





Figure 1s. Rietveld refinement of X-ray powder diffraction patterns of (a) 'Gd₃Tb₃FeSbTe', (b)'Tb₂Dy₄FeSb₂', (c) 'Gd₂Dy₄Fe_{0.75}Mn_{0.25}Sb₂', (d) 'Tb₂Dy₄Fe_{0.75}Mn_{0.25}Sb₂', (e) 'Gd₂Tb₄Fe_{0.75}Mn_{0.25}SbTe', (f) 'Tb₆Fe_{0.75}Mn_{0.25}SbTe', (g) 'Tb₆Co_{0.75}Mn_{0.25}SbTe', (h) 'Tb₆Ni_{0.75}Mn_{0.25}SbTe' and (i) 'Tb₆Ni_{0.5}Mn_{0.5}SbTe'. Inset in Figures b, c, d, e, g, h and i: microstructure of sample. The references of ticks are marked and compositions of alloys are given in Figures.



Figure 2s. Unit cell data of ternary Fe2P-type {Gd-Er}6FeTe2 tellurides: (a) cell parameter *c* vs cell parameter *a* and (b) *c/a* ratio vs unit cell volume *V* (replacing of rare earth leads to monotonic increasing of *a* and *c* cell parameters and unit cell volume *V* with slight decreasing of *c/a* ratio from Gd6FeTe2 to Er6FeTe2) and unit cell data of Ho-based Ho6{Fe, Mn}Sb2 antimonides, Ho6{Mn-Ni}Te2 tellurides and Ho6{Mn, Fe, Co}Bi2 bismuthides: (c) cell parameter *c* vs cell parameter *a* and (d) *c/a* ratio vs unit cell volume *V*. Despite the increasing of atomic radii from Fe to Mn and from Te to Bi (the sequence of atomic radii is given in Figures), the replacing of transition metals leads to increasing of *a* cell parameter and decreasing of *c* cell parameter, unit cell volume *V* and *c/a* ratio in the 'Mn-Fe-Co-Ni' row. The replacing of Sb for Bi leads to expansion of unit cell in *ab*-plane and its slight expansion along *c* axis with increasing of unit cell volume *V* and decreasing of *c/a* ratio, while replacing of Sb for Te corresponds to the expansion of unit cell in *ab*-plane and its compression along *c* axis with decreasing of unit cell volume *V* and *c/a* ratio. The areas of possible transformation of unit cell of the Ho6Fe1-xMnxSb2-y-zTeyBiz quasiternary solid solution are shown in Figures c and d.



Figure 3s. General trends of T_C temperature with *c* cell parameter in (Tb,Gd)₂Dy₄Fe_{1-x}Mn_xSb₂ (grey), Gd_{6-x}Tb_x(Fe,Co)_{1-y}Mn_ySbTe (red) and Tb₆Ni_{1-x}Mn_xSbTe (blue) sub-families. Number labels are the ones used in Tables 1, 2, 3, 4 and Table 1s.



Figure 4s. Arrot Plot representation of magnetization isotherms around T_C for Tb₆Fe_{0.75}Mn_{0.25}SbTe. Positive slopes are indicative of the second order nature of this transition.



Figure 5s. From left to right: Optimized Modified Arrot Plot, M vs. $\mu_0 H$ plot in logarithmic scale for the critical isotherm and representation of the magnetic equation of state for Gd₃Tb₃FeSbTe (top row), Tb₂Dy₄Fe_{0.75}Mn_{0.25}Sb₂ (middle row) and Gd₂Dy₄Fe_{0.75}Mn_{0.25}Sb₂ (bottom row).



Figure 6s. From left to right: Optimized Modified Arrot Plot, M vs. $\mu_0 H$ plot in logarithmic scale for the critical isotherm and representation of the magnetic equation of state for Tb₆Co_{0.75}Mn_{0.25}SbTe (top row), Tb₆Ni_{0.75}Mn_{0.25}SbTe (middle row) and Tb₆Ni_{0.5}Mn_{0.5}SbTe (bottom row).



Figure 7s. Thermal diffusivity (D) as a function of temperature around T_C for $Gd_3Tb_3FeSbTe$, $Tb_2Dy_4Fe_{0.75}Mn_{0.25}Sb_2$, $Gd_2Dy_4Fe_{0.75}Mn_{0.25}Sb_2$ and $Gd_2Tb_4Fe_{0.75}Mn_{0.25}SbTe$.



Figure 8s. Thermal diffusivity (D) as a function of temperature around T_C for Tb₆Co_{0.75}Mn_{0.25}SbTe, Tb₆Ni_{0.75}Mn_{0.25}SbTe and Tb₆Ni_{0.5}Mn_{0.5}SbTe.

N	Alloy	Phase	wt.%	Туре	<i>a</i> (nm)	b (nm)	<i>c</i> (nm)	$\mathbf{R}_{\mathbf{F}}$
1	'Gd ₃ Tb ₃ FeSbTe'	Gd ₃ Tb ₃ FeSbTe ^{a-}	~100	Fe ₂ P	0.82712(2)		0.41051(1)	3.6
2	'Tb ₂ Dy ₄ FeSb ₂ '	Tb ₂ Dy ₄ FeSb ₂	~95	Fe ₂ P	0.81481(5)		0.41703(2)	4.0
		${\sim}Tb_{1.7}Dy_{3.3}Sb_3$	~5	Mn ₅ Si ₃	0.8979(6)		0.6230(3)	5.8
3	'Gd ₂ Dy ₄ Fe _{0.75} Mn _{0.25} Sb ₂ '	$Gd_2Dy_4Fe_{0.75}Mn_{0.25}Sb_2$	~94	Fe ₂ P	0.81646(6)		0.41991(3)	4.2
		$\sim Gd_{1.8}Dy_{3.2}Sb_3$	~6	Mn ₅ Si ₃	0.8937(4)		0.6289(3)	6.2
		$\sim Gd_{0.34}Dy_{0.66}$ b-	< 1	Mg				
4	'Tb ₂ Dy ₄ Fe _{0.75} Mn _{0.25} Sb ₂ '	$Tb_2Dy_4Fe_{0.75}Mn_{0.25}Sb_2$	~96	Fe ₂ P	0.81478(6)		0.41866(3)	4.8
		$\sim Tb_{1.7}Dy_{3.3}Sb_3$	~4	Mn ₅ Si ₃	0.8990(9)		0.6251(4)	4.6
5	'Gd ₂ Tb ₄ Fe _{0.75} Mn _{0.25} SbTe'	$Gd_2Tb_4Mn_{0.25}Fe_{0.75}SbTe$	~96	Fe ₂ P	0.82577(5)		0.41265(3)	5.6
		$Gd_{0.33}Tb_{0.67}$	~3	Mg	0.35421(9)		0.5871(2)	4.3
		$Gd_{1-x}Tb_xSb_{0.5}Te_{0.5}$	~1	NaCl	0.6159(3)			3.9
		Gd _{20.8} Tb _{46.6} Sb _{0.3} Te _{32.3} b-	<1	Sc ₂ Te				
6	'Tb ₆ Fe _{0.75} Mn _{0.25} SbTe'	Tb ₆ Fe _{0.75} Mn _{0.25} SbTe ^{a-}	~100	Fe ₂ P	0.82337(3)		0.41179(1)	4.2
7	' Tb ₆ Co _{0.75} Mn _{0.25} SbTe'	Tb ₆ Co _{0.75} Mn _{0.25} SbTe	~95	Fe ₂ P	0.82547(6)		0.40704(2)	3.7
		Tb	~3	Mg	0.3607(2)		0.5650(3)	3.1
		Tb ₂ Te	~2	Sc ₂ Te	2.1988(13)	0.3965(2)	1.1536(6)	4.2
8	'Tb ₆ Ni _{0.75} Mn _{0.25} SbTe'	Tb ₆ Ni _{0.75} Mn _{0.25} SbTe	~94	Fe ₂ P	0.82927(7)		0.40245(3)	3.9
		Tb	~2	Mg	0.3597(2)		0.5612(4)	3.4
		Tb ₂ Te	~4	Sc ₂ Te	2.1815(12)	0.3971(2)	1.1578(4)	3.7
9	'Tb ₆ Ni _{0.5} Mn _{0.5} SbTe'	Tb ₆ Ni _{0.5} Mn _{0.5} SbTe	~95	Fe ₂ P	0.82840(7)		0.40553(3)	5.4
		$Tb_{64.3}Sb_{21.3}Te_{14.4}$	~4	Mn ₅ Si ₃	0.9292(8)		0.5892(3)	6.6
		$Tb_{49.7}Sb_{21.1}Te_{29.2}$	~1	NaCl	0.6135(8)			4.0

 Table 1s. Alloy's content and unit cell data of phases.

^{a-}Crystallographic data used with permission of JCPDS - International Centre for Diffraction Data (USA), ^{b-} from EDS analysis, only.

Table 2s. Theoretical magnetic critical exponents for several universality models. d dimensionality of the interaction, N number of spin components, β of spontaneous

Universality class	d	Ν	β	γ	δ
Mean-field			0.5	1.0	3.0
2D-Ising	2	1	0.125	1.75	14
3D-Ising	3	1	0.3265	1.237	4.79
3D-XY	3	2	0.348	1.317	4.78
3D-Heisenberg	3	3	0.369	1.396	4.78
Chiral XY	3	2	0.253	1.13	5.47
Chiral-Heisenberg	3	3	0.30	1.17	4.90
Tricritical Mean			0.25	1.0	5.00
Field					
2D-Long range	2	1	0.298	1.393	5.67
3D-Ising spin glass	3	1	0.5	2.9	6.80

magnetization, γ of isothermal susceptibility, δ of critical isotherm [48-50, 52-57].