## Vertical heterostructure of graphite-MoS<sub>2</sub> for gas sensing

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Supplementary Information



**Figure S1: Preparation of the hybrid structure.** Schematic view illustrating the different stages of formation of vertically aligned  $MoS_2$  over nano-thick graphite surface. Initially, there is a polymer-free transfer of graphite film (NGF) over the silica substrate, followed by Mo sputtering and rapid sulfurization. (b) The optical image contrast of the  $MoS_2$  sheet over Silica substrate and graphite are marked by arrows for distinction. (c) SEM micrography of  $MoS_2$  granular sheet over silica. (d) SEM micrograph of  $MoS_2$  granular sheet over wrinkled graphite layers.



**Figure S2: Cross-section view of the hierarchy of the hetero-hybrid structure.** Left-hand side is SEM image of hierarchical architecture as (top) MoS<sub>2</sub>/NGF/SiO<sub>2</sub>/Si substrate. In the series of images from left to right-hand side, there is the elemental distribution in the Z-direction revealing the hierarchy of Si, O, C, Mo, S, respectively. It is worth noting that the elemental mapping of carbon is not from the NGF but a deposited polymeric layer during Energy-

dispersive X-ray (EDX) analysis. The NGF layers sandwiched between  $SiO_2$  and  $MoS_2$  is too thin to resolve in the present setup.



**Figure S3:** (a) AFM topography of NGF surface shows top-view landscale of graphene wrinkles, grain boundary (depression region) and basal plane. (b) High-resolution side view TEM image of NGF and MoS<sub>2</sub> shows horizontal stacking of graphite sheets and vertically aligned MoS<sub>2</sub>.



Figure S4: Appearance of the hybrid heterostructure. SEM micrographs of  $MoS_2$  granular flakes distribution over NGF at different locations (a) basal plane, (b) structural defect and (c) step-edge. The density of  $MoS_2$  varies with the size of Mo NPs, the size of 10 nm Mo film is relatively less populated than that of 20 nm Mo sulfurized film.



Figure S5: Schematic correlation plot of MoS<sub>2</sub> deconvoluting strain and doping

The Raman peak shifts  $\omega$  of  $E_{2g}^1$  and  $A_{1g}$  from the strain-free and doping-free positions can be related to the mechanical strain  $\varepsilon$  and the electrical doping *n* by a linear transformation, *i.e.*,

$$\begin{pmatrix} \omega_{E^{1}2g} \\ \omega_{A_{1g}} \end{pmatrix} = \begin{pmatrix} -2\gamma_{E^{1}2g} & k_{E^{1}2g} \\ -2\gamma_{A_{1g}} & k_{A_{1g}} \end{pmatrix} \begin{pmatrix} \varepsilon \\ n \end{pmatrix}$$
where  $\gamma_{E^{1}2g} = 0.86$  and  $\gamma_{A_{1g}} = 0.15$  are the Grüneisen parameters, while  $k_{E^{1}2g} = -0.33 \times 10^{-13} \, cm^{-1}$ 
and  $k_{A_{1g}} = -2.22 \times 10^{-13} \, cm^{-1}$  are the doping shift rates. With this linear transformation, the strain
and doping axes, as well as iso-strain and iso-doping contours can be constructed. The strain and
doping can then be evaluated by projecting the peak positions onto the strain and doping axes.



Figure S6: Surface potential map of  $MoS_2$  and vertical heterostructure of  $MoS_2$  over graphite. (a) Surface potential map of an  $MoS_2$  over silica substrate. The surface potential reveals the contrast in contact potential difference (CPD, mV/V), indicating the distribution in carrier concentration. The brightest region represents the p-doped region (silica substrate), and the dark region represents the electron-rich region. Here, the edges of  $MoS_2$  reveal the higher electron concentration, indicating a drop in the CPD values confirmed by the average line profile. (b) The surface potential map of  $MoS_2$  over graphite flakes shows higher electron concentration in  $MoS_2$ , indicated through dark colour in the CPD map and a decrease in the surface potential profile. (c) Exposure of the  $MoS_2$ -graphite heterostructure to nitronium ions shows p-doping of  $MoS_2$ , as indicated by the increase in the CPD values. (d) Exposure of the  $MoS_2$ -graphite heterostructure to ammonium ions shows n-type doping in the  $OS_2$ , as there is the drop in the CPD values.



Figure S7: Density functional theory simulations. (a, b) Top views of the graphene and  $MoS_2$  supercells. Optimized structures of the stacking configurations (c) AB, (d) AB1, and (e) AB2 of the S-terminated  $MoS_2$  on graphene. Optimized structures of the stacking configurations (f) AB, (g) AB1 and (h) AB2 of the Mo-terminated  $MoS_2$  on graphene.



**Figure S8: Charge transfer.** Side views of the charge redistributions between the graphene/MoS<sub>2</sub> hybrid structure with Mo-termination (towards the analyte molecules) and (c)  $H_2O$ , (d)  $NH_3$ , and (e)  $NO_2$ . Green and orange isosurfaces (isovalue: 0.001 electrons/Å<sup>3</sup>) represent charge depletion and accumulation, respectively. The gray, yellow, purple, white, red, and blue spheres represent the C, S, Mo, H, O, and N atoms, respectively.

**Table S1:** Adsorption energies  $(E_a)$  of the analyte molecules and charge transfer  $(\Delta q)$  between the graphene/MoS<sub>2</sub> hybrid structure and analyte molecules. Positive (negative)  $\Delta q$  represents charge transfer from the graphene/MoS<sub>2</sub> hybrid structure to the analyte molecules (from the analyte molecules to the graphene/MoS<sub>2</sub> hybrid structure).

	H <sub>2</sub> O		NH <sub>3</sub>		NO <sub>2</sub>	
	Ea	$\Delta q$	$E_{\mathrm{a}}$	$\Delta q$	$E_{\mathrm{a}}$	$\Delta q$
S-	-0.09	0.01	-0.11	-0.01	-0.13	0.21
termination						
Mo-	-1.20	0.10	-1.55	-0.05	-4.42	1.02
termination						



Figure S9: Recovery tests when detecting NO<sub>2</sub>, left side (room temperature) and right (150 °C)



**Figure S10**: Increment in the response from the hybrid sensor during NH<sub>3</sub> exposure for different concentrations at elevated temperatures.



Figure S11: Morphology of exposed  $MoS_2$ . (a) Topography of  $MoS_2$  treated with (a) aqueous solution of NO<sub>2</sub> and (b) NH<sub>3</sub> as NH<sub>4</sub>OH. The topography was taken after drying of the solution and accumulation of salts were monitored. The ammonium salts are distributed uniformly over the basal plane of  $MoS_2$ , while salts of the nitro group mostly appear at the edge of  $MoS_2$ .

Nanomaterial	T° (C)	Carrier gas	Sensitivity coefficient (response/ppm)	RH study	Flow rate (sccm)	Reference
MoS <sub>2</sub> /Graphite	150	Air	1.67	Yes	100	Current work
UFew-Layer Graphene (FLG)	100	Air	0.83	Yes	100	Deokar et al. <sup>1</sup>
MoS <sub>2</sub> /VACNT	RT/100 <sup>a</sup>	Air	0.64	Yes	100	Deokar et al. <sup>2</sup>
MoS <sub>2</sub> /Graphene	RT (UV)	Air	0.08	No	NA	Kumar et al. <sup>3</sup>
MoS <sub>2</sub> /rGO	60	Air	0.02	Yes	500	Zhou et al. <sup>4</sup>
MoS <sub>2</sub> - rGO/CNT	RT (UV)	Air	0.017	No	NA	Ghasemi <sup>5</sup>
MoS <sub>2</sub> /rGO	RT	Nitrogen	0.004	No	1000	Mukherjee et al. <sup>6</sup>

Table S2. Performance comparison in the detection of NO<sub>2</sub> with previously reported sensors.

MoS <sub>2</sub> /rGO	50	Air	0.6	No	175	Kumar et al. <sup>7</sup>
MoS <sub>2</sub> /GA	200	Air	3.4	No	300	Long et al. <sup>8</sup>
Pure MoS <sub>2</sub>	100	Air	1.4	Yes	100	Annanouch et
						al. <sup>9</sup>

## **Reference:**

- 1. G. Deokar, J. Casanova-Cháfer, N. S. Rajput, C. Aubry, E. Llobet, M. Jouiad and P. M. F. J. Costa, *Sensors and Actuators B: Chemical*, 2020, **305**, 127458.
- G. Deokar, P. Vancso, R. Arenal, F. Ravaux, J. Casanova-Cháfer, E. Llobet, A. Makarova, D. Vyalikh, C. Struzzi and P. Lambin, *Advanced Materials Interfaces*, 2017, 4, 1700801.
- 3. R. Kumar, N. Goel and M. Kumar, ACS sensors, 2017, 2, 1744-1752.
- 4. Y. Zhou, G. Liu, X. Zhu and Y. Guo, Sensors and Actuators B: Chemical, 2017, 251, 280-290.
- 5. F. Ghasemi, *Scientific Reports*, 2020, **10**, 11306.
- 6. A. Mukherjee, L. R. Jaidev, K. Chatterjee and A. Misra, *Nano express*, 2020, 1, 010003.
- 7. R. Kumar, N. Goel, A. V. Agrawal, R. Raliya, S. Rajamani, G. Gupta, P. Biswas, M. Kumar and M. Kumar, *IEEE Sensors Journal*, 2019, **19**, 10214-10220.
- 8. H. Long, A. Harley-Trochimczyk, T. Pham, Z. Tang, T. Shi, A. Zettl, C. Carraro, M. A. Worsley and R. Maboudian, *Advanced Functional Materials*, 2016, **26**, 5158-5165.
- 9. F. E. Annanouch, A. Alagh, P. Umek, J. Casanova-Chafer, C. Bittencourt and E. Llobet, *Journal of Materials Chemistry C*, 2022, **10**, 11027-11039.