

1 **The role of oxygen defects on the Electronic, Optical and Phonon Dispersion of LAGO**

2 **Perovskite: A Density Functional Investigation**

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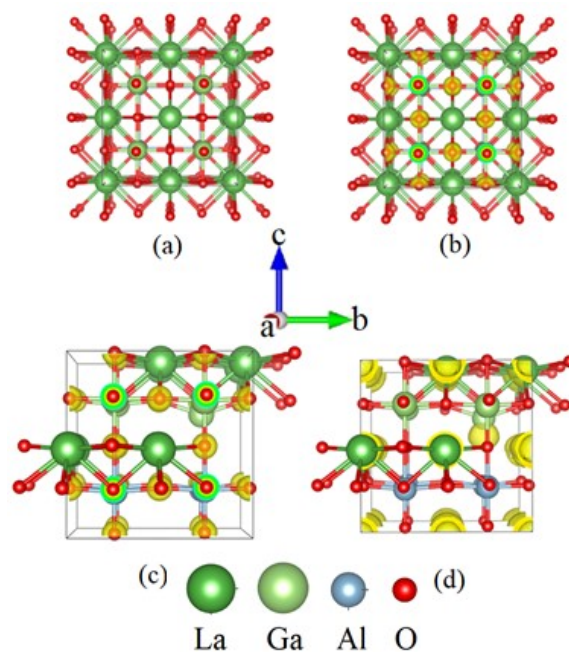
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7 **Supplementary Information**

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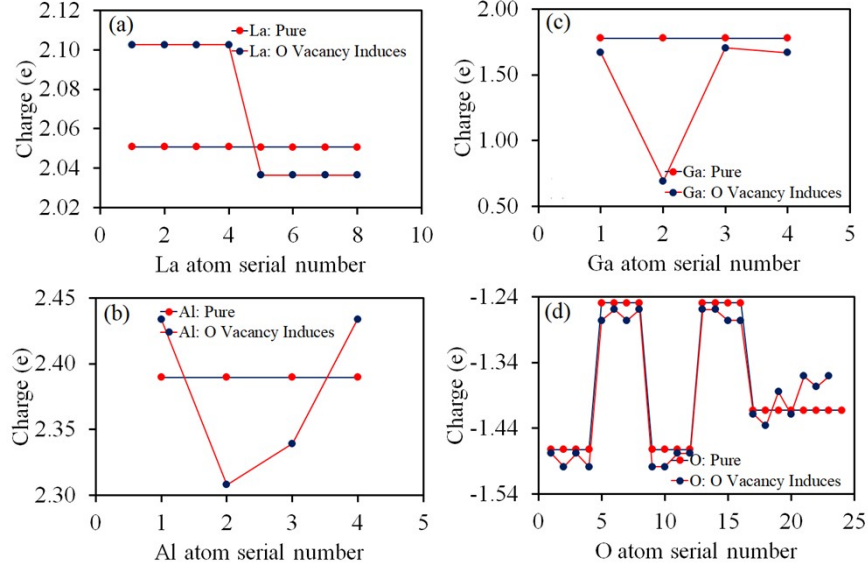
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10 **SI Fig. 1. (a) Structure of $\text{La}_8\text{Al}_4\text{Ga}_4\text{O}_{24}$ supercell, (b) Charge density distribution on the supercell, (c) Charge**
11 **density difference between pure and Oxygen vacancy induced supercell, (d) The yellow spheres is the ELF**
12 **charge at the vacancy sites of the supercell.**

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17 **SI Fig. 2. Bader charges on different atoms. (a) La, (b) Al, (c) Ga, and (d) O.**

18 **SI Table 1: Bader charge on different elements of $\text{La}_8\text{Al}_4\text{Ga}_4\text{O}_{24}$ supercell**

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Elements	Number	Bader charge	Bader charge	Elements	Number	Bader charge	Bader charge
		Pure	O Vacancy			Pure	O Vacancy
La1	1	2.051	2.102	O5	5	-1.25	-1.276
La2	2	2.051	2.102	O6	6	-1.25	-1.260
La3	3	2.051	2.102	O7	7	-1.25	-1.276
La4	4	2.051	2.102	O8	8	-1.25	-1.259
La5	5	2.051	2.037	O9	9	-1.473	-1.499
La6	6	2.051	2.037	O10	10	-1.473	-1.499
La7	7	2.051	2.037	O11	11	-1.473	-1.478
La8	8	2.051	2.037	O12	12	-1.473	-1.478
Al1	1	2.39	2.434	O13	13	-1.25	-1.260
Al2	2	2.39	2.308	O14	14	-1.25	-1.260
Al3	3	2.39	2.339	O15	15	-1.25	-1.276
Al4	4	2.39	2.434	O16	16	-1.25	-1.276
Ga1	1	1.778	1.669	O17	17	-1.413	-1.418
Ga2	2	1.778	0.694	O18	18	-1.413	-1.436
Ga3	3	1.778	1.704	O19	19	-1.413	-1.385
Ga4	4	1.778	1.669	O20	20	-1.413	-1.418
O1	1	-1.473	-1.478	O21	21	-1.413	-1.361
O2	2	-1.473	-1.498	O22	22	-1.413	-1.377
O3	3	-1.473	-1.478	O23	23	-1.413	-1.361
O4	4	-1.473	-1.498	O24	24	-1.413	

37 **Phonon Dynamics**

38 It is known that the phonons are quantized modes of lattice vibrations in transverse and longitudinal modes. When
39 two or more charged particles in a primitive cell move in opposite directions with the center of mass at rest, the result
40 is optical phonons. This model has the highest energy at the gamma point of the lattice. Acoustic phonons are
41 associated with the propagation of the longitudinal lattice vibration mode. They are low-energy vibrations due to the
42 collective motion of the crystal lattice's relatively heavier atoms (metallic). The energy of the acoustic phonons is
43 minimal at the gamma point. Therefore, the curvature of the acoustic phonons and the optical phonons at the gamma
44 point are opposite. The frequency of the acoustic phonons is directly proportional to their wave vector of oscillations.
45 In three-dimensional solids, the two optical branches and one acoustic phonon bunch are related to the transverse and
46 longitudinal modes. The optical phonon frequencies fall in the IR or visible light region. The leading cause of the
47 generation of these high-frequency optical phonons is due to the stretching or bending mode in lattice oscillations.
48 Usually, the optical phonons have a non-linear dispersion relation, which means the frequency is not directly
49 proportional to the wave vector of the oscillation modes. In three-dimensional solids, there can be multiple branches
50 of optical phonons. The distinction between acoustic and optical phonons is primarily based on their energy and the
51 type of lattice vibrations involved. Since thermal conductivity is due to the vibration of atoms in the solids, the acoustic
52 phonons contribute considerably to the thermal properties, such as thermal conductivity and heat capacity of the
53 materials.

54 On the other hand, optical phonons play a crucial role in various optical phenomena of the materials, i.e., absorption
55 and scattering of light and interactions with the bound electrons in materials. Therefore, studying acoustic and optical
56 phonon bands may give valuable information about the materials' thermal, mechanical, and electronic properties.

57 The present study's calculated phonon bands (**Fig. 4**) are discontinuous at most Brillouin zone boundaries. The lattice
58 vibration affects the phonon dispersion and the thermal conductivity at different directions of the crystal lattice,
59 especially at the Brillouin zone boundaries. The thermal conductivity can be higher along specific crystallographic
60 directions with steeper phonon dispersion curves between Brillouin zone boundaries (mainly, about Γ point) and lower
61 along the other direction where the curves have lower slopes. It is to be noted that the phonon bands (in the acoustic
62 region) at the Gamma point in the crystal lattice are symmetrical, and the slope is steeper towards the k-path
63 boundaries. Hence, in the region of the 1st Brillouin zone, the thermal conductivity would be maximum and uniform
64 in all directions. Compared to the pure LAGO, the slopes in the exact Brillouin zone boundaries in Oxygen vacancy-

65 induced LAGO are less. Still, at the same time, this is more widely distributed in the other direction in the K-path.
66 However, the slope varies in different Brillion zone boundaries along the K-path. Concerning **Fig. 4**, careful
67 observation shows that the bands are continuous in the low-frequency region, and within the Brillion zone
68 $\Gamma \rightarrow M \rightarrow X \rightarrow \Gamma \rightarrow R \rightarrow X$ and $R \rightarrow M$, the bands are continuous.
69 Hence, the dispersion is less. A nondegenerate mode has been found at the X point. It stiffens up rapidly away from
70 the X point. It indicates that neighboring chains of oxygen atoms displace in opposite directions. The acoustic phonon
71 spectrum extends to the phonon frequency close to 10 THz in both systems. At the boundary, the bands degenerate
72 and follow a different curvature. The branches are distributed uniformly up to 20 THz along all three directions.
73 However, the phonon band densities differ in pure and vacancy-induced LAGOs. In vacancy-induced LAGO, the
74 optical regions moved downward with a small value of phonon band gap, which is very clear from the corresponding
75 PDOS plots. In the PDOS, the acoustic phonon contributions are mainly from the La, Al, and Ga. The optical PDOS
76 and, hence, the optical phonon bands are due to the oxygen atoms. In the phonon dispersion and PDOS spectrum,
77 the optical modes are divided into two distinct groups, corresponding to the low-frequency and high-frequency modes.
78 The peak corresponding to the high-frequency modes is broader and has a higher intensity than those corresponding
79 to the low-frequency modes. In the vacancies-induced LAGO, the optical phonon bands and PDOS distribution are
80 less wide than the pure LAGO. A clear band appears above 20 THz in the vacancy-induced LAGO, possibly due to
81 the color centers created due to the vacancies.

82 **Thermodynamic Properties**

83 We calculated the Debye temperature from the following equation, where the Debye temperature has a significant role
84 in the variation of C_v with temperature [1]. A detailed discussion of the thermodynamic properties is given in the
85 supplementary materials.

$$86 \quad C_v = 9Nk_B \left(\frac{T}{\Theta_D} \right)^3 \int_0^{x_D} \frac{x^4 \cdot e^x \cdot dx}{(e^x - 1)^2}; \text{ where, } x_D = \frac{T}{\Theta_D}; x = \frac{\hbar\omega}{k_B T}; k_B = \text{Boltzmann constant}, \Theta_D = \text{Debye Temperature}$$

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88 The calculated Debye temperatures for these two systems are 469.92 K and 463.69 K, respectively. The higher the
89 melting point of a substance, the more substantial the atomic bonding force and the higher the Debye temperature.
90 Due to inducing defects, the system bond strengths become weaker and are expected to reduce the melting point. From
91 the calculated values of Debye temperature, the lower value in the induced system supports this. It is known that the

92 degrees of freedom in lattice vibrations play an essential role in the thermodynamic parameters compared to the
 93 electronic degrees of freedom. The temperature dependence of the Helmholtz free energy (F), entropy (S), and
 94 constant-volume heat capacity (C_v) are calculated within the limit of harmonic approximation following the
 95 expression reported earlier.

$$F(T) = E_{Total} + \frac{1}{2} \int g(\omega) \hbar \omega d\omega + k_B T \int g(\omega) \ln \left(1 - e^{-\frac{\hbar \omega}{k_B T}} \right)$$

$$96 \quad S(T) = k_B \left[\int \frac{\frac{\hbar \omega}{k_B T}}{e^{\frac{\hbar \omega}{k_B T}} - 1} g(\omega) d\omega - \int g(\omega) \ln \left(1 - e^{-\frac{\hbar \omega}{k_B T}} \right) d\omega \right],$$

97 As shown in Fig. 5, in pure and vacancy-induced systems, the Gibbs free energy decreases with the increase in
 98 temperature. The Free energy is zero at 321K and 331K in pure and vacancy systems, respectively. Zero Gibb's Free
 99 energy indicates the equilibrium temperature of the system. A small change in the equilibrium temperature between
 100 the pure and vacancy-induced systems also supports the structural change in the system due to vacancies. On the other
 101 hand, the entropy and heat capacity increase from zero per the third law of thermodynamics. The rate of change of
 102 these two parameters is slightly higher in a pure system. It is to be noted that below the Debye temperature, the nature
 103 of the variation of C_v is different in the two systems. The tendency to reach saturation value is faster in vacancy-
 104 induced systems than in pure systems. At lower temperatures (much lower than the Debye temperatures), the heat
 105 capacity C_v is proportional to T³. The contribution in lattice heat capacity at low temperatures is from the lattice's low
 106 frequency or long wavelength acoustic phonon vibration modes. However, C_v tends to the Dulong–Petit classical limit
 107 at high temperatures.

108 **References:**

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 110 <https://doi.org/10.1103/RevModPhys.73.515>

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