## **Supplementary Information**

# **Synergistic carrier and phonon transport advance Ag dynamically-doped** *n***-type PbTe thermoelectrics** *via* **Mn alloying**

Wei Yuan, ‡<sup>a</sup> Qian Deng, ‡<sup>a</sup> Dong Pan, <sup>b</sup> Xiang An, <sup>a</sup> Canyang Zhao, <sup>a</sup> Wenjun Su, <sup>a</sup> Zhengmin He, <sup>a</sup> Qiang Sun<sup>\*c,d</sup> and Ran Ang<sup>\*a,e</sup>

*<sup>a</sup> Key Laboratory of Radiation Physics and Technology, Ministry of Education, Institute of Nuclear Science and Technology, Sichuan University, Chengdu 610064, China*

*<sup>b</sup> State Key Laboratory of Superlattices and Microstructures, Institute of Semiconductors, Chinese Academy of Sciences, P.O. Box 912, Beijing 100083, China*

*<sup>c</sup> State Key Laboratory of Oral Diseases, National Clinical Research Center for Oral Diseases, West China Hospital of Stomatology, Sichuan University, Chengdu, Sichuan 610041, China*

*<sup>d</sup> Sichuan Provincial Engineering Research Center of Oral Biomaterials, Chengdu, Sichuan 610041, China*

*e Institute of New Energy and Low-Carbon Technology, Sichuan University, Chengdu 610065, China*

‡ W. Yuan and Q. Deng contributed equally to this work.

\*Corresponding authors and Emails:  $range@scu.edu.cn$  (RA),  $qiangsun@scu.edu.cn$ (QS)

#### **1. Computational details**

#### **1.1 Single parabolic band model**

The Pisarenko curves and the effective mass of *n*-type PbTe can be modeled by the single parabolic band (SPB) model, and the SPB model is based on the following equations:<sup>1</sup>

Seebeck coefficient:

$$
S = \pm \frac{k_B}{e} \left( \frac{(r+5/2)F_{r+2/3}(\eta)}{(r+3/2)F_{r+1/2}(\eta)} - \eta \right)
$$
 (eq. S1)

Hall carrier concentration:

$$
n_H = 4\pi \left[\frac{2m^* k_B T}{h^2}\right]^{3/2} F_{1/2}
$$
 (eq. S2)

Lorenz number:

$$
L = \frac{k_B^2 3F_0 F_2 - 4F_1^2}{e^2 F_0^2}
$$
 (eq. S3)

where  $k_B$  is the Boltzmann constant,  $\hbar$  is the reduce Planck constant,  $\eta$  is the reduced Fermi level, and *r* denotes the scattering factor and equals -1/2 here assuming that the acoustic scattering mechanism dominates. The Fermi integral is given by:

$$
F_n(\eta) = \int_0^\infty \frac{x^n}{1 + e^{x - \eta}} dx
$$
 (eq. S4)

#### **1.2 Single Kane band model**

The thermoelectric-transport properties of *n*-type PbTe can be modeled by adopting the single Kane band (SKB) model. The transport coefficients are determined by the equations as follows:<sup>2, 3</sup>

Hall carrier concentration:

$$
n_H = \frac{1}{eR_H} = A^{-1} \frac{N_V (2m_b^* k_B T)^{3/2}}{3\pi^2 \hbar^3} \, F_0^{3/2} \tag{eq. S5}
$$

Hall factor:

$$
A = \frac{3K(K+2)\binom{0}{4}F_{-4}^{1/20}F_{0}^{3/2}}{(2K+1)^2\binom{0}{4}F_{-2}^{1/2}}
$$
 (eq. S6)

Hall Carrier mobility:

$$
\mu_H = A \frac{2\pi \hbar^4 e C_l}{m_f^* \left(2m_b^* k_B T\right)^{3/2} E \frac{2}{\det^2} \rho F \frac{3}{2}} \tag{eq. S7}
$$

Power factor:

$$
PF = \frac{2N_V \hbar k_B{}^2 C_l}{\pi E_{def}^2} \frac{1}{m_I^*} \left(\frac{1}{0} F \frac{1}{12} - \xi\right)^{20} F \frac{1}{12}
$$
 (eq. S8)

 ${}^{n}F^{m}_{k}$  has a similar form as the Fermi integral:

$$
{}^{n}F_{k}^{m} = \int_{0}^{\infty} \left(-\frac{\partial f}{\partial \varepsilon}\right) \varepsilon^{n} (\varepsilon + \alpha \varepsilon^{2})^{m} [(1 + 2\alpha \varepsilon)^{2} + 2]^{k/2} d\varepsilon \tag{eq. S9}
$$

Where  $\zeta$  is the reduced chemical potential,  $N_v$  is the band degeneracy ( $N_v=4$  for *n*-type PbTe),<sup>4</sup> *K* is the band anisotropy (*K*=3.6 for *n*-type PbTe),<sup>5</sup>  $k_B$  is the Boltzmann constant,  $\hbar$  is the reduce Planck constant,  $m_l^*$  is the inertial mass  $(m_l^* = 3m_d^*/(N_l^2/3^2 (2K_l^2)^3 + K_l^2/3)^3)$ ,  $c_l^*$  the longitudinal elastic ( $C$ l=7.1×1010 Pa for *n*-type PbTe),<sup>7, 8</sup>  $\varepsilon$  is the reduced energy of electronic state,  $\alpha$  is the reciprocal reduced band gap  $({}^{\alpha} = k_B T / E_g)$ ,  $E_g$  is the band gap ,  $E_g = 0.18eV + 0.0004 eV/K \times T$ ,  $E_{def}$  is the deformation potential coefficient.

#### **1.3 Calculation** of lattice thermal conductivity  $\kappa_{\text{lat}}$

The lattice thermal conductivity  $(\kappa_{lat})$  of the Ag<sub>0.03</sub>Pb<sub>1-*x*</sub>Mn<sub>*x*</sub>Te samples was calculated by the modified Debye-Callaway model, which can be expressed by:

$$
\kappa_{lat} = \frac{k_B}{2\pi^2 v} \left(\frac{k_B}{\hbar}\right)^3 \int_0^{\Theta/T} \tau_{tot}(x) \frac{x^4 e^x}{\left(e^x - 1\right)^2} dx
$$
 (eq. S10)

Here,  $v = 3^{1/3} (v_l^{-3} + 2v_l^{-3})^{-1/3}$  is the average speed of phonon,  $v_l$  and  $v_t$  denote the longitudinal and transverse speeds of sound respectively,  $\hbar$  is the reduced Planck constant,  $\Theta$  is the Debye temperature, *x* is the relation of  $\hbar \omega / k_B T$ ,  $\omega$  is the phonon frequency, and  $\tau_{\text{tot}}$  is the total phonon scattering relaxation time. The total phonon relaxation time  $\tau_{\text{tot}}$  for the Ag<sub>0.03</sub>Pb<sub>1-*x*</sub>Mn<sub>*x*</sub>Te samples consists of Umklapp process, normal process, point defects, grain boundary, and nanoprecipitates  $(U+N+PD+GB+NP)$  by the following equation:<sup>9</sup>

$$
\tau_{tot}^{-1} = \tau_U^{-1} + \tau_N^{-1} + \tau_{PD}^{-1} + \tau_{GB}^{-1} + \tau_{NP}^{-1}
$$
 (eq. S11)

The phonon scattering relaxation time for respective mechanism can be expressed as follows: Umklapp phonon scattering:

$$
\tau_U^{-1} = A_N * \frac{2 k_B \bar{V}^{1/3} \omega^2 \gamma^2 T}{\left(6\pi^2\right)^{\frac{1}{3}} M v^3}
$$
 (eq. S12)

Normal phonon scattering:

$$
\tau_N^{-1} = \frac{2}{\left(\frac{1}{6\pi^2}\right)^{\frac{1}{3}}} \frac{K_B V^{1/3} \omega^2 \gamma^2 T}{M v^3}
$$
 (eq. S13)

Point defect scattering:

$$
\tau_{PD}^{-1} = \frac{V_{00}^{4}}{4\pi^{3}v^{3}} \ast \sum (1 - x_{i}) \left[ \frac{M_{i} - M}{M} \right)^{2} + \varepsilon \left( \frac{a_{i} - a}{a} \right)^{2} \right]
$$
(eq. S14)

Grain boundary scattering:

$$
\tau_{GB}^{-1} = \nu/Gd \tag{eq. S15}
$$

Nanoprecipitates scattering:

$$
\tau_{NP}^{-1} = \nu [(2\pi R^2)^{-1} + (\frac{4}{9}\pi R^2 (\Delta D/D)^2 (\frac{\omega R}{\nu})^4)^{-1}]^{-1} N_p
$$
 (eq. S16)

In above equations,  $\bar{V}$  is the average atomic volume,  $\bar{M}$  is the average atomic mass, *γ* is the Grüneisen parameter,  $A_N$  is the ratio between normal process and Umklapp phonon scattering.

#### **1.4 Calculation of quality factor** *B*

The quality factor *B* of the  $\text{Ag}_{0.03}\text{Pb}_{1-x}\text{Mn}_x$ Te samples was calculated by:<sup>10</sup>

$$
B = \left(\frac{k_B}{e}\right)^2 \frac{8\pi e (2m_e k_B T)^{3/2} \mu_W}{3h^3} T
$$
 (eq. S17)

## **2. Supplementary Figures**



**Figure S1.** *ZT* of the main components of dynamic doped PbTe systems.11-16



**Figure S2.** *κ*lat at 323 K of the main components of dynamic doped PbTe systems.11-16 the Lorenz number calculation methods for representative samples of each system: single parabolic band (SPB) model: PbTe:Cr/Ag2Te, PbTe:Ag, PbTe:Cu/MnTe, PbTe:Ga; single Kane band (SKB) model: PbTe:In/I, PbTe:Cu.



Figure S3. EDS spectra obtained from Ag<sub>2</sub>Te secondary phase.



**Figure S4.** The Hall carrier concentration  $n_H$  and the Hall carrier mobility  $\mu_H$  as a function of *x* in Ag0.03Pb1-*x*Mn*x*Te.



**Figure S5.** DSC curve of  $Ag_{0.03}Pb_{0.95}Mn_{0.05}Te$ , and the peak occurs at 593 K



**Figure S6.** Temperature-dependent effective masses  $m^*$  of PbTe and  $Ag_{0.03}Pb_{1-x}Mn_xTe$  ( $x = 0.02$ , 0.04, 0.05), the black scatter values are from the reported I doped PbTe material.<sup>6</sup>



**Figure S7.** Temperature-dependent power factor PF for *n*-type  $Ag_{0.03}Pb_{1-x}Mn_xTe$  samples.



**Figure S8.** Temperature-dependent electrical thermal conductivity  $\kappa_e$  for *n*-type  $\text{Ag}_{0.03}\text{Pb}_{1-x}\text{Mn}_x\text{Te}$ samples.



**Figure S9.** Temperature-dependent thermal diffusivity (*D*) for *n*-type  $Ag_{0.03}Pb_{1-x}Mn_xTe$  samples.



**Figure S10.** Temperature-dependent heat capacity  $(C_p)$  for *n*-type  $Ag_{0.03}Pb_{1-x}Mn_xTe$  samples.



**Figure S11.** The calculated quality factor *B* at 323 K, 523 K, 623 K and 773 K for *n*-type Ag0.03Pb1-*x*Mn*x*Te samples.



**Figure S12.** Thermal stability and reproducibility of (a) electrical conductivity  $\sigma$ , (b) Seebeck coefficient *S*, (c) total thermal conductivity  $\kappa_{\text{tot}}$ , and (d) figure of merit *ZT* for the high-performance  $n$ -type  $Ag_{0.03}Pb_{0.95}Mn_{0.05}Te$  sample.



**Figure S13.** The average  $ZT(ZT_{avg})$  for  $Ag_{0.03}Pb_{1-x}Mn_xTe$  samples.

### **3. Supplementary tables**

**Table S1.** The sample densities of  $Ag_{0.03}Pb_{1-x}Mn_xTe$  ( $x = 0, 0.01, 0.02, 0.03, 0.04, 0.05, 0.06$ ) measured by Archimedes method.





**Table S2.** The input parameters based on the theoretical simulation of lattice thermal conductivity in this work.

#### **References**

- H. J. Wu, C. Chang, D. Feng, Y. Xiao, X. Zhang, Y. L. Pei, L. Zheng, D. Wu, S. K. Gong, Y. Chen, J. Q. He, M. G. Kanatzidis and L. D. Zhao, *Energy Environ. Sci.*, 2015, **8**, 3298-3312.
- Y. I. Ravich, B. A. Efimova and V. I. Tamarchenko, *Phys. Status Solidi B.*, 1971, **43**, 11-33.
- Y. I. Ravich, B. A. Efimova, I. A. Smirnov and L. S. Stil'bans, *Semiconducting Lead Chalcogenides*, Plenum Press, New York, 1970.
- R. S. Allgaier, *J. Appl. Phys.*, 1961, **32**, 2185-2189.
- H. A. Lyden, *Phys. Rev.*, 1964, **135**, A514-A521.
- Y. Z. Pei, A. D. LaLonde, H. Wang and G. J. Snyder, *Energy Environ. Sci.*, 2012, **5**, 7963-7969.
- Y. I. Ravich, B. A. Efimova and V. I. Tamarchenko, *Phys. Status Solidi B.*, 1971, **43**, 453-469.
- Y. Z. Pei, Z. M. Gibbs, A. Gloskovskii, B. Balke, W. G. Zeier and G. J. Snyder, *Adv. Energy Mater.*, 2014, **4**, 12.
- J. Callaway and H. C. Vonbaeyer, *Phys. Rev. Lett.*, 1960, **5**, 223-223.
- G. J. Snyder, A. H. Snyder, M. Wood, R. Gurunathan, B. H. Snyder and C. N. Niu, *Adv. Mater.*, 2020, **32**, 5.
- Y. Z. Pei, A. F. May and G. J. Snyder, *Adv. Energy Mater.*, 2011, **1**, 291-296.
- Q. Zhang, Q. C. Song, X. Y. Wang, J. Y. Sun, Q. Zhu, K. Dahal, X. Lin, F. Cao, J. W. Zhou, S. Chen, G. Chen, J. Mao and Z. F. Ren, *Energy Environ. Sci.*, 2018, **11**, 933-940.
- H. T. Liu, Q. Sun, Y. Zhong, Q. Deng, L. Gan, F. L. Lv, X. L. Shi, Z. G. Chen and R. Ang, *Nano Energy*, 2022, **91**, 106706.
- L. You, J. Y. Zhang, S. S. Pan, Y. Jiang, K. Wang, J. Yang, Y. Z. Pei, Q. Zhu, M. T. Agne, G. J. Snyder, Z. F. Ren, W. Q. Zhang and J. Luo, *Energy Environ. Sci.*, 2019, **12**, 3089-3098.
- Y. Zhong, H. T. Liu, Q. Deng, F. L. Lv, L. Gan and R. Ang, *ACS Appl. Mater. Interfaces*, 2021, **13**, 52802-52810.
- X. L. Su, S. Q. Hao, T. P. Bailey, S. Wang, I. Hadar, G. J. Tan, T. B. Song, Q. J. Zhang, C. Uher, C. Wolverton, X. F. Tang and M. G. Kanatzidis, *Adv. Energy Mater.*, 2018, **8**, 11.
- Y. Xiao, H. J. Wu, J. Cui, D. Y. Wang, L. W. Fu, Y. Zhang, Y. Chen, J. Q. He, S. J. Pennycook and L. D. Zhao, *Energy Environ. Sci.*, 2018, **11**, 2486-2495.
- Z. S. Wang, H. He, X. D. Cui, H. T. Liu, W. B. Qiu, L. Q. Chen, B. Q. Zhou, J. Tang and R. Ang, *J. Appl. Phys.*, 2019, **125**, 6.
- L. W. Fu, M. J. Yin, D. Wu, W. Li, D. Feng, L. Huang and J. Q. He, *Energy Environ. Sci.*, 2017, , 2030-2040.