Supporting Information

Outstanding High-Temperatures Capacitive Performance in All-Organic Dielectrics Enabled by Synergistic Optimization of Molecular Trap and Aggregation Structure

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Fig. S1 Chemical structures of the BPADA, MPD and HAT-CN.



Fig. S2 Chemical process diagram for preparation of HAT-CN/PEI all-organic composite films.



Fig. S3 Experimental process diagram for preparation of HAT-CN/PEI all-organic composite films.



Fig. S4 The band structure of PEI and HAT-CN were simulated by DFT calculation. PEI is located on the left side, while HAT-CN is located on the right side.



Fig. S5 The FTIR spectra of HAT-CN and the composite films loaded with different content of HAT-CN.



2 Theta (°) Fig. S6 The XRD patterns of HAT-CN and the composite films loaded with different content of HAT-CN.



In(q) Fig. S7 The SAXS curves of the composite films loaded with different content of HAT-CN.



Fig. S8 Simulated snapshots of a) PEI and b,c,d) HAT-CN/PEI composites in molecular dynamics simulation.



Fig. S9 The DSC curves of the composite films loaded with different content of HAT-CN.



Fig. S10 The load-displacement curves of the composite films loaded with different content of HAT-CN.



Fig. S11 The UV-vis absorption spectra of PEI and 0.3 wt% HAT-CN/PEI dissolved in NMP, which are used to determine the difference in solubility of them.



O h1 h12 h24 h7 dFig. S12 The solubility of the films in NMP at different time which studied by UV-visspectra. The ordinate of the bar chart is taken from the highest value of the peak of UV-visabsorption spectra of the samples.



Fig. S13 The contact angles of PEI and 0.3 wt% HAT-CN/PEI determined by water solvent.



Fig. S14 Polarized FTIR spectra of the composite films loaded with different content of HAT-CN.



Fig. S15 UV-vis absorption spectra of all the composite films loaded with different content of HAT-CN.



Fig. S16 The $(\alpha hv)^2 - hv$ plots of the films, which were converted from the UV-vis absorption spectra by Tauc plot, in which α , h, and v are the absorption coefficient, Planck constant, and light frequency, respectively.



Fig. S17 a) Photos of 0.3 wt% HAT-CN/PEI film with dimensions of 15 cm \times 15 cm, showing its flexibility. Cross-sectional SEM images of b) PEI, c) 0.1 wt% HAT-CN/PEI, d) 0.3 wt% HAT-CN/PEI, e) 0.5 wt% HAT-CN/PEI, and f) 0.8 wt% HAT-CN/PEI. The polymer films whose thickness ranging from 9 to 10 μ m



Fig. S18 The TGA curves of HAT-CN. It is shown that HAT-CN can maintain excellent thermal stability till 400 °C, so as to ensure that no reaction occurs during imidization.



Fig. S19 a) Frequency dependence of ε_r and *D* of all samples at room temperatures. b) Temperature dependence of ε_r and *D* of all samples at 1 kHz. c) Frequency dependence of ε_r and *D* of 0.3 wt% HAT-CN/PEI composite film at different temperatures.



Fig. S20 The Weibull statistic of E_b of the samples with increasing electric field at 100 °C.



Fig. S21 Leakage current density of the samples with increasing electric field at a) 100°C, c) 150°C and d) 200°C. b) The hopping distance of the samples at 100 °C. e) Leakage current density of Pure PEI and 0.3 wt% HAT-CN/PEI under different test times at 200°C



0.5 wt% HAT-CN/PEI, e) 0.8 wt% HAT-CN/PEI and f) all the samples at 100 °C and 550 kV mm⁻¹.



Fig. S23 The *D-E* loops of a) PEI, b) 0.1 wt% HAT-CN/PEI, c) 0.3 wt% HAT-CN/PEI, d) 0.5 wt% HAT-CN/PEI, e) 0.8 wt% HAT-CN/PEI and f) all the samples at 150 °C and 550 kV mm⁻¹.



0.5 wt% HAT-CN/PEI, e) 0.8 wt% HAT-CN/PEI and f) all the samples at 200 °C and 450 kV mm⁻¹.



Fig. S25 $U_{\rm e}$ and η of the films at 100 °C.



Fig. S26 Time dependence of U_e under different electric fields of a) BOPP and b) 0.3 wt% HAT-CN/PEI.



Electric Field (kV/mm) Fig. S27 The *D-E* loops of the composite film (Original, After 3 weeks and 15 months) at 200 °C & 450 kV mm⁻¹.