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Supporting Information

Influence of redox engineering on the trade-off relationship between thermopower and electrical conductivity in lanthanum titanium based transition metal oxides

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Weighted carrier properties calculation

The weighted mobility (μ_w) is a measure of the electron mobility weighted by the density of electronic states.¹ The weighted mobility, like the hall mobility, can be defined as a simple function of two measured properties, *S* and σ .¹ The following equation is a simple analytic form for μ_w :¹

$$\mu_{w} = \frac{3h^{3}\sigma}{8\pi e(2m_{e}k_{B}T)^{3/2}} \left[\frac{exp\left(\frac{|S|}{k_{B}/e} - 2\right)}{1 + exp\left(-5\left(\frac{|S|}{k_{B}/e} - 1\right)\right)} + \frac{\frac{3}{\pi^{2}}\frac{|S|}{k_{B}/e}}{1 + exp\left(5\left(\frac{|S|}{k_{B}/e} - 1\right)\right)} \right]$$

Where *h* is Planck's constant (eV · s), σ is the electrical conductivity (S/cm), *e* is the electron charge (eV), k_B the Boltzmann constant (eV/K), *T* is the operating temperature, and *S* is the thermopower (μ V/K).

The weighted carrier properties can be obtained using the calculated μ_w . The weighted carrier concentration (n_w) was calculated using the formula below:²

$$n_w = \frac{\sigma}{\mu_w e}$$

The relaxation time (τ_w) was obtained by using the following equation:³

$$au_w = rac{m_w^* \mu_w}{e}$$

Where m_w^* is the weighted effective mass which was calculated by using the following equation: ⁴

$$\frac{m_w^*}{m_e} = \frac{3eh^2 \times S}{8\pi^2 k_B^2 \times T \times (\pi/3n_w)^{2/3}}$$

Where m_e is the mass of an electron.

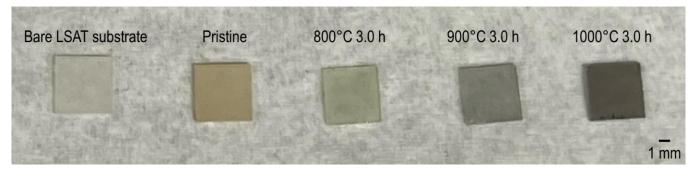


Fig. S1: Picture of the bare LSAT substrate and LCNTO thin films grown on LSAT substrates exposed to different reduction conditions.

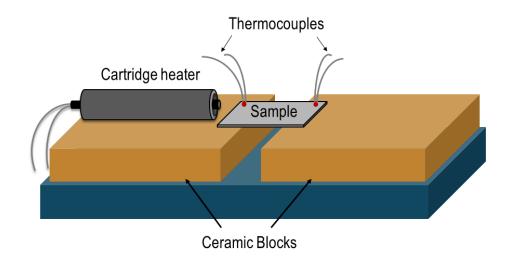


Fig. S2: Schematic of the measurement setup that was used to measure *S*.

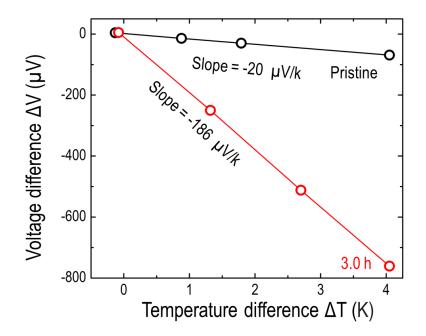


Fig. S3: Voltage difference (ΔV) as a function of Temperature difference (ΔT) measured at room temperature with the pristine and reduced LCNTO films at 900°C for 3.0 h. *S* was extracted from the different slopes ($\Delta V/\Delta T$) of both samples.

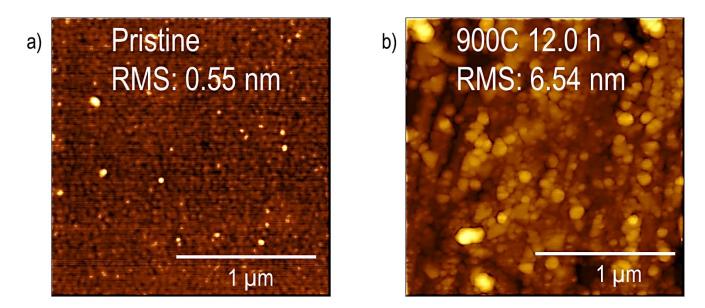


Fig. S4: AFM images of the (a) pristine LCNTO film and (b) LCNTO film reduced at 900°C for 12.0 h.

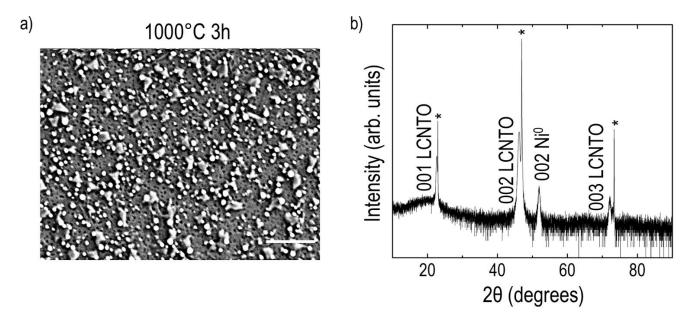


Fig. S5: (a) SEM images of the surface of LCNTO thin films reduced under 1000 °C for 3.0 h (scale bar: 1 μ m). (b) XRD θ -2 θ patterns of the reduced LCNTO thin films at 1000°C for 3.0 h (Substrate peaks are indicated with *).

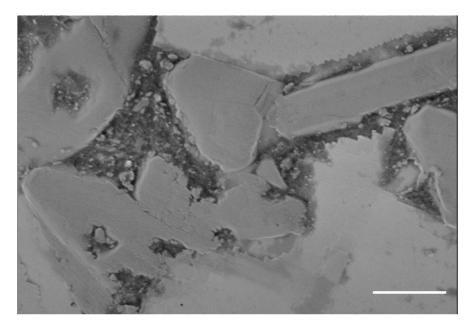


Fig. S6: SEM image of the surface of LCNTO thin films post-annealed in air under 1200 °C for 12.0 h (scale bar: $1 \mu m$).

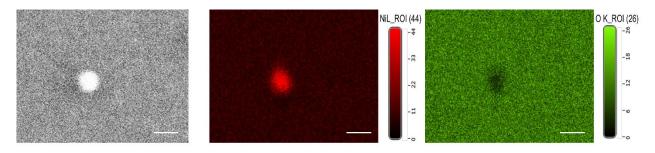


Fig. S7. Top view SEM image of LCNTO film reduced at 900°C for 3.0 h and its corresponding Nickel and Oxygen EDX map (scale bar: 100 nm).

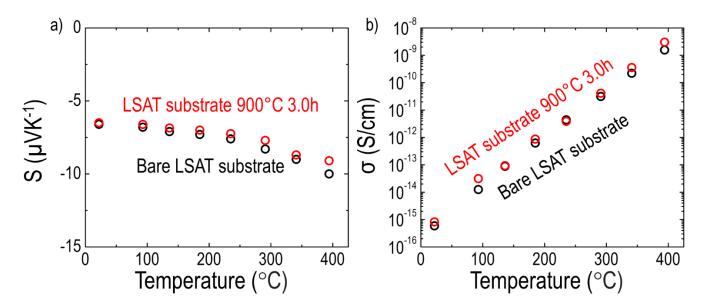


Fig. S8: Temperature dependencies of (a) *S* and (b) σ of the pristine LSAT substrate (black) and reduced LSAT substrate at 900°C for 3.0 h (red) measured from 20 to 400°C.

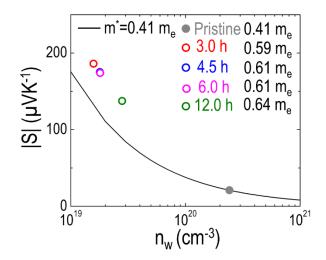


Fig. S9. Pisarenko relation (variation of |S| with n_w) for the LCNTO thin films before and after reduction at 900 °C with varying reduction time at room temperature.

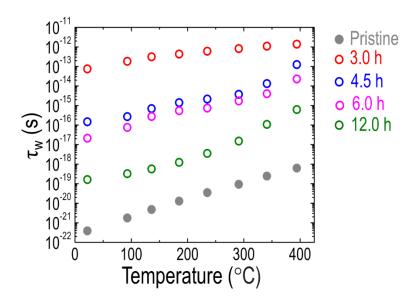


Fig. S10: Temperature dependencies of τ_w of the pristine and reduced LCNTO thin films at 900 °C with varying reduction time.

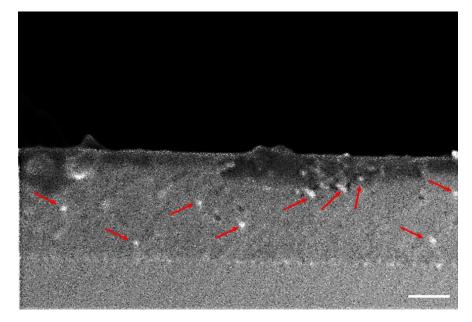


Fig. S11. Cross-sectional SEM image of LCNTO film reduced at 900°C for 12.0 h. The red arrows indicate Ni particles (scale bar: 100 nm).

References

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