Flux-assisted polytypism in the [Na₂Cl]GaQ₂ heterolayered salt-inclusion chalcogenide family

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Physical Measurements

Single-Crystal X-ray Diffraction (SCXRD). SCXRD data were collected at 299(2)–302(2) K on a Bruker D8 QUEST diffractometer equipped with an Incoatec IµS 3.0 microfocus radiation source (MoK α , $\lambda = 0.71073$ Å) and a PHOTON II area detector. The crystals were mounted on a microloop using immersion oil. The raw data reduction and absorption corrections were performed using SAINT and SADABS programs.^{1,2} Initial structure solutions were obtained with SHELXS-2017 using direct methods and Olex2 GUI.³ Full-matrix least-square refinements against $|F|^2$ were performed with SHELXL software.⁴ The crystallographic data and results of the diffraction experiments are summarized in Tables S1 and S4.

Powder X-ray Diffraction (PXRD). PXRD data were collected on a Bruker D2 PHASER diffractometer using CuK α radiation ($\lambda = 1.5418$ Å) over the 2θ range 5–65 ° with a step size of 0.02° for 30 minutes. For high-temperature PXRD, patterns were collected on a Rigaku SmartLab Diffractometer in transmission geometry using a MoK α rotating anode (45 kV, 200 mA; $\lambda = 0.71073$ Å) equipped with an Anton Parr HTK1200N heating stage. A powder sample of [Na₂Cl]GaS₂ was loaded in a quartz capillary with an outer diameter of 1.0 mm and a wall thickness of 0.01mm and sealed. The sample was heated *in situ* in a hot stage that allowed rotation of the capillary during heating. Data were collected over the 2θ range 4–35 ° in 0.01° steps for 30 minutes. An initial PXRD pattern was collected at 25 °C and 100°C, and then the sample was heated at a rate of 10 °C/min from 100 °C to 400 °C, collecting data every 25 °C, and finally to 650 °C, collecting data every 10 °C. The diffraction patterns were analyzed using the Rigaku SmartLab Studio II software.

Energy Dispersive Spectroscopy (EDS). EDS was performed directly on crystals mounted on an SEM stub with carbon tape. Quantitative elemental analysis was carried out using a Tescan Vega-3 SEM instrument equipped with a Thermo EDS attachment. The SEM was operated in the low-vacuum mode with a 30 kV accelerating voltage and a 120 s accumulating time. The EDS results are summarized in Table S3 and Figures S5–S8, S10, and S11.

Thermogravimetric Analysis (TGA). Thermogravimetric and differential thermal analysis (TGA/DTA) measurements were performed on polycrystalline powder samples using a SDT Q600 Thermogravimetric Analyzer and an alumina pan as the sample holder. Samples were heated from room temperature to the target temperature (700 °C or 900 °C) at 10 °C/min under a flow of nitrogen gas (100 mL/min), and the resulting powders were analyzed by PXRD for phase identification post heating.

UV-vis Spectroscopy. UV-vis spectra were recorded using a Perkin-Elmer lambda 35 scanning spectrophotometer. The spectrophotometer was operated in diffuse reflectance mode and was equipped with an integrating sphere. Reflectance data were converted internally to absorbance *via* the Kubelka-Munk function. Spectra were recorded in the 200–900 nm range.

Fourier-Transform Infrared (FTIR) Spectroscopy

Vibrational spectra over the range of 4000–650 cm⁻¹ were recorded using a PerkinElmer Spectrum 100 FT-IR spectrometer equipped with a diamond attenuated total reflectance (ATR) attachment. Final infrared (IR) spectra consist of 20 total averaged scans.

Computational Details. All DFT calculations were carried out using the Vienna Ab-initio Simulation Package (VASP)^{5, 6} utilizing the Projector Augmented-Wave (PAW) potential.⁷

The PBE parametrization of the generalized gradient approximation (GGA) to the exchangecorrelation functional was used for the calculations.⁸ We used the following potentials of the constituent elements: Li_sv (1s² 2s¹), F (2s² 2p⁵), Na_pv (2p⁶ 3s¹), S (3s² 3p⁴), Cl (3s² 3p⁵), K_sv (3s² 3p⁶ 4s¹), Ga (3s² 3p¹), and Br (4s² 4p⁵). All compounds were structurally relaxed (volume, cell shape, and atomic positions) starting from their experimentally known crystal structures, where available, following the DFT settings in the OQMD.^{9, 10} The threshold for energy convergence was set to 10^{-6} eV. The relaxed geometries of the [A_2X]GaS₂ compounds preserve the $P4_2/nmc$ space group during structure relaxation. For [Na₂Cl]GaS₂ composition, geometry relaxation was also done for orthorhombic polymorph while keeping the *Cmcm* space group. The energy cutoff for the DFT calculations was set to 520 eV for static calculations. The optimized lattice constants agree very well (error < 1.2%) with the experimental values (Table S5).

In relation to the current experimental work, we assessed the T = 0 K thermodynamic stabilities of the $[A_2X]GaS_2$ compounds. We calculated their formation energies, $\Delta H_f^{0K} = E(\sigma)$ - $\sum_i n_i \mu_i$ (Table S5), where *E* is the DFT-calculated total energy at 0 K for a compound σ , n_i is the fraction of the *i*-th element with chemical potential μ_i . Utilizing the formation energies of these compositions and all other compounds that are present in each quaternary phase space; we constructed the quaternary convex hull for $[A_2X]GaS_2$ in the OQMD that allowed to estimated decomposition energies (Table S5). Typical decomposition of $[A_2X]GaS_2$ leads to AX and $AGaS_2$. Only $[Na_2Cl]GaS_2$ and $[Na_2Br]GaS_2$ compounds demonstrated stability in the A-X-Ga-S landscape since they have positive decomposition enthalpies estimated by convex hull constriction.

Synthesis

Materials. The following materials were used as received without further purification: S (Fisher Scientific, 99%), Ga (BTC, 99.999%), Se (Alfa Aesar, 99.999%), Na₂S (Alfa Aesar, 95%), NaCl (Fisher Chemical, 99.9%), NaBr (BTC, 99.0%), NaI (Fisher Chemical, 99.9%), dimethylformamide (DMF, Sigma-Aldrich, 99.8%), and acetone (BDH, 99.5%). Synthesis of Na₂Se was performed using a reported procedure.¹¹ For all mentioned reactions, masses of the reagents were mixed inside a silica tube. Typically, the silica tube was flame sealed under a vacuum of $<10^{-4}$ Torr. The sealed tube was placed vertically into a programmable box furnace with the corresponding temperature profile. Once at room temperature, the tube was cut open, and the reaction products were put into a beaker filled with deionized (DI) water or DMF.

[Na₂Cl]GaQ₂ (Q = S and Se). The starting reagents, Ga (0.0540 g), Q, Na₂Q in a 1:2:1 molar ratio, and NaCl/NaI flux (0.054/0.159 g) were used to grow single crystals of [Na₂Cl]Ga Q_2 . The furnace with the tube was ramped up to 650 °C in 3 or 5 h, held at this temperature for 5 or 24 h, and then cooled to 450 or 250 °C at a cooling rate of 40 or 5 °C/h for S- and Se-based compounds, respectively. After the reaction and work up in water were complete, the final product was filtered and washed with distilled water and acetone. Single crystals of [Na₂Cl]Ga Q_2 were placed in immersion oil and mounted for SCXRD. The PXRD pattern for the bulk sample matches the simulated pattern obtained from SCXRD (Figures S13 and S16). The PXRD pattern for [Na₂Cl]GaS₂ contained minor NaGaS₂·H₂O impurity (Figure S13). This correlates with a colling rate chosen for sulfide formation (40 °C/h) versus selenide (5 °C/h),

resulting in precipitation from the flux of NaGaS₂, which converted to NaGaS₂·H₂O after workup in water. For phase pure [Na₂Cl]GaS₂ synthesis, the amount of flux was doubled to 0.108 g and 0.318 g of NaCl and NaI, respectively. The furnace with the tube was ramped up to 650 °C in 5 h, held at this temperature for 48 h, and then cooled to 200 °C at a cooling rate of 5 °C/h. The PXRD pattern for the bulk sample matches the simulated pattern obtained from SCXRD (Figure S14). We also attempted to collect the PXRD pattern for tetragonal polymorph [Na₂Cl]GaS₂ only; for that, we isolated *t*-[Na₂Cl]GaS₂ single crystal, crashed it, and collected data on the resulting powder using phi scan in SC-XRD instrument (Figure S15) that matches well the simulated pattern of *t*-[Na₂Cl]GaS₂.

Na₂GaSe₃ and Na₄Ga₂Se₅. In an attempt to obtain [Na₂Br]GaSe₂ single crystals, we grew crystals of Na₂GaSe₃ and Na₄Ga₂Se₅ instead. The starting reagents, Ga (0.0540 g), Se (0.1230 g), Na₂Se (0.096 g) in a 1:2:1 molar ratio, and NaBr/NaI flux (0.095/0.208 g) were used to obtain single crystals of Na₂GaSe₃ and Na₄Ga₂Se₅. The furnace with the tube was ramped up to 650 °C in 5 h, held at this temperature for 24 h, and then cooled to 250 °C at a cooling rate of 10 °C/h. After the reaction and work up in DMF were complete, the final product was filtered and washed with acetone. The product was placed in immersion oil where the pink rod Na₂GaSe₃ and the colorless block Na₄Ga₂Se₅ single crystals were isolated and mounted for SCXRD. PXRD pattern for the bulk sample consisted of multiple phases: Na₂GaSe₃, Na₄Ga₂Se₅, and NaGaSe₂·*x*H₂O,¹² and no further attempts were made to obtain the phase pure samples.

X-ray Crystal Structure Determination

o-[Na₂Cl]GaS₂. The compound crystallizes in the orthorhombic crystal system, space group *Cmcm*. The asymmetric unit consists of seven Na atoms, two Cl halogens, two Ga atoms, and five S chalcogens. The largest residual electron density peak in the final difference map is 0.58e-/Å³, located 0.68 Å from Na6. The crystallographic data and results of the diffraction experiments are summarized in Table S1.

o-[Na₂Cl_{0.9}I_{0.1}]GaSe₂. The compound crystallizes in the orthorhombic crystal system, space group *Cmcm*. The asymmetric unit consists of seven Na atoms, two Cl and two I halogens, two Ga atoms, and five Se chalcogens. There is a disorder between Cl and I for both Cl crystallographic sites. Positions and thermal ellipsoids for Cl1 and I1, as well as for Cl2 and I2 pairs, were fixed using EXYZ and EADP commands. Occupancies of Cl and I were linked with FVAR command; thus, total site occupancy is equal to 1. Cl1 and Cl2 occupancies were refined to 0.894(5) and 0.921(4), respectively. The largest residual electron density peak in the final difference map is 1.28 e-/Å³, located 1.48 Å from Ga2. The crystallographic data and results of the diffraction experiments are summarized in Table S1. Precession images for (*h0l*) planes generated for the orthorhombic *C* Bravais lattice demonstrated superstructure peaks of low intensity, however, attempts to solve this structure in orthorhombic *C* lattice with double *a* parameter, i.e., 15.194 Å vs. 7.597 Å, that accounts for the superstructure, did not result in a reasonable structure model.

t-[Na₂Cl]GaS₂. The compound crystallizes in the orthorhombic crystal system, space group $P4_2/nmc$. The asymmetric unit consists of three Na atoms, one Cl halogen, one Ga atom, and

three S chalcogens. The largest residual electron density peak in the final difference map is 1.23e-/Å³, located 1.85 Å from Na2. The crystallographic data and results of the diffraction experiments are summarized in Table S1.

t-[Na₂Cl_{0.9}I_{0.1}]GaSe₂. The compound crystallizes in the orthorhombic crystal system, space group $P4_2/nmc$. The asymmetric unit consists of three Na atoms, one Cl and one I halogen, one Ga atom, and three Se chalcogens. There is a disorder between Cl and I. Positions and thermal ellipsoids for Cl1 and I1 were fixed using EXYZ and EADP commands. Occupancies of Cl and I were linked with FVAR command; thus, total site occupancy is equal to 1. Cl1 occupancy was refined to 9075(19). The largest residual electron density peak in the final difference map is 0.97 e-/Å³, located 0.80 Å from Na1. The crystallographic data and results of the diffraction experiments are summarized in Table S1.

Na₂GaSe₃. The compound crystallizes in the monoclinic crystal system, space group C2/c. The asymmetric unit consists of three Na atoms, one Ga atom, and three Se chalcogens. The largest residual electron density peak in the final difference map is 0.70e-/Å³, located 0.98 Å from Na1. The crystallographic data and results of the diffraction experiments are summarized in Table S4.

Na₄Ga₂Se₅. The compound crystallizes in the monoclinic crystal system, space group $P2_1/c$. The asymmetric unit consists of four Na atoms, two Ga atoms, and five Se chalcogens. The largest residual electron density peak in the final difference map is 0.57e-/Å³, located 1.36 Å from Ga₂. The crystallographic data and results of the diffraction experiments are summarized in Table S4.

	0-	<i>t</i> _	0-	<i>t_</i>
compound	[NacCl]GaSa	[NacCl]GaSc	[Na2Clo Jo 1]GaSe2	[Na2Clo Jo 1]GaSe2
empirical				
formula	Na ₂ ClGaS ₂	Na ₂ ClGaS ₂	$Na_2Cl_{0.91}I_{0.09}GaS_2$	$Na_2Cl_{0.91}I_{0.09}GaS_2$
formula weight	215.27	215 27	317 53	317 53
temperature, K	299.85	301.61	299.85	299.08
crystal system	orthorhombic	tetragonal	orthorhombic	tetragonal
space group	Cmcm	$P4_2/nmc$	Cmcm	$P4_2/nmc$
a Å	7 2885(2)	7 2850(2)	7 597(2)	7 5821(2)
h, Å	39.4434(10)	7.2850(2)	41.001(11)	7.5821(2)
c Å	7 2884(2)	19 7163(5)	7 583(2)	20.4761(5)
α°	7.200 1(2)	19.7105(5)	1.505(2)	20.1701(5)
$\beta \circ$			90	
p,				
volume Å ³	2095 29(10)	1046 37(6)	2362 0(11)	1177 13(7)
7	16	8	16	8
$\rho_{\rm colo} g/cm^3$	2 73	2 733	3 572	3 583
$p_{\text{calc}}, g_{\text{calc}}$	$0.05 \times 0.04 \times$	$\frac{2.755}{0.8 \times 0.1 \times 1}$	5.572	5.505
crystal size, mm ³	0.02	0.08	$0.2\times0.05\times0.05$	$0.1\times0.07\times0.02$
2Θ range for	5.684 to	5.962 to	5 A5 A + 55	5 72 4 50 072
data collection, °	64.978	55.944	5.454 to 55	5.73 to 59.972
	$-11 \le h \le 11$,	$-9 \le h \le 9,$	$-9 \le h \le 9,$	$-10 \le h \le 10$,
index ranges	$-59 \le k \le 59,$	$-9-k \leq 9$,	$-53 \le k \le 53,$	$-9 \le k \le 10$,
	–11≤1≤11	$-26 \le l \le 26$	$-9 \le 1 \le 9$	$-28 \le 1 \le 28$
reflections	66207	48407	45040	33459
data/restraints/				
parameters	2158/0/80	736/0/40	1557/0/83	987/0/42
Goodness-of-fit	1.078	1 132	1 1/15	1 171
on F ²	1.070	1.152	1.145	1.1/1
R_1/wR_2	1 34/2 95	1 90/4 91	4 02/11 21	1 66/3 44
[I>=2σ (I)], %	1.54/2.75	1.90/4.91	7.02/11.21	1.00/ 5.11
R_1/wR_2	1 78/3 20	2 07/5 20	4 39/11 73	3 10/4 60
[all data], %	1.70/3.20	2.0773.20	1.59/11.75	5.10/ 7.00
largest diff.	0.58/-0.64	1.23/-0.52	1.28/-1.11	0.97/0.66
peak/hole, e Å ⁻³	0.00, 0.01	1.25, 0.52	1.20/ 1.11	0.277 0.00
$R_{\rm int}, \%$	3.45	7.36	5.08	6.42

Table S1. Crystallographic data for *o*-[Na₂Cl]GaS₂, *t*-[Na₂Cl]GaS₂, *o*-[Na₂Cl_{0.9}I_{0.1}]GaSe₂, and *t*-[Na₂Cl_{0.9}I_{0.1}]GaSe₂.

compound	d(Cl–Na), Å	<i>d</i> (Ga– <i>Q</i>), Å
	2.7696(9) 2.7964(9)	2.2651(3) 2.2758(3)
	2.7704(7) 2.8233(6) × 2	2.2793(3) 2 2791(3)
	$2.8235(0) \times 2$ 2.8286(6) × 2	2.2791(3) 2 2792(3)
o-[Na2C1]GaS2	2.0200(0) 2	2.2792(3)
	2.7673(9)	2.2658(3)
	2.7981(9)	2.2752(3)
	$2.8244(3) \times 2$	2.2791(3)
	$2.8274(3) \times 2$	2.2792(3)
	2.7697(13)	2 2642(5)
t-[Na ₂ Cl]GaS ₂	2.7958(12)	2.2042(3)
<i>t</i> -[INd2CI]CdS2	$2.8225(8) \times 2$	2.2737(4) $2.2770(4) \times 2$
	$2.8280(8) \times 2$	2.2779(4) × 2
	2.804(5)	2.3859(10)
	2.872(5)	2.3860(10)
	$2.922(3) \times 2$	2.3969(8)
	$2.946(3) \times 2$	2.3973(10)
o-[Na ₂ Cl _{0.9} I _{0.1}]GaSe ₂		
	2.818(5)	2.3893(11)
	2.876(5)	2.3939(8)
	$2.9321(12) \times 2$	2.4019(10)
	$2.9322(12) \times 2$	2.4020(10)
	2.813(3)	2.3865(5)
t-[Na2Clool01]GaSe2	2.872(2)	$2.3927(3) \times 2$
· [· · · · · · · · · · · · · · · · · ·	$2.922(2) \times 2$	2.3955(5)
	$2.937(2) \times 2$	2.5955(5)

Table S2. Selected interatomic distances for o-[Na₂Cl]GaS₂, t-[Na₂Cl]GaS₂, o-[Na₂Cl_{0.9}I_{0.1}]GaSe₂, and t-[Na₂Cl_{0.9}I_{0.1}]GaSe₂.



Figure S1. Precession image of (a) (hk0), (b) (h0l), and (c) (0kl) planes for o-[Na₂Cl]GaS₂ (s.g. *Cmcm*) single crystal. The simulated overlay is shown with white circles.



Figure S2. Precession image of (a) (hk0), (b) (h0l), and (c) (0kl) planes for t-[Na₂Cl]GaS₂ (s.g. $P4_2/nmc$) single crystal. The simulated overlay is shown with white circles.



Figure S3. Precession image of (a) (hk0), (b) (h0l), and (c) (0kl) planes for o-[Na₂Cl]GaSe₂ (s.g. *Cmcm*) single crystal. The simulated overlay is shown with white circles.



Figure S4. Precession image of (a) (hk0), (b) (h0l), and (c) (0kl) planes for t-[Na₂Cl]GaSe₂ (s.g. $P4_2/nmc$) single crystal. The simulated overlay is shown with white circles.

Table S3. EDS results for the single crystal of o-[Na₂Cl]GaS₂, t-[Na₂Cl]GaS₂, c-[Na₂Cl_{0.9}I_{0.1}]GaSe₂, t-[Na₂Cl_{0.9}I_{0.1}]GaSe₂, t-[Na₂

o-[Na ₂	Cl]GaS ₂	<i>t</i> -[N	[a ₂ Cl]GaS ₂
element	atom %	elemen	t atom %
Na	29.5(35)	Na	34.3(19)
Cl	17.5(5)	Cl	16.3(3)
Ga	17.1(5)	Ga	15.3(3)
S	35.9(5)	S	34.1(3)
o-[Na ₂ Cl ₀	.9I0.1]GaSe2	$t-[Na_2C]$	$Cl_{0.9}I_{0.1}]GaSe_2$
element	atom %	elemen	t atom %
Na	30.2(9)	Na	31.0(7)
Cl	21.7(6)	Cl	14.1(6)
Ι	2.2(2)	Ι	2.1(4)
Ga	14.9(19)	Ga	16.9(9)
Se	31.0(29)	Se	35.9(34)
Na ₂	GaSe ₃	N	a ₄ Ga ₂ Se ₅
Na	36.6(12)	Na	30.5(39)
Ga	14.8(4)	Ga	18.9(4)
Se	48.6(11)	Se	50.6(9)
Ga Na	S		Ga
	S CI		/\ .

energy, keV

Т

Figure S5. EDS spectra of *o*-[Na₂Cl]GaS₂.



Figure S6. EDS spectra of *t*-[Na₂Cl]GaS₂.



Figure S7. EDS spectra of *o*-[Na₂Cl_{0.9}I_{0.1}]GaSe₂.



Figure S8. EDS spectra of *t*-[Na₂Cl_{0.9}I_{0.1}]GaSe₂.

compound	Na ₂ GaSe ₃	Na4Ga2Se5
empirical formula	Na ₂ GaSe ₃	Na4Ga2Se5
formula weight	352.58	626.2
temperature, K	299.61	298.93
crystal system	monoclinic	monoclinic
space group	C2/c	$P2_{1}/c$
<i>a</i> , Å	21.8190(4)	8.3169(3)
b, Å	8.3340(2)	6.4160(2)
<i>c</i> , Å	6.86160(10)	20.7332(7)
α , °	90	90
β , °	100.6730(10)	91.1838(13)
γ, °	90	90
volume, Å ³	1226.12(4)	1106.11(6)
Z	8	4
$\rho_{\rm calc}, {\rm g/cm^3}$	3.82	3.76
crystal size, mm ³	0.3 imes 0.1 imes 0.01	0.1 imes 0.05 imes 0.02
2Θ range for data collection, °	5.244 to 59.99	4.898 to 56.586
	$-30 \le h \le 30,$	$-11 \le h \le 11$,
index ranges	$-11 \le k \le 11,$	$-8 \le k \le 8,$
	$-9 \le 1 \le 9$	$-27 \le l \le 27$
reflections collected	24947	47434
data/restraints/ parameters	1798/0/57	2740/0/101
Goodness-of-fit on F ²	1.076	1.102
R_1/wR_2 [I>=2 σ (I)], %	1.97/3.84	1.40/2.97
R_1/wR_2 [all data], %	2.63/4.18	1.71/3.10
largest diff. peak/hole, e Å ⁻³	0.70/-0.59	0.57/-0.70
$R_{\rm int}$, %	6.18	6.41

Table S4. Crystallographic data for Na_2GaSe_3 and $Na_4Ga_2Se_5$.



Figure S9. View of (*a*) Na₂GaSe₃ and (*c*) Na₄Ga₂Se₅ structures. Purple, blue, and yellow spheres and blue tetrahedra represent Na, Ga, and Se atom and GaSe₄ polyhedra, respectively. Na₂GaSe₃ structure consists of (*b*) the GaSe₃⁻ layers that are built with (*b*) the corner-shared-GaSe₄ chains connected through (*a*) diselenides with a 2.375 Å Se–Se bond distance;¹³ Na atoms are located between the GaSe₃²⁻ layers and the chains. Na₄Ga₂Se₅ structure is isostructural to Na₄Ga₂S₅¹⁴ and consists of (*d*) the Ga₂Se₅⁴⁻ ribbons constructed through (*d*) the corner-sharing between GaSe₄ tetrahedra and the edge-sharing Ga₂Se₆ dimer; Na atoms fill the voids between the Ga₂Se₅⁴⁻ ribbons.



Figure S10. EDS spectra of Na₂GaSe₃.



Figure S11. EDS spectra of Na₄Ga₂Se₅.



AB'C'D' AB'C'Figure S12. View of (a) AMQ_2 (KInS₂ structure type)¹⁵ and (c) (DEA)₃In₅S₉ (DEA = diethylammonium)¹⁶ structures. Purple, blue, and yellow spheres and blue tetrahedra represent A, M (Ga or In), Q atoms, and MQ_4 polyhedra, respectively. DEA ions were not determined from the SCXRD.¹⁶ We analyzed the stacking motifs presented in the literature for the T_n-corner-sharing square net layers for the In and Ga chalcogenide-based materials,¹⁷ and revealed only two stacking examples given in the literature: (b) A'B' in KInS₂ structure type and (d) A'B'C'D' for (DEA)₃In₅S₉ (DEA = diethylammonium) structure.^{15, 16, 18} Blue, purple, green, and yellow square-net present unique T_n-corner-sharing layers where a node is a T_n supertetrahedral cluster. Both A'B' and A'B'C'D' stacking differ from the AB and ABCD stacking found for [Na₂Cl]Ga Q_2 polymorphs (Figure 1). As DEA plays a template role in the (DEA)₃In₅S₉ structure formation, ClNa₆-based salt-inclusion templates novel stacking motifs for the GaS₂⁻ layers arrangement.



Figure S13. PXRD pattern for a product of reaction targeted $[Na_2Cl]GaS_2$ (black). Simulated pattern for *o*- $[Na_2Cl]GaS_2$ (red). Peak at 9.9° correspond to $NaGaS_2 \cdot H_2O$.¹⁹



Figure S14. PXRD pattern of experimental [Na₂Cl]GaS₂ (black) and simulated *o*-[Na₂Cl]GaS₂ (red). Blue lines correspond to peak positions of the simulated *t*-[Na₂Cl]GaS₂ pattern.



Figure S15. PXRD pattern of experimental t-[Na₂Cl]GaS₂ (black) and simulated t-[Na₂Cl]GaS₂ (blue) and o-[Na₂Cl]GaS₂ (red). Dash lines correspond to peak positions of the simulated o-[Na₂Cl]GaS₂ pattern that differ from the simulated t-[Na₂Cl]GaS₂ pattern.



Figure S16. PXRD pattern of experimental $[Na_2C1]GaSe_2$ (black) and simulated *o*- $[Na_2C1]GaSe_2$ (red). Blue lines correspond to simulated t- $[Na_2C1]GaSe_2$ pattern peak positions.



Figure S17. TGA(black)/DSC(red) plot of [Na₂Cl]GaS₂ in the 20–650 °C temperature range.



Figure S18. PXRD pattern of $[Na_2Cl]GaS_2$: before (black) and after TGA/DSC measurement in the 20–650 °C temperature range (red). A new peak at 20.4° can be seen in the post-heating sample.



Figure S19. TGA(black)/DSC(red) plot of [Na₂Cl]GaS₂ in the 20–900 °C temperature range.



Figure S20. PXRD pattern for samples of the [Na₂Cl]GaS₂ annealed and quenched at different temperatures (black). Simulated pattern for *o*-[Na₂Cl]GaS₂ (blue). Blue, pink, and green box highlight the most intense peaks of [Na₂Cl]GaS₂, NaGaS₂·H₂O, and NaCl phases, respectively.



Figure S21. (*top*) Contour plot 2D image of the PXRD patterns for $[Na_2Cl]GaS_2$ in the sealed quartz glass capillary as a function of temperature. The HT-PXRD patterns were collected under Mo K α irradiation. After 610 °C intensity of peaks corresponding to $[Na_2Cl]GaS_2$ reduces in comparison to the background, which may correlate with sample amorphization. The peak at ~4.1 ° is due to the capillary. (*bottom*) PXRD pattern of simulated *o*- $[Na_2Cl]GaS_2$ (red).

Table S5. DFT-calculated lattice constants,^{*} enthalpies of formation at 0 K, and decomposition enthalpies for $[A_2X]$ GaS₂ (A = Li, Na, and K; and X = F, Cl, and Br).

compound	a, Å	b, Å	<i>c</i> , Å	ΔH_f^{0K} , eV/atom	$\Delta H^{0K}_{decom}, kJ/mol$
o-[Na ₂ Cl]GaS ₂	7.37 (+1.1%)	39.78 (+0.9%)	7.37 (1.1%)	-1.41589	15.82
t-[Na ₂ Cl]GaS ₂	7.1	37 1%)	19.89 (+0.9%)	-1.41586	15.83
t-[Na ₂ F]GaS ₂	7.	20	19.04	-1.65	-24.9
<i>t</i> -[Na ₂ Br]GaS ₂	7	48	20.27	-1.32	10.0
t-[Li ₂ F]GaS ₂	7.	06	17.35	-1.66	-101.8
t-[Li ₂ Cl]GaS ₂	7.	13	18.24	-1.36	-57.3
t-[Li ₂ Br]GaS ₂	7.	20	18.63	-1.27	-53.1
t-[K ₂ F]GaS ₂	7.	35	21.25	-1.72	-30.6
t-[K ₂ Cl]GaS ₂	7.	76	22.04	-1.45	-38.7
t-[K ₂ Br]GaS ₂	7.	90	22.57	-1.34	-57.5

* - The values within the parentheses indicate errors in the lattice constants with respect to the experimental values.



Figure S22. UV-vis spectra of [Na₂Cl]GaS₂.



Figure S23. IR spectra of $[Na_2Cl]GaS_2$ after soaking in 0.001 M (red) and 0.01M (blue) UO_2^{2+} solution for 24 hours.

Table S6. EDS results for the single crystal of $[Na_2Cl]GaS_2$ soaked in 0.01 M UO₂²⁺ solution for 3 and 24 hours.

$[Na_2Cl]GaS_2(3h)_1$		
element	atom %	
Na	34.6(21)	
Cl	15.9(12)	
Ga	16.4(17)	
S	32.8(13)	
U	0.3(2)	

[Na ₂ Cl]GaS ₂ (3h)_2		
element	atom %	
Na	22.1(17)	
Cl	8.6(7)	
Ga	28.9(29)	
S	36.6(17)	
U	3.8(7)	



Figure S24. EDS spectra of $[Na_2Cl]GaS_2$ soaked in 0.01 M UO₂²⁺ solution for 3 hours. The insert shows the area chosen for EDS analysis.



Figure S25. EDS spectra of $[Na_2Cl]GaS_2$ soaked in 0.01 M UO₂²⁺ solution for 3 hours. The insert shows the area chosen for EDS analysis.



Figure S26. EDS spectra of [Na₂Cl]GaS₂ soaked in 0.01 M UO₂²⁺ solution for 24 hours.



Figure S27. SEM images of $[Na_2Cl]GaS_2$ single crystals in UO_2^{2+} solution.



Figure S28. PXRD pattern for samples of the $[Na_2Cl]GaS_2$ single crystals soaked in UO_2^{2+} solution for 24 hours (black). Simulated pattern for *o*- $[Na_2Cl]GaS_2$ (red).



Figure S29. PXRD pattern for samples of the $[Na_2Cl]GaS_2$ single crystals soaked in 0.01 M UO_2^{2+} solution for 24 hours (black). Simulated pattern for *o*- $[Na_2Cl]GaS_2$ with the preferred orientation along [040] (red). We were not able to assign peaks at 14.6°, 23.0°, 23.4°, and 27.6 ° to a specific phase. From the literature analysis of reports on uranyl exchange with layered chalcogenides, we found an example of reversible uranyl species intercalation between layers for K₂MnSn₂S₆ (KMS-1)²⁰ material. In KMS-1, Potassium ions were substituted by UO_2^{2+} , which resulted in an interlayer distance shrinkage from 8.5 Å to 7.4 Å. In the case of $[Na_2Cl]GaS_2$ material, the distance between GaS_2^- layers is significantly higher (9.9 Å vs. 8.5 Å), and uranyl species integration is likely to start structure degradation decomposition and results in an amorphous product.

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