Supporting Information

Stepwise oxidation strategy for the synthesis of amorphous $V_2O_5@V_2CT_x$ nanohybrid cathodes toward high-performance aqueous Zn-ion batteries

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First-Principle DFT Calculations

All computations are carried out in the framework of density generalized function theory, using a projector augmented plane-wave approach, implemented in the Vienna Vienna ab-initio simulation package [1]. The generalized gradient approximation proposed by Perdew, Burke, and Ernzerhof was chosen as the exchange-correlation potential [2]. The plane wave cut-off energy was set to 450 eV. The energy criterion was set as 10⁻⁵ eV in iterative solution of the Kohn-Sham equation. The Brillouin zone integration was performed using a $3\times3\times3$ k-mesh for VO₂ $2\times2\times2$ supercell, 5×5×3 k-mesh for Zn₃OH₂V₂O₇ and Zn₃OH₂V₂O₇·2H₂O, and 3×3×3 k-mesh for the amorphous $V_2O_5 2 \times 2 \times 2$ supercell, which is obtained through the AIMD method we have used the NVT ensemble with a timestep of 1 fs for 3000 fs at 1250 K. All the structures were relaxed until the residual forces on the atoms have declined to less than 0.02 eV/Å. Zn ion diffusion barriers and transition states in VO₂ 2×2×2 supercell, $Zn_3OH_2V_2O_7$, $Zn_3OH_2V_2O_7$, $2H_2O$ and V_2O_5 2×2×2 was determined using the climbing image nudged elastic band (CINEB) method [3, 4]. All calculations of this work were performed through the MedeA VASP 6.



Figure S1. (a) SEM image and (b) XRD pattern of V₂CT_x.

The weak diffraction peaks located at around 11.2° appeared after chemical etching, which is assigned to the (002) plane of V₂CT_x MXene. Importantly, the position of this peak tends to shift slightly depending on the amount of enclosed water intercalated between the MXene layers, which in turn may change with different synthesis conditions. The small peaks of 13.5° and 41.5° around 20 indicate the presence of a small amount of unreacted crystalline V₂AlC MAX phase (JCPDS No. 290101). This small amount of unreacted MAX phase appears in almost all MXenes synthesized by HF etching, and due to its more ordered structure compared to the produced MXene, its XRD peak is relatively sharp. In addition, SEM shows the formation of the typical accordion-like multilayered MXene structure of V_2 CT_x.



Figure S2. SEM elemental mapping images of the VO2(M)@V2CTx-600 materials



Figure S3. V 2p spectrum of V_2CT_x MXene.



Figure S4. GCD curves of (a) the $VO_2(M)@V_2CT_x$ -400 and (b) $VO_2(M)@V_2CT_x$ -800 electrodes in the 1 cycle and 2nd cycle.



Figure S5. TEM analysis of $VO_2(M)@V_2CT_x$ -600 cathode at selected voltage.



Figure S6. (a) GCD curves and (b) XRD pattern of the $VO_2(M)@V_2CT_x$ -600 electrode in 2 M $Zn(CF_3SO_3)_2$ in acetonitrile.



Figure S7. Predominant V5+ species in aqueous solutions at 25 °C as a function of vanadiumconcentration([V5+])andpH.5



Figure S8. (a, b) SEM images of the bare Zn. (c, d) SEM images and corresponding elementmapping(e)ofZn@ZnMoO₄.



Figure S9. Digital photographs of $Zn@ZnMoO_4$ and bare Zn immersed in 2M $Zn(CF_3SO_3)_2$ electrolyte.



Figure S10. Sketches of the cell models of $a-V_2O_5@V_2CT_x//without Zn$ anode, amorphous $V_2O_5@V_2CT_x//with Zn$ anode and the amorphous $V_2O_5@V_2CT_x//$ with $Zn@ZnMoO_4$ anode batteries and their corresponding XRD pattern and SEM image after 2 weeks of standing.



Figure S11. (a) XRD pattern, (b-d) SEM images, and (e-h) SEM element mapping of theamorphous $V_2O_5@V_2CT_x$ cathodeafter 2000^{th} cycle.



Figure S12. (a) CV curves, and (b) relationship between the logarithm of peak current andlogarithmofscanrateofthe $VO_2(M)@V_2CT_x$ -600.



Figure S13. (a, b) initial state, transition state and final state of immigration path of Zn in $VO_2(M)$ and $Zn_3OH_2V_2O_7$, respectively.

In the ZVO crystal structure, Zn atoms occupy the octahedral sites in a close-packed layer of O atoms; thus, Zn atoms in the layers are difficult to move. Although the $[ZnO_6]$ layers are connected by pyrovanadatem groups and forms an open framework, it exhibits much higher Zn^{2+} diffusion barriers than those of than $VO_2(M)$ and amorphous V_2O_5 .



Figure S14. GCD curves of the amorphous $V_2O_5@V_2CT_x$ cathode at 3 A g⁻¹ in H2SO4 and Zn(CF3SO3)2 electrolyte.

Cathode	Potential	Specific	Rate	Cycling	Ref.	
	window	capacity	capacity	performance		
Amorphous	0 2-1 8 V	651 mA h	302 mA h	405 mA h g ⁻¹ at 2	This work	
	0.2 1.0 1	σ ⁻¹ at 0.1	σ ⁻¹ at 20 Δ	Δ g ⁻¹ after 2000		
V205@V2C1x		g at 0.1	g at 20 A			
		Ag	8	(020/ rotantian)		
				(93% retention)		
Amorphous	0.2-1.8 V	489 mA h	237 mA h	240 mA h g⁻¹ at	Adv. Energy	
V_2O_5 /Graphene		g ⁻¹ at 0.1	g ⁻¹ at 20 A	30 A g ⁻¹ after	Mater. 2020,	
		A g ⁻¹	g ⁻¹	3000 cycles.	2000081	
				(87% retention)		
N-doped KMn ₈ O ₁₆	0.8-1.8 V	298 mA h	106 mA h	262 mA h g ⁻¹ at 1	Adv. Sci. 2022,	
with abundant		g ⁻¹ . at 0.1	g ⁻¹ at 10 A	A g ⁻¹ after 2500	9, 2106067	
oxygen vacancy		A g ⁻¹	g ⁻¹	cycles.		
				(91% retention)		
CuMn-PBA	0.45-1.8	108 mA h	52 mA h	104 mA h g ⁻¹ at 1	Angew. Chem.	
	V	g ⁻¹ . at 0.1	g ⁻¹ at 2 A	A g ⁻¹ after 2000	Int. Ed. 2022,	
		A g ⁻¹	g ⁻¹	cycles.	61, e202212031	
				(96.8% retention)		
CNT-CaMO	1-1.8 V	366 mA h	104 mA h	no obvious	Dalton Trans.,	
		g ⁻¹ . at 0.1	g ⁻¹ at 2 A	capacity fading	2022, 51, 9477–	
		A g ⁻¹	g ⁻¹	during 6000	9485	
				cycles at 3 A g ⁻¹ .		
d-VOH@CT	0.2-1.6 V	386 mA h	168 mA h	- 194 mA h g ⁻¹ at 5	J. Mater. Chem.	
		g ⁻¹ . at 0.2	g ⁻¹ at 10 A	 A g ⁻¹ after 2000	A, 2022, 10.	
		Α g ⁻¹	g ⁻¹	cycles.	13428–13438	
			o	(900/ rotontion)		

Table S1. Electrochemical performance comparisons of the amorphous $V_2O_5@V_2CT_x$ with therecently reported Zn-ion battery cathodes.

Amorphous	0.2-1.8 V	544 mA h	144 mA h	135 mA h g ⁻¹ at	J. Power
$V_2O_5@Ti_3C_2T_x$		g ⁻¹ at 0.5	g ⁻¹ at 30 A	30 A g ⁻¹ after	Sources, 2022,
		A g ⁻¹	g ⁻¹	1000 cycles.	544, 231883
Mn ²⁺ -intercalated	0.2-1.6 V	403 mA h	235.5 mA	287.6 mA h g ⁻¹ at	Energy Stor.
$V_{10}O_{24}.nH_2O$		g ⁻¹ at 0.5	h g ⁻¹ at 22	10 A g ⁻¹ after	Materials, 2022,
		A g ⁻¹	A g ⁻¹	25000 cycles.	45 568-577
				(92.9% retention)	
Activated V_2CT_x	0.2-1.8 V	386 mA h	358 mA h	283.7 mA h g ⁻¹ at	Adv. Funct.
		g ⁻¹ at 1 A	g ⁻¹ at 30 A	30 A g ⁻¹ after	Mater. 2021,
		g ⁻¹	g ⁻¹	2000 cycles.	31, 2008033
				(79.2% retention)	
VSe ₂ @V ₂ CT _x	0.1-1.6 V	302 mA h	132 mA h	158 mA h g ⁻¹ at 2	ACS Nano 2022,
		g ⁻¹ at 0.1	g-1 at 2 A	A g ⁻¹ after 600	16, 2711–2720
		A g ⁻¹	g ⁻¹	cycles.	
				(93.1% retention)	
VS ₂ /MXene	0.2-1.2 V	213.4 mA	208.7 mA	195 mA h g ⁻¹ at 5	J. Power
		h g ⁻¹ at	h g ⁻¹ at 5	A g ⁻¹ after 2400	Sources, 2022,
		0.2 A g ⁻¹	A g ⁻¹	cycles.	545, 231944
				(93.4% retention)	

References

- 1 G. Kresse, D. Joubert, *Phy. Revi. B* 1999, **59**, 1758-1775.
- 2 J.P. Perdew, K. Burke, M. Ernzerhof, Phy. Rev. Lett. 1996, 77, 3865-3868.
- 3 G. Henkelman, B. P. Uberuaga; H. Jónsson, J. Chem. Phys. 2000, 113, 9901-9904.
- 4 G. Mills, H. Jónsson, G.K. Schenter, Surf. Sci. 1995, 324, 305-337.
- 5 Y. Kim, Y. Park, M. Kim, J. Lee, K.J. Kim, J.W. Choi, Nat. Commun. 2022, 13, 2371.