# **Supporting Information**

#### Over 19.2% efficiency of layer-by-layer organic photovoltaics enabled by

## highly crystalline material as energy donor and nucleating agent

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### **Experimental section**

#### **Device Fabrication**

The patterned indium tin oxide (ITO) coated glass substrates (15  $\Omega$  per square) were cleaned via sequential sonication in detergent, de-ionized and ethanol and then blow-dried by high-purity nitrogen. All pre-cleaned ITO substrates were treated by oxygen plasma for 1 minute (min) to improve their work function and clearance. Then, the poly(3,4-ethylenedioxythiophene): poly(styrene sulfonate) (PEDOT:PSS, purchased from H.C. Starck co. Ltd.) solution was spin-coated on ITO substrates at 5000 round per minute (rpm) for 40 s and dried at 150 °C for 15 min in atmospheric air. Then ITO substrates coated with PEDOT:

PSS films were transferred into a high-purity nitrogen-filled glove box. The polymer donor PM1 and small molecule accepter L8-BO were purchased from Solarmer Materials Inc. The PM1:D18A with different mass ratios (1:0, 1:0.15, 1:0.3, 1:0.5, 1:1) were dissolved in chloroform to prepare the blend donor solutions. The acceptor L8-BO was dissolved in chloroform to prepare the acceptor solutions with the concentration of 8 mg/ml, and 10 mg/ml TCB was added as an additive. The donor layer was spin coated from donor solutions on top of the PEDOT: PSS layer, the donor layers were thermal annealed at 80 °C for 8 min. The acceptor layer was spin coated from acceptor solutions at 2700 rpm for 40 s on the top of the donor layer. Then, the whole active layers were thermally annealed at 100 °C for 5 min. PNDIT-F3N was purchased from eflex PV limited company. The PNDIT-F3N was dissolved in methanol with the addition of 0.25 vol% acetic acid to prepare a 0.5 mg/ml solution. The prepared PNDIT-F3N solution was spin-coated onto the active layers at 2000 rpm for 30 s. Finally, 100 nm Ag was deposited by thermal evaporation. The active area is approximately 3.8 mm<sup>2</sup>, defined by the overlapping area of ITO anode and Ag cathode.

#### **Devices Measurement**

The current-voltage (*J-V*) curves of LbL OPVs were measured in a high-purity nitrogenfilled glove box using a Keithley 2400 source meter. AM 1.5G irradiation at 100 mW cm<sup>-2</sup> was provided by an XES-40S2 (SAN-EI Electric Co., Ltd.) solar simulator (AAA grade,  $70 \times 70$ mm<sup>2</sup> photobeam size), which was calibrated by standard monocrystalline silicon reference solar cells. The external quantum efficiency (EQE) spectra of OPVs were measured in air conditions by a Zolix Solar Cell Scan 100. The absorption spectra of films were measured with a Shimadzu UV-3101 PC spectrometer. Photoluminescence (PL) spectra of films were measured by a HORIBA Fluorolog®-3 spectrofluorometer system.

Transient photovoltage (TPV) and transient photocurrent (TPC) were conducted with the Paioscarrier measurement system (FLUXiM AG, Switzerland). A high-power white LED is utilized as light source for TPV and TPC. The integrated power of the LED is 72mW cm<sup>-2</sup>, and the spectrum distribution is mainly in the wavelength range of 440-470 nm and 540-630 nm, and the peak value located at 460 nm and 550 nm. Electrochemical impedance spectroscopy was measured by a ZAHNER CIMPS electrochemical workstation, Germany. Grazing incidence wide angle X-ray scattering (GIWAXS) measurements were accomplished at PLS-

II 9A U-SAXS beamline of the Pohang Accelerator Laboratory in Korea.

The contact angle images were obtained using a surface contact angle tester (Zhongchen, JC2000D1, China). The contact angles of neat PM1, D18A, and L8-BO films and PM1:D18A blend films were measured based on  $H_2O$  and  $CH_2I_2$ . The surface energy of films was calculated according to the contact angles by using Wu's model. According to the surface energy of films, the interfacial energy between various films can be evaluated by the following equation (1):

$$\gamma_{AB} = \gamma_A + \gamma_B - \frac{4\gamma_A^d \gamma_B^d}{\gamma_A^d + \gamma_B^d} - \frac{4\gamma_A^p \gamma_B^p}{\gamma_A^p + \gamma_B^p}$$
(1)

Here,  $\gamma_{AB}$  is the interfacial energy between material A and B;  $\gamma_A$  and  $\gamma_B$  are the surface energy of the pure materials; superscript *d* and *p* represent the dispersion and polar components calculated by using the contact angles.

The energy loss ( $E_{loss}$ ) of typical LbL OPVs were investigated.  $V_{OC}^{rad}$  is the voltage of radiative limit, which can be obtained according to the following equation:

$$V_{OC}^{rad} = \frac{kT}{q} ln^{[ro]} (\frac{J_{SC}}{J_{0}^{rad}} + 1) \cong \frac{kT}{q} ln^{[ro]} (\frac{q \cdot \int_{0}^{+\infty} EQE_{PV}(E) \cdot \phi_{AM \ 1.5G}(E) \cdot dE}{q \cdot \int_{0}^{+\infty} EQE_{PV}(E) \cdot \phi_{BB}(E) \cdot dE}$$
(2)

Here, *k*, *T*, and *q* represent the Boltzmann constant, temperature of samples, and elementary charge, separately.  $\int_{0}^{rad}$  is the saturation current density calculated by considering only the blackbody radiation of the real absorption profile.

The photogenerated current density  $(J_{ph})$  versus effective voltage  $(V_{eff})$  characteristic was were measured in a high-purity nitrogen-filled glove box. The  $J_{ph}$  is represented as  $J_L$ - $J_D$ , where  $J_L$  and  $J_D$  represent the current densities under one-sun illuminated and dark conditions, separately. The  $V_{eff}$  is represented as  $V_0$ - $V_a$ , in which  $V_0$  and  $V_a$  represent the voltage at  $J_{ph} = 0$ mA cm<sup>-2</sup> and applied voltage, separately. The  $J_{ph}$  can be defined as saturated photocurrent density  $(J_{sat})$  at the relatively large  $V_{eff}$ , in which situation all the photogenerated excitons are assumed to be dissociated into free carriers and then be collected by the electrodes.

The structure of sole-electron devices is ITO/ZnO/active layer/PNDIT-F3N/Al and the

structure of sole-hole devices is ITO/PEDOT: PSS/active layer/MoO<sub>3</sub>/Ag. The fabrication conditions of the active layer films are same with those for the OPVs. The charge mobilities are generally described by the Mott-Gurney equation (2):

$$J = \frac{9}{8} \varepsilon_r \varepsilon_0 \mu \frac{V^2}{L^3} \tag{3}$$

where *J* is the current density,  $\varepsilon_0$  is the permittivity of free space (8.85×10<sup>-14</sup> F/cm),  $\varepsilon_r$  is the dielectric constant of used materials,  $\mu$  is the charge mobility, *V* is the applied voltage and L is the active layer thickness. The  $\varepsilon_r$  parameter is assumed to be 3 as a typical value for organic materials. In organic materials, charge mobility is usually field dependent and can be described by the disorder formalism, typically varying with electric field, E=V/L, according to the equation (3):

$$\mu = \mu_0 exp[0.89\gamma \sqrt{\frac{V}{L}}]$$
(4)

where  $\mu_0$  is the charge mobility at zero electric field and  $\gamma$  is a constant. Then, the Mott-Gurney equation can be described by (4):

$$J = \frac{9}{8} \varepsilon_r \varepsilon_0 \mu_0 \frac{V^2}{L^3} exp[0.89\gamma \sqrt{\frac{V}{L}}]$$
<sup>(5)</sup>

In this case, the charge mobilities were estimated using the following equation (5):

$$\ln\left(\frac{JL^3}{V^2}\right) = 0.89\gamma \sqrt{\frac{V}{L}} + \ln\left(\frac{9}{8}\varepsilon_r\varepsilon_0\mu_0\right)$$
(6)

#### Additional experimental results

**Table S1.** Contact angles, surface energy  $(\gamma)$ , dispersion  $(\gamma^d)$  and polar  $(\gamma^p)$  components of individual films, and interfacial energy  $(\gamma_{X/Y})$  between two films.

Film (X)	Contact angle H <sub>2</sub> O (deg)	Contact angle $CH_2I_2$ (deg)	γ (mN m <sup>-1</sup> )	$\gamma^d$ (mN m <sup>-1</sup> )	$\gamma^p$ (mN m <sup>-1</sup> )	Film (Y)	γx/y (mN m <sup>-1</sup> )
PM1	95.51	43.62	31.89	30.85	1.03	L8-BO	1.92
D18A	79.74	44.09	33.47	26.51	6.96	L8-BO	8.89
PM1:D18A (1:0.3)	94.40	48.14	29.75	28.22	1.51	L8-BO	3.21
L8-BO	90.27	33.39	43.10	42.44	0.66		



Figure S1. PL and absorption spectra for L8-BO film.



**Figure S2.** TA 2D plots of (a) neat PM1 and (b) D18A films pumped at 400 nm. TA 2D plots of (c) neat L8-BO films pumped at 800 nm, and (d) D18A/L8-BO LbL films pumped at 400 nm. TA 2D plots of (e) D18A/L8-BO, (f) PM1/L8-BO, (g) PM1:D18A/L8-BO LbL films pumped at 800 nm.



**Figure S3.** The TA spectra at different time delay of (a) PM1 films, and (b) D18A films pumped at 400 nm, (c) L8-BO films pumped at 800 nm.



**Figure S4.** The TA spectra at different time delay of (a) D18A/L8-BO films pumped at 400 nm. (b) D18A/L8-BO films, (c) PM1/L8-BO, and (d) PM1:D18A/L8-BO films pumped at 800 nm.

PM1:D18A	${J_{ph}}^{*}$	$J_{ph}{}^{\#}$	$J_{sat}$	$P_{diss}$	$P_{coll}$
	$(mA cm^{-2})$	$(mA cm^{-2})$	$(mA cm^{-2})$	(%)	(%)
1:0	26.28	24.14	28.10	93.5	85.9
1:0.3	27.02	25.02	28.11	96.1	89.0
1:1	26.42	23.89	28.48	92.8	83.9

**Table S2.** Device parameters obtained from  $J_{ph}$ - $V_{eff}$  curves of LbL OPVs.

Table S3. The fitted parameters of LbL OPVs.

PM1:D18A	$R_{OS}(\Omega)$	$R_{CT}\left(\Omega ight)$	CPE-T (nF)	CEP-P
1:0	34.6	34.6	26.4	0.956
1:0.3	31.6	27.1	28.9	0.975
1:1	36.7	47.9	16.6	0.949



**Figure S5.** The  $\ln(Jd^3/V^2)$  versus  $(V/d)^{0.5}$  curves of (a) hole-only devices and (b) electron-only devices.

Table S4. The  $\mu_h$ ,  $\mu_e$  and  $\mu_h/\mu_e$  values of layered films.

PM1:D18A	$\mu_h (cm^2 V^{-1} s^{-1})$	$\mu_{e} (cm^{2}V^{-1}s^{-1})$	$\mu_h/\mu_e$
1:0	5.8×10 <sup>-4</sup>	4.9×10 <sup>-4</sup>	1.18
1:0.3	7.1×10 <sup>-4</sup>	6.5×10-4	1.09
1:1	8.9×10 <sup>-4</sup>	7.7×10 <sup>-4</sup>	1.16

**Table S5.** The different vector (q) values of diffraction peaks and crystal coherence length

 (CCL) of layered films.

	Diffraction vector (Å <sup>-1</sup> )		Crystal correlation length (Å)		
TWILDIOA	IP (100)	OOP (010)	IP (CCL <sub>100</sub> )	OOP (CCL <sub>010</sub> )	
1:0	0.30	1.70	64.45	20.00	
1:0.3	0.31	1.71	65.34	21.38	
1:1	0.32	1.72	71.22	24.43	



Figure S6. (a) PCE decay under 65 °C conditions, (b) PCE decay under AM 1.5G illumination of 100 mW cm<sup>-2</sup> condition.