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### Supporting Information

# Heteroanionic $[\text{VO}_x\text{S}_{4-x}]$ Groups: Tetrahedral Units with Large Bire-fringence for Mid-infrared Nonlinear Optical Crystals

Shengzi Zhang,<sup>†ab</sup> Linfeng Dong,<sup>†ad</sup> Bohui Xu,<sup>ad</sup> Huige Chen,<sup>ad</sup> Hao Huo,<sup>ad</sup> Fei Liang,<sup>a</sup> Rui Wu,<sup>c</sup> Pifu Gong,<sup>\*a</sup> and Zheshuai Lin<sup>\*ad</sup>

**Abstract:** Tetrahedral groups can possess wide bandgap and large second harmonic generation (SHG) response in nonlinear optical (NLO) materials, but they usually exhibit small optical birefringence owing to low structural anisotropy, which would deteriorate the NLO performance. Herein the  $[\text{VO}_x\text{S}_{4-x}]$  ( $x = 0 - 4$ ) tetrahedra that combine  $\text{V}^{5+}$  cation with oxygen-sulfur hybrid anions are highlighted as a type of good structural groups to overcome the above problem. We systematically investigate the optical and NLO properties of all reported  $\text{V}^{5+}$ -based thiovanadates theoretically. The calculated results reveal the largest birefringence ( $\sim 0.37$  in  $\text{Ba}_5\text{V}_2\text{O}_4\text{S}_8$ ) among the pure tetrahedra-based compounds and indicate the potential of  $[\text{VO}_x\text{S}_{4-x}]$  units in enhancing birefringence and well-balanced NLO performance suitable for mid-infrared (mid-IR) region. Furthermore,  $\text{Na}_3\text{VOS}_3$  in this family is selected for experimental synthesis and measurements to verify our strategy and calculations. This work demonstrates that the  $[\text{VO}_x\text{S}_{4-x}]$  ( $x = 0 - 4$ ) tetrahedral groups are a good type of birefringent and NLO unit, but was long-termly overlooked, for the exploration of mid-IR NLO materials.

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## Experimental Procedures

### **Single-crystal structure determination:**

The Single-crystal structure determination was performed on Rigaku AFC10 diffractometer equipped with a graphite-monochromated  $K\alpha$  ( $\lambda = 0.71073 \text{ \AA}$ ) radiation. A crystal of  $\text{Na}_3\text{VOS}_3$  with a size of  $0.15 \times 0.2 \times 0.2 \text{ mm}^3$  was picked for measurement. Diffraction data were collected using the Crystalclear software in the range of  $3.4340^\circ \leq \theta \leq 28.725^\circ$ , and the data was subjected to absorption correction using the XPREP program. The crystal structure framework is determined using the SHELX direct method, with structure analysis and refinement carried out using SHELXS and OLEX2 software. Differential Fourier synthesis is used to determine the specific location of each atom. Finally, the coordinates and anisotropy parameters of all atoms are fitted by F2-based full matrix least squares fitting until the final convergence. The PLATON program was used to examine the symmetry of the structure, which revealed no high symmetry. As a result, the crystal structure of the compound was determined. Table S1 lists the data for structural refinement.

### **Reagents:**

The following reagents used in the experiments were purchased directly from Aladdin Industries without further purification, including sodium sulphide (99.999%), vanadium (99.999%), vanadium pentoxide (99.99%), sulphur (99.99%).

### **Synthesis:**

Polycrystalline powders of  $\text{Na}_3\text{VOS}_3$  were obtained via high-temperature solid-state synthesis in an anhydrous and oxygen-free environment. The reagents were weighed and ground in stoichiometric ratios in a dry, inert argon glove box with oxygen and water vapour content below 0.1 ppm. The raw materials were sealed in quartz tubes under a high vacuum of  $10^{-3} \text{ Pa}$  and then placed in a temperature-programmed tube furnace. The quartz tubes were heated to  $350^\circ\text{C}$  at a rate of  $13^\circ\text{C/h}$  and maintained at this temperature for 100 hours. After the reaction, the tube furnace was turned off and allowed to cool naturally to room temperature.

### **Crystal Growth:**

Single crystals of  $\text{Na}_3\text{VOS}_3$  were grown using  $\text{Li}_2\text{O}$  as a flux and oxygen source. The reagent V: S:  $\text{Li}_2\text{O}$ :  $\text{Na}_2\text{S}$  was weighed to be 1: 5: 1.2: 2, fully ground, and packed into a quartz glass tube. The tube was then sealed in high vacuum and heated slowly to  $450^\circ\text{C}$  at a rate of  $10^\circ\text{C/hour}$ . The temperature was maintained at  $450^\circ\text{C}$  for 72 hours before being cooled to  $100^\circ\text{C}$  at a rate of  $3^\circ\text{C/hour}$ . Finally, the tube was naturally cooled to room temperature. After completion of the reaction, numerous red needle-like crystals adhered to the walls of the quartz tube. Cleaning the reaction products with DMF (Dimethyl Formamide) resulted in well-crystallized red needle-like crystals, some of which were up to a centimeter long. The crystal picture is shown in Figure 3b. The crystals are slightly deliquescent when left in the air for a long time.

### **Powder X-ray diffraction:**

Polycrystalline  $\text{Na}_3\text{VOS}_3$  powders were subjected to X-ray diffraction measurements using a Bruker D8 Focus powder diffractometer. The scanning range was  $5^\circ$  to  $70^\circ$  with a step length of  $0.02^\circ$  and a scanning speed of  $0.1 \text{ s/step}$ . The diffraction seam and the receiving seam were both 1.0 mm. The Cu target  $K\alpha$  radiation with a wavelength of  $\lambda = 1.5418 \text{ \AA}$  was used, and the operating voltage and current were 40 kV and 40 mA, respectively. The X-ray diffraction pattern agrees well with the fitted pattern of the solved  $\text{Na}_3\text{VOS}_3$  crystal structure, indicating that a pure sample was obtained (Fig. 3a).

### **UV-vis-NIR diffuse reflectance spectra:**

The Cary 7000 UV-vis-NIR spectrometer with diffuse integrating sphere attachment was used to collect the UV-vis-NIR diffuse reflection spectrum of  $\text{Na}_3\text{VOS}_3$  powder at 298 K in the wavelength range of 200–2000 nm. With  $\text{BaSO}_4$  as the reference material, the reflection spectrum was converted into the absorption spectrum through Kubelka-Munk function, from which one can obtain the experimental band gap of  $\text{Na}_3\text{VOS}_3$ .

### **Power SHG measurement:**

The Kurtz–Perry method was adopted to measure the SHG effect of  $\text{Na}_3\text{VOS}_3$  powder. Standard sieve was used to divide the pure polycrystalline  $\text{Na}_3\text{VOS}_3$  powder into five particle sizes (20–41  $\mu\text{m}$ , 41–74  $\mu\text{m}$ , 74–105  $\mu\text{m}$ , 105–150  $\mu\text{m}$ , 150–212  $\mu\text{m}$ ), which were used for test separately. The fundamental frequency light used in our measurement was generated by a Ho: Tm: Cr: YAG laser with a Q switch. The wavelength of the fundamental frequency light is 2090 nm (10 ns, 1 Hz). By measuring samples of different particle sizes, the relative intensities of the frequency doubling light and the fundamental frequency light were recorded on an oscilloscope. Ten values were taken for each particle size and averaged, and a reference material  $\text{AgGaS}_2$  of comparable particle size was tested in the same way.

### **Energy-dispersive X-ray spectroscopy:**

Semiquantitative EDS analyses were conducted using an energy-dispersive-X-ray (EDX)-equipped Hitachi S-4300 scanning electron microscope on the crystals that had been measured by single-crystal XRD.

### **Raman spectroscopy:**

Raman spectroscopy was performed on inVia-Reflex instrument with a line of 532 nm of solid-state laser. The spectral range of the sample is between 100 and  $800 \text{ cm}^{-1}$ .

### **Photoluminescence measurements:**

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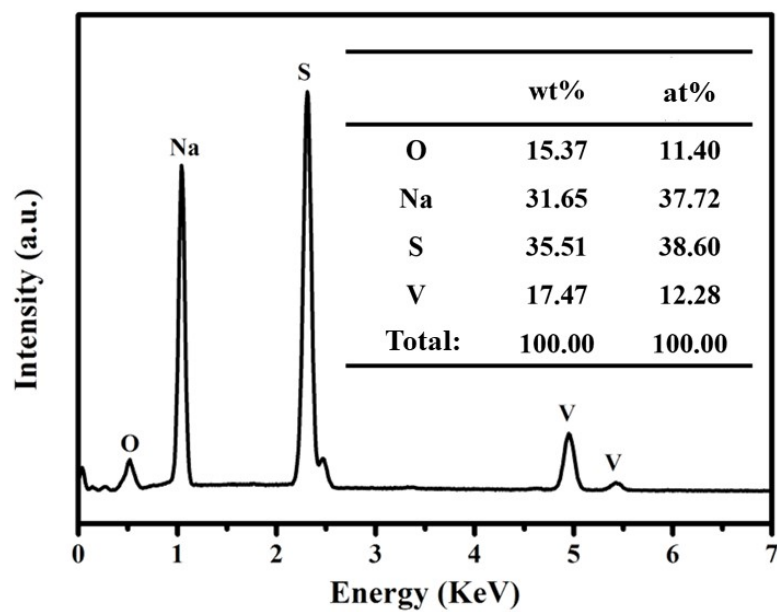
The photoluminescence spectra of  $\text{Na}_3\text{VOS}_3$  were obtained using an inVia-Reflex Raman spectrometer from Renishaw, UK, by switching its regular operating mode.

### **Computational method:**

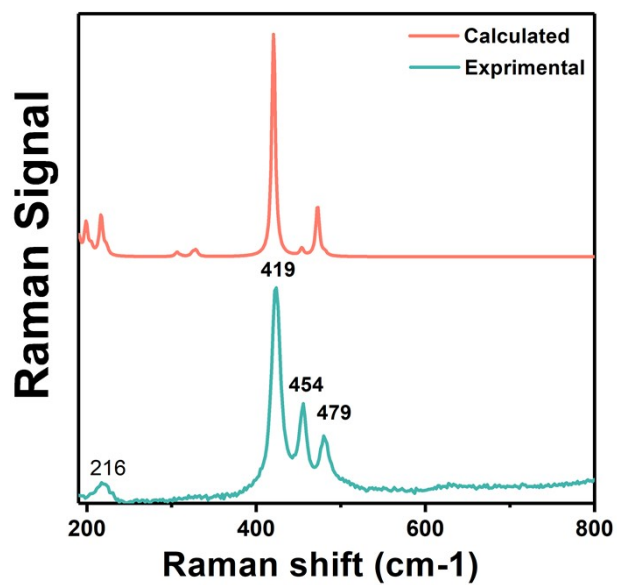
First-principles calculations were performed on  $\text{Na}_3\text{VOS}_3$  using the CASTEP package and the density functional plane-wave pseudopotential method.<sup>1-3</sup> Geometric optimization was carried out using the Broyden Fletcher Goldfarb Shanno (BFGS) minimization method.<sup>4</sup> The convergence criteria for energy, maximum force, maximum stress, and maximum displacement were set at  $1 \times 10^{-5}$  eV, 0.03 eV/Å, 0.05 GPa and  $10^{-3}$  Å, respectively. The local-density approximation (LDA) was chosen for all calculations. The plane wave cutoff energy<sup>5</sup> was set at 700 eV and the Monkhorst-Pack k point density in the Brillouin zone was set at  $2 \times 2 \times 4$ . Norm-conserving pseudopotentials<sup>6</sup> were employed, with the valence electrons of Na  $2s^2 2p^6 3s^1$ , O  $2s^2 2p^4$ , S  $3s^2 3p^4$ , and V  $3d^3 4s^2$ .<sup>7</sup> As the exchange-correlation energy functional is discontinuous, LDA typically underestimates the band gap.<sup>8</sup> Therefore, a scissors factor correction was applied to calculate the optical properties. The SHG coefficient  $d_{ij}$  can be obtained by using the electron band structure modified by the scissors factor.<sup>9</sup> Additionally, a band-resolved  $d_{ij}$  and SHG density analysis is performed to demonstrate the SHG contributions of electronic orbitals and structural motifs in real space<sup>10</sup> To accurately evaluate the band gaps of the compounds, including the designed ones, we used the hybrid sx-LDA functionals.<sup>11</sup> The scissors factors of the designed compounds were set as the difference between the sx-LDA and LDA band gaps. According to our previous theoretical calculations, the parameter settings mentioned above can accurately predict the energy band gaps of chalcogenide compounds.<sup>12, 13</sup>

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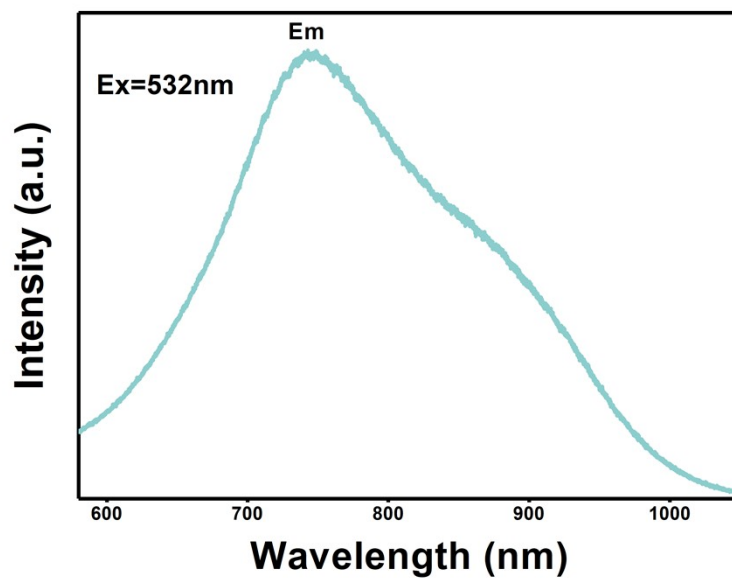
## Results and Discussion



**Figure S1.** Energy dispersive spectrometer (EDS) image of  $\text{Na}_3\text{VOS}_3$  crystal  
The results showed a Na/V/O/S molar ratio of 3.3:1.1:3.3:1.0.

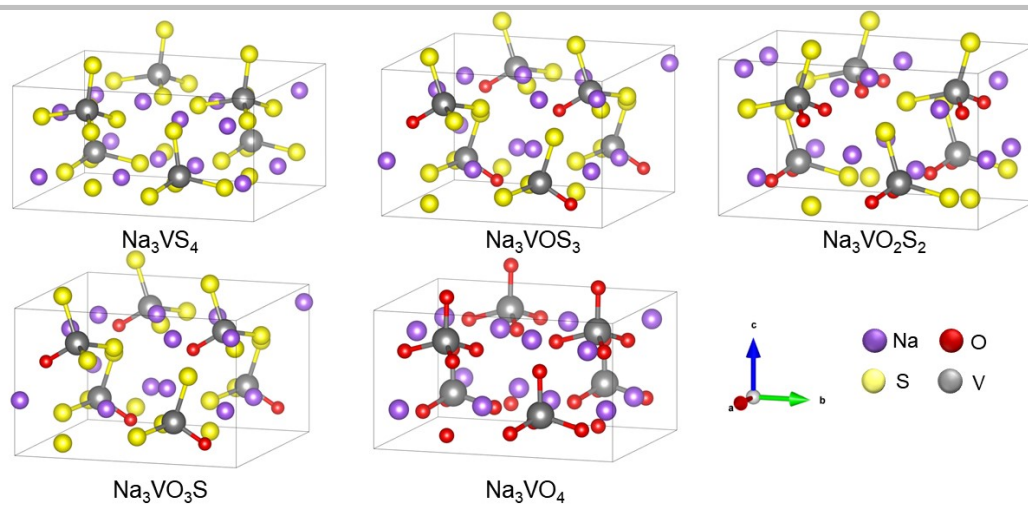


**Figure S2.** Comparison of the experimental raman spectra of  $\text{Na}_3\text{VOS}_3$  and calculated spectra.  $\text{a}_3\text{VOS}_3$  exhibits significant absorption peaks at 216, 419, 454, and 479  $\text{cm}^{-1}$ , which can be attributed to the characteristic absorption of the V-O bond, the respiratory vibration of the  $[\text{VOS}_3]$  tetrahedron, and the asymmetric stretching vibration, respectively.



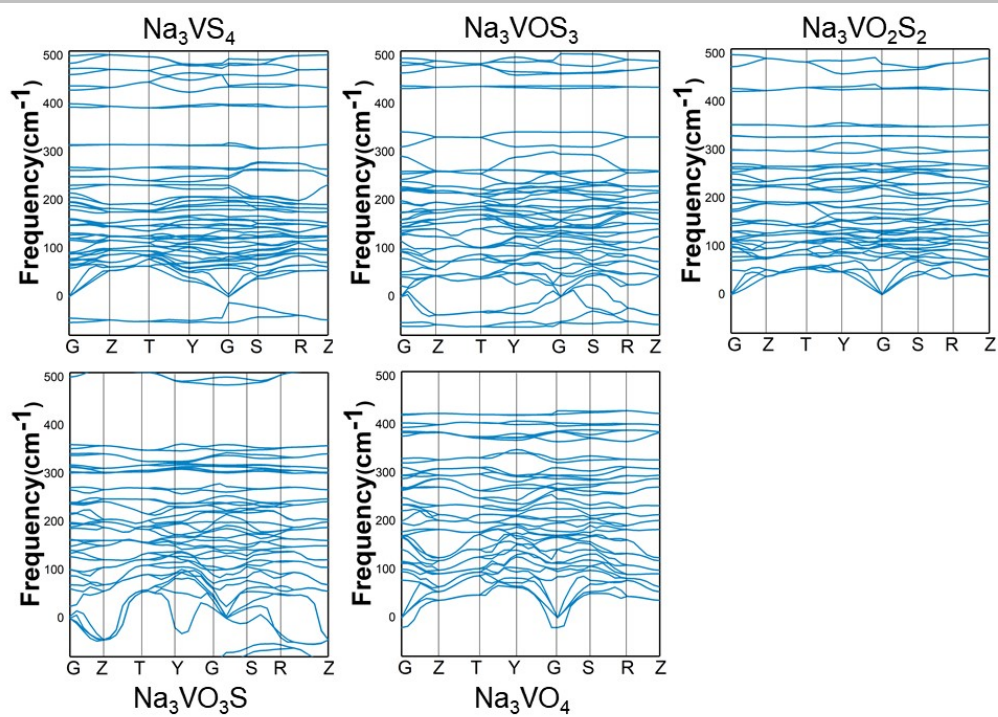
**Figure S3.** Photoluminescence spectroscopy of  $\text{Na}_3\text{VOS}_3$ . The excitation light was set at 532 nm and the emission wavelength was measured to be around 710 nm.

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**Figure S4.** Designed a series  $\text{Na}_3[\text{VO}_x\text{S}_{4-x}]$  ( $x = 0 - 4$ ) compounds using  $\beta\text{-Na}_3\text{VOS}_3$  as the parent structure.

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**Figure S5.** calculated phonon spectra  $\text{Na}_3[\text{VO}_x\text{S}_{4-x}]$  ( $x = 0 - 4$ )



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**Table S1.**  $\Delta n$  and energy band gap data for tetrahedral sulfides, halides<sup>14, 15</sup>

Compounds	Eg	$\Delta n @1\mu\text{m}$	Compounds	Eg	$\Delta n @1\mu\text{m}$	Compounds	Eg	$\Delta n @1\mu\text{m}$
a-ZnS	3.24	0.000	Cu <sub>2</sub> HgGeS <sub>4</sub>	1.12	0.024	CuCl	2.600	0.000
b-ZnS	3.28	0.003	Cu <sub>2</sub> ZnSnS <sub>4</sub>	1.19	0.141	CuBr	2.700	0.000
AgAlS <sub>2</sub>	3.18	0.047	AgCd <sub>2</sub> GaS <sub>4</sub>	2.32	0.059	CuI	3.300	0.000
AgGaS <sub>2</sub>	2.76	0.034	LiCd <sub>2</sub> GaS <sub>4</sub>	2.61	0.017	CuI	3.400	0.013
LiAlS <sub>2</sub>	4.68	0.014	LiZn <sub>2</sub> GaS <sub>4</sub>	3.1	0.017	LiI	6.200	0.013
LiGaS <sub>2</sub>	3.77	0.014	CuZn <sub>2</sub> AlS <sub>4</sub>	2.12	0.006	ZnCl <sub>2</sub> -a	5.100	0.014
Ag <sub>2</sub> GeS <sub>3</sub>	1.45	0.092	Ag <sub>4</sub> CdGe <sub>2</sub> S <sub>7</sub>	2.45	0.021	ZnCl <sub>2</sub> -b	5.000	0.029
Cu <sub>2</sub> SiS <sub>3</sub>	2.31	0.112	Ag <sub>4</sub> HgGe <sub>2</sub> S <sub>7</sub>	2.11	0.058	CuGaCl <sub>4</sub>	3.800	0.005
Cu <sub>2</sub> GeS <sub>3</sub>	1.57	0.093	Li <sub>4</sub> CdGe <sub>2</sub> S <sub>7</sub>	4.23	0.015	CuAlCl <sub>4</sub>	4.600	0.000
Cu <sub>2</sub> SiS <sub>3</sub>	2.34	0.045	GeS <sub>2</sub>	3.85	0.05	CuAlBr <sub>4</sub>	4.400	0.000
Cu <sub>2</sub> SnS <sub>3</sub>	1.09	0.078	Ga <sub>2</sub> S <sub>3</sub>	3.01	0.041	CuGaI <sub>4</sub>	3.200	0.002
Cu <sub>3</sub> SbS <sub>4</sub>	0.83	0.072	HgGa <sub>2</sub> S <sub>4</sub>	3.02	0.078	Cu <sub>2</sub> HgI <sub>4</sub> -a	2.500	0.032
Cu <sub>3</sub> AsS <sub>4</sub>	0.89	0.23	CdGa <sub>2</sub> S <sub>4</sub>	3.41	0.042	Cu <sub>2</sub> HgI <sub>4</sub> -b	2.000	0.024
Li <sub>3</sub> PS <sub>4</sub>	3.68	0.016	ZnGa <sub>2</sub> S <sub>4</sub>	3.6	0.029	Ag <sub>2</sub> HgI <sub>4</sub> -a	2.800	0.031
Cu <sub>3</sub> PS <sub>4</sub>	3.07	0.048	Hg <sub>2</sub> SiS <sub>4</sub>	2.92	0.118	Ag <sub>2</sub> HgI <sub>4</sub> -b	2.500	0.003
Ag <sub>3</sub> PS <sub>4</sub>	2.88	0.039	Hg <sub>2</sub> GeS <sub>4</sub>	2.8	0.096	Ag <sub>2</sub> CdI <sub>4</sub>	3.600	0.004
Li <sub>2</sub> CdGeS <sub>4</sub>	3.42	0.023	Hg <sub>2</sub> SnS <sub>4</sub>	2.4	0.022	Ag <sub>2</sub> ZnI <sub>4</sub>	3.700	0.022
Li <sub>2</sub> CdSnS <sub>4</sub>	3.11	0.018	Zn <sub>2</sub> GeS <sub>4</sub>	3.96	0.006			
a-Cu <sub>2</sub> ZnSiS <sub>4</sub>	2.01	0.052	Zn <sub>3</sub> (PS <sub>4</sub> ) <sub>2</sub>	3.19	0.035			
Li <sub>2</sub> MnGeS <sub>4</sub>	3.07	0.01	InPS <sub>4</sub>	3.44	0.023			
a-Li <sub>2</sub> MnSnS <sub>4</sub>	3	0.011	AlPS <sub>4</sub>	3.47	0.217			
Ag <sub>2</sub> CdGeS <sub>4</sub>	2.05	0.037	BPS <sub>4</sub>	3.02	0.361			
Li <sub>2</sub> ZnSnS <sub>4</sub>	3.32	0.015	LiZnPS <sub>4</sub>	3.76	0.072			
Ag <sub>2</sub> ZnSiS <sub>4</sub>	3.05	0.034	AgZnPS <sub>4</sub>	3.21	0.051			
b-Cu <sub>2</sub> ZnSiS <sub>4</sub>	2.39	0.035	LiBSiS <sub>4</sub>	3.87	0.03			
Cu <sub>2</sub> CdSnS <sub>4</sub>	1.05	0.134	LiBGeS <sub>4</sub>	3.95	0.031			

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**Table S2.** Representative works on photocatalytic methane coupling

Identification code	Na <sub>3</sub> VOS <sub>3</sub>
Empirical formula	Na <sub>2</sub> O <sub>0.67</sub> S <sub>2</sub> V <sub>0.67</sub>
Formula weight	154.73
Temperature/K	293(2)
Crystal system	orthorhombic
Space group	Cmc2 <sub>1</sub>
a/Å	9.5667(7)
b/Å	11.8659(9)
c/Å	5.8806(4)
α/°	90
β/°	90
γ/°	90
Volume/Å <sup>3</sup>	667.55(8)
Z	6
Calculated density (g/cm <sup>3</sup> )	2.309
Absorption coefficient (mm <sup>-1</sup> )	2.503
F(000)	448.0
2θ range for data collection/°	6.868 to 57.45
Independent reflections	878 [ <i>R</i> <sub>int</sub> = 0.0735, <i>R</i> <sub>sigma</sub> = 0.0469]
Final R indexes [ <i>I</i> > 2σ( <i>I</i> )] <sup>a</sup>	<i>R</i> <sub>1</sub> = 0.0331, <i>wR</i> <sub>2</sub> = 0.0842
Final R indexes [all data]	<i>R</i> <sub>1</sub> = 0.0345, <i>wR</i> <sub>2</sub> = 0.0853

$${}^a R_1 = \frac{\sum ||F_0| - |F_c||}{\sum |F_0|} \text{ for } F_0^2 > 2\sigma(F_0^2). \quad wR_2 = \left\{ \frac{\sum [w(F_0^2 - F_c^2)^2]}{\sum wF_0^4} \right\}^{1/2} \text{ for all data. } w^{-1} = \sigma^2(F_0^2) + (zP)^2, \text{ where}$$

$$P = \frac{(\text{Max}(F_0^2, 0) + 2F_c^2)}{3}.$$

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**Table S3.** Fractional atomic coordinates ( $\times 10^4$ ) and equivalent isotropic displacement parameters ( $\text{\AA}^2 \times 10^3$ ) for  $\text{Na}_3\text{VOS}_3$ .  $U_{\text{eq}}$  is defined as 1/3 of the trace of the orthogonalized  $U_{ij}$  tensor.

Atom	x	y	z	U(eq)
V001	5000	7980.4(6)	3468.5(19)	17.0(3)
S002	5000	6194.9(12)	2516(3)	26.9(4)
S003	6899.1(12)	8770.8(10)	2303.1(17)	29.3(3)
Na04	5000	6187(2)	7692(5)	30.6(6)
Na05	7054(2)	8805.9(18)	7507(4)	39.4(5)
O006	5000	8036(4)	6287(9)	27.3(11)

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**Table S4.** anisotropic displacement parameters ( $\text{\AA}^2 \times 10^3$ ) for  $\text{Na}_3\text{VOS}_3$ . The anisotropic displacement factor exponent takes the form:-  
 $2\pi^2[h^2a^2U_{11}+2hka^*b^*U_{12}+\dots]$ .

Atom	$U_{11}$	$U_{22}$	$U_{33}$	$U_{23}$	$U_{13}$	$U_{12}$
V001	21.4(4)	16.8(4)	12.9(4)	-0.3(4)	0	0
S002	37.9(8)	17.7(6)	25.0(7)	-2.9(5)	0	0
S003	25.3(5)	32.1(6)	30.4(7)	-0.8(4)	4.7(5)	-6.9(4)
Na04	40.4(15)	24.3(13)	27.0(14)	5.1(9)	0	0
Na05	37.2(11)	46.1(12)	34.9(12)	-0.3(9)	-5.7(10)	-7.7(8)
O006	39(3)	29(3)	15(2)	-0.2(15)	0	0

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### Author Contributions

P. G., Z. L. and L. D. conceived the project and designed the research. S. Z. prepared and experimentally characterized Na<sub>3</sub>VOS<sub>3</sub> samples. L. D. performed the density functional theory calculations. B. X., H. C., H. H., F. L., R. W. and P. G. provided assistance during the discussion of the results. L. D., P. G., and Z. L. wrote the manuscript. P. G., and Z. L. supervised the project. All the authors discussed the results and commented on the manuscript at all stages.