## **Supporting Information**

## **A novel 2D VC<sup>4</sup> as a promising Na-host material for Na-ion batteries: computational insights**

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## **Computational Details**

The particle swarm optimization (PSO) method within the evolutionary algorithm as implemented in the Crystal structure Analysis by Particle Swarm Optimization (CALYPSO) code<sup>1</sup> was employed to find the lowest energy structures of  $VC<sub>4</sub>$  monolayers. Unit cells containing 1, 2, and 4 formula units (f.u.) were considered. In the first step, random structures with certain symmetries are built, in which atomic coordinates are generated by the crystallographic symmetry operations. Local optimizations using the VASP code were done with the conjugate gradients method and stopped when Gibbs free energy changes became smaller than  $1\times10^{-5}$  eV per cell. After processing the first-generation structures, 60% of them with lower Gibbs free energies are selected to construct the next generation structures by PSO. 40% of the structures in the new generation are randomly generated. A structure fingerprinting technique of bond characterization matrix is applied to the generated structures, so that identical structures are strictly forbidden. These procedures significantly enhance the diversity of the structures, which is crucial for structural global search efficiency. In most cases, structural searching simulations for each calculation were stopped after generating  $1000 \sim 1200$  structures (e.g., about  $20 \sim 30$  generations).



Fig. S1 Structures of VC<sub>4</sub> after 10 ps steps of AIMD at (a) 500 K and (b) 1000 K.



Fig. S2 Adsorption of Na on monolayer VC<sub>4</sub> at (a)  $A_1$ -site, (b)  $A_2$ -site, (c)  $A_3$ -site, (d)  $A_4$ -site, (e) A<sub>2</sub>-site, (f) A<sub>4</sub>-site, (g) A<sub>1</sub>-site, and (h) A<sub>3</sub>-site.



Fig. S3 Diffusion energy barrier and their corresponding path of  $\text{Na}_x \text{VC}_4$  at  $x = 2$  and 5.



Fig. S4 Partial density of states of (a)  $Na<sub>1</sub>VC<sub>4</sub>$ , (b)  $Na<sub>2</sub>VC<sub>4</sub>$ , (c)  $Na<sub>3</sub>VC<sub>4</sub>$ , (d)  $Na<sub>4</sub>VC<sub>4</sub>$ , and (e)  $Na<sub>5</sub>VC<sub>4</sub>$ .



Fig. S5 Energy fluctuations versus time steps of  $\text{Na}_x \text{VC}_4$  ( $x = 1, 2, 3, 4$ , and 5) at 300 K after 10 ps.



Fig. S6 Structural snapshots of  $\text{Na}_x\text{VC}_4$  ( $x = 1, 2, 3, 4$ , and 5) at 300 K after 10 ps.



Fig. S7 Optimized geometry of various defects in  $VC_4$  monolayer. (a)  $V_{C1}$ , (b)  $V_{C2}$ , (C)  $V_{V}$ , (d)  $V_{VC}$  vacancy.



Fig. S8 (a) The inequivalent adsorption sites of 2D VC<sub>4</sub> with Vv vacancy, named as  $B_1$ ,  $B_2$ ,  $B_3$ ,  $B_4$ , and  $B_5$ . The side view of  $B_2$  and  $B_3$  are denoted as  $B_{2/3}$  due to their close positions in the z direction. The considered migration paths of Na diffusion on the 2D  $VC_4$  with Vv vacancy. (b) The corresponding diffusion energy barrier profiles of path I and path II.



Fig. S9 Average voltages of different Na concentrations in  $\text{Na}_x\text{V}_{0.917}\text{C}_4$  ( $x = 1-5$ ).



Fig. S10 Optimized structures of  $\text{Na}_x\text{V}_{0.917}\text{C}_4$  ( $x = 1-5$ ).

Material	Voltage (V)	Diffusion barrier (eV)	Capacity $(mAh g-1)$	Ref.
Li-Cu $C_6$	0.34	0.41	1188	$\overline{2}$
$Li-Mo2C$	0.14	0.13	526	$\overline{3}$
$Li-VC2$	0.40	0.09	1430	$\overline{4}$
$K-SnC$	0.41	0.17	410	5
$K-PC6$	0.64	0.29	780	$\sqrt{6}$
$Na-CP_3$	0.53	0.35	2298	$\tau$
$Na-ZrC2$	0.25	0.02	932	$\,$ 8 $\,$
$Na-TiC_3$	0.18	0.20	1278	9
$Li-NiC_3$	0.17	0.50	1698	$10\,$
$Na-V2C$	0.49	0.14	300	$\overline{11}$
Na-Twin- Graphene	1.09	0.198	496	12
$Na_xVC_4$	0.51	0.32	1353	This work

Table S1 Comparative study of the calculated voltage, energy barrier, and Na storage capacity of VC<sup>4</sup> monolayer with previously reported 2D materials.



Table S2 Adsorption energies of Na at different sites on  $V_{0.917}C_4$  monolayer.

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