Beyond Metals: Theoretical Discovery of Semiconducting MAX Phases and their Potential Application in Thermoelectrics

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Calculation of scattering rates and relaxation times

The momentum relaxation rate for acoustic deformation potential scattering is given by [1, 2]

$$\frac{1}{\tau_{ac}(k)} = \frac{e^2 k_B T E_D^2 k^2}{3\pi \hbar^2 c_{el} v(k)} [3 - 8c^2(k) + 6c^4(k)],$$
(S1)

where c_{el} is the spherically averaged elastic constant and E_D is acoustic deformation potential.

The momentum relaxation rate for ionized impurity scattering is given by [1, 2]

$$\frac{1}{\tau_{ii}(k)} = \frac{e^4 N}{8\pi\epsilon_0^2 \hbar^2 k^2 v(k)} [D(k)ln(1 + \frac{4k^2}{\beta^2}) - B(k)],$$
(S2)

where ϵ_0 is the dielectric constant, β is the inverse screening length and N, the concentration of ionized impurity. The expressions for β , D(k) and B(k) are given in [3]. This scattering mechanism is important at low temperatures and high carrier concentrations. $\frac{1}{\tau_{el}(k)}$ is the sum of momentum relaxation rates of all elastic scattering processes considered (Matthiessen's rule) *i.e.* $\frac{1}{\tau_{el}(k)} = \frac{1}{\tau_{ac}(k)} + \frac{1}{\tau_{ii}(k)}$. For inelastic POP scattering, the *out scattering* operator S_o is given by [1]

$$S_o = (N_{po} + 1 - f^-)\lambda_0^- + (N_{po} + f^+)\lambda_0^+$$
(S3)

and the *in scattering* operator S_i is given by [1]

$$S_{i} = (N_{po} + f)\lambda_{i}^{-}g^{-} + (N_{po} + 1 - f)\lambda_{i}^{+}g^{+}$$
(S4)

Here, λ_o and λ_i are the rates of *out* scattering and *in scattering* respectively. The expressions for which are provided in [3]. Also, the subscript plus and minus denotes the scattering by absorption or emission respectively. N_{po} is the number of phonons and is given by [1]

$$N_{po} = \frac{1}{exp(\hbar\omega_{po}/k_BT) - 1}.$$
(S5)

The POP scattering rate is calculated iteratively using Rode's formalism.



FIG. S1: The HSE06 functional projected band structures for all structures..



FIG. S2: Ternary phase diagrams for Sc₂SC, Sc₂SeC, Y₂SeC, and Sc₃AuC₂.





FIG. S3: The projected phonon dispersion curve.



FIG. S4: The phonon projected density of states for all structures.



FIG. S5: The Grünisen parameter of all structures.



FIG. S6: The average Grünisen parameter of all structures.



FIG. S7: Lattice thermal conductivity decomposed on various components.



FIG. S8: The accumulative kappa Vs mean free path for all structures.



FIG. S9: The band structure of (a) Sc_2SC , (b) Y_2SC , (c) Sc_3AuC_2 , and (d) Y_3AuC_2 using appropriate U values.

The band structure of four MAX phases obtained using GGA+U method [4] is given in Figure S9. As a representation, the ab-initio parameters of Sc_2SC and Y_2SC used in the transport code, AMMCR [3, 5, 6] are given in Table S1.

Parameters	Sc ₂ SC	Y ₂ SC	
Acoustic deformation potentials, $D_A(eV)$:			
$D_{A,LA}$	7.90	5.74	
$D_{A,TA}$	7.90	5.74	
$D_{A,ZA}$	6.80	4.36	
Elastic modulus, $C_A(dynes/cm^2)$			
$C_{A,LA}$	2.76 x 10 ¹²	$2.20 \ge 10^{12}$	
$C_{A,TA}$	2.76 x 10 ¹²	$2.20 \ge 10^{12}$	
$C_{A,ZA}$	2.68 x 10 ¹²	2.17 x 10 ¹²	
Polar optical phonon frequency $\omega_{pop}(THz)$			
$\omega_{pop,LO}$	6.87	5.67	
$\omega_{pop,TO}$	6.87	5.67	
$\omega_{pop,ZO}$	8.76	7.45	
High frequency dielectric constant, ϵ_{∞}	11.49	9.62	
Low frequency dielectric constant, ϵ_0	26.34	23.50	

TABLE S1: Material parameters used for Sc_2SC and Y_2SC



FIG. S10: Electron mobility and conductivity of (a,b) Sc_2SC (first row) and (c,d) Y_2SC (second row) MAX phases. The filled and empty squares represent the property calculated using Rode's method and RTA respectively. (e) The bar plot representing in-plane and out-of-plane mobility of Sc_2SC , Y_2SC , Sc_3AuC_2 and Y_3AuC_2 at 300 K and an electron concentration of 5×10^{15} cm⁻³.



FIG. S11: (a) Total scattering rate and (b) contribution of LO, TO and ZO modes to the scattering rate of Sc₂SC at 300 K for an electron concentration of 5×10^{15} cm⁻³. (c) Total scattering rate of Sc₂SC as a function of temperature. (d) Average relaxation time of Sc₂SC, Y₂SC, Sc₃AuC₂ and Y₃AuC₂.

In Figure S11, the scattering rate of Sc₂SC for an electron concentration of 5×10^{15} cm⁻³ is provided. Also, the average relaxation time for all the four MAX phases are shown.

The electron mobility and conductivity of four semiconducting MAX phases are calculated using AMMCR code, which employs Rode's iterative method. These values are compared with those obtained using relaxation time approximation (RTA). For Sc₂SC and Y₂SC, the corresponding plots are given in Figure S10.

	T = 300K			T = 500K				
	S	$S^2\sigma/\tau$	κ _l	zT	S	$S^2\sigma/\tau$	κ _l	zT
Sc ₂ SC								
p – type	584.97	0.46	12.89	0.006	376.10	0.96	7.49	0.037
n – type	-583.72	0.44	12.89	0.006	-377.27	1.2	7.49	0.041
Y ₂ SC								
p-type	1300.27	1.05	7.69	0.02	795.10	2.10	4.53	0.085
n – type	-1279.11	0.53	7.69	0.01	-762.79	1.17	4.53	0.05
Sc ₂ SeC								
p-type	235.22	0.35	12.16	0.009	176.26	0.65	7.08	0.043
n – type	-241.04	0.48	12.16	0.011	-195.30	3.95	7.08	0.063
Y ₂ SeC								
p-type	592.20	0.49	4.89	0.023	378.71	1.00	2.94	0.15
n – type	-566.37	0.50	4.89	0.02	-367.70	1.04	2.94	0.14
Sc ₃ AuC ₂								
p-type	694.50	2.96	28.45	0.02	406.50	4.01	16.73	0.073
n – type	-580	1.63	28.45	0.01	-351.20	2.84	16.73	0.047
Y ₃ AuC ₂								
p-type	556.73	0.4	2.21	0.07	319.15	0.55	1.3	0.27
n-type	-604.07	0.78	2.21	0.14	-375.30	1.09	1.3	0.50

TABLE S2: Calculated S($\mu V K^{-1}$), PF = $S^2 \sigma / \tau (10^{14} m W / m K^2 s)$, $\kappa_l (Wm^{-1} K^{-1})$ and zT for all structures along z-axis at the different tempretures.

	T = 300K				T = 500K			
	S	$S^2\sigma/\tau$	κ _l	zT	S	$S^2\sigma/\tau$	κ _l	zT
Sc ₂ SC								
p-type	522.45	0.45	11.01	0.01	315.90	0.93	6.31	0.042
n – type	-639.30	1.69	11.01	0.026	-424.50	3.23	6.31	0.14
Y ₂ SC								
p-type	1257.20	2.47	6.36	0.045	735.17	4.84	3.72	0.19
n – type	-1313.17	1.42	6.36	0.032	-811.53	2.74	3.72	0.16
Sc ₂ SeC								
p – type	180.13	0.40	9.38	0.013	135.41	0.83	5.48	0.065
n – type	-296.34	1.85	9.38	0.055	-234.05	3.42	5.48	0.25
Y ₂ SeC								
p-type	522.80	1.78	5.88	0.023	318.77	3.28	3.46	0.13
n – type	-632.18	1.55	5.88	0.072	-415.65	2.91	3.46	0.33
Sc ₃ AuC ₂								
p – type	758.25	8.48	8.27	0.17	495.75	12.22	4.92	0.66
n – type	-632.83	1.74	8.27	0.042	-379.92	3.77	4.92	0.22
Y ₃ AuC ₂								
p-type	635.90	5.28	2.88	0.57	408.07	8.45	1.77	1.73
n-type	-556.28	1.98	2.88	0.25	-331.85	3.56	1.77	0.88

TABLE S3: Calculated S($\mu V K^{-1}$), PF = $S^2 \sigma / \tau (10^{14} m W / m K^2 s)$, $\kappa_l (Wm^{-1} K^{-1})$ and zT for all structures along ab-lattice at the different tempretures.



FIG. S12: Seebeck and power factor and zT parameter for all structures.

Coordinates of Sc₂SC

POSCAR of Sc2SC

1.0

3.4391777515 0.000000000 0.000000000 -1.7195888758 2.9784153010 0.0000000000 0.000000000 0.000000000 11.9092426300 Sc S C

422

Direct

Coordinates of Y₂SC

POSCAR of Y2SC

1.0

3.7083451748 0.000000000 0.000000000 -1.8541725874 3.2115211274 0.0000000000 0.000000000 0.000000000 12.8999385834 Y S C

422

Direct

Coordinates of Sc₂SeC

POSCAR of Sc2SeC

1.0

3.4966380596 0.000000000 0.000000000 -1.7483190298 3.0281773875 0.0000000000 0.000000000 0.000000000 12.3083686829 Sc Se C 4 2 2

Direct

Coordinates of Y₂SeC

POSCAR of Y2SeC

1.0

3.7715880871 0.000000000 0.000000000 -1.8857940435 3.2662910960 0.0000000000 0.000000000 0.000000000 13.3069391251 Y Se C 4 2 2

Direct

Coordinates of Sc₃AuC₂

POSCAR of Sc3AuC2

1.0

3.3747251034 0.000000000 0.000000000 -1.6873625517 2.9225976703 0.000000000 0.000000000 0.000000000 19.1152229309 Sc Au C

SC AU

624

Direct

0.00000000 0.0000000 0.00000000 0.0000000 0.0000000 0.50000000 0.33332986 0.666666985 0.130154997 0.6666666985 0.333332986 0.630155027 0.6666666985 0.333332986 0.869844973 0.333332986 0.6666666985 0.369845003 0.00000000 0.00000000 0.25000000 0.333332986 0.666666985 0.925032973 0.666666985 0.333332986 0.425033003 0.666666985 0.333332986 0.074966997 0.333332986 0.666666985 0.574967027

Coordinates of Y₃AuC₂

POSCAR of Y3AuC2

1.0

3.6488833427 0.000000000 0.000000000 -1.8244416714 3.1600256703 0.000000000 0.000000000 0.00000000 20.2883396149

Y Au C

624

Direct

0.00000000 0.0000000 0.00000000 0.0000000 0.0000000 0.50000000 0.33332986 0.666666985 0.135432005 0.666666985 0.333332986 0.635432005 0.666666985 0.333332986 0.864567995 0.333332986 0.666666985 0.364567995 0.00000000 0.00000000 0.25000000 0.00000000 0.00000000 0.750000000 0.333332986 0.666666985 0.922215998 0.666666985 0.333332986 0.422215998 0.666666985 0.333332986 0.077784002 0.333332986 0.666666985 0.577784002

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Some additional discussion regarding relaxation time:

Though there is currently no experimental data available on semiconducting MAX phases as they have not been synthesized. Nevertheless, we have a reliable estimation of the relaxation time in metallic MAX phases based on our previous estimation using experimental results for electrical conductivity, which can be found in the supporting information file of Physical Chemistry Chemical Physics 16, 7841(2014). Below, we have summarized the results.

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M ₂ AX	a (Å)	c (Å)	Experimental	Calculated	Estimated
Phase			σ (1/μΩ.m)	σ/ τ (1/Ω.m.s)	τ (s)
Ti ₂ AlC	3.069,	13.737,	2.7 ^{1,2} , 2.8 ¹	2.797×10^{20}	1.036×10^{-14} ,
	3.051	13.637		20	0.999×10 ⁻¹⁴
Ti ₂ AlN	2.997,	13.644,	$3.2^3, 4.00^1$	2.700×10^{20}	0.844×10^{-14}
	2.989 ¹	13.614 ¹		20	0.675×10 ⁻¹⁴
Ti ₂ SC	3.205,	11.257,	1.923	1.450×10^{20}	0.754×10 ⁻¹⁴
T: C.C	3.004	12 001	1 A ³	3.005×10^{20}	0.008×10^{-14}
H2GeC	3.07 ⁴	12.991, 12.93^{4}	4.4	5.595~10	0.908~10
TiaInC	3.148.	14.208.	5.000 ⁶	2.367×10^{20}	0.473×10 ⁻¹⁴
11/11C	3.1344	14.0774			
Ti ₂ SnC	3.178,	13.777,	4.545 ⁷	3.944×10 ²⁰	0.868×10^{-14}
1120110	3.163 ⁴	13.6794			
V ₂ AIC	2.906,	13.104,	4.000 ⁸	3.095×10 ²⁰	0.774×10^{-14}
	3.1 ⁴	13.83 ⁴			
Cr ₂ AlC	2.847,	12.671,	1.389 ⁸ , 1.4 ²	2.998×10 ²⁰	2.129×10 ⁻¹⁴ ,
	2.863 ²	12.814 ²			2.158×10^{-14}
Zr ₂ SnC	3.390,	14.684,	3.571 ⁷ , 7.00 ⁹	4.475×10^{20}	1.253×10^{-14} ,
	3.358 ⁴	14.57 ⁴			0.639×10 ⁻¹⁴
Zr ₂ PbC	3.418,	14.926,	2.778 ⁷	4.455×10^{20}	1.604×10^{-14}
	3.38 ⁴	14.66 ⁴	10	20	
Nb ₂ AlC	3.116,	13.932,	3.448 ¹⁰	4.174×10^{20}	1.210×10^{-14}
	3.14	13.8 ⁴	7 0	20	14
Nb ₂ SnC	3.268,	13.888,	2.222, 4.00	6.070×10 ²⁰	2.732×10^{-14}
	3.214	13.802	6	20	1.517×10 ⁻¹⁴
Hf ₂ InC	3.324,	14.890,	5.263°	3.270×10 ²⁰	0.621×10^{-14}
	3.309	14.723	7 0	20	14
Hf ₂ SnC	3.345,	14.449,	2.222', 2.4	5.379×10 ²⁰	2.421×10^{-14}
	3.320	14.388			2.241×10 ⁻¹⁴
Hf ₂ PbC	3.379,	14.680,	13.333'	5.391×10 ²⁰	0.404×10^{-14}
	3.55'	14.46'	a a 12		1 2 2 2 1 2 1 4
Ta ₂ AlC	3.093,	13.957	3.91	4.803×10 ²⁰	1.228×10 ⁻¹⁴
	3.079	13.860			

As seen in the above Table, we conducted an extensive search in the literature and gathered data on the electrical conductivity (σ) for 16 different M2AX systems, as listed in the table below. Using Boltzmann theory and first-principles calculations, we determined the σ/τ for these M₂AX systems and estimated the relaxation time (τ) by fitting the experimental data of σ . The estimated relaxation time ranges from 0.4×10^{-14} s to 2.7×10^{-14} s, as shown in the table. Despite limitations in the number of conductivity measurements and the scattered nature of the data for different M₂AX systems, which involve various transition metals (M), interlayer elements (A), and X atoms (carbon or nitrogen), we can conclude that the average relaxation time of approximately 1.251×10^{-14} s can be considered a reliable value for the relaxation time in all the considered M₂AX systems with similar hexagonal structures. This value is comparable to the relaxation time obtained for semiconducting MAX phases through scattering theory.

Moreover, we have tested the Rode's method of estimating mobility on other known bulk semiconductors like CdS and ZnSe. The results are in good agreement with available/existing experimental data and are published [Computer Physics Communications 2559, 107697 (2021) and J. Phys.: Condens. Matter 31, 345901 (2019)]. These mobilities are obtained from the determination of scattering rate for each scattering mechanism that is significant to the material. The average relaxation time is calculated from these scattering rates. If the mobilities match well with experimental data, we expect the relaxation time to be correct.

The mobility of CdS as a function of temperature for a carrier concentration of $6.9 \times 10^{15} \text{ cm}^{-3}$ [Computer Physics Communications 2559, 107697 (2021)] is given in the following figure. The figure shows good qualitative and quantitative agreement between experimental and calculated curves. The contribution of each scattering mechanism towards mobility is also provided. In Figure 1(b), the red curve corresponds to experimental data and the black line represents the total mobility due to acoustic deformation potential, piezoelectric, polar optical phonon and ionized impurity scattering.



Figure 1: (a) Calculated and experimental measured mobility of CdS as a function of temperature for a specific carrier concentration. (b) The contribution of different scattering mechanisms towards mobility for the same [Computer Physics Communications 2559, 107697 (2021), Phys. Rev. 169, 577 (1968)].

For ZnSe, the electron mobility is calculated for three different samples with different carrier concentrations over a wide temperature range. The donor and acceptor concentrations of the samples are taken as shown in Table 1. As can be seen in the mobility versus temperature plots, the Rode's method gives mobility which are in better agreement with experimental values compared to those obtained using relaxation time approximation.



Figure 2: Calculated and experimentally measured mobility of ZnSe with temperature variation at different carrier concentrations [J. Phys.: Condens. Matter 31, 345901 (2019)]. The details about donor and acceptor concentrations are given in Table 1.

Table 1: Carrier of different experimentally fabricated n-ZnSe samples.

Sample	$N_{\rm D}$ - $N_{\rm A}$ (cm ⁻³)	Donor N _D (cm ⁻³)	Acceptor N _A (cm ⁻³)
a [#]	1×10^{15}	2.9×10^{15}	1.9×10^{15}
b [\$]	1.1×10^{16}	6×10^{16}	$4.9 imes 10^{16}$
c [*]	6.3×10^{15}	7.5×10^{15}	1.2×10^{15}

[[#]J. Appl. Phys. 42, 1204 (1971), ^{\$}J. Phys.: Condens. Matter 18, 8113 (2006), ^{*}Phys. Status Solidi b 238, 45 (2003)]

Finally, the acoustic deformation potential is calculated using standard procedure followed in other theoretical works based on Bardeen and Shockley proposition (Phys. Rev. 80, 72 (1950)). E_D is the deformation potential constant. It represents the shift of the conduction band edge (i.e. conduction band minima for electrons) induced by the strains and is defined



Figure 3: Linear fit of conduction band edge of Sc₂SC MAX phase compound as a function of small strain.

as $E_D = \partial E_{CBM} / \partial \epsilon$. The strain ϵ is given by $(l-l_0)/l_0$, where l and l_0 are the strained and equilibrium lattice parameters respectively. For Sc₂SC MAX phase, the shift of conduction band edge as a function of strain (for the strain in the x-direction) is given above.

function used for fitting: $f(x) = E(\varepsilon) = a\varepsilon + b$

Final set	of parameters	Asymptotic Standard Error				
a	= -7.89714	+/- 0.07763	(0.983%)			
b	= 6.09046	+/- 0.0007763	(0.01275%)			
$E_{1x} = \partial E / \partial \epsilon = -7.897$						

Please note that it is the polar optical phonon (POP) scattering rate which is most dominant in the MAX phases considered. The Rode's method of estimating mobility is found to be good for polar

semiconductors as explained in Refs. [Computer Physics Communications 2559, 107697 (2021) and J. Phys.: Condens. Matter 31 345901 (2019)].

To verify the absence of imaginary modes, the inset shows a zoomed-in view of the phonon frequencies at the Gamma point region. It clearly demonstrates that there are no imaginary phonon modes in these structures.



Phonon spectra of the investigated MAX phase compounds. The inset shows the zoomed-in view at the zone centers.