

## Supporting Information

# Layerwise Replacement Method to Achieve High Solar-to-Hydrogen efficiency for Photocatalytic Water Splitting: A First Principles Study

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## Computational details

DFT-1/2 is based on Slater's half-occupation technique, with the -1/2 denoting an intermediate position between the initial ground state and the ionized state.<sup>1-2</sup> The DFT-1/2 method introduces the computed atomic self-energy potential into solids, where it is incorporated into the pseudopotential as a common pseudopotential for calculating electronic structures.<sup>3-4</sup> The self-energy potential requires correction before its introduction. The self-energy potential's reach is primarily constrained by the cutoff radius  $r_{\text{cut}}$ . The potential is trimmed by a function as

$$\Theta = \begin{cases} \left[1 - \left(\frac{r}{CUT}\right)^n\right]^3, & r \leq CUT \\ 0, & r > CUT \end{cases} \quad \text{S(1)}$$

Since  $r_{\text{cut}}$  is variable, selecting an appropriate value for  $r_{\text{cut}}$  results in the maximum presentation of the bandgap, where the correction is maximized. In this work, we determine the appropriate  $r_{\text{cut}}$  for the atoms through testing. We set the parameter  $n$  to 8, a value also used in other articles.

The formation energy of all the structures are evaluated by the following formula:

$$E_f = \frac{E_{total} - \sum_i n_i \mu_i}{\sum_i n_i} \quad \text{S (2)}$$

where  $E_{total}$  and  $n_i$  represent the total energy and the number of  $i$ -th atom of the structures in a unit cell, and  $\mu_i$  is the cohesive energy of the  $i$ -th atom in its bulk crystal.

The calculation formula of plane-averaged charge density differences  $\Delta\rho_{(z)}$  is as follow:

$$\Delta\rho_{(z)} = \rho_{Al_2S_2Se-t} - \rho_{Al} - \rho_S - \rho_{Se} \quad \text{S (3)}$$

$$\Delta\rho_{(z)} = \rho_{Al_2TeSe_2-t} - \rho_{Al} - \rho_{Te} - \rho_{Se} \quad \text{S (4)}$$

The formula for calculating the photoabsorption coefficient ( $\alpha_\omega$ ), denoted as (S5), incorporates the frequency ( $\omega$ ), with  $\varepsilon'(\omega)$  and  $\varepsilon''(\omega)$  representing the real and imaginary parts of the dielectric function, respectively. These parameters were obtained using the HSE06 hybrid functional.

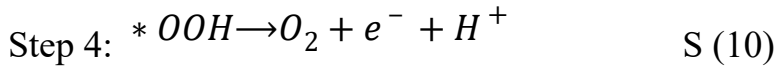
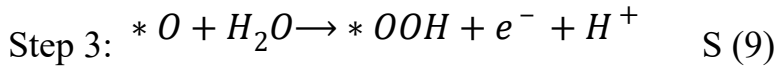
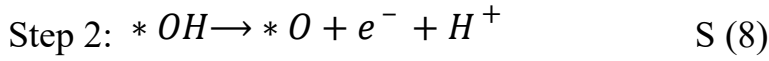
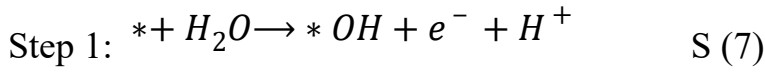
$$\alpha_{\omega} = \frac{\sqrt{2}\omega}{2} \left[ \sqrt{(\varepsilon'(\omega))^2 + (\varepsilon''(\omega))^2} - \varepsilon'(\omega) \right]^{1/2} \quad \text{S (5)}$$

The Gibbs free energy change ( $\Delta G$ ) in the water redox reactions is defined as:

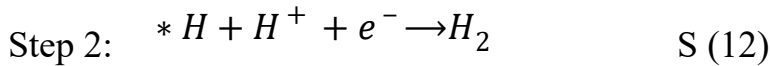
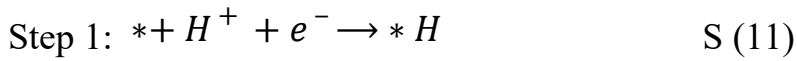
$$\Delta G = \Delta E + \Delta E_{ZPE} - T\Delta S \quad \text{S (6)}$$

where  $\Delta E$ ,  $\Delta E_{ZPE}$ , and  $\Delta S$  represent the energy difference of adsorption, zero-point energy and entropy, respectively. T is system temperature (298 K).

The OWS process comprises two half-reactions: HER, which involves a two-electron reaction, and OER, which is a four-electron reaction in an acidic solution. The OER process can be written as:



The HER process can be described as:



Then, the  $\Delta G$  for OER are achieved as:

$$\text{Step 1: } \Delta G = G_{*OH} + \frac{1}{2}G_{H_2} - G_{H_2O} - G_* - U_h \quad \text{S (13)}$$

$$\text{Step 2: } \Delta G = G_{*O} + \frac{1}{2}G_{H_2} - G_{*OH} - U_h \quad \text{S (14)}$$

$$\text{Step 3: } \Delta G = G_{*OOH} + \frac{1}{2}G_{H_2} - G_{H_2O} - G_{*O} - U_h \quad \text{S (15)}$$

$$\text{Step 4: } \Delta G = G_{*} + \frac{1}{2}G_{H_2} + G_{*O} - G_{*OOH} - U_h \quad \text{S (16)}$$

The  $\Delta G$  for HER is calculated by:

$$\Delta G = G_{*H} - \frac{1}{2}G_{H_2} - G_{*} - U_e \quad \text{S (17)}$$

the  $\mu_\alpha$  of 2D materials can be expressed as the following equation: <sup>5-6</sup>

$$\mu_\alpha = \frac{e\hbar^3 C_\alpha}{k_B T |m_\alpha^*| m_d (E_\alpha^l)^2} \quad \text{S (18)}$$

$$C_\alpha = \frac{2 \partial^2 E_{total}}{S_0 \partial (\Delta l_\alpha / l_\alpha^0)^2} \quad \text{S (19)}$$

$$E_\alpha^l = \frac{\Delta E_{ks}}{\Delta l_\alpha / l_\alpha^0} \quad \text{S (20)}$$

Here,  $e$  denotes the electron charge,  $\hbar$  represents the reduced Planck constant,  $k_B$  stands for the Boltzmann constant,  $T$  denotes the temperature (set to  $T = 300\text{K}$ ),  $m_\alpha^*$  corresponds to the effective mass in the  $\alpha$  direction of the material without strain (where  $\alpha$  is either the  $x$  or  $y$  direction),  $m_d$  is the average effective mass ( $m_d = \sqrt{m_x^* m_y^*}$ ).  $C_\alpha$  denotes the elastic modulus, with  $S_0$  represents the area of the optimized 2D unit cell without strain,  $E_{total}$  represents the total energy of the system,  $l_\alpha^0$  corresponds to the lattice constant without strain, and  $\Delta l_\alpha$  signifies the change in lattice constant. The deformation potential is calculated as S (18).

## Other supplementary contents

The HSE06 Hybrid functionals typically yield more accurate bandgaps, but their computational demands are higher.<sup>1</sup> The DFT-1/2 method, introduced in recent years as a correction function for DFT, mitigates the problem of underestimated bandgaps in standard DFT by rectifying the self-energy of valence band holes, all while maintaining rapid computational speed.<sup>3, 7</sup> Previously, DFT-1/2 has successfully predicted the bandgaps of various semiconductor materials, ranging from metal oxides to halide perovskites, with accuracy comparable to GW but at lower computational cost.<sup>8</sup> In this work, the computational results indicate that, except for  $\text{Al}_2\text{S}_2\text{Te-t}$  and  $\text{Al}_2\text{TeS}_2\text{-m}$ , the  $E_g$  of other materials calculated with PBE are approximately 0.8 eV smaller than those obtained with HSE06, while the difference between the HSE06 band gap and the DFT-1/2 corrected  $E_g$  is around 0.2 eV. Considering the fast computation speed and extremely low cost of the DFT-1/2 method, for large systems with similar structures, we can use the DFT-1/2 method to evaluate the  $E_g$ . The significant difference in  $E_g$  predicted for  $\text{Al}_2\text{S}_2\text{Te-t}$  and  $\text{Al}_2\text{TeS}_2\text{-m}$  monolayers using the HSE06 and DFT-1/2 methods may be due to their metallic nature as predicted by PBE.

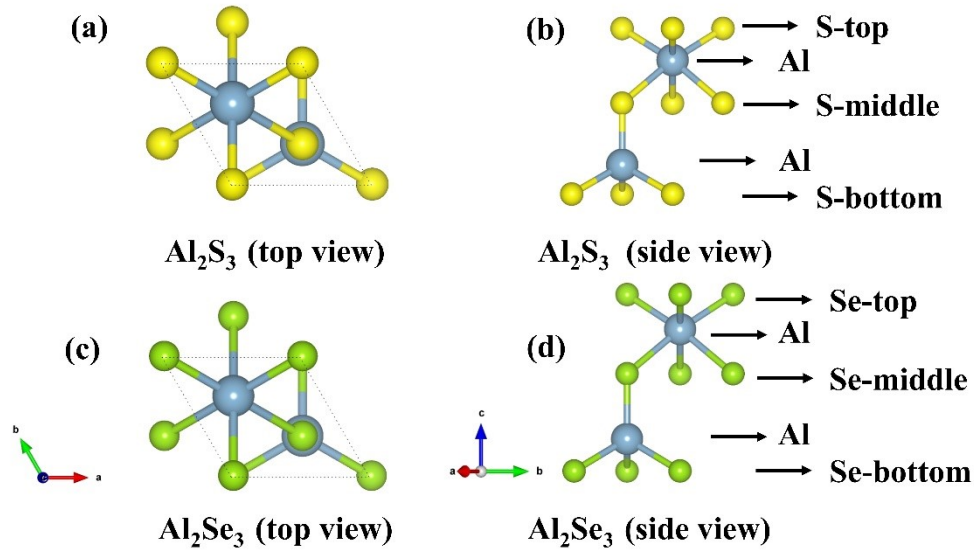


Fig. S1. The unit cells of  $\text{Al}_2\text{S}_3$  and  $\text{Al}_2\text{Se}_3$  monolayers.

Tab. S1. The lattice constant ( $a$ ), formation energy ( $E_f$ ), band gap ( $E_g$ ), and electrostatic potential difference ( $\Delta\Phi$ ) of other 10 types of  $\text{Al}_2\text{X}_2\text{X}'$  monolayers except for  $\text{Al}_2\text{S}_2\text{Se-t}$  and  $\text{Al}_2\text{TeSe}_2\text{-m}$ .

	$a(\text{\AA})$	$E_f$ (eV/atom)	$E_g$ (eV)/ PBE	$E_g$ (eV)/ DFT-1/2	$E_g$ (eV)/ HSE06	$\Delta\Phi$
$\text{Al}_2\text{S}_2\text{Te-t}$	3.750	-0.896	—	0.211	1.195	0.165
$\text{Al}_2\text{TeS}_2\text{-m}$	3.70	-0.841	—	0.829	0.403	1.892
$\text{Al}_2\text{S}_2\text{Te-b}$	3.774	-0.864	1.791	2.415	2.465	2.285
$\text{Al}_2\text{S}_2\text{Se-t}$	3.638	-1.108	1.154	2.091	1.839	2.230
$\text{Al}_2\text{SeS}_2\text{-m}$	3.618	-1.084	1.427	2.178	2.202	2.202
$\text{Al}_2\text{S}_2\text{Se-b}$	3.640	-1.096	2.431	3.360	3.255	2.287
$\text{Al}_2\text{Se}_2\text{S-t}$	3.692	-1.009	1.813	2.653	2.575	2.123
$\text{Al}_2\text{SSe}_2\text{-m}$	3.716	-1.035	1.679	2.595	2.361	2.109
$\text{Al}_2\text{Se}_2\text{S-b}$	3.691	-1.020	1.237	2.219	1.973	2.075
$\text{Al}_2\text{Se}_2\text{Te-t}$	3.891	-0.795	0.253	1.050	0.807	1.584
$\text{Al}_2\text{TeSe}_2\text{-m}$	3.854	-0.762	0.522	1.323	1.157	1.714
$\text{Al}_2\text{Se}_2\text{Te-b}$	3.900	-0.779	1.959	2.377	2.728	1.999

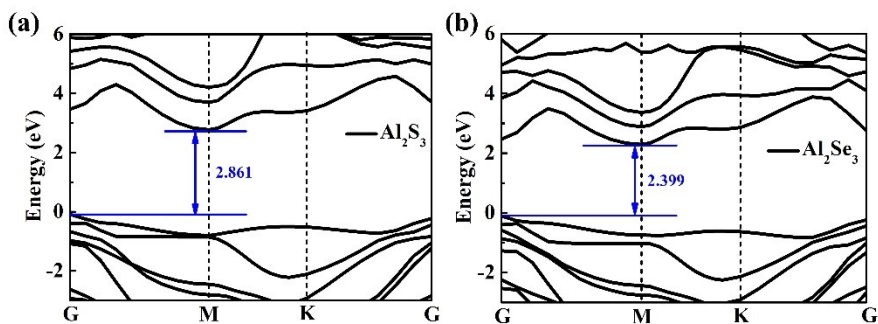


Fig. S2. The electronic band structures of  $\text{Al}_2\text{S}_3$  and  $\text{Al}_2\text{Se}_3$  monolayers at the HSE06 level.

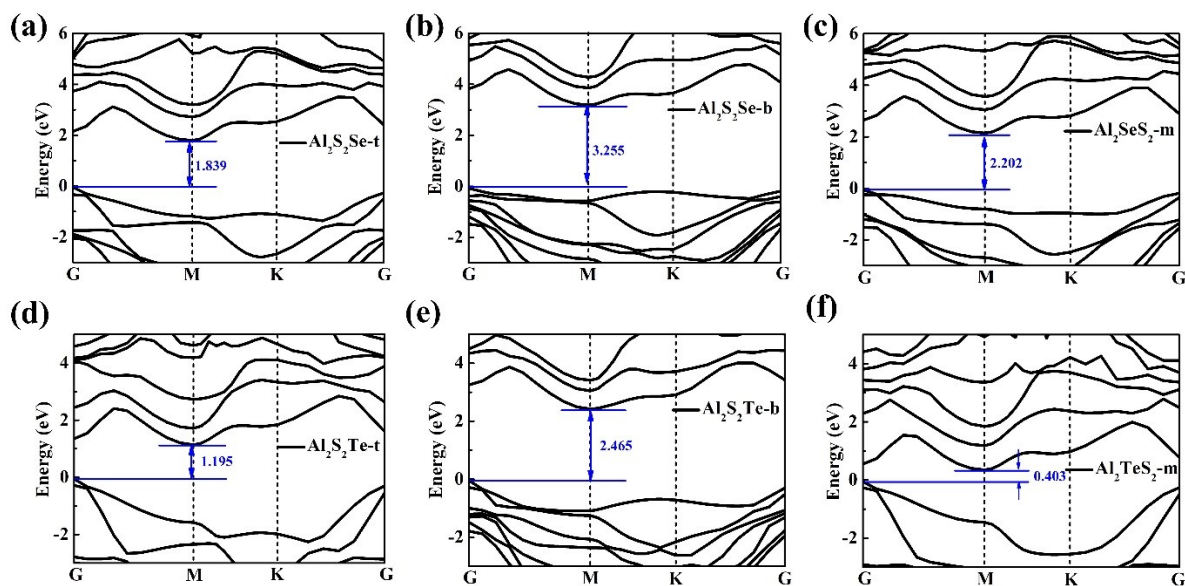


Fig. S3 (a-f). The electronic band structures of  $\text{Al}_2\text{S}_2\text{Se-t}$ ,  $\text{Al}_2\text{S}_2\text{Se-b}$ ,  $\text{Al}_2\text{SeS}_2\text{-m}$ ,  $\text{Al}_2\text{S}_2\text{Te-t}$ ,  $\text{Al}_2\text{S}_2\text{Te-b}$  and  $\text{Al}_2\text{TeS}_2\text{-m}$  monolayers at the HSE06 level.

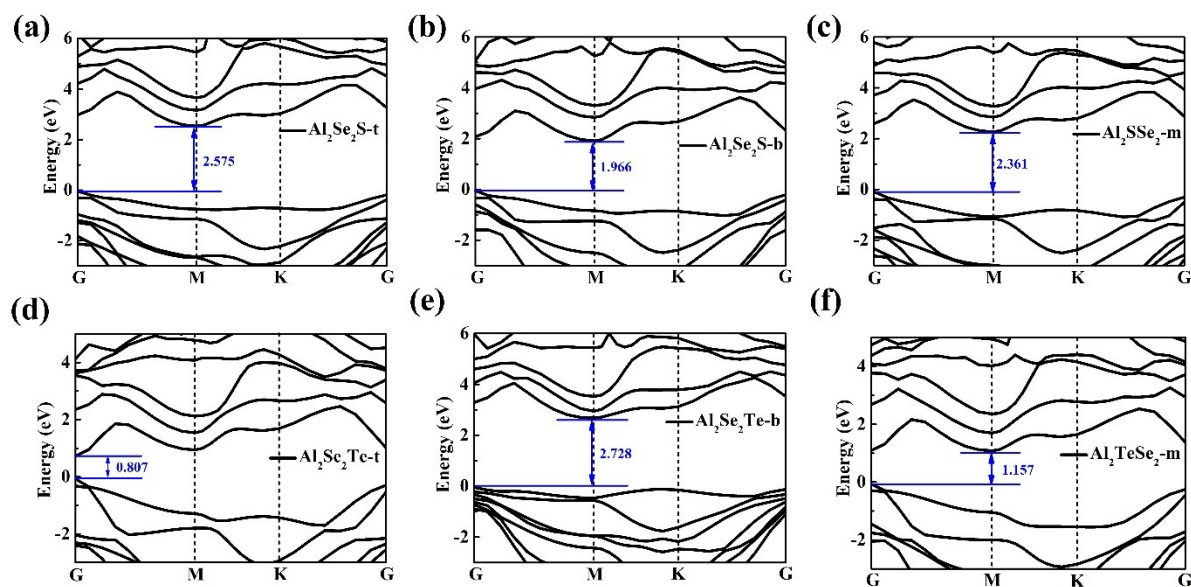


Fig. S4 (a-f). The electronic band structures of  $\text{Al}_2\text{Se}_2\text{S-t}$ ,  $\text{Al}_2\text{Se}_2\text{S-b}$ ,  $\text{Al}_2\text{SSe}_2\text{-m}$ ,  $\text{Al}_2\text{Se}_2\text{Te-t}$ ,  $\text{Al}_2\text{Se}_2\text{Te-b}$  and  $\text{Al}_2\text{TeSe}_2\text{-m}$  monolayers at the HSE06 level.

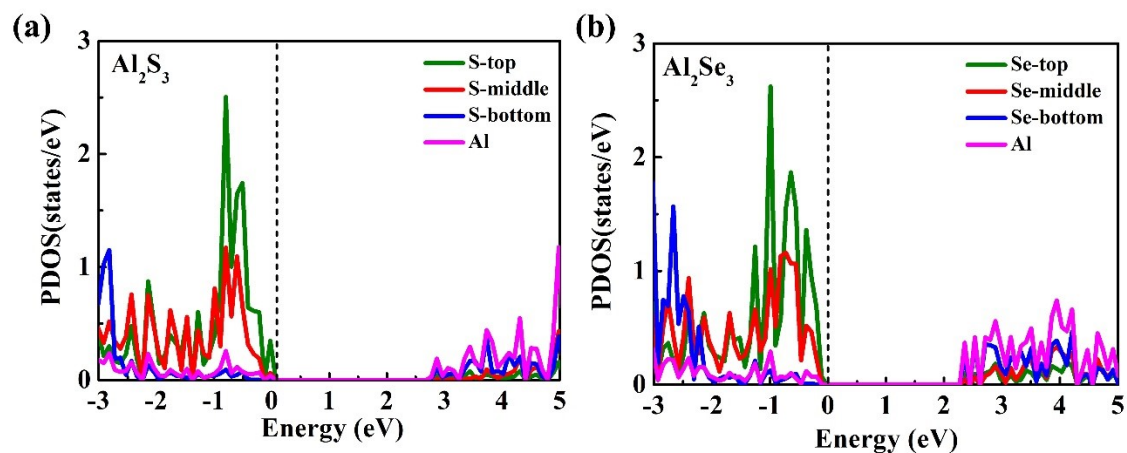


Fig. S5 (a-b) The PDOS of  $\text{Al}_2\text{S}_3$  and  $\text{Al}_2\text{Se}_3$  monolayers at the HSE06 level

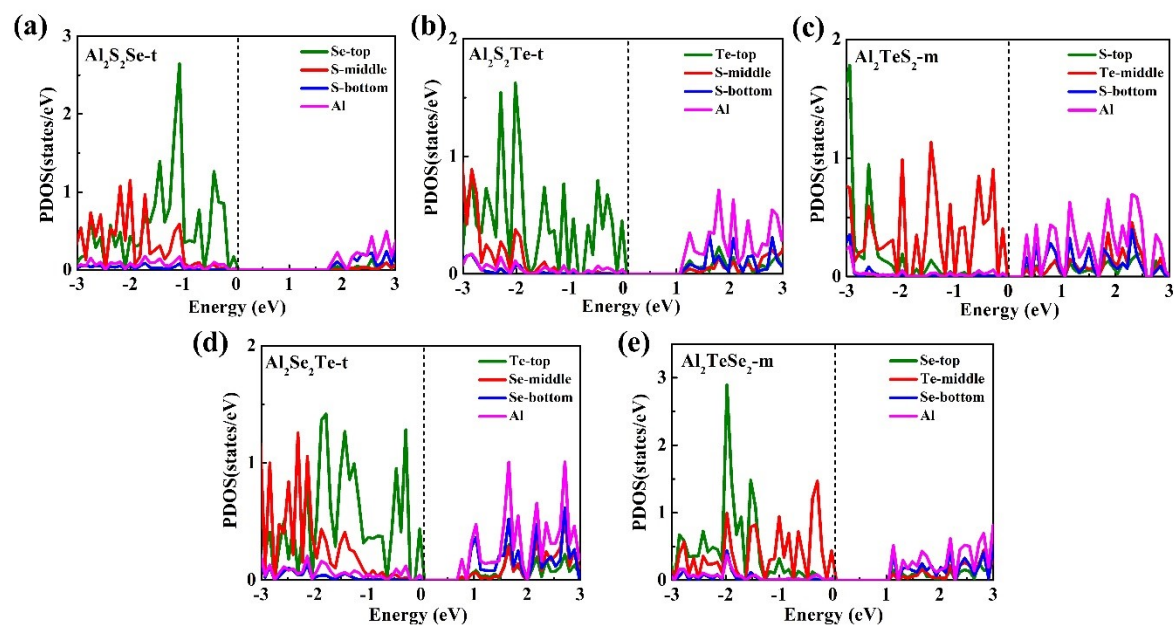


Fig. S6 (a-e) The PDOS of  $\text{Al}_2\text{S}_2\text{Se-t}$ ,  $\text{Al}_2\text{S}_2\text{Te-t}$ ,  $\text{Al}_2\text{TeS}_2\text{-m}$ ,  $\text{Al}_2\text{Se}_2\text{Te-t}$  and  $\text{Al}_2\text{TeSe}_2\text{-m}$  monolayers at the HSE06 level.



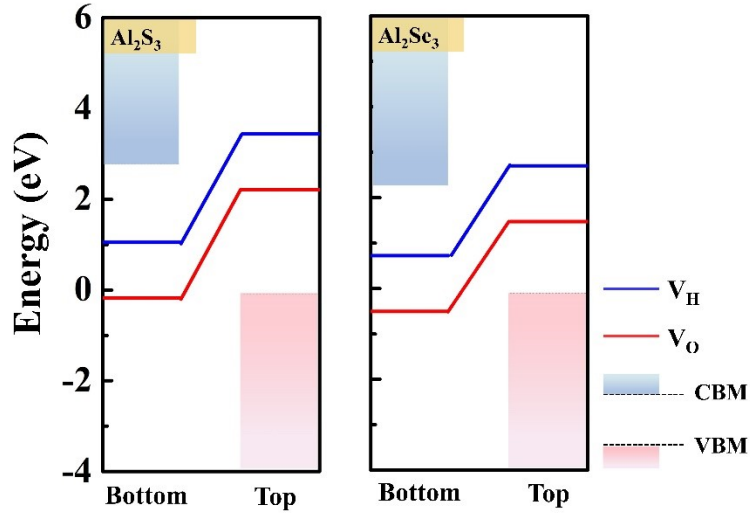


Fig. S7. The band alignment relative to the water redox potentials for  $\text{Al}_2\text{S}_3$  and  $\text{Al}_2\text{Se}_3$  monolayers.

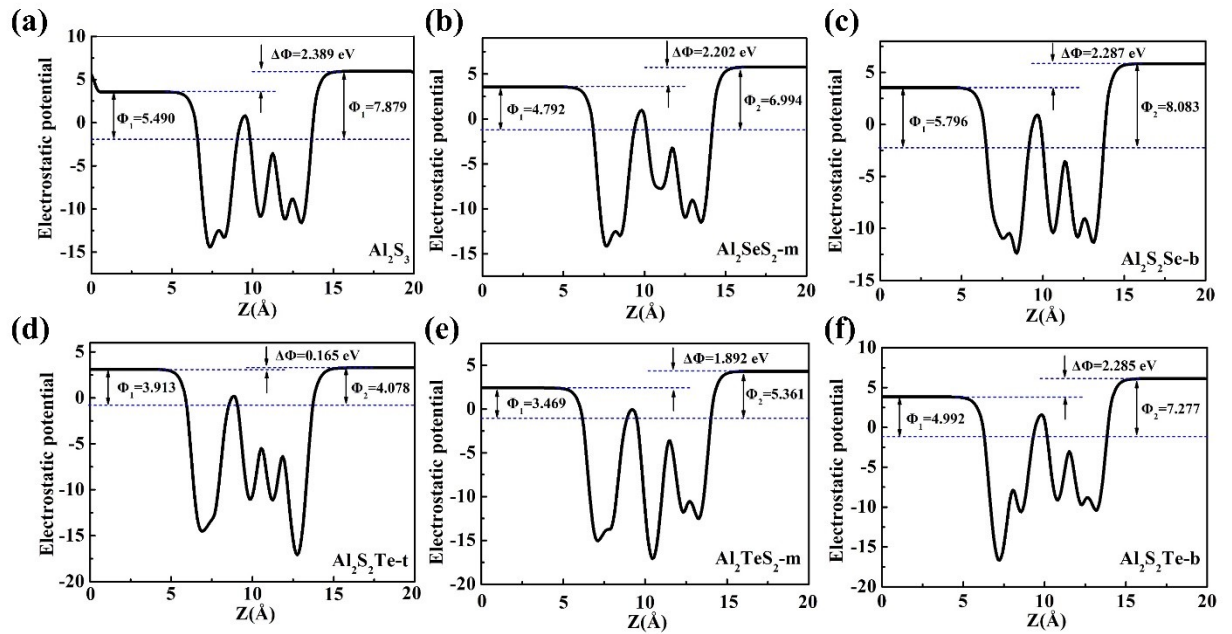


Fig. S8. The electrostatic potentials, Fermi level and work functions of  $\text{Al}_2\text{S}_3$ ,  $\text{Al}_2\text{SeS}_2\text{-m}$ ,  $\text{Al}_2\text{S}_2\text{Se-b}$ ,  $\text{Al}_2\text{S}_2\text{Te-t}$ ,  $\text{Al}_2\text{TeS}_2\text{-m}$  and  $\text{Al}_2\text{S}_2\text{Te-b}$  monolayers along the Z-direction.

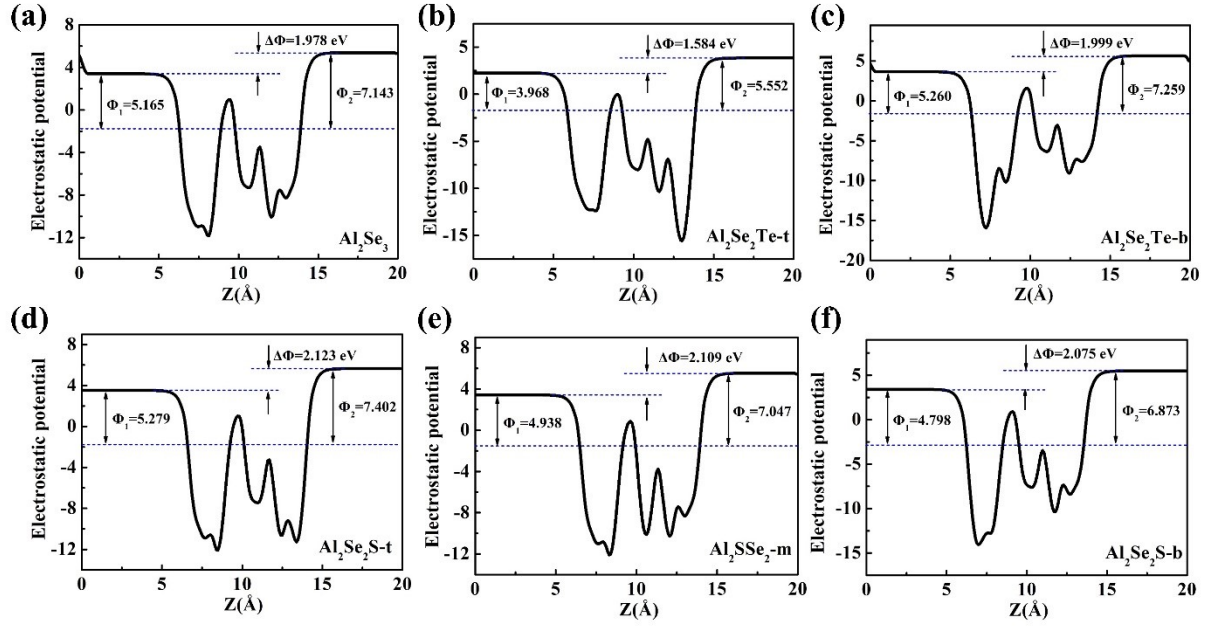


Fig. S9. The electrostatic potentials, Fermi level and work functions of  $\text{Al}_2\text{Se}_3$ ,  $\text{Al}_2\text{Se}_2\text{Te-t}$ ,  $\text{Al}_2\text{Se}_2\text{Te-b}$ ,  $\text{Al}_2\text{Se}_2\text{S-t}$ ,  $\text{Al}_2\text{SSe}_2\text{-m}$  and  $\text{Al}_2\text{Se}_2\text{S-b}$  monolayers along the Z-direction.

Table S2. The overpotentials  $\chi(\text{H}_2)$  and  $\chi(\text{O}_2)$ , energy conversion efficiency of light absorption ( $\eta_{abs}$ ), carrier utilization efficiency ( $\eta_{cu}$ ), STH efficiency ( $\eta_{STH}$ ), and corrected STH ( $\eta'_{STH}$ ).

	$\chi_{\text{H}_2}$	$\chi_{\text{O}_2}$	$\eta_{abs}$	$\eta_{cu}$	$\eta_{STH}$	$\eta'_{STH}$
$\text{Al}_2\text{S}_3$	1.723	2.297	8.5%	38.5%	3.3%	3.1%
$\text{Al}_2\text{Se}_3$	1.570	1.577	21.0%	43.6%	9.2%	8.0%
$\text{Al}_2\text{S}_2\text{Te-t}$	1.668	-1.538	76.6%	1.2%	0.9%	0.9%
$\text{Al}_2\text{TeS}_2\text{-m}$	1.320	-0.255	99.3%	46.0%	45.7%	20.2%
$\text{Al}_2\text{S}_2\text{Te-b}$	1.870	1.650	18.9%	42.9%	8.1%	7.0%
$\text{Al}_2\text{S}_2\text{Se-t}$	1.664	1.176	44.2%	51.6%	22.8%	16.1%
$\text{Al}_2\text{SeS}_2\text{-m}$	1.729	1.382	28.2%	46.2%	13.0%	10.5%
$\text{Al}_2\text{S}_2\text{Se-b}$	1.827	2.484	3.3%	35.0%	1.1%	1.1%
$\text{Al}_2\text{Se}_2\text{S-t}$	1.691	1.777	15.6%	41.6%	6.5%	5.9%
$\text{Al}_2\text{SSe}_2\text{-m}$	1.765	1.474	22.4%	44.1%	9.9%	8.4%
$\text{Al}_2\text{Se}_2\text{S-b}$	1.553	1.254	38.0%	49.5%	18.8%	14.3%
$\text{Al}_2\text{Se}_2\text{Te-t}$	1.199	-0.038	90.7%	42.1%	38.2%	20.6%
$\text{Al}_2\text{TeSe}_2\text{-m}$	1.305	0.336	78.8%	49.9%	39.3%	25.9%
$\text{Al}_2\text{Se}_2\text{Te-b}$	1.865	1.632	11.4%	39.9%	4.5%	4.3%

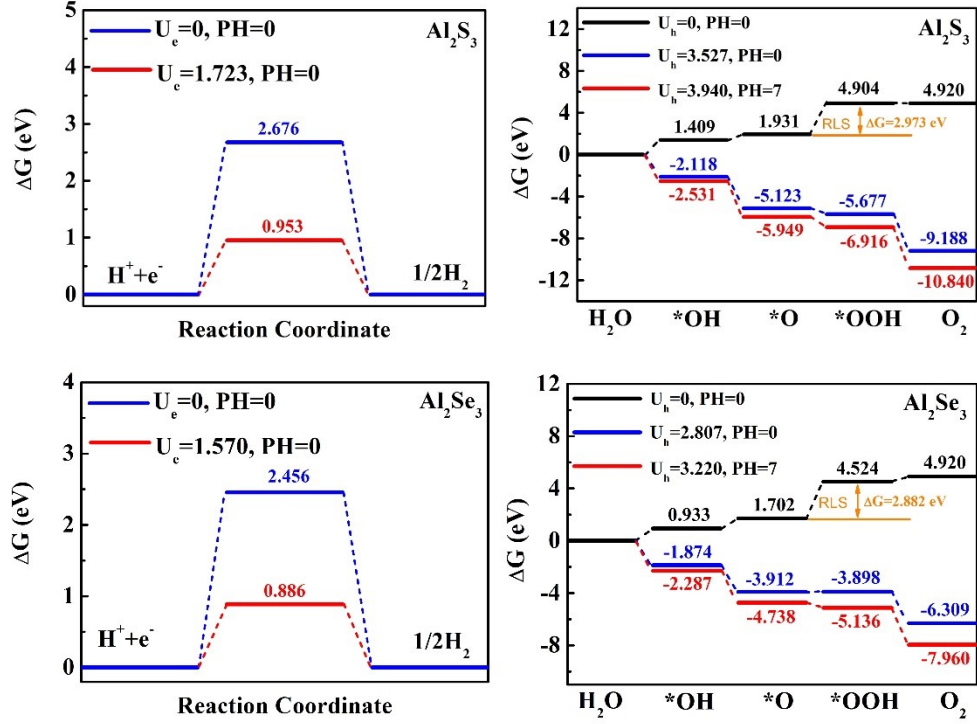


Fig. S10. The Gibbs free energy change diagrams for the OER and HER on the  $\text{Al}_2\text{S}_3$  and  $\text{Al}_2\text{Se}_3$  monolayers.

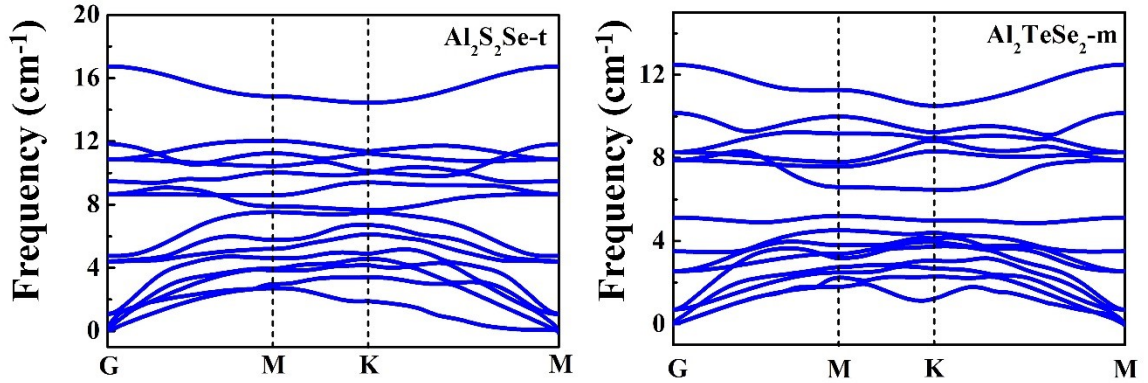


Fig. S11. Phonon band structure of  $\text{Al}_2\text{S}_2\text{Se-t}$  and  $\text{Al}_2\text{TeSe}_2\text{-m}$  monolayers.

Tab. 2. Effective mass  $m_\alpha^*$ , elastic modulus  $C_\alpha$ , deformation potential  $E_\alpha^l$ , and carrier mobility  $\mu_\alpha$  of  $\text{Al}_2\text{S}_2\text{Te-b}$ ,  $\text{Al}_2\text{SeS}_2\text{-m}$ ,  $\text{Al}_2\text{S}_2\text{Se-b}$ ,  $\text{Al}_2\text{Se}_2\text{S-t}$ ,  $\text{Al}_2\text{SSe}_2\text{-m}$ ,  $\text{Al}_2\text{Se}_2\text{S-b}$ ,  $\text{Al}_2\text{Se}_2\text{Te-t}$  and  $\text{Al}_2\text{Se}_2\text{Te-b}$  monolayers.

Material	Direction	$m_\alpha^* (m_e)$		$E_\alpha^l$ (eV)		$C_\alpha$ (J/m <sup>2</sup> )	$\mu_\alpha$ (cm <sup>2</sup> /(s*V))	
		e	h	e	h	h=e	e	h
$\text{Al}_2\text{S}_2\text{Te-b}$	x	0.61	0.73	2.50	11.45	161.43	1415.96	50.90

	y	<b>0.65</b>	<b>0.68</b>	<b>3.09</b>	<b>11.98</b>	<b>176.17</b>	<b>951.44</b>	<b>54.09</b>
<b>Al<sub>2</sub>SeS<sub>2</sub>-m</b>	x	<b>0.62</b>	<b>6.44</b>	<b>3.69</b>	<b>8.99</b>	<b>232.21</b>	<b>1200.39</b>	<b>3.28</b>
	y	<b>0.38</b>	<b>1.31</b>	<b>6.63</b>	<b>9.92</b>	<b>241.97</b>	<b>625.72</b>	<b>13.79</b>
<b>Al<sub>2</sub>S<sub>2</sub>Se-b</b>	x	<b>0.66</b>	<b>2.20</b>	<b>3.21</b>	<b>6.30</b>	<b>210.00</b>	<b>1123.81</b>	<b>18.24</b>
	y	<b>0.51</b>	<b>3.59</b>	<b>5.73</b>	<b>7.45</b>	<b>223.97</b>	<b>493.91</b>	<b>8.53</b>
<b>Al<sub>2</sub>Se<sub>2</sub>S-t</b>	x	<b>0.62</b>	<b>8.49</b>	<b>3.78</b>	<b>8.23</b>	<b>212.34</b>	<b>1054.75</b>	<b>1.57</b>
	y	<b>0.37</b>	<b>2.95</b>	<b>6.14</b>	<b>9.46</b>	<b>222.30</b>	<b>694.38</b>	<b>3.59</b>
<b>Al<sub>2</sub>SSe<sub>2</sub>-m</b>	x	<b>0.63</b>	<b>2.32</b>	<b>3.92</b>	<b>10.12</b>	<b>190.54</b>	<b>849.83</b>	<b>9.98</b>
	y	<b>0.38</b>	<b>1.26</b>	<b>6.76</b>	<b>10.75</b>	<b>206.41</b>	<b>513.74</b>	<b>17.74</b>
<b>Al<sub>2</sub>Se<sub>2</sub>S-b</b>	x	<b>1.05</b>	<b>1.60</b>	<b>3.81</b>	<b>8.38</b>	<b>215.23</b>	<b>327.41</b>	<b>11.98</b>
	y	<b>0.81</b>	<b>7.26</b>	<b>7.01</b>	<b>10.63</b>	<b>225.58</b>	<b>130.85</b>	<b>1.72</b>
<b>Al<sub>2</sub>Se<sub>2</sub>Te-t</b>	x	<b>0.78</b>	<b>1.48</b>	<b>10.53</b>	<b>11.72</b>	<b>179.12</b>	<b>76.88</b>	<b>17.25</b>
	y	<b>0.42</b>	<b>0.80</b>	<b>11.42</b>	<b>12.37</b>	<b>190.85</b>	<b>130.43</b>	<b>30.50</b>
<b>Al<sub>2</sub>Se<sub>2</sub>Te-b</b>	x	<b>1.10</b>	<b>2.64</b>	<b>3.44</b>	<b>5.50</b>	<b>189.11</b>	<b>186.41</b>	<b>19.21</b>
	y	<b>2.52</b>	<b>2.60</b>	<b>5.69</b>	<b>7.82</b>	<b>202.25</b>	<b>31.80</b>	<b>10.33</b>

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