

# **ELECTRONIC SUPPORTING INFORMATION**

## **(ESI)**

### **(TeCl<sub>4</sub>)<sub>4</sub>(TiCl<sub>4</sub>) with Isolated Te<sub>4</sub>Cl<sub>16</sub> and TiCl<sub>4</sub> Molecules and Second-Harmonic-Generation**

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## 1. Analytical Techniques

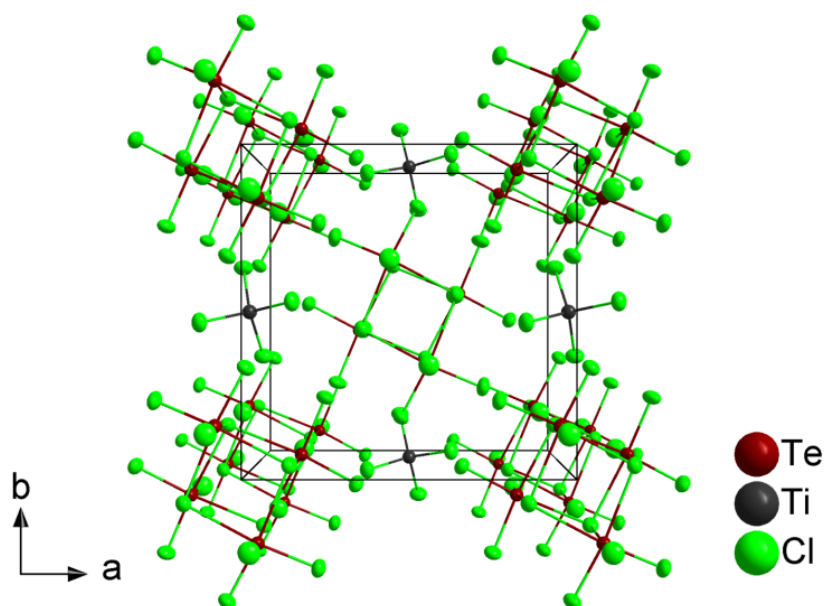
*X-ray powder diffraction (XRD).* X-ray powder diffraction analysis (XRD) of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$  was performed on a Stoe Stadi-P diffractometer (Stoe, Germany) using  $\text{Cu-K}\alpha$  radiation ( $\lambda = 154.06 \text{ pm}$ ) with a Ge-monochromator. Rietveld refinements were performed with the program TOPAS-Academic (Version 5), using the *cif*-data to confirm the phase purity of the title compound. The refinement was carried out by a simple axial model.

*Fourier-transform infrared (FT-IR)* spectra of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$  was recorded on a Bruker Vertex 70 FT-IR spectrometer (Bruker, Germany). The samples were measured in reflection with a Platinum A 225 ATR unit.

*Thermogravimetry (TG)* of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$  was carried out on a Netzsch STA 449 F3 Jupiter device (Netzsch, Germany), using  $\alpha\text{-Al}_2\text{O}_3$  as a crucible material and reference. Buoyancy effects were corrected by baseline subtraction based on a blank measurement. The samples were measured under dried nitrogen up to  $1200 \text{ }^\circ\text{C}$  with a heating rate of  $5 \text{ K/min}$ . The Netzsch software PROTEUS Thermal Analysis (Version 5.2.1) was used for graphical illustration.

## 2. Material Characterization

The crystallographic data and refinement details of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$  are listed in Table S1. The data of the Rietveld refinement are summarized in Table S2. The unit cell of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$  is shown in Figure S1.



**Figure S1.** Unit cell of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$ .

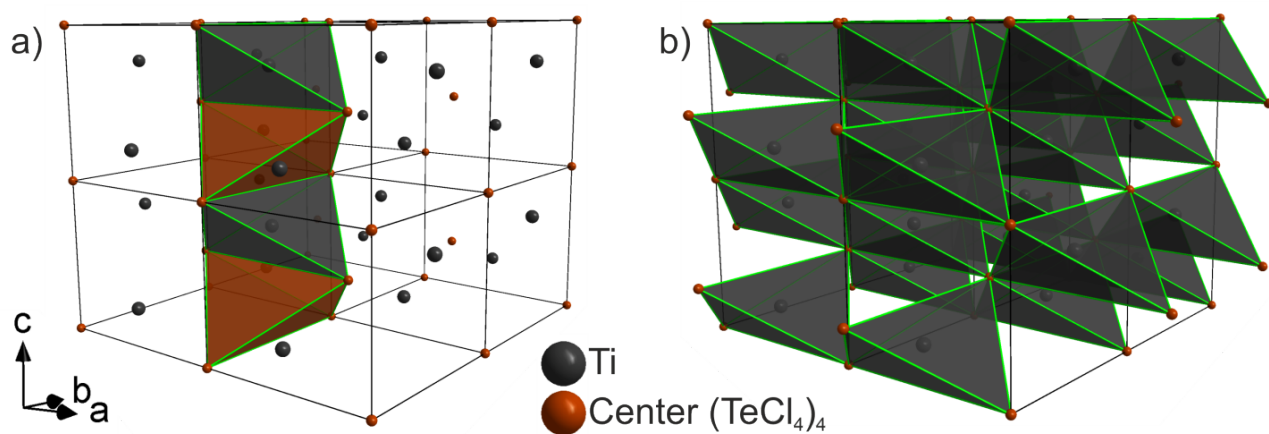
**Table S1.** Crystallographic data and refinement details of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$  (recorded at 210 K).

<b>Compound</b>	<b><math>(\text{TeCl}_4)_4(\text{TiCl}_4)</math></b>
W ( $\text{g}\cdot\text{mol}^{-1}$ )	1267.30
Space group	$\bar{I}4$
Flack parameter	0.50(4)
<i>a</i> (pm)	1210.4(2)
<i>c</i> (pm)	1015.2(2)
V ( $10^6 \text{ pm}^3$ )	1487.2(4)
Z	2
$\rho_{\text{calc.}}$ ( $\text{g}\cdot\text{cm}^{-3}$ )	2.83
$\mu$ ( $\text{mm}^{-1}$ )	5.93
$\lambda(\text{Mo-K}\alpha)$ (pm)	71.073
T (K)	210
Observed reflections	3863
Independent reflections	1960
$R_1$ ( $I \geq 2\sigma_I$ )	0.0191
$R_1$ (all data)	0.0211
$wR_2$ ( $I \geq 2\sigma_I$ )	0.0435
$wR_2$ (all data)	0.0438
Goof	0.997

**Table S2.** Rietveld refinement details of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$  (recorded at 20 °C)

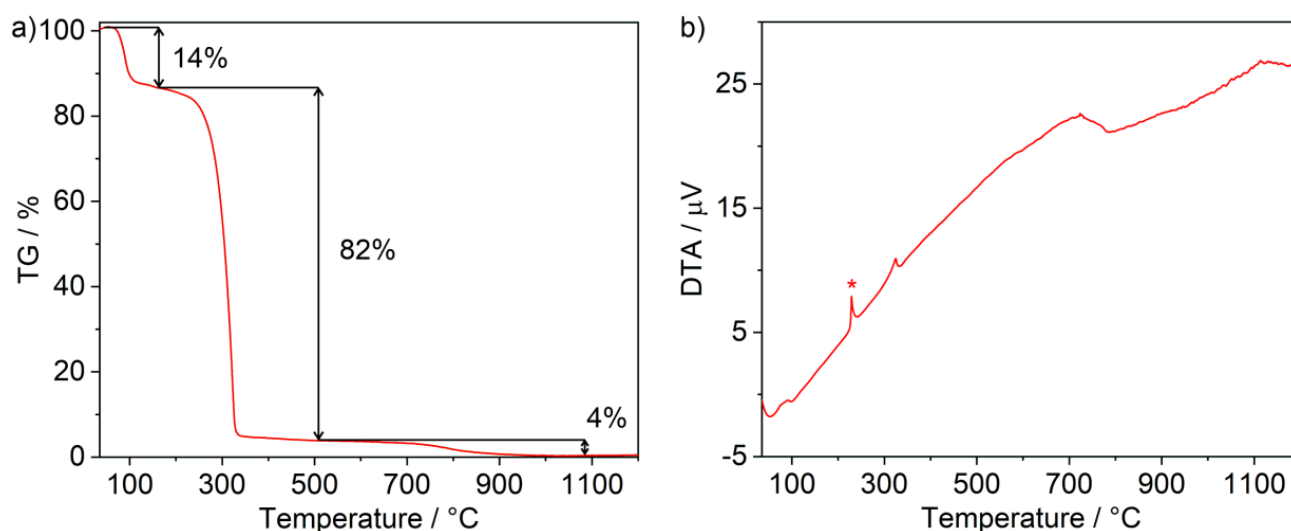
<b>Compound</b>	<b><math>(\text{TeCl}_4)_4(\text{TiCl}_4)</math></b>
<i>a</i> (pm)	1222.2
<i>c</i> (pm)	1013.8
$R_{\text{wp}}$	0.0183

In regard of the body-centred cubic (*bcc*) packing of  $(\text{TeCl}_4)_4$  with  $\text{TiCl}_4$  in half of the tetrahedral voids (Figure S2), it must be noticed that these tetrahedral voids are distorted as not all edges have the same length.<sup>S1</sup> Compressing a single axis to a tetragonal lattice changes the distortion so that all edges of the tetrahedra become identical (995.1(1) pm). The distortion is related to angles of 98.6(1)° being much smaller than the tetrahedral angle (109.5°). The distance between the center of the  $\text{TiCl}_4$  tetrahedra and the tetrahedral corners is 656.2(1) pm.



**Figure S2.** Reduced building units of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$  in a  $(2 \times 2 \times 2)$  supercell showing (a) selected and (b) all occupied tetrahedral voids.

The purity of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$  was confirmed by X-ray powder diffraction with Rietveld refinement (*see main paper: Figure 2*), Fourier-transform infrared (FT-IR) spectroscopy (*see main paper: Figure 4a*) and thermogravimetry (TG, Figure S3a) and differential thermal analysis (DTA, Figure S3b). Thermally, the title compound shows three-step decomposition with the first step at 100-150 °C with 14 % mass loss, the second step at 200-400 °C with a mass loss of 82 %, and the third step beginning at 700 °C with a mass loss of 4 %. These mass losses can be attributed to the sublimation of  $\text{TiCl}_4$  (calculated: 15 %) and the sublimation of  $\text{TeCl}_4$  (calculated: 85 %). The third minor mass loss of 4 % can be ascribed to the sublimation of elemental tellurium as an impurity, which was already indicated by XRD (*see main paper: Figure 2*).



**Figure S3.** Thermal properties of  $(\text{TeCl}_4)_4(\text{TiCl}_4)$ : a) TG and b) DTA (\*melting point of  $\text{TeCl}_4$ ).

### 3. Calculations

Atomistic model calculations were carried out within the framework of DFT<sup>S2</sup> and the pseudopotential method using the CASTEP simulation package.<sup>S3</sup> Norm-conserving pseudopotentials were generated “on the fly” using the parameters provided with the CASTEP distribution. These pseudopotentials have been extensively tested for accuracy and transferability.<sup>S4</sup> The pseudopotentials were employed in conjunction with plane waves up to a kinetic energy cutoff of 630 eV. The calculations were carried out with the PBE exchange-correlation functional.<sup>S5</sup> Monkhorst-Pack grids<sup>S6</sup> were used for Brillouin zone integrations with a distance of  $< 0.025 \text{ \AA}^{-1}$  between grid points. A dispersion correction according to Grimme *et al.* was used.<sup>S7</sup> Convergence criteria included an energy change of  $< 5 \times 10^{-6} \text{ eV/atom}$ , a maximal force of  $< 0.008 \text{ eV/\AA}$ , and a maximal deviation of the stress tensor  $< 0.02 \text{ GPa}$  from the imposed stress tensor. Optical properties were computed as described by Refson *et al.*<sup>S8</sup> Phonon frequencies were obtained from density functional perturbation theory (DFPT) calculations. Raman intensities and NLO properties were computed using DFPT with the ‘2n+1’ theorem approach.<sup>S9</sup> It should be stressed that all calculations were carried out in the athermal limit, i.e., the influence of temperature and zero-point motion were not taken into account.

Band structure calculations resulted in a band gap of 2.9 eV. However, the apparent agreement with the experimentally determined optical band gap of 2.8 eV (*see main paper: Figure 5*) is somewhat fortuitous as DFT-GGA-PBE calculations are known to underestimate the band gap systematically. The top-most valence bands and the lower-most conduction bands are nearly dispersionless.

### References

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