Supporting Information for

Microstructure Parameters-Dependent Non-Collinear Magnetic Structure in Scandium-Doped M-Type Hexaferrite Nanocrystal

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Morphology and crystal Structure



Figure S1. Morphology of $BaFe_{12-x}Sc_xO_{19}$ ($0 \le X \le 3.47$) nanofibers. (a-e) SEM images of $BaFe_{12-x}Sc_xO_{19}$ nanofibers with x=0, 0.28, 0.91, 1.61, 1.86 respectively. (f) Line graph of the average diameter acquired using the Nano Measure software.

BaFe_{12-x}Sc_xO₁₉ nanofibers with different Sc concentration nanofibers were systematically studied. Figure S1a-e shows representative SEM images of BaFe_{12-x}Sc_xO₁₉ (x=0, 0.28, 0.65, 0.91, 1.22, 1.46, 1.61, 1.86, 2.31, 3.04). It is clearly seen that nanofibers consist of single nanoparticles arranged in a chain-like manner along the axial direction. The average diameter is approximately 60 nm as shown in Figure S1f. Note that the chemical composition of Ba: Fe: Sc of BaFe12-xScxO19 nanofibers is confirmed by the EDX spectra (Figure S2), and the Fe concentration of all samples was demonstrated to deviate from the nominal composition of raw materials.



Figure S2. EDX patterns of prepared BaFe_{12-x}Sc_xO₁₉ ($0 \le x \le 3.04$) nanofibers.



Figure S3. Room temperature X-ray diffraction patterns of $BaFe_{12-x}Sc_xO_{19}$ ($0 \le x \le 3.47$) nanofibers.

The XRD pattern of the BaFe_{12-x}Sc_xO₁₉ nanofibers indicates a *hcp* structure with *P63/mmc* space group, which is well maintained with increasing Sc content up to 3.04 (Figure S3). The diffraction peaks characteristic of M-type barium ferrite exhibits a gradual leftward shift with increasing Sc concentration, signifying an expansion of the lattice. This shift is attributed to the larger ionic radius of Sc³⁺ compared to Fe³⁺. As the value of *x* increments from 0 to 3.04, this shift is approximately 0.4° to the left. Note that, the BaFe₂O₄ impurity spinel phase was found as doping content increased to 3.47, indicating 3.04 is an upper limit for the Sc concentration. This can be attributed to hexagonal lattice expansion and chemical potential variation of the *hcp* structure.

Figure S4. (a)-(d) Locally enlarge HAADF-TEM image and ABF-TEM image, and line intensity profiles for the atomic columns respectively at $4f_1$ and 12k sites $BaFe_{12}O_{19}$ of $BaFe_{11.39}Sc_{1.61}O_{19}$. (e)-(f) Theoretical high-resolution HAADF-TEM images along the [110] zone axis of the $BaFe_{12-x}Sc_xO_{19}$, and line intensity profiles for the atomic columns at $4f_1$ and 12k sites.

x	2 <i>a</i>	2b	$4f_2$	<i>a</i> (Å)	<i>c</i> (Å)	c/a
0	0	0	0	5.892	23.204	3.938
0.28	0.21	0	0.07	5.901	23.279	3.945
0.91	0.40	0.12	0.39	5.917	23.439	3.961
1.46	0.46	0.27	0.73	5.927	23.532	3.971
1.61	0.48	0.32	0.81	5.931	23.561	3.973
1.86	0.52	0.98	0.36	5.944	23.651	3.979

Table S1. Detailed refinement parameters including the substituted amount of Fe³⁺ by Sc³⁺ at different sites (2*a*, 4*f*₂, 2*b*), and lattice-parameter values (*a*, *c*, and *c/a*) with different Sc-substitution amounts.



Figure S5. Room temperature hysteresis loops of $BaFe_{12-x}Sc_xO_{19}$ (3.47 $\leq x \leq 12$) nanofibers.

The presence of $BaFe_2O_4$ and other impurity phases is likely to attenuate the saturation magnetization and coercivity of our specimens when *x* further increased from 3.47 to 5.20, as indicated by Figure S5.



Figure S6. (a) Hysteresis loops of $BaFe_{12-x}Sc_xO_{19}$ with x=3.04 (b) Temperature dependence of coercivity and saturation magnetization.

Sublattice	Coordination	$B_{hf}(T)$	IS (mm/s)	QS (mm/s)
4 <i>f</i> ₂ (Fe4)	octahedral	51.0(1)	0.34(1)	0.10(2)
2a (Fe1)	octahedral	48.7(2)	0.27(3)	0.33(5)
$4f_1$ (Fe3)	tetrahedral	48.2(4)	0.31(5)	0.03(9)
12k (Fe5)	octahedral	41.5(4)	0.30(1)	0.41(1)
2 <i>b</i> (Fe2)	bi-pyramidal	38.7(3)	0.48(4)	2.15(7)

Table S2. Detailed refinement parameters of $BaFe_{12}O_{19}$.

Table S3. Detailed refinement parameters of $BaFe_{12-x}Sc_xO_{19}$ with x=1.61 and 1.86.

<i>x</i> value	$B_{hf}(T)$	IS (mm/s)	QS (mm/s)	
1.61	29.1(0)	0.37 (0)	0.18(8)	_
1.86	23.7(9)	0.39(3)	0.16(3)	