# Supporting Information:

## High-pressure observation of elusive iodoplumbic acid in different hydroniumhydrate solid forms

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#### **General microscopy**

For observing and controlling the growth of compounds **3**, **4** and  $\beta$ -Pbl<sub>2</sub> single crystals, Olympus MVX10 microscope, equipped with MV PLAPO 1xFF objective was used, with a total optical magnification ranging from 4× to 80×, equipped with an Olympus Digital Camera SC 30/MVX-TV1XC. All crystal dimensions were measured with Olympus cellSens imaging software.

#### **Pseudosymmetry in 1**

Similarly as in SiC polytypes,<sup>10</sup> at least two structure can be found for the layered structure of **1**. Several independent single-crystal diffraction experiments have been made to identify the correct space group. Depending on the shift in the position of water molecules, occupying the interlayer space and arranged in a honeycomb formation, two different symmetries can be refined: monoclinic space group C2/m (mC) and trigonal space group R-3m (hR). However, the careful analysis of systematic absences suggests that the crystal belongs to mC group, which can be confused with the strained hexagonal lattice. This strain causes broadening of reflections along  $c^*$ , during the transformation from mC to hR leads to differentiation in the unit-cell parameters of about 0.04 Å and described by the matrix:

(-1/2)	0.5	$0 \langle a_c \rangle$	$(a_r)$
0	- 1	$0 \  b_c \ $	$=  b_r $
\ 1.5	0	$3/(c_c)$	$\langle c_r \rangle$

Where c and r indexes indicate *m*C to *h*R lattices, respectively.

## **High-pressure experiments**

For each high-pressure experiment a liquid solution of PbI<sub>2</sub> dissolved in concentrated aqueous hydroiodic acid (HI, 57% by weight) was loaded to a modified Merrill-Bassett diamond anvil cell (DAC).<sup>1</sup> The DAC anvils were supported directly on the steel discs with conical windows (the culet size was 0.8 mm, type 1A diamonds, a steel gasket 0.15 mm thick, the hole diameter 0.45 mm).<sup>2</sup> Pressure in the DAC was determined by the ruby fluorescence (R1 ruby line) shift with a Photon Control spectrometer affording an accuracy of 0.02 GPa.<sup>3,4</sup>

The single-crystals of **3** and **4** were obtained in isochoric conditions: after the polycrystalline mass precipitated, the DAC with was heated using a hot-air gun till all but one grain melted. Then the single crystal grew as the DAC was cooled slowly to room temperature. Temperature inside DAC was determined using a dual-method approach: a laser thermometer directed at the diamond anvil's surface and a thermocouple mounted directly on the anvil. The high thermal conductivity of diamond ensured that the laser thermometer readings provide a reliable estimation of the DAC's internal temperature, which is cross-verified with the thermocouple measurements. The progress and experimental details on growing the single crystals of **3** and **4** are shown in Figures S1 and S2.



**Figure S1**. Single crystal of **3** grown *in situ* in a DAC in isochoric conditions at (a) 350 K; (b) 320 K and; (c) 300 K / 0.11 GPa. The ruby sphere for pressure calibration lies at the upper-right edge of the chamber



**Figure S2.** Single crystal of **4** grown *in situ* in a DAC in isochoric conditions at (a) 420 K; (b) 360 K and; (c) 300 K / 2.63 GPa. The ruby sphere for pressure calibration lies at the lower-edge of the chamber.

Single crystals of  $\beta$ -Pbl<sub>2</sub> were obtained similarly to those of **3** and **4**, however due to the phase transformation at room temperature at 1.2 GPa, the single crystal grown at 2.05 GPa was kept at 320 K and the X-ray diffraction experiment was performed at this temperature. Temperature inside DAC was determined using a dual-method approach: a laser thermometer directed at the diamond anvil's surface and a thermocouple mounted directly on the anvil. The high thermal conductivity of diamond ensured that the laser thermometer readings provide a reliable estimation of the DAC's internal temperature, which is cross-verified with the thermocouple measurements.



**Figure S3.** Isochoric growth (a-d) of the pink single-crystal of  $\beta$ -Pbl<sub>2</sub> in the DAC at 2.05 GPa. The  $\beta$ -Pbl<sub>2</sub> crystals display a reverse solubility phenomena resulting in crystal growth upon heating from 350 to 380 K.

Diffraction data were collected at 295 K for **1**, **3** and **4** and at 320 K for  $\beta$ -Pbl<sub>2</sub>, by using a KM-4 CCD diffractometer with the graphite-monochromated MoK $\alpha$  radiation. The DAC was centered by the gasket-shadow method.<sup>5</sup> The CrysAlisCCD and CrysAlisRED programs<sup>6</sup> were used for collecting the data, determination of the UB-matries, and for initial data reductions and Lp corrections. Reflection intensities were corrected for the DAC and sample absorption; the gasket shadowing and the reflections of diamond-anvils were eliminated.<sup>7</sup> All structures were solved by direct methods, and refined with anisotropic displacement parameters, with programs ShelXS and ShelXL<sup>8</sup> using Olex2 interface.<sup>9</sup> Details of structure refinements and crystal data are given in Table S1.

## Table S1. Detailed crystallographic data for 1, 3, 4 and $\beta$ -PbI<sub>2</sub>

Compo	ound	<b>1</b> (H <sub>3</sub> O) <sub>2-x</sub> Pb <sub>x</sub> I <sub>2</sub> ·(2-2x) H <sub>2</sub> O	<b>3</b> (H₃O)Pbl₃·4H₂O	<b>3</b> (H₃O)Pbl₃·4H₂O	<b>3</b> (H₃O)Pbl₃·4H₂O	<b>4</b> (H₃O)Pbl₃·3H₂O	$\beta$ -Pbl <sub>2</sub>
CSD Nu	umber	2071140	-	2071141	2071142	2071143	2071144
Press	sure	0.1 MPa	0.20 GPa	0.50 GPa	1.20 GPa	2.63 GPa	2.05 GPa
Temper	rature	300 K	300 K	300 K	300 K	300 K	320 K
Formula	weight	227.79	667.89	667.89	667.89	651.89	691.48
Wavelen	gth (Å)	0.71073	0.71073	0.71073	0.71073	0.71073	0.71073
Crystal s	system	monoclinic	monoclinic	monoclinic	monoclinic	monoclinic	monoclinic
Space	group	C2/c	l2/m	l2/m	l2/m	P2 <sub>1</sub> /m	C2/m
Unit cell dimensions	<i>a</i> (Å)	4.5598(3)	16.92(14)	16.205(2)	16.119(13)	9.57(3)	14.06(4)
	b (Å)	11.1985(18)	4.4666(7)	4.51695(14)	4.5261(9)	4.513(3)	4.4260(12)
Unit cell angles	c (Å)	90	17.31(2)	17.184(14)	16.88(4)	12.98(11)	10.540(6)
	α (°)	118.023(19)	90.0	90	90	90	90
	β (°)	90	112.0(4)	111.50(4)	111.25(17)	95.9(6)	93.08(12)
	γ (°)	355.86(9)	90.0	90	90	90	90
Volume	e (Å <sup>3</sup> )	1068.10(14)	1213(10)	1170.3(10)	1147(3)	558(5)	654.9(17)
Z/2	<u>_</u> '	4/0.5	4/0.5	4/0.5	4/0.5	2/0.5	2/0.25
Molecular vo	lume ( <i>V/Z</i> )	267.025	303.25	292.575	286.75	279	327.45
Calculated (g/cr	l density n³)	4.252	3.657	3.791	3.866	3.883	7.013
Absorption	n (mm <sup>-1</sup> )	27.566	24.068	22.303	22.746	23.395	52.550
F(00	00)	383.0	-	1124.0	1124.0	546.0	1128.0
Crystal siz	ze (mm)	0.1 × 0.1 × 0.05	0.445 × 0.127 × 0.068	0.32 × 0.101 × 0.065	0.30 × 0.10 × 0.06	0.30 × 0.20 × 0.15	0.30 × 0.10 × 0.10
2θ-range collecti	for data on (°)	8.246 to 57.384	7.0648 to 49.3486	5.404 to 52.382	8.758 to 51.294	4.28 to 55.116	3.87 to 52.736
Min/max ind	ices: <i>h, k, l</i>	-10 ≤ h ≤ 10, -6 ≤ k ≤ 6, -15 ≤ l ≤ 15	-4 ≤ h ≤ 4, -5 ≤ k ≤ 5, -19 ≤ l ≤ 19	-19 ≤ h ≤ 19, -5 ≤ k ≤ 5, -5 ≤ l ≤ 5	-16 ≤ h ≤ 16, -5 ≤ k ≤ 5, -7 ≤ l ≤ 8	-10 ≤ h ≤ 8, -5 ≤ k ≤ 5, -8 ≤ l ≤ 6	-8 ≤ h ≤ 8, -5 ≤ k ≤ 5, -12 ≤ l ≤ 12
Reflect. Colle	cted/unique	3208/523	8331/-	6503/335	1022/323	788/287	2000/258
Rir	nt	0.0541	0.2558	0.0592	0.1061	0.1665	0.2469
Refinemen	t method			Full-matrix leas	st-squares on F <sup>2</sup>		
Completer	ness (%)	99.7	25.8	39.5	36.6	23.3	42.7
Data/restrains	/parameters	523/0/21	-	335/0/42	323/6/42	287/39/38	258/17/29
Goodness-o	of-fit on F <sup>2</sup>	1.019	-	0.857	1.035	0.962	1.067
Final R <sub>1</sub> /wF	R <sub>2</sub> (I>2σ1)	0.0364 /0.0794	-	0.0329/0.0832	0.0773/ 0.1686	0.0851/0.2064	0.0818/0.1946
R <sub>1</sub> /wR <sub>2</sub> (all data)		0.0492 /0.0877	-	0.0352/0.0853	0.1013/0.1834	0.2011/0.2788	0.1309/0.2211
Largest diff. peak/hole (e.Å <sup>-3</sup> )		2.72/-0.75	-	0.66/-0.55	0.88/-1.01	0.96/-1.02	2.77/-1.63

 $w=1/(\sigma^2 F_o^2 + w_1^2 * P^2 + w_2 * P)$ , where  $P=(Max(F_o^2, 0) + 2 * F_c^2)$ 

## **Ball-Milling experiments**

### Milling Synthesis of $[H_3O]_{2x}[Pb_{1-x}I_2] \cdot (2-2x)H_2O(1)$

For the ball-milling synthesis, 0.461 g (1 mmol) of PbI<sub>2</sub> was added to one half of zirconia (ZrO<sub>2</sub>) 10 mL jars, followed by the addition of 67  $\mu$ L (0.5 mmol) of aqueous hydroiodic acid (HI 57%) and one 3.5 g zirconia ball. The jar was carefully sealed and placed on a MM400 mixer mill and the reaction mixture was milled for 30 minutes at an oscillation rate of 30 Hz. Reaction completion afforded a bright yellow product, which was left to dry in dark for 1h. The resulting solid product was scraped off the jar walls and subjected to powder X-ray diffraction (PXRD) analysis, revealing the formation of compound **1**. Compound **1** can be also isolated when performing the 1:1 stoichiometric reaction by milling 0.230 g (0.5 mmol) of PbI<sub>2</sub> with 67  $\mu$ L (0.5 mmol) of aqueous hydroiodic acid (HI 57%), respectively.

## <u>Milling Synthesis of $(H_3O)_2Pb_3I_8 \cdot 6H_2O$ (2)</u>

For the ball-milling synthesis 0.230 g (0.5 mmol) of Pbl<sub>2</sub> was added to one half of zirconia (ZrO<sub>2</sub>) 10 mL jars, followed by the addition of 201  $\mu$ L (1.5 mmol) of aqueous hydroiodic acid (HI 57%) and one 3.5 g zirconia ball. The jar was carefully sealed and placed on a MM400 mixer mill and the reaction mixture was milled for 30 minutes at an oscillation rate of 30 Hz. Reaction completion afforded a bright yellow product, which was left to dry in dark overnight. The resulting product was scraped off the jar walls and subjected to powder X-ray diffraction (PXRD) analysis, revealing the formation of compound **2**.



**Figure S4**. Powder X-ray diffraction patterns: a) experimental pattern of  $PbI_2$  (Sigma Aldrich) b) simulated pattern of  $[H_3O]_{2x}[Pb_{1-x}I_2] \cdot (2-2x)H_2O(1)$  and c) experimental pattern of the milling product from the 1.5PbI<sub>2</sub>:HI stoichiometric reaction.



**Figure S5.** Powder X-ray diffraction patterns: a) experimental pattern of  $PbI_2$  (Sigma Aldrich) b) simulated pattern of  $(H_3O)_2Pb_3I_8\cdot 6H_2O$  (**2**) and c) experimental pattern of the milling product from the  $PbI_2$ :3HI stoichiometric reaction.



**Figure S6.** The ORTEP drawing of  $(H_3O)_2$ -xPb<sub>x</sub> $I_2$ ·(2-2x) $H_2O$  (1) crystal structure collected at ambient pressure.



**Figure S7.** The ORTEP drawing of  $(H_3O)PbI_3 \cdot 4H_2O$  (**3**) crystal structure determined at 0.5 GPa.



**Figure S8.** The ORTEP drawing of  $(H_3O)PbI_3 \cdot 4H_2O$  (**3**) crystal structure observed at 1.2 GPa.



**Figure S9.** The ORTEP drawing of  $(H_3O)PbI_3 \cdot 3H_2O$  (4) crystal structure collected at 2.63 GPa



**Figure S9.** The ORTEP drawing of the  $\beta$ - PbI<sub>2</sub> crystal structure obtained at 2.05 GPa.



**Figure S10.** Histogram of Pb(II)–I bond distances in 6-coordinate complexes in crystals deposited in the Cambridge Structural Database (November 2021 release).<sup>11</sup>

## References

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