <sub>2</sub> /3D-ordered hollow carbon sphere composite	100	692.5	100	[18]
Nanoflaky MnO <sub>2</sub> /carbon nanotube nanocomposites	200	620	50	[19]
Hierarchically layered mesoporous MnO <sub>2</sub> micropolyhedron	200 1000	866 797	60 300	This work

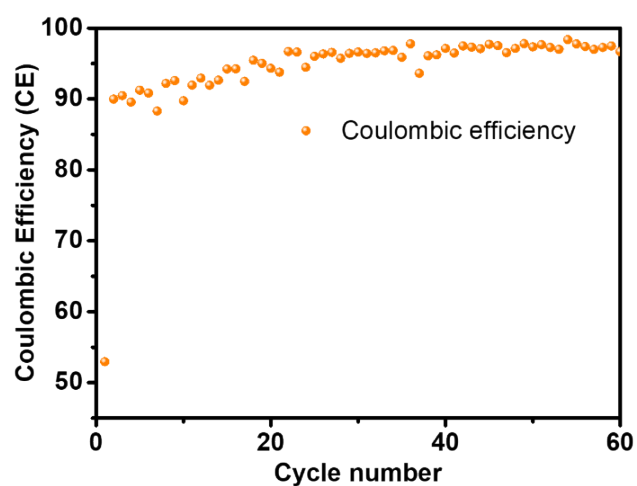


Fig. S8 Coulombic efficiency of the MnO<sub>2</sub> electrode when cycled at 200 mA g<sup>-1</sup>.

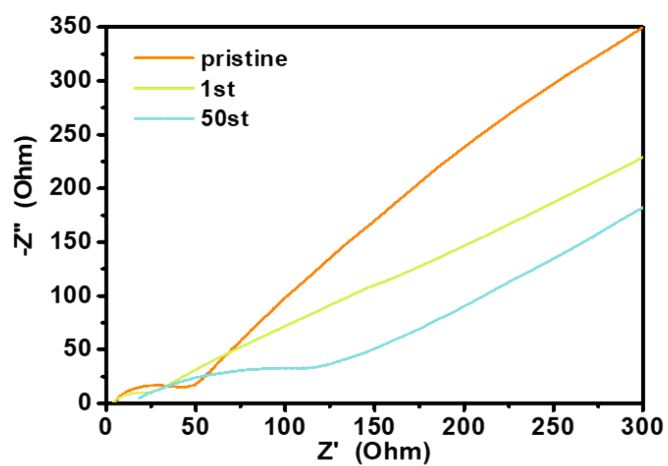
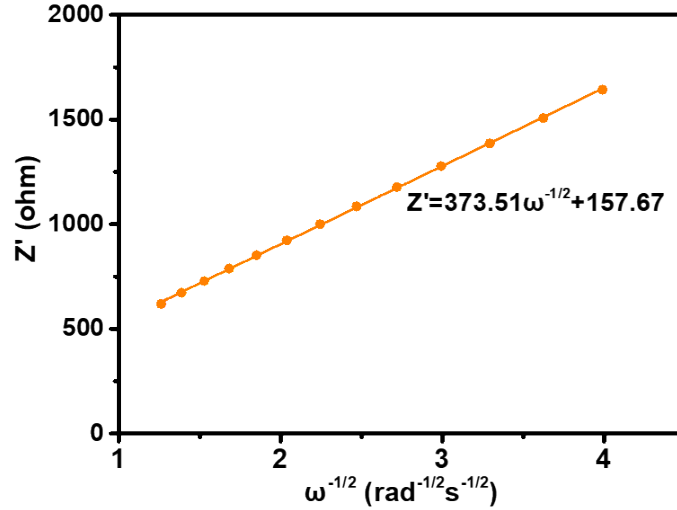


Fig. S9 Nyquist plots of our MnO<sub>2</sub> anode at the pristine state, after 1 cycle and after 50 cycles, respectively.

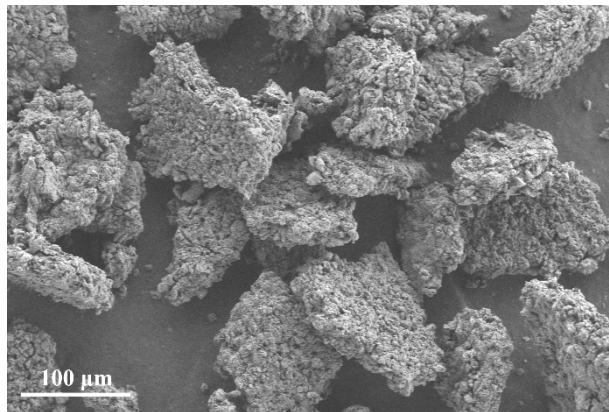


**Fig. S10** Relationship between  $Z'$  and  $\omega^{-1/2}$  for the  $\text{MnO}_2$  electrode after 50 cycles, the slope of the fitting line can be adopted to calculate the ion diffusion coefficient. The Li-ion diffusion coefficient ( $D$ ) in the electrode complies with the following formulas:<sup>[20]</sup>

$$D = \frac{R^2 T^2}{2A^2 n^4 F^4 C^2 \sigma^2} \quad (1)$$

$$Z' = \sigma \omega^{-1/2} \quad (2)$$

where  $R$ ,  $T$ ,  $A$ ,  $n$ ,  $F$ ,  $C$ , and  $\sigma$  correspond to gas constant ( $8.314 \text{ J mol}^{-1} \text{ K}^{-1}$ ), absolute temperature ( $298 \text{ K}$ ), electrode area ( $1.77 \text{ cm}^2$ ), electron transfer number per formula (ca. 4), Faraday constant ( $96485 \text{ C mol}^{-1}$ ), concentration of Li-ions ( $1 \times 10^{-3} \text{ mol cm}^{-3}$ ), and Warburg factor respectively. The Warburg factor  $\sigma$  can be obtained after fitting the slope of  $Z'$  versus  $\omega^{-1/2}$  and was determined to be 373.51.



**Fig. S11** Panoramic SEM images of  $\text{MnO}_2$  electrodes at charged state after long-term cycling.

### Supplementary references

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