## - **Supporting Information -**

# Mechanical Properties and Molecular Adhesion Exhibited by Inorganic–Organic Composite Elastomers

*Naoki Yamashita,<sup>a</sup> Ryohei Ikura,a,b Kenji Yamaoka,a,b Nobu Kato,<sup>c</sup> Masanao Kamei,<sup>c</sup> Kentaro Ogura,<sup>c</sup> Minoru Igarashi,<sup>c</sup> Hideo Nakagawa, d and Yoshinori Takashimaa,b,e\**

- a. Department of Macromolecular Science, Graduate School of Science, Osaka University, 1-1 Machikaneyamacho, Toyonaka, Osaka 560-0043, Japan
- b. Forefront Research Center, Graduate School of Science, Osaka University,1-1 Machikaneyamacho, Toyonaka, Osaka 560-0043, Japan
- c. Shin-Estu Chemical Co., Ltd., Silicone-Electronics Materials Research Center, 1-10, Hitomi, Matsuida-machi, Annaka-shi, Gunma 379-0224, Japan
- d. Shin-Etsu Chemical Co., Ltd., 4-1 Marunouchi, 1-chome, Chiyoda-ku, Tokyo 100-0005, Japan
- e. Innovative Catalysis Science Division, Institute for Open and Transdisciplinary Research Initiatives, Osaka University, 1-1 Yamadaoka, Suita, Osaka 565-0871, Japan

## **Content**



#### <span id="page-2-0"></span>**S1. Materials**

Thiol-modified polydimethylsiloxane (PDMS-SH) was obtained from Shin-Etsu Chemical Co., Ltd. Allyl alcohol and ethyl acetate were purchased from FUJIFILM Wako Pure Chemical Corporation. 2-Hydroxy-2-methylpropiophenone (IRGACURE 1173) was purchased from Sigma-Aldrich Co. Ethyl acrylate (EA) was purchased from Nacalai Tesque, Inc. Phenylbis(2, 4, 6-trimethylbenzoil)phosphine oxide was purchased from Tokyo Chemical Industry Co., Ltd. Deuterated chloroform (CDCl<sub>3</sub>) was purchased from Eurisotop. Triacetylated 6-acrylamido methylether-β-cyclodextrin (TAcβCDAAmMe) was purchased from Kyoeisha Chemical Co., Ltd. *N*-(1-adamantyl)acrylamide (AdAAm) was purchased from Yushiro Chemical Industry Co., Ltd.

#### <span id="page-2-1"></span>**S2. Measurements**

**Nuclear magnetic resonance (NMR) spectroscopy:** <sup>1</sup>H NMR spectra were recorded at 400 MHz with a JEOL ECS-400 NMR spectrometer at 25 °C.  ${}^{1}$ H- ${}^{1}$ H nuclear Overhauser effect spectroscopy (NOESY) NMR spectra were recorded at 600 MHz with an Agilent VNS600 NMR spectrometer at 25 °C. Chemical shift values were referenced to the CHCl<sub>3</sub> value ( $\delta$  = 7.26 ppm) for <sup>1</sup>H NMR.

**Gel permeation chromatography (GPC):** Number-average molecular weight (*M*n), weight average molecular weight (*M*w), and molecular weight distribution (*M*w/*M*n) were measured by GPC in THF at 40 °C with two columns (Tosoh TSK gel SuperHZM-N  $\times$ 2). The molecular weights of the samples were calculated based on calibration curve got by polystyrene standards.

**Tensile test:** Tensile tests of the elastomers were performed using an Autograph AG-X plus (Shimadzu Co.) at a deformation rate of 1.0 and 0.1 mm/s. Rectangular test pieces (20 mm  $\times$  5 mm  $\times$  0.5 mm) were used for the tensile test. The toughness was calculated from the integral of the stress–strain curve. The Young's modulus was calculated from the initial slope of the stress-strain curve at a range of 1-6% strain.

**Cyclic tensile test:** Cyclic tensile tests were performed using an Autograph AG-X plus (Shimadzu Co.). Rectangular test pieces (20 mm  $\times$  5 mm  $\times$  0.5 mm) were continuously stretched and recovered without interval, where the maximum strains were set to 140%, 280%, 420%, 560%, 700%, 840%, and 980% with a deformation rate of 1 mm/s

**Stress relaxation test:** Stress relaxation tests were performed using an Autograph AG-X plus (Shimadzu Co.). The test pieces were stretched until 100%. Then, the strain was held, and the stress was recorded for 1000 seconds.

**X-ray scattering measurements:** The internal structures of the elastomers were determined by smallangle X-ray scattering (SAXS) measurements at the BL19B2 beam line in SPring-8, Nishi-harima, Japan. The power of the incident X-ray beams for BL19B2 was 18 keV. The sample-to-detector lengths for SAXS measurements was 3 m. The length of the scattering vector  $q$  in the SAXS measurements was 0.1-3 nm<sup>-1</sup>, where  $q = 4\pi \sin \lambda (2\theta$  and  $\lambda$  are the scattering angle and the wavelength, respectively).

**Fourier transform infrared (FT-IR) spectroscopy:** FT-IR spectra were acquired in ATR mode through diamond crystals with  $N_2$  gas flow (JASCO FT/IR-410).

#### <span id="page-4-0"></span>**S3. Preparation of PDMS-CD**



**Scheme S1.** Preparation of PDMS-CD.

PDMS-SH (4500 mg, SH group: 5.6 mmol, 10 eq.), TAcβCDAAmMe (1200 mg, 0.55 mmol, 1 eq.), allyl alcohol (290 mg, 5.0 mmol, 9 eq.), and IRGACURE 1173 (92 mg, 0.56 mmol, 1 eq.) were dissolved in ethyl acetate (20 mL). The solution was irradiated by UV light with a high-pressure Hg lamp  $(\lambda = 253$  and 365 nm) for 2 hours.



Figure S1. <sup>1</sup>H NMR spectrum of PDMS-CD (CDCl<sub>3</sub>, 400 MHz, 25 °C).



**Figure S2.** FT-IR spectra of PDMS-SH, TAcγCDAAmMe, and PDMS-CD.

### <span id="page-6-0"></span>**S4. Preparation of PDMS-CD/P(EA-Ad) (***x***)**



**Scheme S2.** Preparation of PDMS-CD/P(EA-Ad) (*x*).

EA, AdAAm, and BAPO were added to ethyl acetate solution of PDMS-CD (4800 mg, including 1200 mg of PDMS-CD). The solution was irradiated by visible light with a LED ( $\lambda = 420$  nm) for 1 hour. The products were dried at r.t. for 24 hours and then at 80 ℃ in vacuum for 24 hours. Table S1 summarize amounts of reagents in the reactions.

X /mol%	EA		AdAAm		<b>BAPO</b>	
	/mg	/mmol	/mg	/mmol	/mg	/mmol
	900	9.0	23	0.11	19	0.045
2	900	9.0	46	0.22	19	0.045
5	900	9.0	120	0.56	19	0.045

**Table S1.** Amount of each reagent for PDMS-CD/P(EA-Ad) (*x*).



**Figure S3.** <sup>1</sup>H NMR spectrum of PDMS-CD/P(EA-Ad) (1) (CDCl<sub>3</sub>, 400 MHz, 25 °C).



**Figure S4.** <sup>1</sup>H NMR spectrum of PDMS-CD/P(EA-Ad) (2) (CDCl<sub>3</sub>, 400 MHz, 25 °C).



**Figure S5.** <sup>1</sup>H NMR spectrum of PDMS-CD/P(EA-Ad) (5) (CDCl<sub>3</sub>, 400 MHz, 25 °C).



**Figure S6.** FT-IR spectra of PDMS-CD, EA, AdAAm, and PDMS-CD/P(EA-Ad) (5).

#### <span id="page-9-0"></span>**S5. Preparation of PDMS-CD/PEA**



**Scheme S3.** Preparation of PDMS-CD/PEA.

EA (900 mg, 9.0 mmol) and BAPO (19 mg, 0.045 mmol) were added to ethyl acetate solution of PDMS-CD (4800 mg, including 1200 mg of PDMS-CD). The solution was irradiated by visible light with a LED ( $\lambda = 420$  nm) for 1 hour. The products were dried at r.t. for 24 hours and then at 80 °C in vacuum for 24 hours.



Figure S7. <sup>1</sup>H NMR spectrum of PDMS-CD/PEA (CDCl<sub>3</sub>, 400 MHz, 25 °C).



**Figure S8.** FT-IR spectra of PDMS-CD, EA, and PDMS-CD/PEA.

#### <span id="page-11-0"></span>**S6. Preparation of PDMS/P(EA-Ad) (5)**



**Scheme S4.** Preparation of PDMS/P(EA-Ad) (5).

PDMS-SH (4500 mg, SH group: 5.6 mmol, 10 eq.), allyl alcohol (330 mg, 5.7 mmol, 10 eq.), and IRGACURE 1173 (92 mg, 0.56 mmol, 1 eq.) were dissolved in ethyl acetate (20 mL). The solution was irradiated by UV light with a high-pressure Hg lamp (the dominant wavelength,  $\lambda = 253$  and 365 nm) for 2 hours. The exposure enargy is 2.05 mW/cm<sup>2</sup> at  $\lambda$  = 365 nm. EA (900 mg, 9.0 mmol), AdAAm (120 mg, 0.56 mmol), and BAPO (19 mg, 0.045 mmol) were added to obtained ethyl acetate solution of PDMS-AAl (4600 mg, including 960 mg of PDMS-AAl). The solution was irradiated by visible light with a LED  $(\lambda = 420 \text{ nm})$  for 1 hour. The products were dried at r.t. for 24 hours and then at 80 °C in vacuum for 24 hours.



**Figure S9.** <sup>1</sup>H NMR spectrum of PDMS/P(EA-Ad) (5) (CDCl<sub>3</sub>, 400 MHz, 25 °C).



**Figure S10.** FT-IR spectra of PDMS-SH, EA, AdAAm, and PDMS/P(EA-Ad) (5).

#### <span id="page-13-0"></span>**S7. Preparation of PDMS/PEA**



**Scheme S5.** Preparation of PDMS/PEA.

PDMS-SH (4500 mg, SH group: 5.6 mmol, 10 eq.), allyl alcohol (330 mg, 5.7 mmol, 10 eq.), and IRGACURE 1173 (92 mg, 0.56 mmol, 1 eq.) were dissolved in ethyl acetate (20 mL). The solution was irradiated by UV light with a high-pressure Hg lamp (the dominant wavelength,  $\lambda = 253$  and 365 nm) for 2 hours. The exposure enargy is 2.05 mW/cm<sup>2</sup> at  $\lambda$  = 365 nm. EA (900 mg, 9.0 mmol) and BAPO (19 mg, 0.045 mmol) were added to obtained ethyl acetate solution of PDMS-AAl (4600 mg, including 960 mg of PDMS-AAl). The solution was irradiated by visible light with a LED ( $\lambda$  = 420 nm) for 1 hour. The products were dried at r.t. for 24 hours and then at 80 ℃ in vacuum for 24 hours.



Figure S11. <sup>1</sup>H NMR spectrum of PDMS/PEA (CDCl<sub>3</sub>, 400 MHz, 25 °C).



**Figure S12.** FT-IR spectra of PDMS-SH, EA, and PDMS/PEA.

<span id="page-15-0"></span>**S8. Investigation of molecular weight by GPC**



**Figure S13.** GPC chart of PDMS-SH and PDMS-CD.

<span id="page-16-0"></span>**S9. Confirmation of inclusion complex formation by NOESY NMR** 



**Figure S14.** 2D NOESY NMR spectrum of PDMS-CD/P(EA-Ad) (5) (CDCl3, 600 MHz, 25 ℃, mixing time  $\tau = 1000$  ms).

#### <span id="page-17-0"></span>**S10. Investigation of deformation behaviors in cyclic tensile test**

We investigated the effect of the reversible cross-links with cyclic tensile tests of PDMS-CD/P(EA-Ad) (5). The analysis revealed three distinct modes of molecular motion in PDMS-CD/P(EA-Ad) (5): phase separation, exchange of reversible cross-links, and extension of polymer chains. The detailed discussion is as follows:

#### **(i) Phase separation of PEA and PDMS domains in the low strain region**

The stretching curves from  $2<sup>nd</sup>$  cycle exhibit significantly lower stress compared to the 1st cycle, indicating inadequate recovery of mechanical properties due to substantial structural changes caused by stress-induced phase separation. Previous research on composites with movable cross-links connecting dissimilar polymers also showed stress-induced phase separation at low strains (0–120%) in SAXS profiles of composite elastomers under tensile deformation<sup>[1]</sup>. Therefore, the upward convex shape of the stress-strain curve for PDMS-CD/P(EA-Ad) (5) at low strains is attributed to stress-induced phase separation.



**Figure S15.** A cyclic tensile test for PDMS-CD/P(EA-Ad) (5).

#### **(ii) Exchange of reversible cross-links in the middle strain region**

In the  $2<sup>nd</sup>$  through  $5<sup>th</sup>$  cycles, the decrease in the stress of stretching curves is moderate, indicating the effective recovery of the mechanical properties due to reformation and exchanging of the host-guest complex. Thus, the S-shaped curve with a wide range of small slopes in middle strain region can be attributed to the stress relaxation resulting from the exchange of reversible cross-links.

#### **(iii) Extending polymer chains in the high strain region**

The downward convex stress-strain curves at high strain is characteristic of stress-hardening due to the extension of polymer chains. In the  $5<sup>th</sup>$  through  $7<sup>th</sup>$  cycles, the significant decrease in the stress of stretching curves indicates the inadequate recovery during this timescale.

#### <span id="page-18-0"></span>**S11. Investigation of the network structure by swelling tests**

To characterize the network structures, we conducted swelling tests using chloroform as the swelling solvent. The polymers were immersed in excess chloroform for two days. The swelling ratio (*Q*) was determined by the following equation:

Swelling ratio (Q) = 
$$
\frac{W - W_0}{W_0} \times 100\%
$$
 (S1)

where *W* is the weight of the swollen polymer and *W*<sup>0</sup> is the initial weight of the polymer before swelling.



**Figure S16. (a)** Experimental procedure of the swelling test. **(b)** Swelling ratios of PDMS-CD/PEA and PDMS-CD/P(EA-Ad)  $(x)$   $(x = 1, 2,$  and 5).

#### <span id="page-19-0"></span>**S12. Evaluation of the stress dispersion property by stress relaxation tests**

Curve fitting was carried out on the obtained stress *σ* versus relaxation time *t* curves using the Kohlrausch–Williams–Watts models, as described by the following equation.

$$
\sigma = \sigma_r \exp\left\{-\left(\frac{t}{\tau}\right)^{\beta}\right\} + \sigma_{\infty} \tag{S2}
$$

In the above equation,  $\sigma_r$  is the relaxable stress,  $\sigma_\infty$  is the residual stress,  $\tau$  is the time constant, and  $\beta$  is the stretching exponent.



**Figure S17.** Stress relaxation curves of PDMS-CD/P(EA-Ad)  $(x)$  when  $x = (a)$  1,  $(b)$  2,  $(c)$  5,  $(d)$ PDMS-CD/PEA, and **(e)** PDMS/P(EA-Ad) (5) with fitting curves (raw data: cyan solid line, fitting curves: brown dashed line).

	Relaxable component			<b>Residual component</b>	
	$\sigma$	T/S		$\sigma_{\scriptscriptstyle{\alpha}}$	
PDMS-CD / P(EA-Ad) (1)	0.33	17	0.33	0.67	
PDMS-CD / P(EA-Ad) (2)	0.37	21	0.37	0.63	
PDMS-CD / P(EA-Ad) (5)	0.61	19	0.36	0.39	
PDMS-CD / PEA	0.32	16	0.33	0.68	
PDMS / $P(EA-Ad)$ (5)	0.50	81	0.45	0.50	

**Table S2.** Fitting parameters obtained for PDMS-CD/P(EA-Ad) (*x*), PDMS-CD/PEA, and PDMS/P(EA-Ad) (5) by using the KWW models.

#### <span id="page-21-0"></span>**S13. Tensile test at different tensile rates**

The tensile tests of PDMS-CD/PEA and PDMS-CD/P(EA-Ad)  $(x)$  ( $x = 1, 2,$  and 5) were performed at 0.1 and 1.0 mm/s. The tensile test of all composites at 0.1 mm/s show decrease in toughness. Too slow tensile speed decreases the elasticity of reversible cross-links connecting the PDMS and PEA domains, impairing the cooperative deformation of these domains to decrease the toughness. In our previous work<sup>R1</sup>, composites with movable cross-links also exhibited reduced toughness under slower tensile deformation. The decreases in the elasticity of reversible cross-links also leads to the reduction in Young's modulus. PDMS-CD/P(EA-Ad) (5) shows the largest decrease in the Young's modulus. PDMS-CD/PEA and PDMS-CD/P(EA-Ad) (*x*) ( $x = 1, 2$ , and 5) show similar time constant ( $\tau$ ), which contributes to the similar dependency of mechanical properties on tensile speed.



**Figure S18. (a)** Stress‒strain curves and **(b)** plots of the toughness and Young's modulus values of PDMS-CD/PEA and PDMS-CD/P(EA-Ad)  $(x)$  ( $x = 1, 2,$  and 5) with tensile rate of 1 or 0.1 mm/s.

#### <span id="page-21-1"></span>**S14. References**

[1]. R. Ikura, S. Murayama, J. Park, Y. Ikemoto, M. Osaki, H. Yamaguchi, A. Harada, G. Matsuba and Y. Takashima *Molecular Systems Design & Engineering* 2022, **7**, 733-745.