

## Supporting Information

### **Tailoring Selenium Vacancies in MoSe<sub>2</sub> through Oxygen Passivation for Room-Temperature NO<sub>2</sub> Sensing Enhancement**

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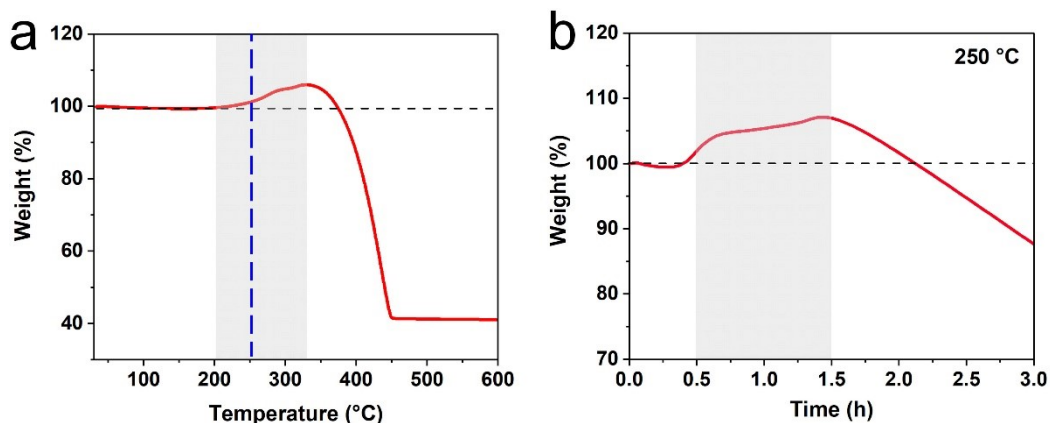
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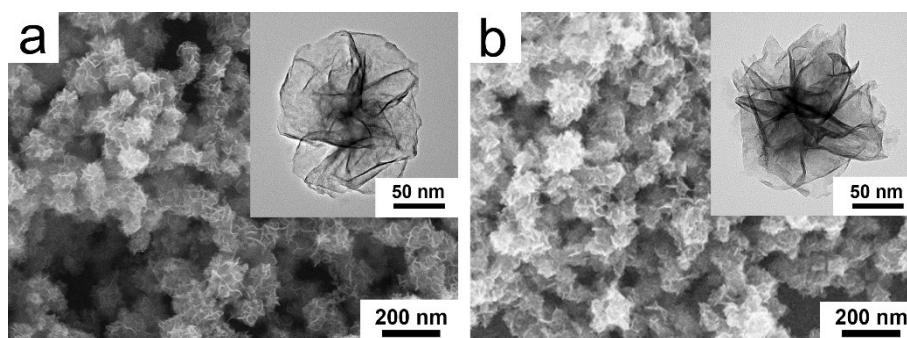
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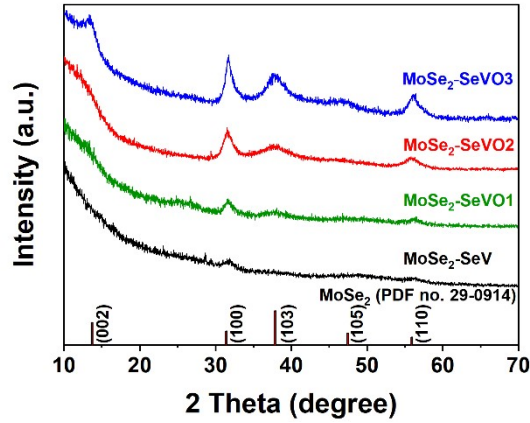
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**Fig. S1.** TGA curves of MoSe<sub>2</sub>-SeV (a) record in the temperature from 30 to 600 °C with a heating rate of 10 °C/min, and (b) record at 250 °C for 3 h. According to the TGA curve in Fig. S1a, the annealing temperature was determined to be 250 °C, at which the weight of MoSe<sub>2</sub>-SeV is increased slowly possibly due to the embed of oxygen atoms at Se vacancy sites. The relatively low annealing temperature would facilitate the following precise control of annealing time. As for the annealing time, the weight of MoSe<sub>2</sub>-SeV increases within 1.5 h at 250 °C in Fig. S1b, therefore the times of 0.5 to 1.5 h were chosen to tailoring the vacancy density of MoSe<sub>2</sub>-SeV. Higher temperatures and longer annealing time will result in a loss of the material weight due to the gasification of SeO<sub>2</sub> during the oxidation of MoSe<sub>2</sub>.



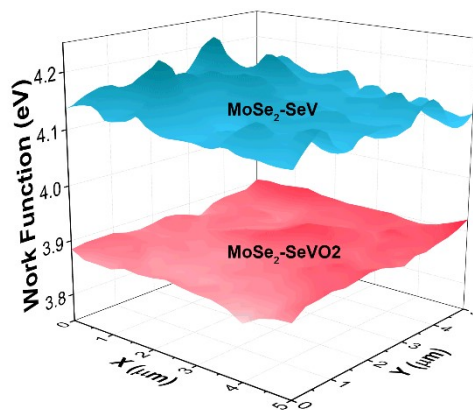
**Fig. S2.** SEM and HRTEM (insets) images of (a) MoSe<sub>2</sub>-SeVO1, and (b) MoSe<sub>2</sub>-SeVO3 nanoflowers. The nanoflower morphology of MoSe<sub>2</sub>-SeV remains intact after annealing for 0.5 and 1.5 h.



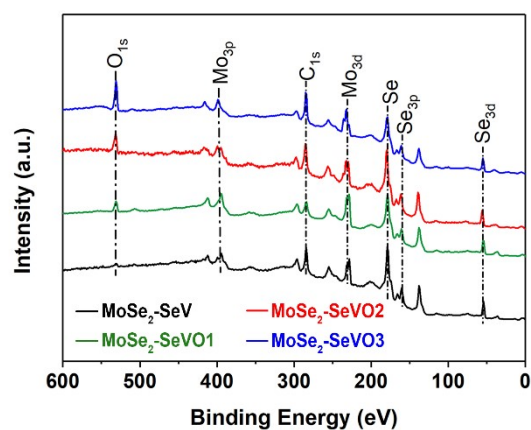
**Fig. S3.** XRD patterns of the MoSe<sub>2</sub>-SeV, MoSe<sub>2</sub>-SeVO1, MoSe<sub>2</sub>-SeVO2, and MoSe<sub>2</sub>-SeVO3. The crystal structures of MoSe<sub>2</sub>-SeV are assigned to the hexagonal 2H-MoSe<sub>2</sub> phase ((PDF no. 29-0914) before and after oxygen passivation, and the crystalline degree is increased with the annealing time which can be ascribed to the decrease of SeV density brought by oxygen passivation.

**Table S1.** The Raman A<sub>1g</sub> peak frequencys, the corresponding Se vacancy contents, and the percentages of Se vacancies filled up by O of each sample.

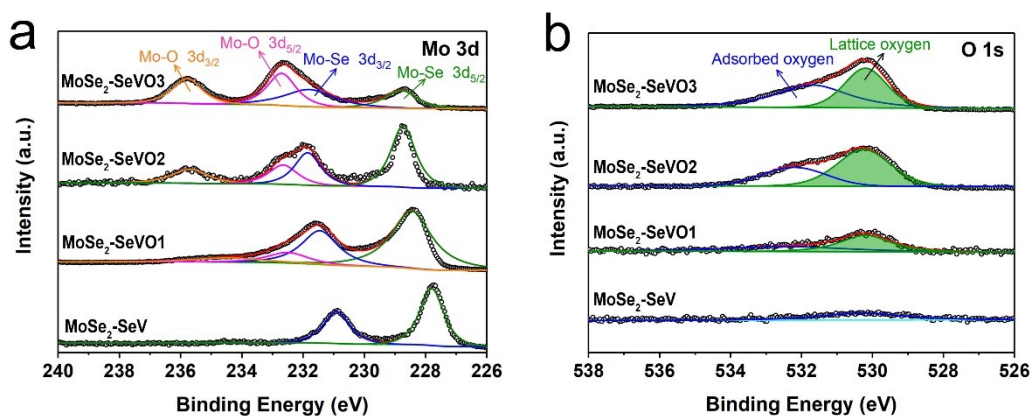
Samples	A <sub>1g</sub> frequency (cm <sup>-1</sup> )	SeV content	O filled SeV
MoSe <sub>2</sub> -SeV	229	21%	--
MoSe <sub>2</sub> -SeVO1	232	16%	5%
MoSe <sub>2</sub> -SeVO2	233	13%	8%
MoSe <sub>2</sub> -SeVO3	235	10%	11%



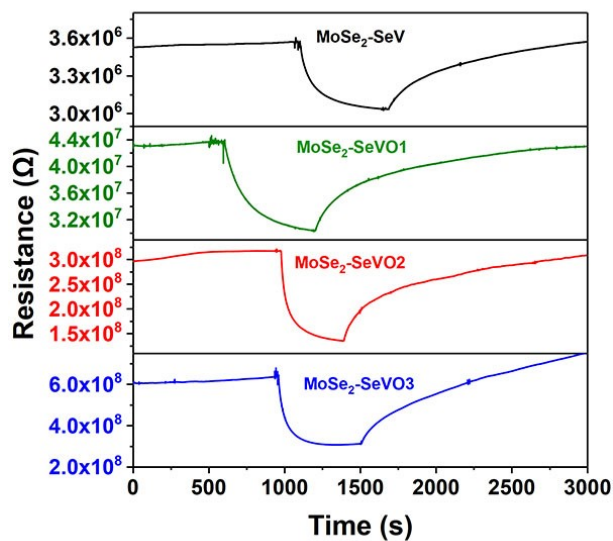
**Fig. S4.** Work function maps of MoSe<sub>2</sub>-SeV and MoSe<sub>2</sub>-SeVO2 measured by KPFM. The observed smaller work function of MoSe<sub>2</sub>-SeVO2 (3.87 eV) than that of MoSe<sub>2</sub>-SeV (4.15 eV) manifests the higher electron density of MoSe<sub>2</sub>-SeVO2 after oxygen passivation.



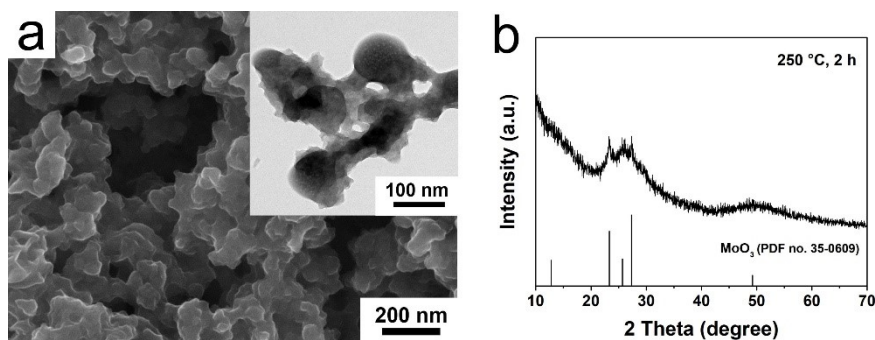
**Fig. S5.** XPS survey of the  $\text{MoSe}_2\text{-SeV}$ ,  $\text{MoSe}_2\text{-SeVO1}$ ,  $\text{MoSe}_2\text{-SeVO2}$ , and  $\text{MoSe}_2\text{-SeVO3}$ . The intensity of O 1s increases with the annealing time, indicating the oxygen element content of  $\text{MoSe}_2$  increases through oxygen passivation.



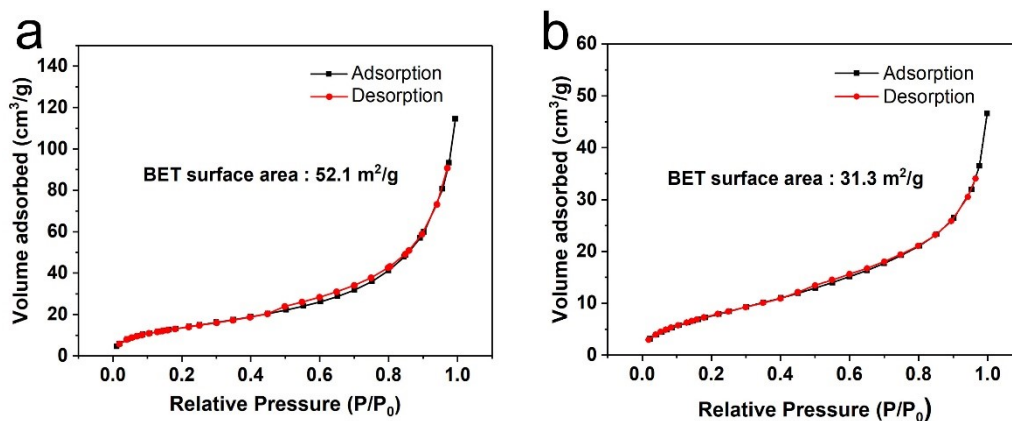
**Fig. S6.** (a) Mo 3d and (b) O 1s XPS spectra of the  $\text{MoSe}_2\text{-SeV}$ ,  $\text{MoSe}_2\text{-SeVO1}$ ,  $\text{MoSe}_2\text{-SeVO2}$ , and  $\text{MoSe}_2\text{-SeVO3}$ . The contents of Mo–O bond and lattice oxygen increase with the annealing time.



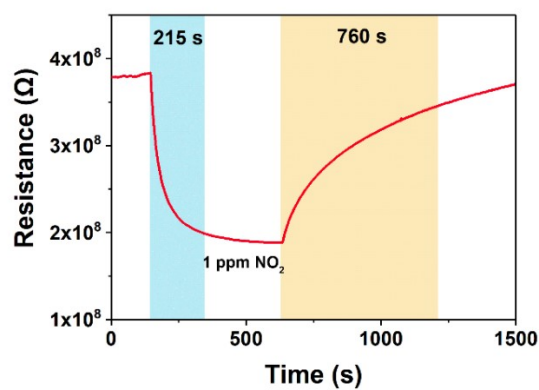
**Fig. S7.** The real-time resistance curves of the MoSe<sub>2</sub>-SeV, MoSe<sub>2</sub>-SeVO1, MoSe<sub>2</sub>-SeVO2, and MoSe<sub>2</sub>-SeVO3 towards 1 ppm NO<sub>2</sub> at room temperature.



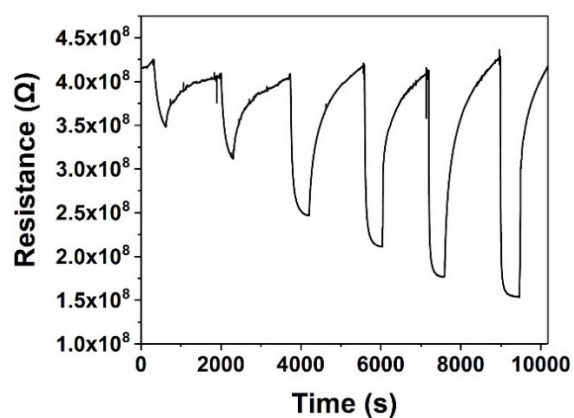
**Fig. S8.** (a) SEM and TEM (inset) images and (b) XRD pattern of the MoSe<sub>2</sub>-SeV annealed at 250 °C for 2 h. The heterostructure of MoSe<sub>2</sub>-SeV nanoflower collapses and the becomes to MoO<sub>3</sub> (PDF no. 35-0609), indicating the surface of the MoSe<sub>2</sub> starts to be oxidized and the lattice structure gradually changes to MoO<sub>3</sub>.



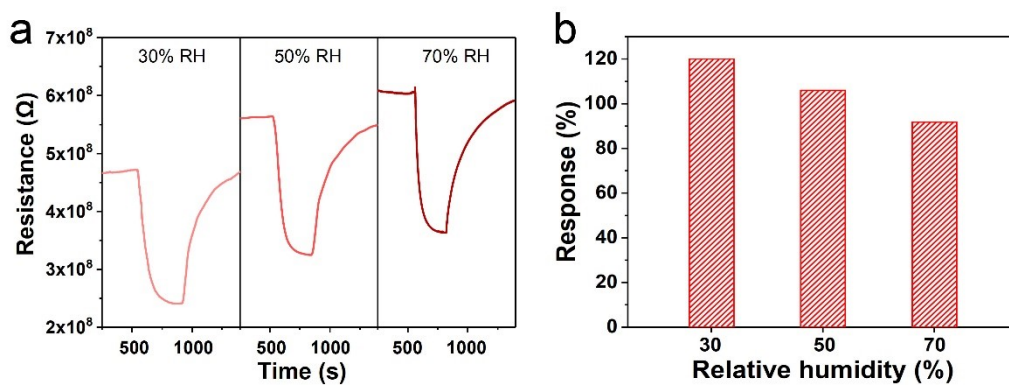
**Fig. S9.** The nitrogen adsorption and desorption isotherms of (a) MoSe<sub>2</sub> nanoflower, and (b) collapsed MoSe<sub>2</sub> nanoflower after 2 h annealing.



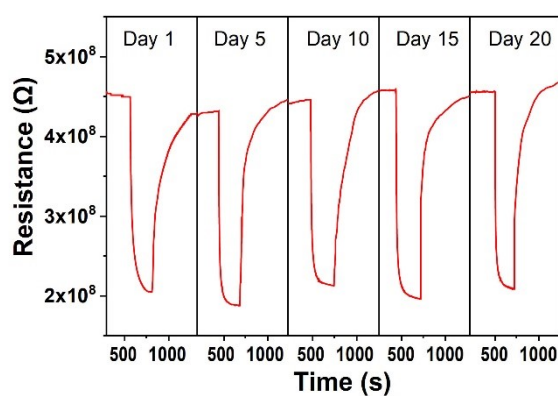
**Fig. S10.** The response and recovery curve of the MoSe<sub>2</sub>-SeVO<sub>2</sub> sensor towards 1 ppm NO<sub>2</sub>, from which the response/recovery time of MoSe<sub>2</sub>-SeVO<sub>2</sub> is calculated to be 215/760 s.



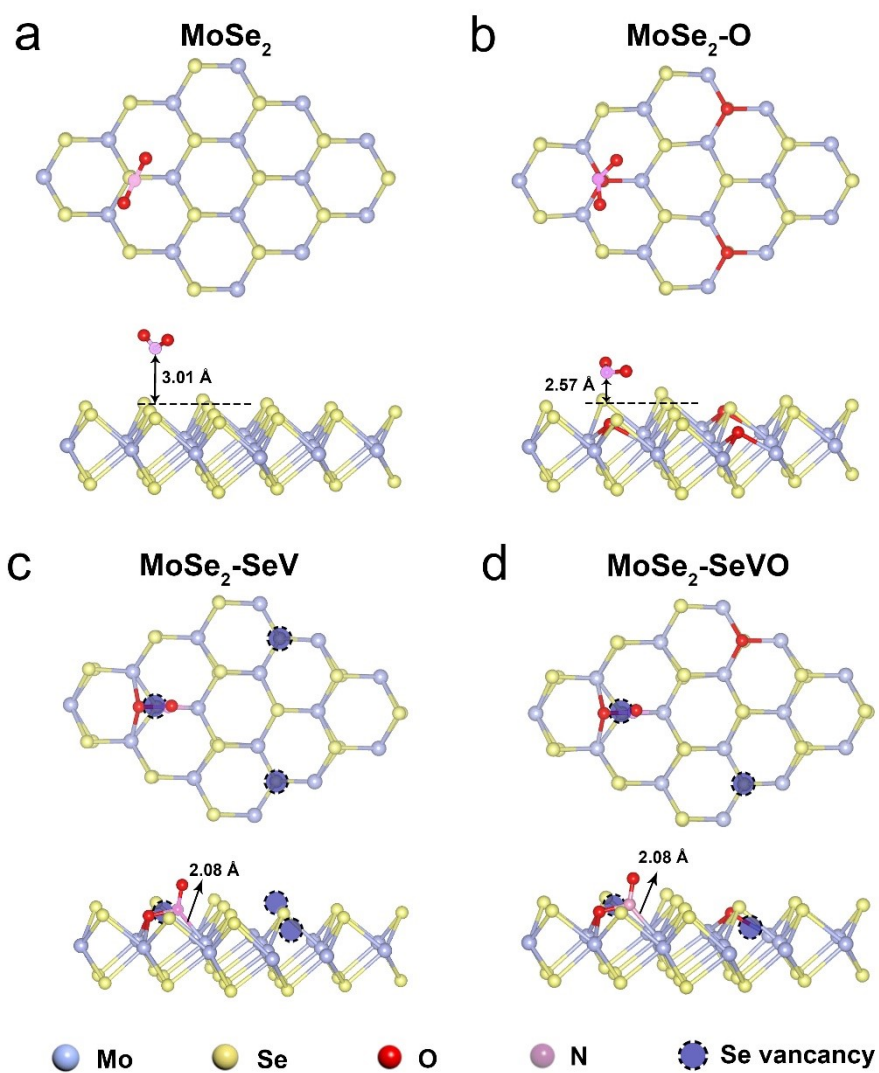
**Fig. S11.** The corresponding real-time resistance curve from the dynamic response test in the sensing studies.



**Fig. S12.** The sensing (a) resistance curves and (b) response of the  $\text{MoSe}_2\text{-SeVO}_2$  sensor to 1 ppm  $\text{NO}_2$  at room temperature. The higher baseline resistance of the  $\text{MoSe}_2\text{-SeVO}_2$  sensor in higher humidity could be explained by the interaction between the water molecule (as electron donor) and the  $\text{MoSe}_2\text{-SeVO}_2$  (p-type electron acceptor).

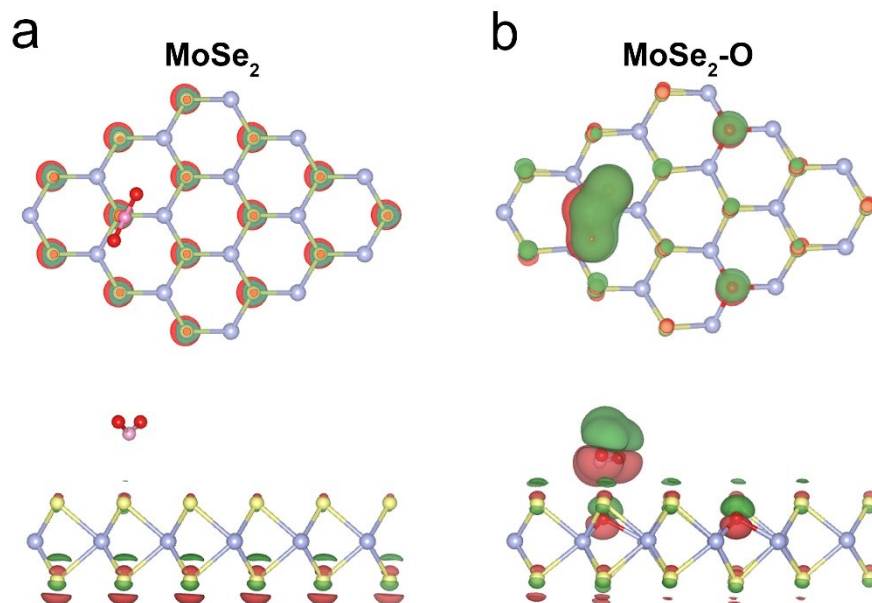


**Fig. S13.** The real-time resistance curves of the  $\text{MoSe}_2\text{-SeVO}_2$  sensor towards 1 ppm  $\text{NO}_2$  at room temperature during stability test of for 20 days.



**Fig. S14.** Top and side views of the most stable adsorption configurations of  $\text{NO}_2$  on (a)  $\text{MoSe}_2$ , (b)  $\text{MoSe}_2\text{-O}$ , (c)  $\text{MoSe}_2\text{-SeV}$ , and (d)  $\text{MoSe}_2\text{-SeVO}$ .





**Fig. S15.** The top and side views of the charge density difference (CDD) plots for adsorbed  $\text{NO}_2$  on (a)  $\text{MoSe}_2$  and (b)  $\text{MoSe}_2\text{-O}$ . Red/green regions represent the areas of electron accumulation/depletion. The isosurface value is  $0.18 \text{ e Bohr}^{-3}$ .