

Electronic Supplementary Information

Reticulated carbon nanofiber-encapsulated SnO₂ hollow spheres as high performance anode material for lithium-ion batteries

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1. Experimental

1.1 Synthesized SnO₂ hollow spheres

K₂SnO₃·3H₂O (1.5 mmol, 0.484 g, AR) and (NH₂)₂CO (8 mmol, 0.48 g, AR) were dissolved in deionized water, and the solution was stirred with a magnetic stirrer until complete dissolution. Anhydrous ethanol was then added to the solution in a 5:3 (water: ethanol) ratio, yielding a colorless, transparent solution. After stirring for approximately 30 min, a white turbid solution was formed. This solution was transferred to a PTFE liner, placed in a reactor, and heated in an electric blast drying oven at 190°C for 24 h. After the reaction, the system was allowed to cool naturally to room temperature. The reaction products were washed by centrifugation three times using water and ethanol, respectively. The washed product was then dried in a vacuum drying oven at 60°C for 8 h. The final SnO₂ hollow sphere powder was obtained.

1.2 Synthesis of SnO₂@CF

1.2 g (12%) of PAN was dissolved in 8.2 g of DMAC and stirred magnetically at room temperature for 8 hours. After complete dissolution, 0.6 g of hollow spheres of SnO₂ powder was added, and the mixture was stirred magnetically for 24h. A homogeneous SnO₂@PAN blend was obtained as the spinning solution. The solution was loaded into a 5 mL syringe equipped with a 23-gauge metal conductive needle. The SnO₂@PAN nanofiber precursor was collected by placing a nonwoven fabric receiver beneath the needle. Adjust the distance between the metal needle and the collector so that the distance between the two is 17-25 cm. The voltage setting is 15 kV positive and 2 kV negative. The syringe advancement rate was set to 0.8 mL h⁻¹. SnO₂@PAN nanofiber precursor was obtained after 5h of electrospinning. The obtained SnO₂@PAN nanofiber precursor was then pre-oxidized in a muffle furnace, with the temperature increased at a rate of 1°C min⁻¹: first to 180°C (held for 1h), then to 200°C (held for 1h), 230°C (held for 1h), and 250°C (held for 1h), followed by a final reduction to 180°C (held for 1h). Finally, the precursor was heated in a nitrogen atmosphere at 2°C min⁻¹ to 500°C and held for 2h. The resulting material was carbonized SnO₂@CF.

1.3 Materials characterizations

Crystallographic structure of the samples is determined using X-ray diffractometer (XRD, Bruker D8 advanced) with $\text{CuK}\alpha$ radiation in the 2θ range from 10 to 90° . The Raman spectrometer (Horiba LabRAM HR Evolution) was employed to analyze the material and identify the type of carbon. Transmission electron microscopy (TEM, JEM-F200) and field emission scanning electron microscopy (FESEM, ZEISS SUPRA-55) were employed to examine the morphology and microstructure of the materials for further analysis. Energy dispersive spectroscopy (EDS, QUANTAX) analysis was conducted to obtain elemental mappings. Energy dispersive X-ray spectroscopy (EDS) and high-resolution TEM (HRTEM, JEM-F200) were employed to observe the crystal lattice spacing. X-ray photoelectron spectroscopy (XPS) was performed using a Thermo Scientific K-Alpha spectrometer, with data processed using the “Avantage.exe” software.

1.4 Electrochemical measurement

The electrochemical properties of the electrode materials were evaluated by assembling a CR2032 button battery. The active material, Super-P and PVDF were weighed in the ratio of 7:2:1. they were then placed in a mortar for initial grinding to ensure that the components were initially mixed homogeneously. Next, 400 μL of NMP was added, followed by continued grinding for 20 min to obtain a homogeneous electrode slurry. The above slurry was uniformly coated onto the copper foil using a flatbed heated coater, wherein the thickness of the squeegee was set to 10 μm . After drying in a vacuum oven for 12 h at 60°C ., the coated copper foils were cut into 10 mm electrode sheets. The active substance loading of each copper foil was about $\sim 1.9\text{ mg cm}^{-2}$. The assembly of the CR2032 half-battery was carried out in a glove box filled with argon atmosphere. The electrolyte used was 0.8 M LiPF_6 in DEC:EC=1:1 Vol%. The counter electrode was a lithium metal sheet with a radius of 16 mm. The assembly is then sealed using a battery packaging machine. The encapsulated batteries were left to stand for 8h and then electrochemical tests were carried out. Cycling performance

tests were performed on a CT 4008-5VmA-164 (Shenzhen Xinwei Electronics) with a voltage range of 0.01- 3.0 V. Cyclic voltammetry (CV) curves were obtained from an electrochemical workstation (CHI 660 E). Electrochemical impedance spectroscopy (EIS) was performed on an electrochemical workstation (CHI 660 E) in the frequency range of 100 kHz to 0.01 Hz. The EIS data processing software was “ZView2.exe”.

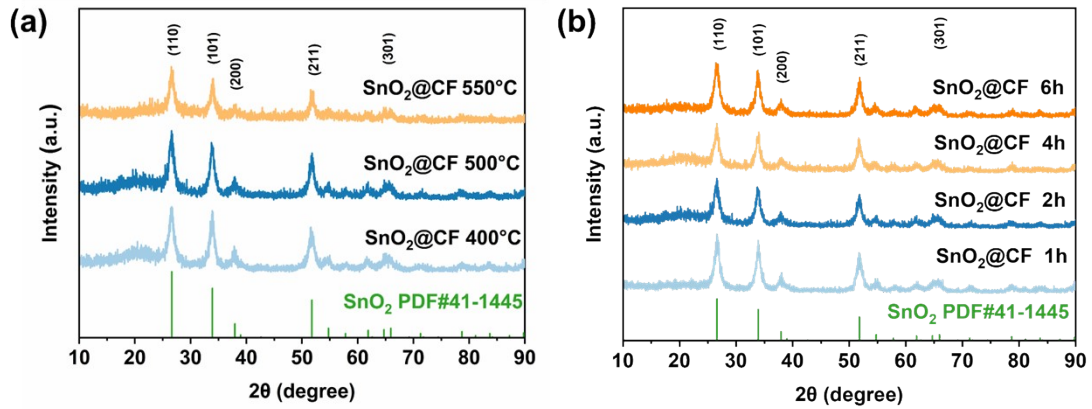


Figure S1 SnO₂@CF prepared at different heat treatment (a) temperatures and (b) times.

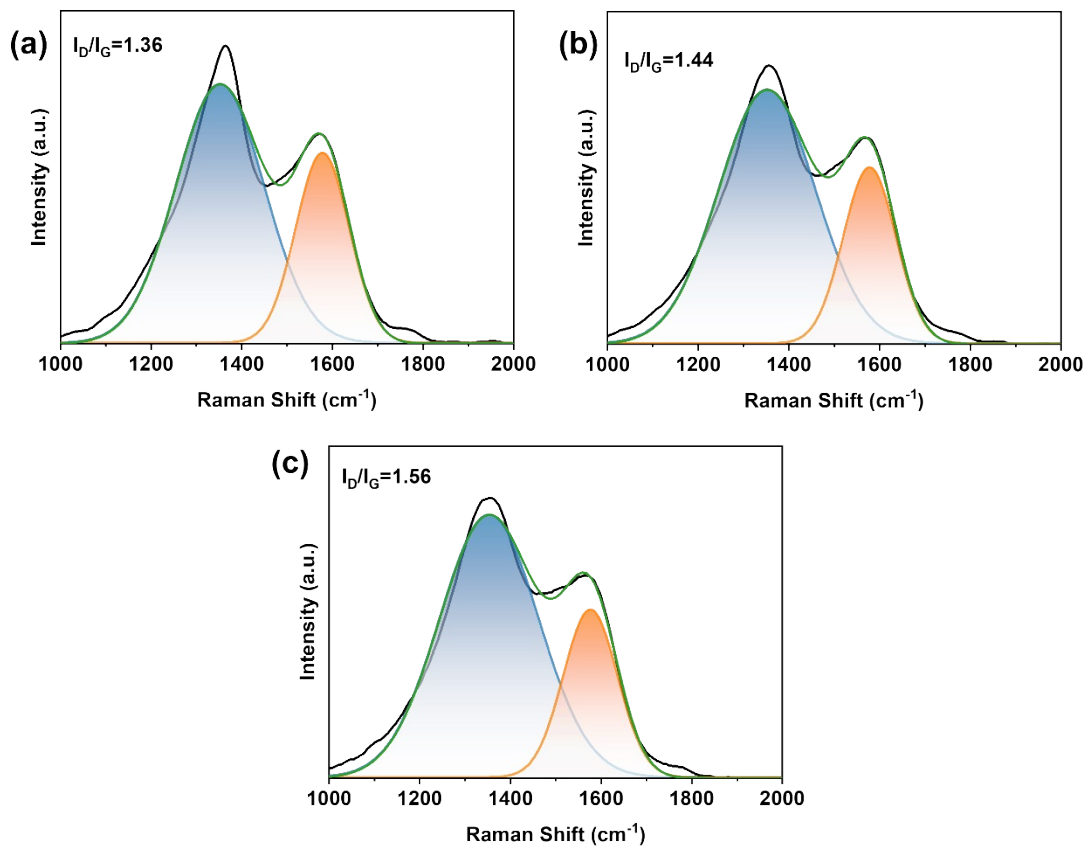


Figure S2 The intensity ratio of I_D/I_G was obtained by peak-fitting.

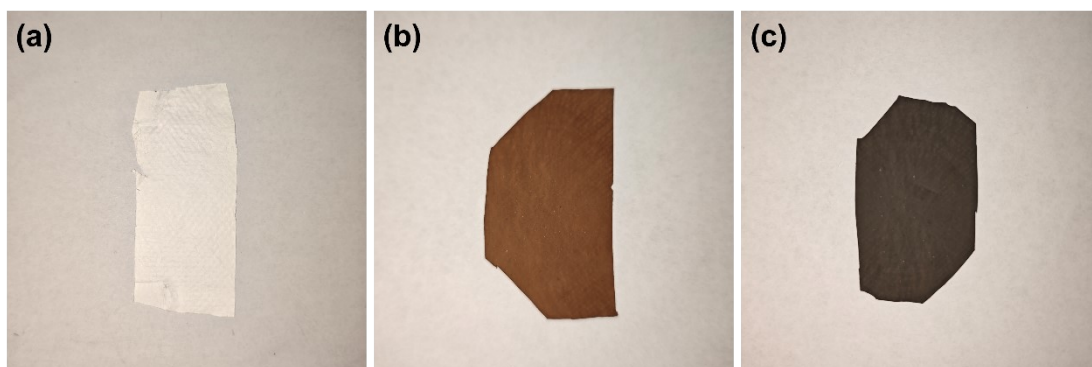


Figure S3 The digital photos: (a) The SnO₂@CF precursors. (b) The pre-oxidized product. (c) The SnO₂@CF.

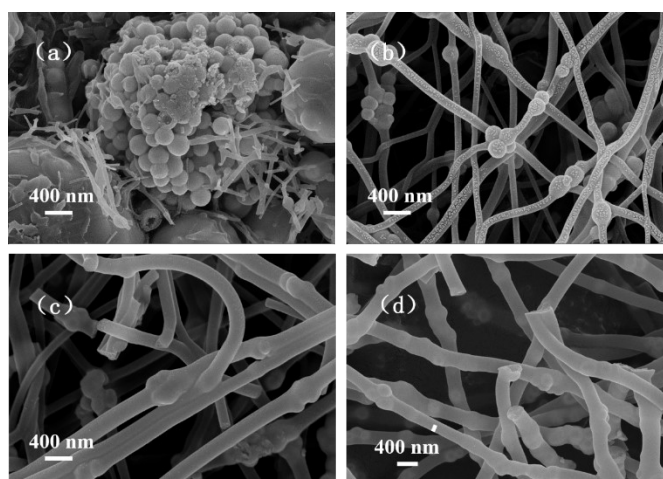


Figure S4 The SEM images of SnO₂@CF with different PAN additions: (a) PAN-8%, (b) PAN-10%, (c) PAN-12%, (d) PAN-14%.

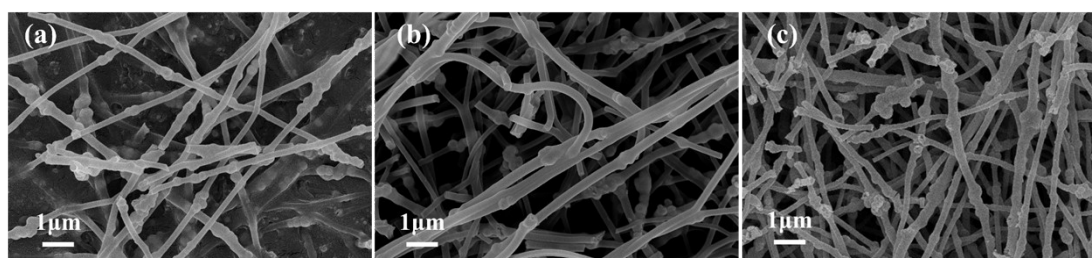


Figure S5 The SEM images of SnO₂@CF at different heat treatment temperatures: (a) 400°C, (b) 500°C, (c) 550°C.

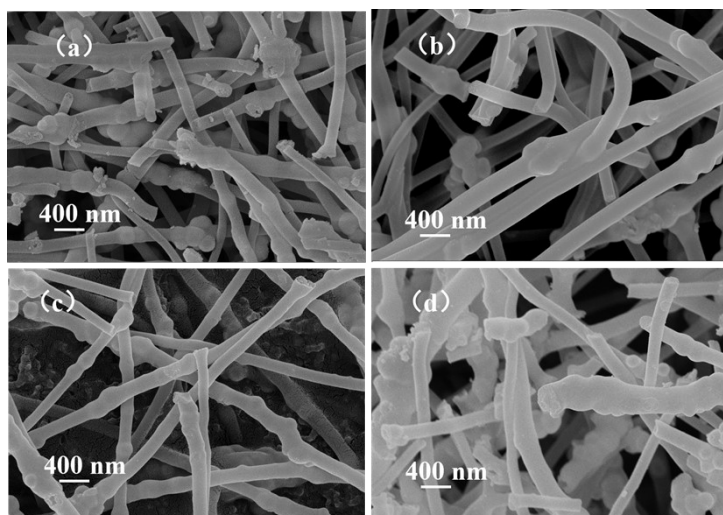


Figure S6 SEM images of SnO₂@CF with different heat treatment times: (a) 1h, (b) 2h, (c) 4h, (d) 6h.

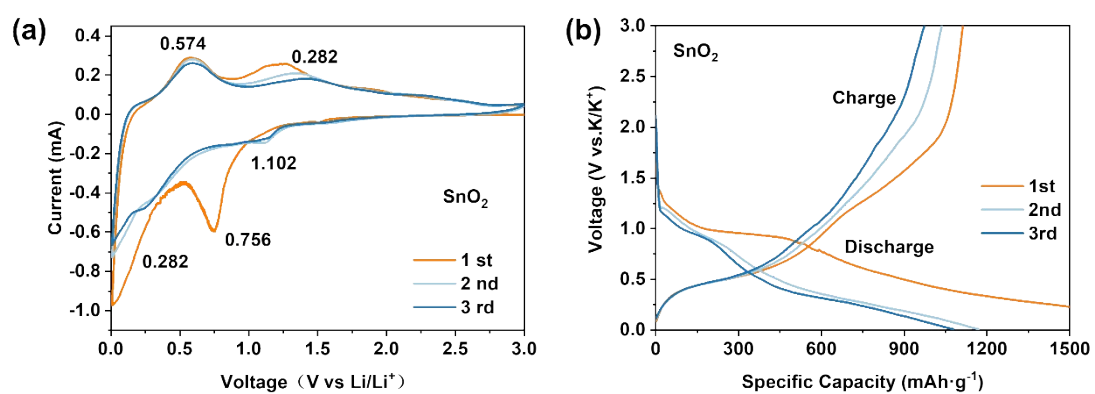


Figure S7 (a) CV curve of SnO₂ at a scan rate of 0.2 mV s⁻¹. (b) The constant current discharge/charge curves of SnO₂ at 0.2C.

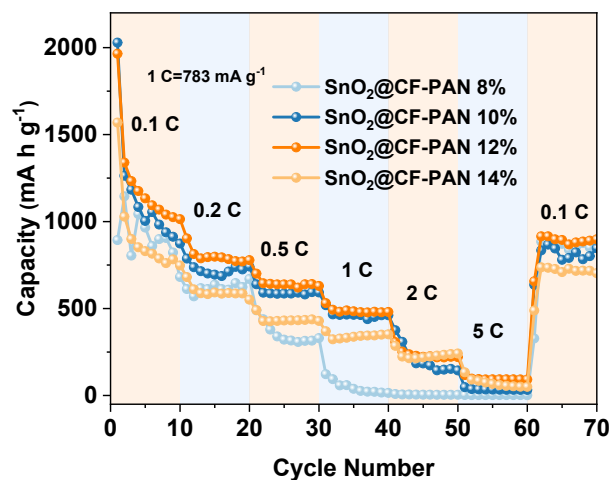


Figure S8 Rate capability of SnO₂@CF with different PAN additions.

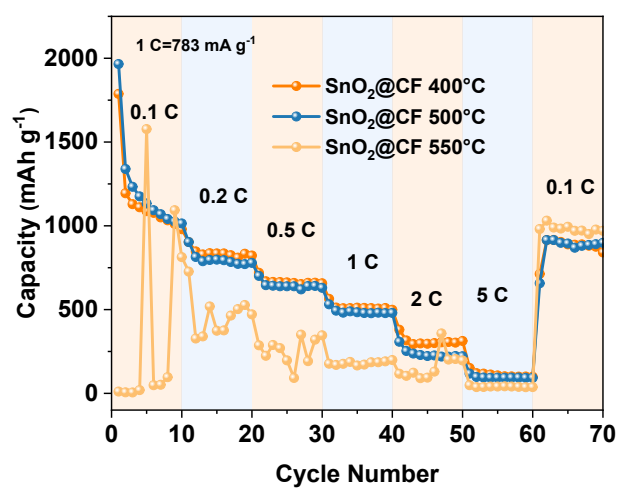


Figure S9 Rate capability of SnO₂@CF with different heat treatment temperatures.

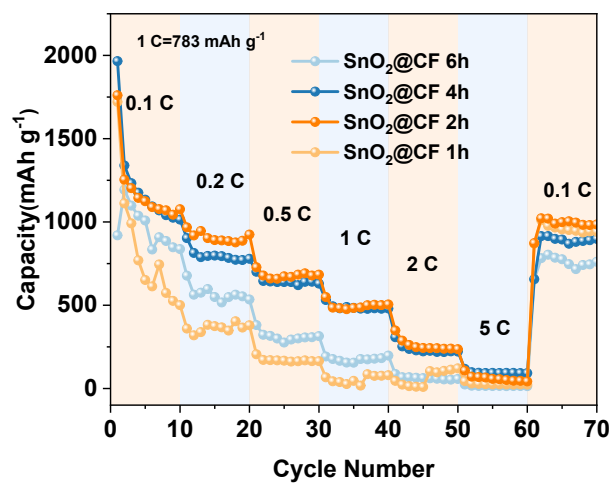


Figure S10 Rate capability of SnO₂@CF with different heat treatment times.

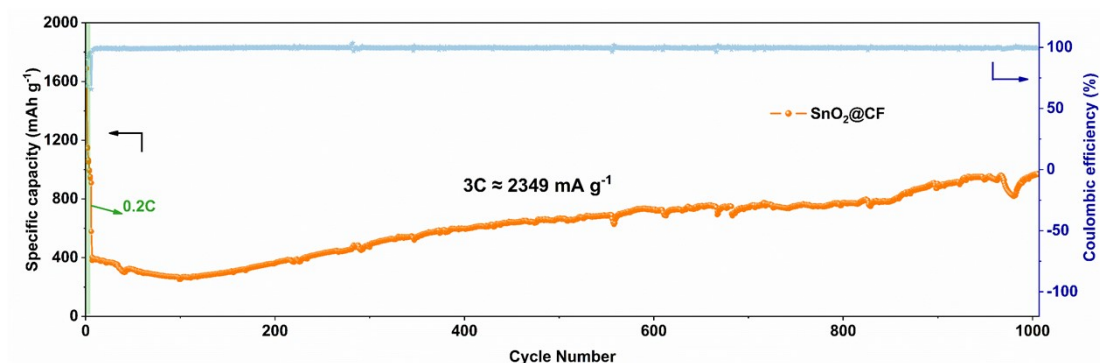


Figure S11 Long-term cycling stability of SnO₂@CF at 3C.

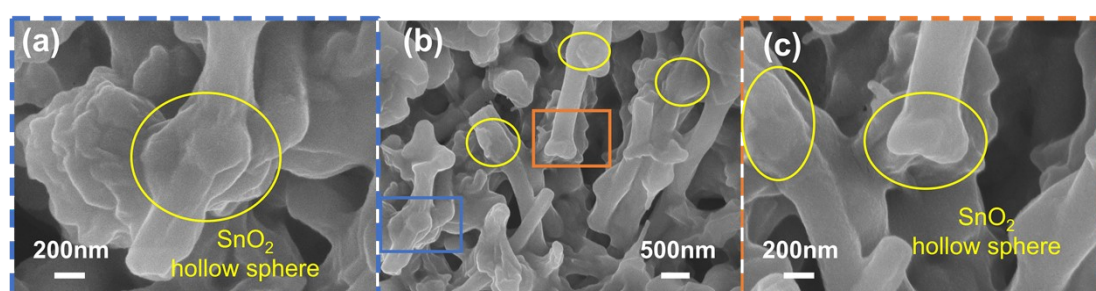


Figure S12 The post-mortem SEM images of SnO₂@CF after 250 cycles at 0.2C.

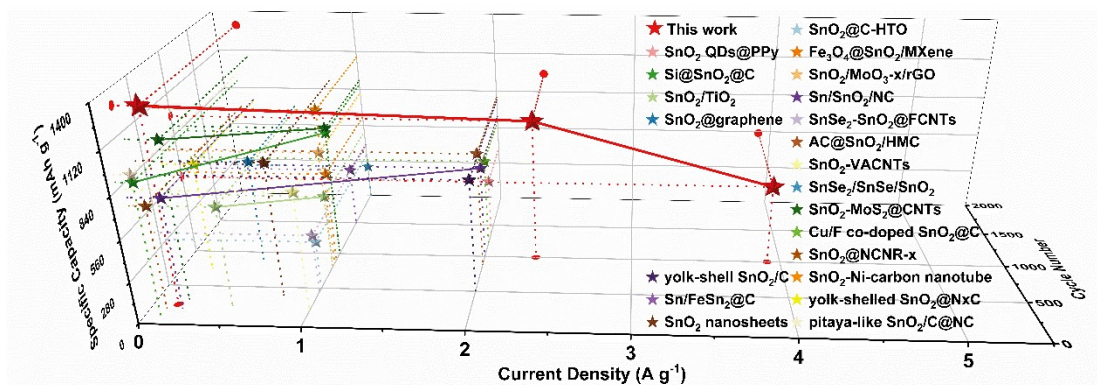


Figure S13 (f) Comparison of the cycling performance of SnO₂-based electrodes and the designed SnO₂@CF electrode.

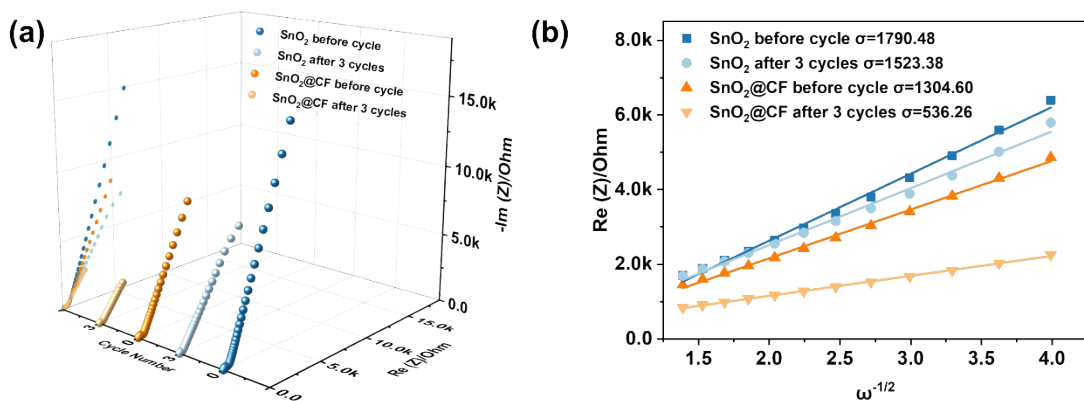


Figure S14 (a) Nyquist plots of the SnO₂@CF and SnO₂ with before and after cycling tests. (b) The fitting results of σ value.

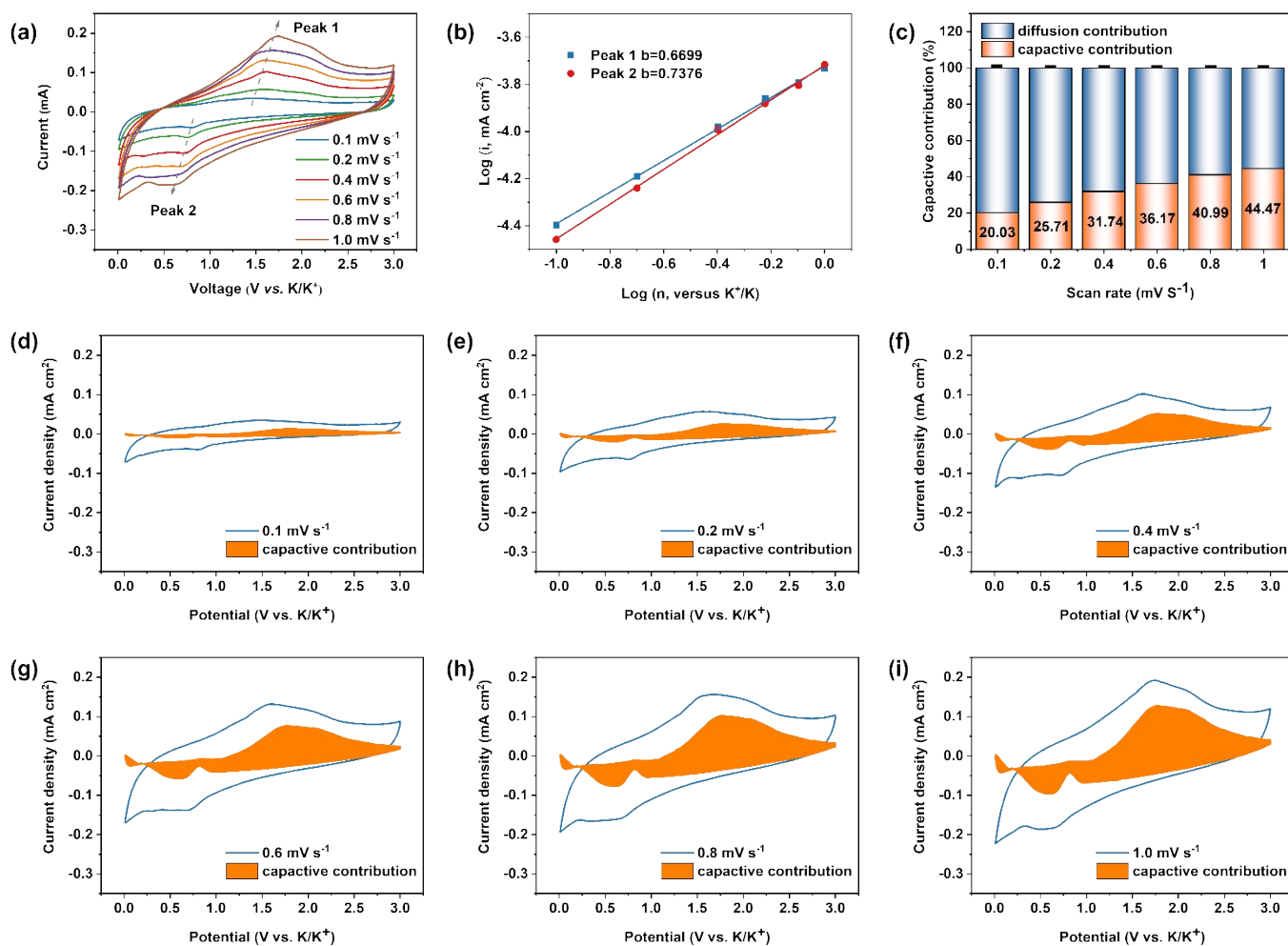


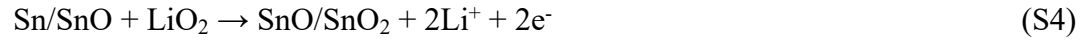
Figure S15 (a) CV curves of SnO₂@CF at gradually increasing scan rates. (b) the b -value of the linear regression. (c) The percentages of capacitive/diffusion contribution.

The capacitive contributions at different scan rates: (d) 0.1 mV s⁻¹, (e) 0.2 mV s⁻¹, (f) 0.4 mV s⁻¹, (g) 0.6 mV s⁻¹, (h) 0.8 mV s⁻¹, and (i) 1.0 mV s⁻¹.

The CV curves of SnO₂@CF, measured at scan rates of 0.1, 0.2, 0.4, 0.6, 0.8, and 1.0 mV s⁻¹, are shown in **Fig. S15(a)**. The shape of all CV curves was well preserved, and the peak current increased with the scan rate, indicating a rapid electrochemical response of the electrode material to the embedding/de-embedding of Li⁺. **Eq. (S6)** describes the relationship between scan rate and peak current. The logarithm is taken to obtain the **Eq. (S7)**. When b approaches 0.5, the electrode is considered to exhibit diffusion-controlled behavior. Conversely, if b approaches 1.0, the electrode is considered to exhibit capacitance-controlled behavior. **Fig. S15(b)** shows the peaks observed in both the cathodic and anodic processes, corresponding to b -values of 0.6699 and 0.7376, respectively, demonstrate that the charge storage is attributed to the synergistic effect of capacitive-controlled and diffusion-controlled processes.

In addition, the contribution ratio of diffusion control and capacitance control can be described by **Eq. (S8)**. In **Eq. (S8)**, k_1v represents the non-Faradic response, while $k_2v^{0.5}$ corresponds to the Faradic response. By taking the logarithm of both sides of **Eq. (S3)** and dividing by $v^{0.5}$, **Eq. (S9)** is obtained, allowing for the quantitative evaluation of the capacitive contribution. The capacitive contribution distribution and its percentage share were evaluated for other scan rates (**Fig. S15(c)**). Additionally, the capacitive contributions at scan rates of 0.1, 0.2, 0.4, 0.6, 0.8 and 1.0 mV s⁻¹ were found to be 20.03%, 25.71%, 31.74%, 36.17%, 40.99%, and 44.47%, respectively (**Fig. S15(d)-(i)**). Therefore, the Li storage in SnO₂@CF is dominated by the diffusion contribution.

Equation:



$$Z_{real} = R_e + R_{ct} + \sigma\omega^{1/2} \quad (\text{S5})$$

$$i = av^b \quad (0.5 < b < 1.0) \quad (\text{S6})$$

$$\log(i) = b \log(v) + \log(a) \quad (\text{S7})$$

$$i = k_1v + k_2v^{0.5} \quad (\text{S8})$$

$$i / v^{0.5} = k_1v^{0.5} + k_2 \quad (\text{S9})$$

Table S1 Comparison results of cycling performances between the SnO₂@CF and other recently reported SnO₂-based anode materials for LIBs.

Electrodes	Current density (A g ⁻¹)	Cycle number	Specific capacity (mA h g ⁻¹)	Ref.
SnO₂@CF	0.16 (0.2C)	250	1301	This work
	2.3 (3C)	1000	973	
	3.9 (5C)	1000	548	
SnO ₂ QDs@PPy	2.0	1360	399	1
Si@SnO ₂ @C	1.0	1000	874.3	2
	0.1	100	888.7	
SnO ₂ /TiO ₂	0.5	250	673.7	3
	1.0	800	510.5	
SnO ₂ @graphene	0.5	900	703.1	4
SnO ₂ @C-HTO	1.0	500	330	5
Fe ₃ O ₄ @SnO ₂ /MXene	1.0	900	626.1	6
SnO ₂ /MoO _{3-x} /rGO	1.0	800	813	7
Sn/SnO ₂ /NC	0.2	200	747.9	8
	2.0	1000	644.1	
SnSe ₂ -SnO ₂ @FCNTs	1.0	450	400	9
AC@SnO ₂ /HMC	2.0	800	792.1	10
SnO ₂ -VACNTs	1.0	200	800	11
SnSe ₂ /SnSe/SnO ₂	1.0	2000	233	12
SnO ₂ -MoS ₂ @CNTs	0.2	320	1069.3	13
	1.0	1000	904.5	
Cu/F co-doped SnO ₂ @C	2.0	1200	612	14
SnO ₂ @NCNR-x	0.1	200	694.2	15
SnO ₂ -Ni-carbon nanotube	1.0	850	1078.2	16
yolk-shell SnO ₂ /C	2.0	500	771	17
Sn/FeSn ₂ @C	1.0	1500	409	18
SnO ₂ nanosheets	0.782	400	901.2	19
yolk-shelled SnO ₂ @NxC	0.2	750	740	20
pitaya-like SnO ₂ /C@NC	0.1	100	936.8	21

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