Supporting Information for

Thermal conductivity and balanced performance in infrared nonlinear optical multicomponent chalcogenides Li_xAg_{1-x}Ga_vIn_{1-v}Se₂

L.I. Isaenko ^{a,c,*}, Bohui Xu ^{b,d}, K.E. Korzhneva ^a, Pifu Gong ^b, D.A. Samoshkin ^e, A.F. Kurus ^{a,c}, Zheshuai Lin ^{b,d,*}

a. Sobolev Institute of Geology and Mineralogy SB RAS, Novosibirsk 630090 Russia Email: lyudmila.isaenko@mail.ru (L. I.)

b. Functional Crystals Lab, Technical Institute of Physics and Chemistry, Chinese Academy of Sciences, Beijing 100190, China

Email: zslin@mail.ipc.ac.cn

c. Novosibirsk State University, Novosibirsk 630090 Russia

d. Centre of Materials Science and Optoelectronics Engineering, University of Chinese Academy

of Sciences, Beijing 100049, China

e. Kutateladze Institute of Thermophysics SB RAS, Novosibirsk 630090, Russia

*Corresponding author

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1. Experimental method

1.1. Theoretical calculations

All values of the lattice heat capacity k_l at 300 K used for ML modeling with TL-CGCNN algorithm were obtained using data from Ref. [1-4],¹⁻⁴ containing not only CIF structural files, but also certain basic properties of solid solutions both calculated using the PBE functional and obtained experimentally. TL-CGCNN algorithm offers instant and high-performance k_l research with a small expenditure of machine time. Our previous study has determined reliable computational parameters for the TL-CGCNN algorithm in convolutional neural network of ML model, which were used for high-performance screening and pre-experimental design on the NLO chalcogenides with high thermal conductivity.⁵ The TLC GC model was used to predict the thermal conductivity of elementary compounds, assuming symmetrical positions of *A* and *B* in the *ABC*₂ system. For AgGa_yIn_{1-y}Se₂, the *A*-position is fixed as Ag and the ratio of Ga/In units in the *B*-position is controlled. The model of high-performance screening and pre-experimental design were considered in Ref. [6]. Density functional theory implemented in the CASTEP package⁶ is used to calculate the SHG coefficients, energy band gaps and birefringence values in our studied multicomponent chalcogenides, and the computational details are presented in Ref. [7]. ⁷

1.2. Measurement of thermal conductivity

A non-stationary short-term heating method, i.e., the laser flash method, is used to directly measure temperature conductivity k_{τ} , which a physical parameter characterizes the rate of change (equalization) of the temperature of a substance in nonequilibrium thermal processes. Consequently, the thermal conductivity k of the studied material can be determined by taking into account the known values of the specific heat capacity C_p and density ρ , as well as k_{τ} , using the formula $k = k_T \cdot C_p \cdot \rho$.

The laser flash measurements at a given temperature *T* are carried out in a series of two laser "flashes" with the interval of 3 minutes after thermostating the sample on an automated LFA-427 device from NETZSCH (Germany) in an atmosphere of high-purity argon (99.992) in the temperature range 23-300 °C. The interval between the "flashes" was 3 minutes. The description of the measurement method and the experimental setup are represented in Ref.

[8] and [9].^{8, 9} It is important to note that all samples had the same thickness, since thickness is an important parameter for the laser "flash" method.

Based on the laser flash method, we determine the thermal conductivity of LiInSe₂, LiGaSe₂, Li_{0.8}Ag_{0.2}InSe₂, Li_{0.5}Ag_{0.5}GaSe₂, AgGaSe₂, AgInSe₂ crystals. **Table S1** shows the values of thermal conductivity, heat capacity and density of the listed crystals. In Ref. [10],¹⁰ experimental data of measuring the heat capacity of LiInS₂, LiInSe₂, LiGaS₂, LiGaSe₂ and LiGaTe₂ crystals in the temperature range from 180 to 460 K were obtained. The thermal conductivity values for AgGaSe₂ and AgInSe₂ are taken from other sources. ^{1, 2}

Table S1. Heat capacity, density and thermal conductivity for LiGaSe2, LiInSe2, Li0.8Ag0.2InSe2,Li0.5Ag0.5GaSe2, AgGaSe2 and AgInSe2

Crystal	Heat capacity,	Density, g/cm ³	Thermal conductivity,	
	J/8 K		VV/III K	
LiGaSe ₂	0.41 10	4.21 11	4.57 ¹¹	
LiInSe ₂	0.34 10	4.47 ¹²	4.9 ¹³	
Li _{0.5} Ag _{0.5} GaSe ₂	0.34 (exp.)	4.69 (exp.)	1.23 (exp.)	
Li _{0.8} Ag _{0.2} InSe ₂	0.33 (exp.)	4.48 (exp.)	2.1 (exp.)	
AgGaSe ₂	0.29 ¹	5.68 ¹⁴	1.1 ¹⁵	
AgInSe ₂	0.26 ²	5.84 ¹⁶	0.99 17	

Solid	Commenciation	Space	k _L , W*m ⁻¹ *K ⁻¹		4	F
solution	Composition	group	cal.	exp.	a _{ij} , pm ⁺ v -	Eg, ev
Li _x Ag ₁₋	AgGaSe ₂	I-42d	0.99	1.1 ¹⁶	39.5	1.8
	Li _{0.125} Ag _{0.865} GaSe ₂	I-42d	1.09			
	Li _{0.25} Ag _{0.75} GaSe ₂	I-42d	1.37			
	Li _{0.5} Ag _{0.5} GaSe ₂	I-42d	2.14	1.23	26	2.11
	Li _{0.6} Ag _{0.4} GaSe ₂	I-42d	2.57		38	2.08
	Li _{0.8} Ag _{0.2} GaSe ₂	I-42d	3.06		43	2.22
	Li _{0.9} Ag _{0.1} GaSe ₂	I-42d	3.61		31	2.28
	$Li_{0.98}Ag_{0.02}GaSe_2$	Pna2 ₁	3.24			
	LiGaSe₂	Pna2 ₁	4.51	4.57 ¹¹	9.9	3.47
	AgInSe ₂	I-42d	0.96	0.99 17	37	1.22
Li _x Ag _{1-x} InSe ₂	Li _{0.2} Ag _{0.8} InSe ₂	I-42d	1.42		14.47	1.41
	Li _{0.37} Ag _{0.63} InSe ₂	I-42d	1.69		21.8	
	Li _{0.55} Ag _{0.45} InSe ₂	Pna2 ₁	2.21		17.6	1.92
	$Li_{0.78}Ag_{0.22}InSe_2$	Pna2 ₁	3.06		28.8	2.2
	$Li_{0.81}Ag_{0.19}InSe_2$	Pna2 ₁	3.41		26.3	2.27
	LiInSe ₂	Pna2 ₁	4.6	4.9 ¹⁵	11.7	2.86

Table S2. Lattice thermal conductivity (k_L), nonlinearity coefficient (d_{ij}) and band gap (*Eg*) for solid solutions $Li_xAg_{1-x}GaSe_2$ and $Li_xAg_{1-x}InSe_2$

Composition	Cases group	k _L , W*m ⁻¹ *K ⁻¹		
Composition	space group	cal	exp	
AgInSe ₂	I-42d	0.96	0.99 17	
AgGa _{0.2} In _{0.8} Se ₂	I-42d	0.93		
AgGa _{0.4} In _{0.6} Se ₂	I-42d	0.91		
AgGa _{0.5} In _{0.5} Se ₂	I-42d	0.89		
AgGa _{0.665} In _{0.335} Se ₂	I-42d	0.94		
AgGa _{0.82} In _{0.18} Se ₂	I-42d	0.93		
AgGaSe ₂	I-42d	0.99	1.1 ¹⁶	

Table S3. Lattice thermal conductivity (k_L) for solid solution AgGa_yIn_{1-y}Se₂

Table S4. Molecular weight and average Li(Ag)-Se or Ga(In)-Se bond lengths for LixAg1-xGaSe2,LixAg1-xInSe2 and AgGayIn1-ySe2 solid solutions

		Space group	Molocularwoight	Bond lenght
Solid solution	Composition			Li(Ag)-Se or
			a.e.m.	Ga(In)-Se, Å
	AgGaSe ₂	I-42d	335.51	2.6416
	Li _{0.125} Ag _{0.865} GaSe ₂	I-42d	321.88	2.5927
	Li _{0.25} Ag _{0.75} GaSe ₂	I-42d	310.28	2.5885
	Li _{0.5} Ag _{0.5} GaSe ₂	I-42d	285.04	2.5898
Li _x Ag _{1-x} GaSe ₂ ³	Li _{0.6} Ag _{0.4} GaSe ₂	I-42d	274.95	2.5755
	Li _{0.8} Ag _{0.2} GaSe ₂	I-42d	254.77	2.5593
	Li _{0.9} Ag _{0.1} GaSe ₂	I-42d	244.67	2.5387
	Li _{0.98} Ag _{0.02} GaSe ₂	Pna2 ₁	236.6	2.5588
	LiGaSe₂	Pna2 ₁	234.58	2.5509
	AgInSe ₂	I-42d	380.61	2.6354
	Li _{0.2} Ag _{0.8} InSe ₂	I-42d	360.42	2.6154
	Li _{0.37} Ag _{0.63} InSe ₂	I-42d	343.26	2.6039
Li _x Ag _{1-x} InSe ₂ ⁴	Li _{0.55} Ag _{0.45} InSe ₂	Pna2 ₁	325.1	2.5823
	Li _{0.78} Ag _{0.22} InSe ₂	Pna2 ₁	301.88	2.5603
	Li _{0.81} Ag _{0.19} InSe ₂	Pna2 ₁	298.85	2.559
	LilnSe ₂	Pna2 ₁	279.68	2.5498
AgGa _y In _{1-y} Se ₂	AgGaSe ₂ ³	I-42d	335.51	2.6416
	AgGa _{0.82} In _{0.18} Se ₂ ¹⁸	I-42d	343.63	2.6105
	AgGa _{0.665} In _{0.335} Se ₂ ¹⁸	I-42d	350.62	2.6274
	AgGa _{0.5} In _{0.5} Se ₂ ¹⁹	I-42d	358.06	2.59
	AgGa _{0.4} In _{0.6} Se ₂ ²⁰	I-42d	362.57	2.6068
	AgGa _{0.2} In _{0.8} Se ₂ ²⁰	I-42d	371.59	2.6104
	AgInSe ₂ ⁴	I-42d	380.61	2.6354

			-		
Solid solution	Composition	Electronegativity of atoms (χ), eV	Electronegativity difference (Δχ), eV	The nature of the chemical bond	
	AgGaSe ₂		1.3		
	Li _{0.125} Ag _{0.865} GaSe ₂		1.43		
	Li _{0.25} Ag _{0.75} GaSe ₂		1.53	polar covalent	
	Li _{0.5} Ag _{0.5} GaSe ₂	χ(Li)=1, χ(Ag)=1.9,	1.75		
Li _x Ag _{1-x} GaSe ₂	Li _{0.6} Ag _{0.4} GaSe ₂	χ(Ga)=1.6,	1.84		
	Li _{0.8} Ag _{0.2} GaSe ₂	χ(Se)=2.4	2.02		
	Li _{0.9} Ag _{0.1} GaSe ₂		2.11	lonio	
	Li _{0.98} Ag _{0.02} GaSe ₂		2.18	IONIC	
	LiGaSe ₂		2.2		
	AgInSe ₂		1.2		
	Li _{0.2} Ag _{0.8} InSe ₂		1.38	polar covalent	
Li _x Ag _{1-x} InSe ₂	Li _{0.37} Ag _{0.63} InSe ₂		1.53		
	Li _{0.55} Ag _{0.45} InSe ₂	$\chi(LI)=1, \chi(Ag)=1.9,$	1.7		
	Li _{0.78} Ag _{0.22} InSe ₂	$\chi(m)=1.7, \chi(Se)=2.4$	1.9		
	Li _{0.81} Ag _{0.19} InSe ₂		1.93		
	LiInSe ₂		2.1	Ionic	
AgGa _y In _{1-y} Se ₂	AgGaSe ₂ ³		1.3		
	AgGa _{0.82} In _{0.18} Se ₂ ¹⁸		1.28		
	AgGa _{0.665} In _{0.335} Se ₂ ¹⁸	χ(Ag)=1.9,	1.27		
	AgGa _{0.5} In _{0.5} Se ₂ ¹⁹	χ(Ga)=1.6,	1.25	polar covalent	
	AgGa _{0.4} In _{0.6} Se ₂ ²⁰	χ(In)=1.7, χ(Se)=2.4	1.24		
	AgGa _{0.2} In _{0.8} Se ₂ ²⁰		1.22		
	AgInSe ₂ ⁴		1.2		

Table S5. Electronegativity of atoms, electronegativity difference and the nature of chemicalbonding for solid solutions $Li_xAg_{1-x}GaSe_2$, $Li_xAg_{1-x}InSe_2$ and $AgGa_yIn_{1-y}Se_2$



Figure S1. Structure changes from tetragonal (I-42d) to rhombic (Pna21) and the trend of the main features.



Figure S2. Decrease of molecular weight for solid solutions (a) Li_xAg_{1-x}InSe₂, (b) Li_xAg_{1-x}GaSe₂, (c) AgGa_yIn_{1-y}Se₂.



Figure S3. Phonon desperation curve and phonon DOS of (a) $LiGaSe_2$, (b) $LiAgGa_2Se_4$, (c) AgGaSe₂.

References

- H. J. Hou, F. J. Kong, J. W. Yang, L. H. Xie and S. X. Yang, First-principles study of the structural, optical and thermal properties of AgGaSe2, *Phys. Scr.*, 2014, 89, 065703.
- H. Neumann, J. Łażewski, P. T. Jochym and K. Parlinski, Ab initio heat capacity and atomic temperature factors of chalcopyrites, *Phys. Rev. B*, 2007, 75, 224301.
- L. Isaenko, L. Dong, A. Kurus, Z. Lin, A. Yelisseyev, S. Lobanov, M. Molokeev, K. Korzhneva and A. Goloshumova, Li_xAg₁₋ xGaSe₂: Interplay Between Lithium and Silver in Mid-Infrared Nonlinear Optical Chalcogenides, *Adv. Opt. Mater.*, 2022, 10, 2201727.
- 4. L. Isaenko, L. Dong, K. Korzhneva, A. Yelisseyev, S. Lobanov, S. Gromilov, M. S. Molokeev, A. Kurus and Z. Lin, Evolution of Structures and Optical Properties in a Series of Infrared Nonlinear Optical Crystals $Li_xAg_{1-x}InSe_2$ ($0 \le x \le 1$), *Inorg. Chem.*, 2023, **62**, 15936-15942.
- Q. Wu, L. Kang and Z. Lin, A Machine Learning Study on High Thermal Conductivity Assisted to Discover Chalcogenides with Balanced Infrared Nonlinear Optical Performance, *Adv. Mater.*, 2024, 36, 2309675.
- S. J. Clark, M. D. Segall, C. J. Pickard, P. J. Hasnip, M. I. J. Probert, K. Refson and M. C. Payne, First principles methods using CASTEP, 2005, 220, 567-570.
- Z. Lin, X. Jiang, L. Kang, P. Gong, S. Luo and M.-H. Lee, First-principles materials applications and design of nonlinear optical crystals, *J. Phys. D: Appl. Phys.*, 2014, 47, 253001.
- W. J. Parker, R. J. Jenkins, C. P. Butler and G. L. Abbott, Flash Method of Determining Thermal Diffusivity, Heat Capacity, and Thermal Conductivity, J. Appl. Phys., 1961, 32, 1679-1684.
- 9. I. V. Savchenko and S. V. Stankus, Thermal conductivity and thermal diffusivity of tantalum in the temperature range from 293 to 1800 K, *Thermophys. and Aeromech.*, 2008, **15**, 679-682.
- 10. V. A. Drebushchak, L. I. Isaenko, S. I. Lobanov, P. G. Krinitsin and S. A. Grazhdannikov, Experimental heat capacity of LiInS₂, LiInS₂, LiGaS₂, LiGaS₂, and LiGaTe₂ from 180 to 460 K, *J. Therm. Anal. Calorim.*, 2017, **129**, 103-108.
- M. S. Yaseen, G. Murtaza and R. M. Arif Khalil, Ab-initio study of Li based chalcopyrite compounds LiGaX₂ (X= S, Se, Te) in tetragonal symmetry: A class of future materials for optoelectronic applications, *Current Applied Physics*, 2018, 18, 1113-1121.
- A. Kargar, H. Hong, J. Tower, A. Gueorguiev, H. Kim, L. Cirignano, J. F. Christian, M. R. Squillante and K. Shah, LilnSe₂ for Semiconductor Neutron Detectors, 2020, 8.
- A. P. Yelisseyev, A. S. Titov, K. M. Lyapunov, V. A. Drebushchak, L. I. Isaenko and S. I. Lobanov, Thermal and thermooptic parameters of LiInSe₂ single crystals, *J. Cryst. Growth*, 2005, **275**, e1679-e1684.
- 14. R. L. Aggarwal and T. Y. Fan, Thermal diffusivity, specific heat, thermal conductivity, coefficient of thermal expansion, and refractive-index change with temperature in AgGaSe₂, *Appl. Opt.*, 2005, **44**, 2673-2677.
- 15. S. Sharma, A. S. Verma and V. K. Jindal, Ab initio studies of structural, electronic, optical, elastic and thermal properties of silver gallium dichalcogenides (AgGaX₂: X=S, Se, Te), *Mater. Res. Bull.*, 2014, **53**, 218-233.
- H. Hahn, G. Frank, W. Klingler, A.-D. Meyer and G. Störger, Untersuchungen über ternäre Chalkogenide. V. Über einige ternäre Chalkogenide mit Chalkopyritstruktur, 1953, 271, 153-170.
- P. Qiu, Y. Qin, Q. Zhang, R. Li, J. Yang, Q. Song, Y. Tang, S. Bai, X. Shi and L. Chen, Intrinsically High Thermoelectric Performance in AgInSe₂ n-Type Diamond-Like Compounds, 2018, 5, 1700727.
- V. V. Badikov, G. M. Kuz'micheva, V. L. Panyutin, V. B. Rybakov, V. I. Chizhikov, G. S. Shevyrdyaeva and S. I. Shcherbakov, Preparation and Structure of AgGa_{1-x}In_xSe₂ Single Crystals, *Inorg. Mater.*, 2003, **39**, 1028-1034.
- J. E. Avon, K. Yoodee and J. C. Woolley, Solid solution, lattice parameter values, and effects of electronegativity in the (Cu_{1-x}Ag_x)(Ga_{1-v}In_v)(Se_{1-z}Te_z)₂ alloys, *J. Appl. Phys.*, 1984, **55**, 524-535.
- S.-R. Hahn and W.-T. Kim, Anomalous composition and temperature dependence of the energy gap of AgGa_{1-x}In_xSe₂ mixed crystals, *Phys. Rev. B*, 1983, **27**, 5129-5131.