

# A density functional theory study of two-dimensional bismuth selenite: layer-dependent electronic, transport and optical properties with spin-orbit coupling†

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## AFFILIATIONS

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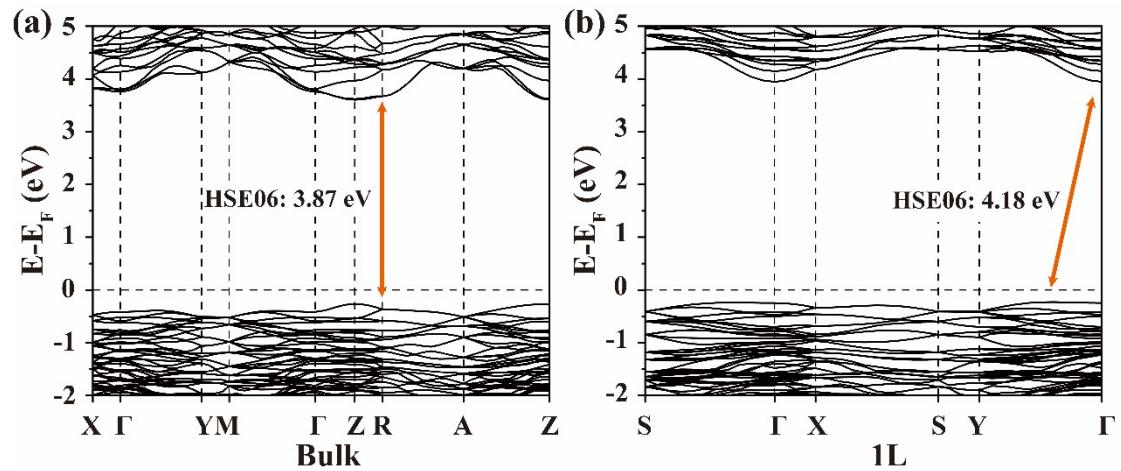
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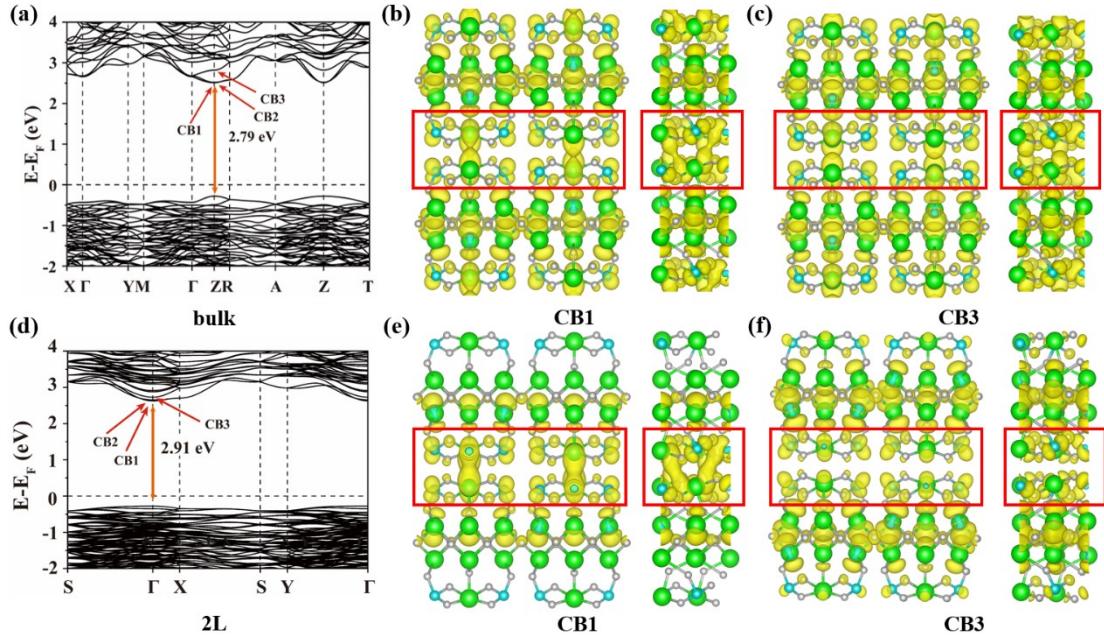
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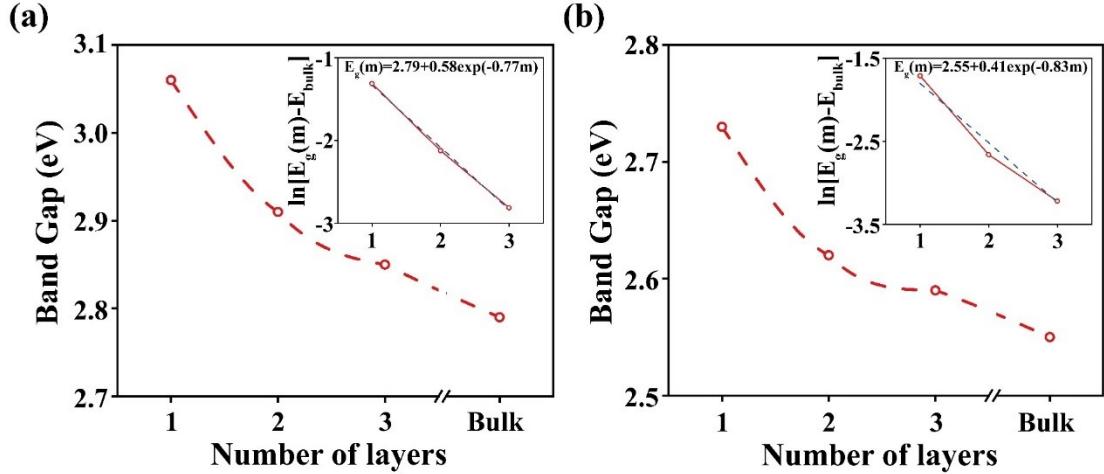
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**FIG. S1.** Electronic structures of (a) bulk and (b) 1L  $\text{Bi}_2\text{SeO}_5$  calculated with HSE06 functional.



**FIG. S2.** (a, d) Electronic structures of bulk/2L  $\text{Bi}_2\text{SeO}_5$ . CB1, CB2 and CB3 are marked. Spatial distribution of wavefunction for the lowest CBs using an isosurface of  $0.0008 \text{ e}\text{\AA}^{-3}$  for bulk (b, c) and 2L (e, f)  $\text{Bi}_2\text{SeO}_5$ . For CB1, it is obvious to see that the interlayer charge distribution mimics the bonding-like behavior while antibonding-like for CB3. Besides, the perpendicular charge distribution always mixes some in-plane components, indicating the contribution of in-plane  $p$  orbitals.



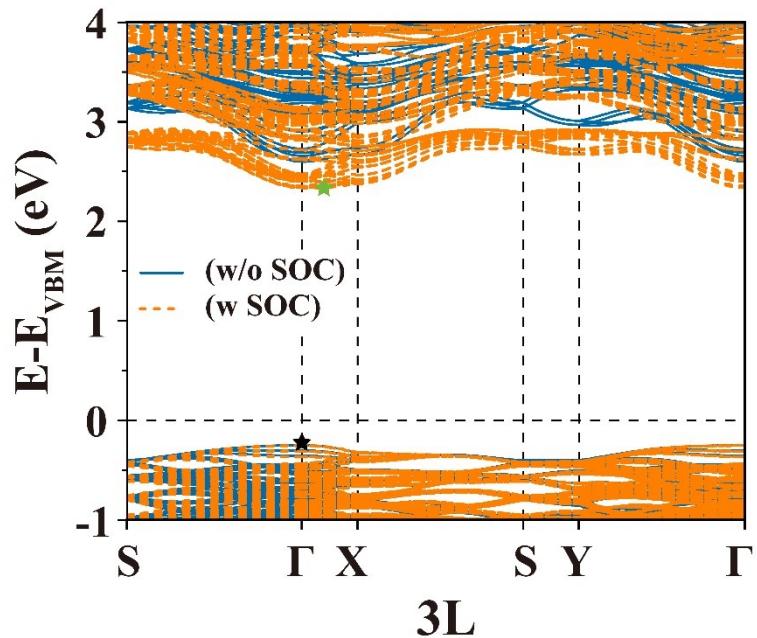
**FIG. S3.** The variation of bandgap  $E_g(m)$  as a function of layer number  $m$ . Inset: The blue dash line indicates the linear fitting. For (a), SOC is not considered while for (b), SOC is included.

### Note 1. The quantum confinement effect of 2D $\text{Bi}_2\text{SeO}_5$ .

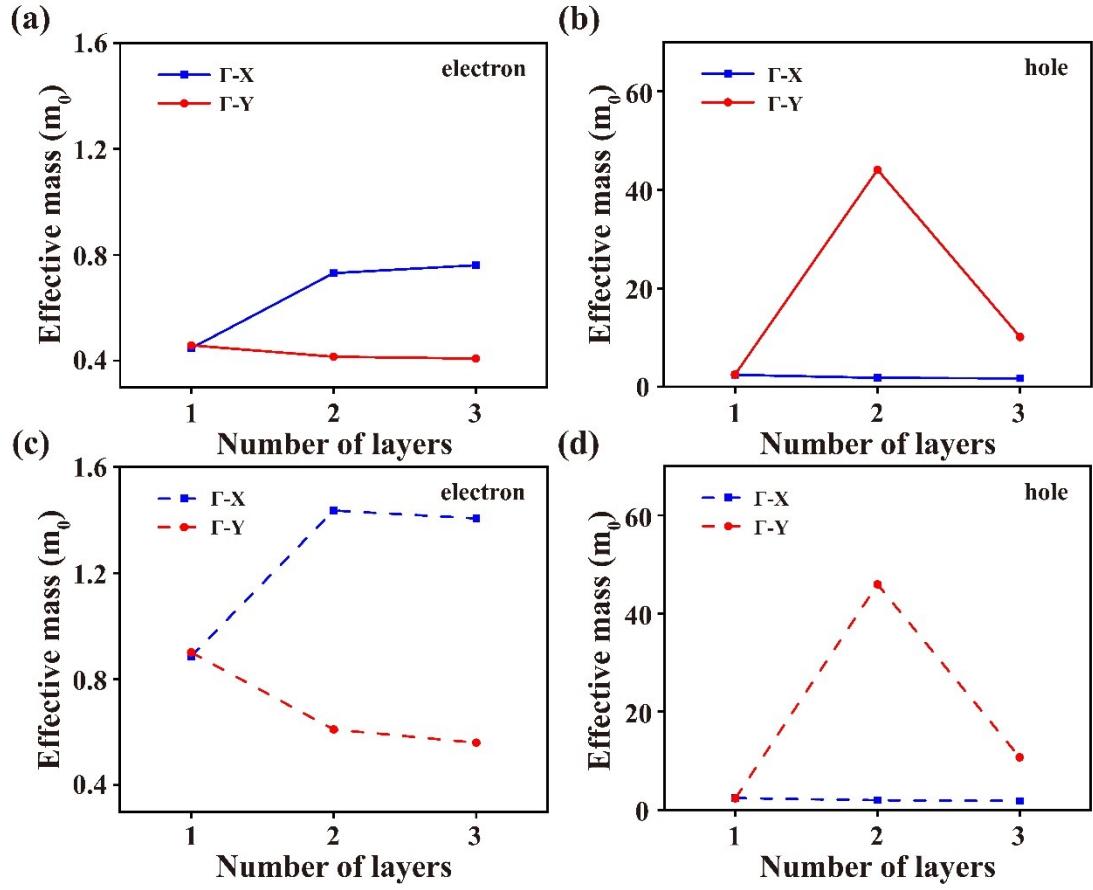
To qualitatively explore the quantum confinement effect, we calculate the relationship between the bandgap ( $E_g$ ) and the layer number ( $m$ ). As shown in **Fig. S3-inset**, an obvious linear dependence between  $\ln[E_g(m)-E_g(\text{bulk})]$  and  $m$  can be observed. Such behavior indicates that  $E_g$  versus  $m$  follows the equation below,

$$E_g(m) = E(\text{bulk}) + \alpha \exp(-\beta m). \quad (1)$$

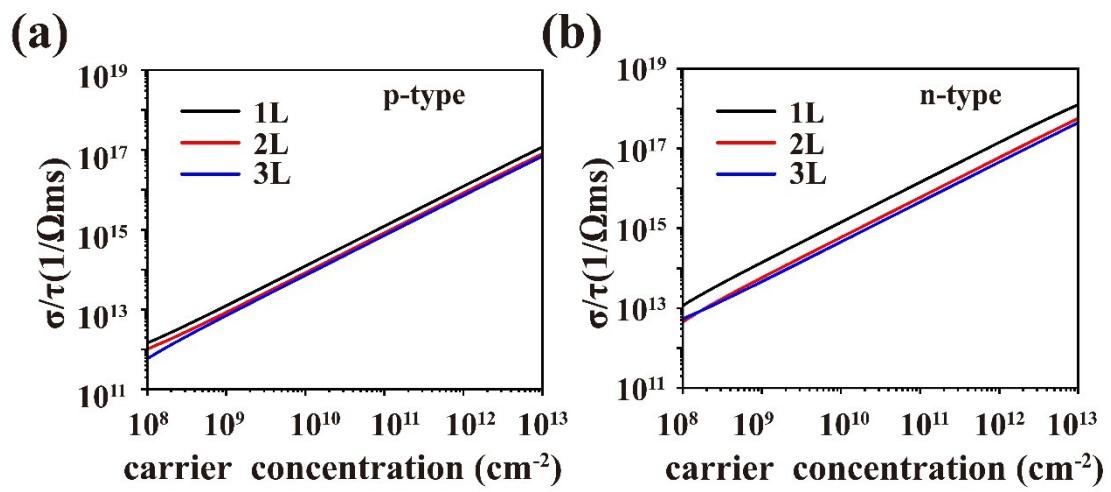
where  $\alpha$  and  $\beta$  are the characteristic parameters. The fitted curves are shown in **Fig. S3**, without considering SOC, the values of  $\alpha$  and  $\beta$  are 0.58 and 0.77, respectively. When SOC is taken into account, the values of  $\alpha$  and  $\beta$  are 0.41 and 0.83, respectively. This indicates that SOC plays a crucial role in the bandgap. With the above expression, it is convenient to predict the bandgap of multilayer  $\text{Bi}_2\text{SeO}_5$  without the detailed calculation, which can accelerate the selection of 2D  $\text{Bi}_2\text{SeO}_5$  with specific bandgap.



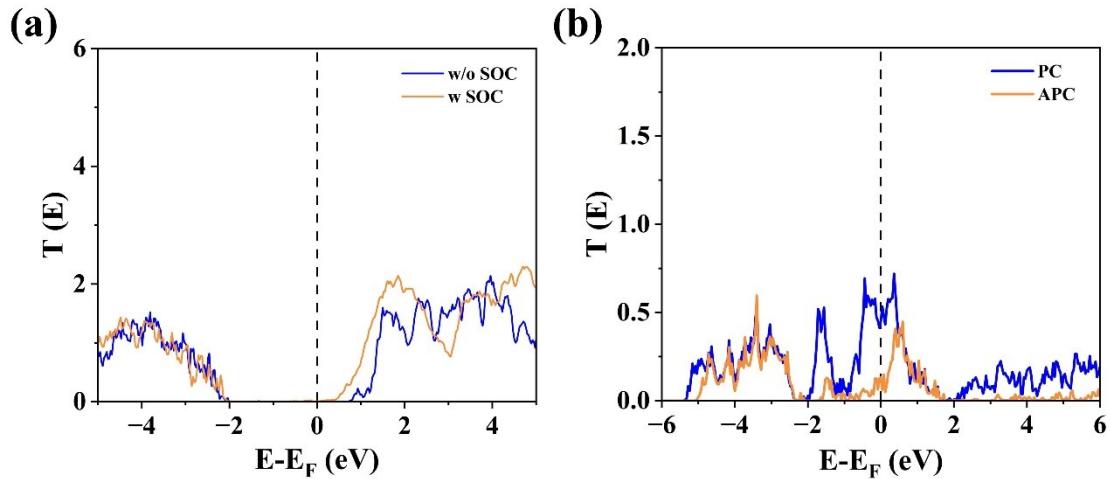
**FIG. S4.** Electronic band structure of 3L  $\text{Bi}_2\text{SeO}_5$  with SOC (orange dashed lines). For easy comparison, we plot the bands without SOC together (blue solid lines). Green and black pentagrams represent CBM and VBM, respectively.



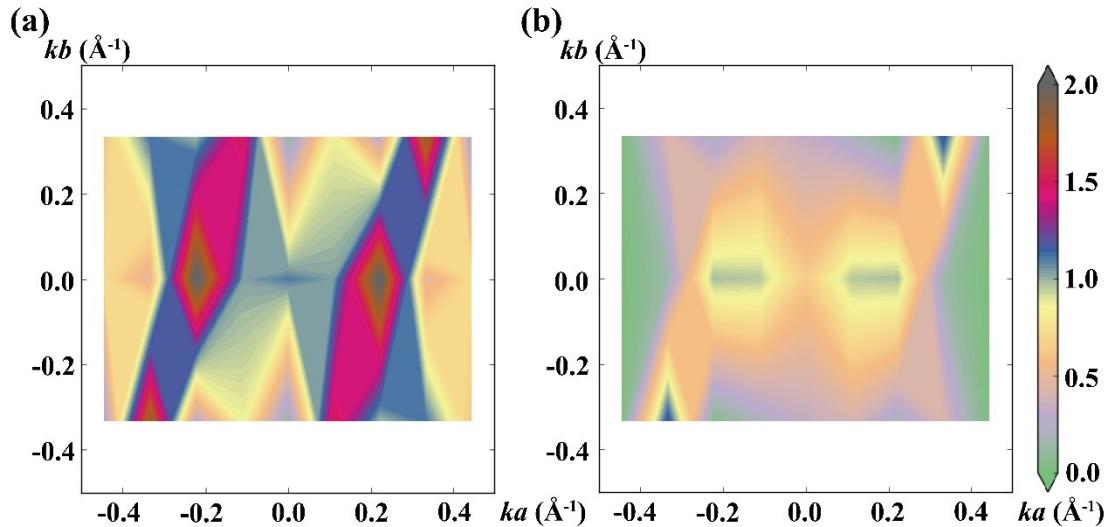
**FIG. S5.** Effective mass of (a, c) electrons and (b, d) holes along the  $\Gamma$ -X (blue) and  $\Gamma$ -Y (red) directions versus the number of layers. SOC is included for (c, d) with dashed lines.



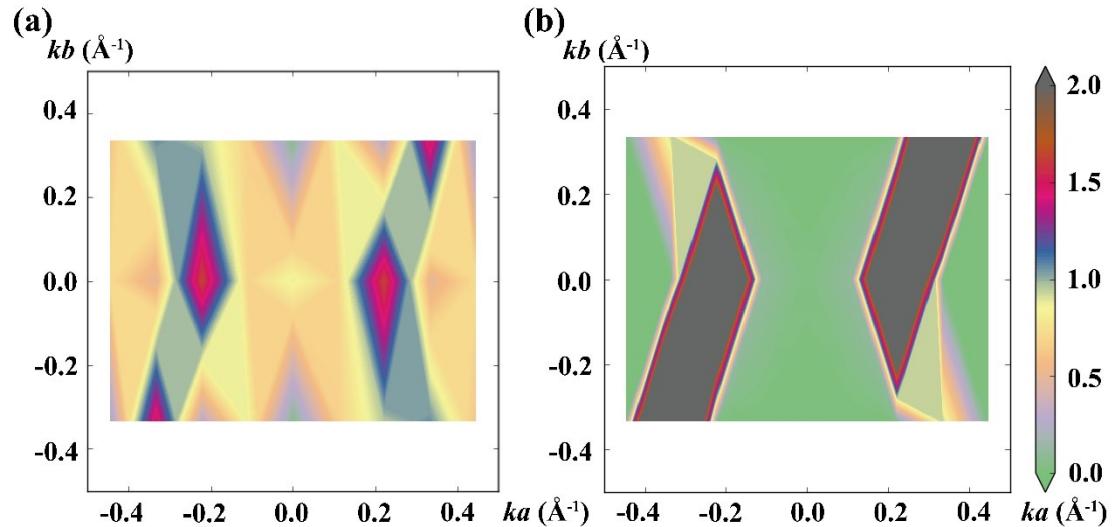
**FIG. S6.** The variation of (a-b)  $\sigma/\tau$  with different doping concentration at 300 K for 2D  $\text{Bi}_2\text{SeO}_5$ .



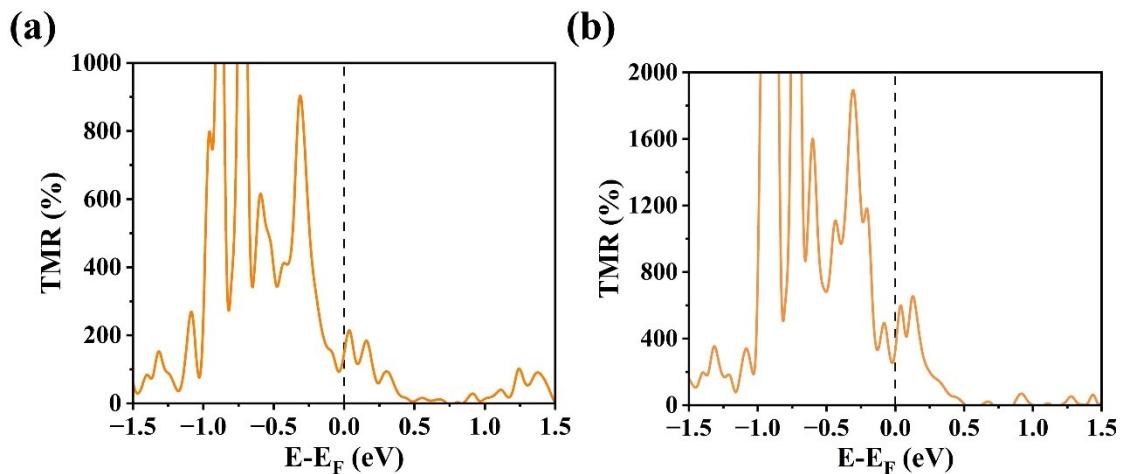
**FIG. S7.** (a) Transmission spectra of the Al/2L Bi<sub>2</sub>SeO<sub>5</sub>/Al device. (b) Transmission spectra of the FeS<sub>2</sub>/2L Bi<sub>2</sub>SeO<sub>5</sub>/FeS<sub>2</sub> MTJ.



**FIG. S8.**  $k$ -resolved transmission coefficients for the FeS<sub>2</sub>/1L Bi<sub>2</sub>SeO<sub>5</sub>/FeS<sub>2</sub> MTJ for the (a) PC and (b) APC states.  $E = 0$  eV.



**FIG. S9.**  $k$ -resolved transmission coefficients for the FeS<sub>2</sub>/2L Bi<sub>2</sub>SeO<sub>5</sub>/FeS<sub>2</sub> MTJ for the (a) PC and (b) APC states.  $E = 0$  eV.



**FIG. S10.** TMR as a function of chemical potential ( $E$ ) for (a) FeS<sub>2</sub>/1L Bi<sub>2</sub>SeO<sub>5</sub>/FeS<sub>2</sub> and (b) FeS<sub>2</sub>/2L Bi<sub>2</sub>SeO<sub>5</sub>/FeS<sub>2</sub> MTJs.

**TABLE S1.** Lattice constants and volume of bulk Bi<sub>2</sub>SeO<sub>5</sub> calculated by using different

System	a (Å)	b (Å)	c (Å)	V (Å <sup>3</sup> )
Exp. <sup>1</sup>	11.44	16.28	5.49	1022.48
PBE <sup>2</sup>	11.64	16.68	5.57	1081.444
LDA <sup>3</sup>	11.09	16.12	5.43	970.7254
DFT-D2 <sup>4</sup>	11.41	16.47	5.52	1037.333
DFT-D3 <sup>5</sup>	11.49	16.52	5.55	1053.472
DFT-D3 <sup>6</sup>	11.37	16.41	5.52	1029.931
DFT-D4 <sup>7</sup>	11.65	16.69	5.57	1083.022
TS <sup>8</sup>	11.53	16.58	5.56	1062.891
TS/HI <sup>9,10</sup>	11.42	16.47	5.53	1040.123
MBD@rSC <sup>11,12</sup>	11.46	16.50	5.54	1047.559
MBD@rSC/FI <sup>13,14</sup>	11.64	16.68	5.57	1081.444
dDsC <sup>15,16</sup>	11.38	16.43	5.52	1032.093
DFT-ulg <sup>17</sup>	11.43	16.48	5.54	1043.55
vdW-DF <sup>18</sup>	11.71	16.80	5.63	1107.579
vdW-DF2 <sup>19</sup>	11.68	16.87	5.66	1115.255
optPBE-vdW <sup>20</sup>	11.51	16.57	5.57	1062.314
optB88-vdW <sup>20</sup>	11.37	16.44	5.54	1035.552
optB86b-vdW <sup>21</sup>	11.33	16.37	5.52	1023.806
rev-vdW-DF2 <sup>22</sup>	11.35	16.40	5.53	1029.354
vdW-DF-cx <sup>23</sup>	11.28	16.31	5.50	1011.872
rVV10 <sup>24</sup>	11.23	16.32	5.52	1011.67
SCAN+rVV10 <sup>25</sup>	11.21	16.18	5.46	990.3228

computational methods.

**TABLE S2.** Relative differences between DFT and experimental results (unit: %).  $A_0$ ,  $b_0$ ,  $c_0$  and  $V_0$  are the lattice constants and volume measured in the experiment.

System	$(a - a_0)/a_0$	$(b - b_0)/b_0$	$(c - c_0)/c_0$	$(V - V_0)/V_0$
DFT-D3	-0.61	0.80	0.55	0.73
dDsC	-0.52	0.92	0.55	0.94
optB88-vdW	-0.61	0.98	0.91	1.28
optB86b-vdW	-0.96	0.55	0.55	0.13
rev-vdW-DF2	-0.79	0.74	0.73	0.67

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