## Carbon layer-confined MoS<sub>2</sub>/Ni<sub>3</sub>S<sub>2</sub> heterostructure with

## enhanced sodium and potassium storage performance

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## **Experimental section**

*Materials synthesis*: Firstly, the NiMoO<sub>4</sub> precusor material was synthesized by a simple one-pot hydrothermal process. In detail, 0.25 g glucose was uniformly dispersed in 60 ml DI water under continuous stirring for 0.5 h. Then, 0.8724 g Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O, 0.5297 g (NH<sub>4</sub>)<sub>6</sub>Mo<sub>7</sub>O<sub>24</sub>·4H<sub>2</sub>O and 1.4019 g hexamethylenetetramine were dissolved in the above solution and stirred for 1h. Next, the mixed solution was transferred to a 100 ml Teflon-lined stainless steel autoclave and kept temperature at 200 °C for 24 h. After cooling to room temperature, the brown sediment was washed several times by centrifugation using DI water and ethanol, and dried under a vacuum at 60 °C for 12 h. Secondly, the sulfur powder and the obtained NiMoO<sub>4</sub> precusor were put at the upstream and downstream of the tube furnace, respectively, and annealed at 500 °C for 2 h under H<sub>2</sub>/Ar atmosphere to obtain the final MoS<sub>2</sub>/Ni<sub>3</sub>S<sub>2</sub>@C material. For comparison, Ni<sub>3</sub>S<sub>2</sub>/C, MoS<sub>2</sub>/C and MoS<sub>2</sub>/Ni<sub>3</sub>S<sub>2</sub> materials were also synthesized via the similar process just without Mo or Ni sources or glucose additives.

*Materials characterizations*: The phase purity and structure of the obtained materials were verfied by X-ray diffraction (XRD) with Cu K $\alpha$  radiation. The morphology and microstructure were probed by the scaning electron microscopy (SEM, SU 8020, HITACHI), transmission electron microscopy (TEM, Tecnai G2F20), and highresolution transmission electron microscopy (HRTEM). The elemental mapping measurement was carried out in the energy-dispersive X-ray spectroscope (EDS) attached to TEM. Raman spectroscopy measurements were performed on a Horiba Jobin Yvon T6400 with a 514.5 nm laser excitation. X-ray photoelectron spectroscopy (XPS) was collected by a Thermo Scientific ESCALAB 250 with Al K $\alpha$  as the excitation source. The specific surface area and pore size distribution were provided by N<sub>2</sub> adsorption and desorption test (Autosorb-iQ-Cx) at 77 K using the Bruauer-Emmett-Teller (BET) and Barret-Joyner-Halenda (BJH) methods. Thermogravimetric analysis (TGA) was implemented by the Pyris 1 Thermogravimetric Analyzer. The electrical conductivity measurements were carried out on a commercial Physical Property Measurement System (PPMS, 1.8 K $\leq$ T $\leq$ 400 K, 0 T $\leq$ H $\leq$ 9 T) using the four-probe method.

Electrochemical measurements: Electrochemical performance was evaluated using CR2032 coin-type cells, which were assembled in an argon-filled glove box with moisture and oxygen less than 0.1 ppm. The working electrode was prepared by mixing the as-prepared materials, Super P, and sodium carboxymethyl cellulose (CMC) in a weight ratio of 7:2:1 into a certain amount of DI water. The uniformly mixed slurry was pasted onto Cu foil and then dried in a vacuum oven at 80 °C overnight. The loading mass of the working electrodes is about 1-2 mg/cm<sup>2</sup>. For SIBs, Na foil was employed as the counter electrode, glass microfiber filters as the separator, and 1 M NaClO<sub>4</sub> dissolved in PC (propylene carbonate) and EC in a volume ratio of 1:1 with 5% FEC (fluoroethylene carbonate) as the electrolyte. For PIBs, K foil was served as the counter electrode, glass microfiber filters as the separator, and 3 M KFSI in DME (1, 2-Dimethoxyethane) as the electrolyte. Galvanostatic charge and discharge (GCD) measurements were carried out on a LAND battery test system in a voltage window of 0.01-3.0 V. Cyclic voltammetry (CV) measurements from 0.01 to 3.0 V at different scan rates and electrochemical impendence spectra (EIS) at the frequency range of  $10^5$ -0.01 Hz were performed using a CHI 660E electrochemical workstation.

*DFT Calculation*: The density functional theory (DFT) calculations were performed by the Vienna Ab-initio Simulation Package (VASP).<sup>[1]</sup> According to the generalized gradient approximation (GGA), the projector-augmented wave (PAW) pseudopotential with Perdew-Burke-Ernzerhof (PBE) parametrization were used.<sup>[2]</sup> The cut-off energy for the wave function is set as 450 eV, and the Brillouin zone was sampled via a Γcentered *k*-mesh scheme with a 0.02 ×  $2\pi$  Å<sup>-1</sup> space. The energy and force were converged to  $1.0 \times 10^{-5}$  eV/atom and 0.02 eV/ Å, respectively. A 20 Å vacuum layer thickness was applied to avoid virtual interaction. The adsorption energy of Na on MoS<sub>2</sub> or MoS<sub>2</sub>/Ni<sub>3</sub>S<sub>2</sub> heterostructure was calculated by  $E_{ads}=E_{total}-E_{Na}-E_{host}$ , where the  $E_{total}$  is the total energy of Na-absorbed MoS<sub>2</sub> or MoS<sub>2</sub>/Ni<sub>3</sub>S<sub>2</sub> heterostructure,  $E_{Na}$  is the energy of isolated Na atom and  $E_{host}$  is the energy of MoS<sub>2</sub> or MoS<sub>2</sub>/Ni<sub>3</sub>S<sub>2</sub> heterostructure framework.



Fig. S1. XRD pattern of the NiMoO<sub>4</sub> precursor material.



Fig. S2. a-c) SEM, d-f) TEM and g-i) HRTEM images of the  $MoS_2/Ni_3S_2$ ,  $MoS_2/C$  and  $Ni_3S_2/C$  samples.



Fig. S3. XRD patterns of the  $MoS_2/Ni_3S_2$ ,  $MoS_2/C$  and  $Ni_3S_2/C$  samples.



Fig. S4. Raman spectrum of the  $MoS_2/Ni_3S_2$  sample.



Fig. S5. XPS survey spectrum the MoS<sub>2</sub>/Ni<sub>3</sub>S<sub>2</sub>@C material.



Fig. S6. a) Isothermal  $N_2$  adsorption and desorption curves and b) pore size distribution of the  $MoS_2/Ni_3S_2$  sample.

**Detailed process about TGA calculation of the MoS**<sub>2</sub>/Ni<sub>3</sub>S<sub>2</sub>@C sample: In the temperature range from 50 to 700 °C in air, the main part of the weight loss is composed of three parts: the oxidation of MoS<sub>2</sub>, Ni<sub>3</sub>S<sub>2</sub> and carbon layer. The total weight loss is about 46.5 wt.% in the experimental process. Assumed to the final product is MoO<sub>3</sub> and NiO, the corresponding reaction equations can be written as follows:

$$2\text{MoS}_2 + 7\text{O}_2 \rightarrow 2\text{MoO}_3 + 4\text{SO}_2 \tag{1}$$

$$2Ni_3S_2 + 7O_2 \rightarrow 6NiO + 4SO_2 \tag{2}$$

$$C + O_2 \rightarrow CO_2 \tag{3}$$

The carbon content of the  $Ni_3S_2/MoS_2/C$  heterostructure material was calculated based on the following procedures:

Assumed to molar mass of  $MoS_2 = M_1$ ,  $Ni_3S_2 = M_2$ ,  $MoO_3 = M_3$ ,  $NiO = M_4$ , the total content of the heterostructure material =  $m_0$ , the content of  $MoS_2$  in the heterostructure =  $m_1$ ,  $Ni_3S_2$  content in the heterostructure =  $m_2$ ,  $MoO_3$  content in the heterostructure =  $m_3$ , NiO content in the heterostructure =  $m_4$  and carbon content in the heterostructure =  $m_5$ .

Due to the final product is MoO<sub>3</sub> and NiO, so  $m_3 + m_4 = 0.535m_0$ .

According to Equation (1) and (2),  $m_3$  and  $m_4$  can be deduced from  $m_3 = M_3 \times m_1/M_1 = 0.90m_1$ , and  $m_4 = M_4 \times 3m_2/M_2 = 0.90m_2$ , respectively.

Combining the EDS results, the atomic ratio of Ni and Mo is about 2:1, we can calculate that  $m_1/m_2$  is about 1:1. So, we can obtain  $0.9m_1+0.90m_2 = 0.535m_0$ ,  $m_1 = m_2 = 0.3m_0$ , Thus, carbon content of the heterostructure  $m_5 = m_0 - 0.6m_0 = 0.4m_0$ .

So, the contents of MoS<sub>2</sub>, Ni<sub>3</sub>S<sub>2</sub> and carbon in the heterostructure material are about 30

wt%, 30 wt% and 40 wt%, respectively.



Fig. S7. The SEM images of (a) before cycling and (b) after 200 cycles for the  $MoS_2/Ni_3S_2@C$  electrode for SIBs.



Fig. S8. EIS comparison of the for the  $MoS_2/Ni_3S_2@C$  and  $MoS_2/Ni_3S_2$  electrodes for SIBs.



Fig. S9. The relationship of temperature dependent electrical conductivity at the range of 250-340 K for (a)  $MoS_2/Ni_3S_2$  and (b)  $MoS_2/Ni_3S_2@C$  electrodes.



Fig. S10. Rate performance comparison of this work with the previously reported  $MoS_2$ -based heterostructure anodes for SIBs.

Electrodes	Cycling performance	Rate performance	Loading mass	Ref.
	(mAh g <sup>-1</sup> /n/A g <sup>-1</sup> )	(mAh g <sup>-1</sup> /A g <sup>-1</sup> )	(mg cm <sup>-2</sup> )	
Bi <sub>2</sub> S <sub>3</sub> /MoS <sub>2</sub>	427.7/100/0.5	325.5/10	1.0-1.2	[6]
MnS-MoS <sub>2</sub>	214/500/1.0	78.3/10	1.2	[7]
NB-NiMoS	420/200/0.5	309/10	1.4	[8]
Cu <sub>2</sub> S@carbon@MoS <sub>2</sub>	300/200/0.3	297/3	1.0	[9]
MoS <sub>2</sub> -NiS	391/700/2.0	342/10	1.5	[3]
G/NiS <sub>2</sub> -MoS <sub>2</sub>	509.6/500/0.5	424.5/2	1.0	[4]
Cu <sub>2</sub> S/MoS <sub>2</sub> ⊂Carbon	336.2/300/1.0	260.2/10	1.0	[10]
NiS/MoS <sub>2</sub> /C	335/190/1.0	398/5	1.0-2.0	[5]
MoS <sub>2</sub> /Ni <sub>3</sub> S <sub>2</sub> @C	400/200/1.0	318.5/10	1.0-2.0	This work

Table S1 The detailed comparison of the sodium storage performance this work with the previously reported  $MoS_2$ -based heterostructure anodes for SIBs.



Fig. S11. The cycling performance of the  $MoS_2/Ni_3S_2$  electrode for PIBs at 1 A g<sup>-1</sup>.



Fig. S12. The rate performance of the  $MoS_2/Ni_3S_2$  electrode for PIBs.



Fig. S13. The fitted *b* values at the different redox peaks for the  $MoS_2/Ni_3S_2@C$  heterostructure electrode in SIBs.



Fig. S14. *Ex-situ* XRD patterns of the  $MoS_2/Ni_3S_2@C$  heterostructure electrode for SIBs at the different discharge/charge states.



Fig. S15. *Ex-situ* XRD patterns of the  $MoS_2/Ni_3S_2@C$  heterostructure electrode for PIBs at the different discharge/charge states.



Fig. S16. EDS-mapping images of the  $MoS_2/Ni_3S_2@C$  electrode heterostructure electrode for SIBs at the full de-sodiation state.



Fig. S17. EDS-mapping images of the  $MoS_2/Ni_3S_2@C$  electrode heterostructure electrode for PIBs at the full de-potassiation state.

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