## Supporting Information for

# A Nonsolvolytic Fluorine/LiNO<sub>3</sub>-containing Electrolyte for Stabilizing Dynamic Interfaces in Li||LiMn<sub>2</sub>O<sub>4</sub> Batteries

Tian Tang<sup>1</sup>, J.X. Kent Zheng<sup>2</sup>, Nyalaliska W. Utomo<sup>3</sup>, Lynden A. Archer<sup>3\*</sup>

1. Department of Materials Science and Engineering, Cornell University, Ithaca, NY, 14853, USA.

2. Department of Physics, Massachusetts Institute of Technology, Cambridge, MA, 02139, USA.

3. Robert Frederick Smith School of Chemical and Biomolecular Engineering, Cornell University, Ithaca, NY, 14853, USA.

\* Corresponding author: <u>laa25@cornell.edu</u>

### Materials and methods

#### **Materials**

LiPF<sub>6</sub> and lithium bis(trifluoromethanesulfonyl)imide (LiTFSI), LiClO<sub>4</sub>, LiMn<sub>2</sub>O<sub>4</sub> were purchased from Sigma Aldrich. LiBF<sub>4</sub>, LiNO<sub>3</sub>, 750  $\mu$ m lithium foil were purchased from Alfa Aesar. Plain carbon cloth 1071 was purchased from Fuel Cell Store. Ketjen Black carbon was purchased from AkzoNobel. The cathodes are fabricated according to the protocol we reported in an earlier work (LiMn<sub>2</sub>O<sub>4</sub> content = 80 wt%).<sup>1</sup> Areal mass loading of the cathodes is ~3 mg/cm<sup>2</sup>.

#### **Characterization of materials**

Scanning electron microscopy (SEM) was carried out on Zeiss Gemini 500 Scanning Electron Microscope equipped with Bruker energy dispersive spectroscopy (EDS) detector. Scanning transmission electron microscopy analyses were performed on a Titan Themis Z G3 Cs-Corrected S/TEM system at MIT.nano Characterization Facility. X-ray diffraction patterns were collected on a Rigaku SmartLab system with a Cu K $\alpha$  incident irradiation. Synchrotron scattering is performed at 5-BM-C at Advanced Photon Source, Argonne. The electrolyte conductivity was measured through dielectric relaxation spectroscopy (DRS). Differential Scanning Calorimetry (DSC) was adopted to evaluate thermal transitions on TA Instruments Q2000. The thermal transitions were measured under nitrogen flow at a fixed ramp rate of 10 °C/min.

#### **Electrochemical measurements**

Galvanostatic charge/discharge performance of coin cells were tested on Neware battery test systems at room temperature at 1C during 3.0-4.3 V (vs. Li/Li<sup>+</sup>) unless specified. Electrochemical studies were performed using CR2032 coin cells. The area of electrodes in this study is 0.71 cm<sup>2</sup>. Electrodes are separated by Celgard 3501. ~100 µL electrolyte is added into each cell using pipette. Electrolytes were prepared by mixing 1 M of a Li salt, 0.5 M LiNO<sub>3</sub> and ethylene carbonate. The coin cell (CR2032) was assembled in an argon-filled glove box. In a lithium plating/stripping Coulombic efficiency measurement (using a Li||Cu cell), 1 mAh/cm<sup>2</sup> lithium metal is plated on the substrate from the electrolyte of interest at 1mA/cm<sup>2</sup>. **Coulombic efficiency** =  $\frac{stripping capacity}{plating capacity} \times 100\%$ .



**Supplementary Figure 1**. Voltage profile of a Li||LiMn<sub>2</sub>O<sub>4</sub> cell with LiPF<sub>6</sub>-based electrolyte charged at 0.1C. This small current density means that any electrolyte decomposition can contribute a faradic current on this order of magnitude. In this case, the faradic current for any electrolyte decomposition can contribute a larger amount of the total current density compared with using a large current density. Considering the absence of corrosion features upon visual examination and the fact that LiMn<sub>2</sub>O<sub>4</sub> is known to be catalytic, we tend to attribute that endless charging plateau to electrolyte decomposition.



**Supplementary Figure 2**. STEM-EDS characterization of the electrode material. The HAADF image shown on the left and the EDS mapping shown on the right demonstrate a uniform dispersion of LiMn<sub>2</sub>O<sub>4</sub> particles within a percolated carbon network (ketjen black).



**Supplementary Figure 3**. High magnification STEM characterization of the defects found in  $LiMn_2O_4$  cycled in  $LiPF_6$  electrolyte. Clearly, the images show an edge dislocation with an extra (111) half plane.



**Supplementary Figure 4**. HAADF-STEM image of LiMn<sub>2</sub>O<sub>4</sub> cycled in LiTFSI electrolyte. The particle shows a clean, well-defined (111) terminated facet.



**Supplementary Figure 5**. Synchrotron diffraction patterns LiMn<sub>2</sub>O<sub>4</sub> electrodes. Patterns of the LiMn<sub>2</sub>O<sub>4</sub> electrodes before cycling (black) and after 200 cycles (red), respectively.



**Supplementary Figure 6**. Evaluation of Li plating/stripping in the LiClO<sub>4</sub>-based EC electrolyte. (A) CE values over cycling. (B) Representative potential profile a plating/stripping cycle.



**Supplementary Figure 7**. Evaluation of Li plating/stripping in the LiPF<sub>6</sub>-based EC electrolyte. (A) CE values over cycling. (B) Representative potential profile a plating/stripping cycle.



**Supplementary Figure 8**. Evaluation of Li plating/stripping in the LiTFSI-based EC electrolyte. (A) CE values over cycling. (B) Representative potential profile a plating/stripping cycle.



**Supplementary Figure 9.** DSC curve of LiTFSI-based EC electrolyte. The thermal history is removed by doing a ramp up to 70 °C, then cooling to -120 °C and having a subsequent heating to 70 °C. The data shown is the second cooling and third heating. The crystallization loop is observed and this artifact is due to the large exotherm of EC crystallization. The competition between self-heating and crystallization process leads to the loop. The melting temperature Tm obtained from the curve is 27.7 °C.



**Supplementary Figure 10**. Temperature-dependent conductivities of LiTFSI-based EC electrolyte probed through DRS. The ionic conductivities of the LiTFSI-based electrolyte show no clear inflection of the conductivities curve to at least 60 °C. This proves there is no decomposition of LiTFSI in this controlled atmosphere.

#### References

1 Zheng, J. *et al.* Nonplanar Electrode Architectures for Ultrahigh Areal Capacity Batteries. *ACS Energy Letters* **4**, 271-275 (2018).