## **Supplementary information**

## Sodium chloride-assisted CVD enables controlled synthesis of large single-layered MoS<sub>2</sub>

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Fig S1. The morphology of MoS<sub>2</sub> growth on sapphire substrate influenced by (a-d) the gas flow rates, (e-h) the distance between MoO<sub>3</sub> with the substrate and (i-j) the substrate

placement modes.

As seen in Fig.S1 (a-d), with the rise of gas flow rates from 50-200 sccm (Standard cubic centimeter per minute), the size and quality of the MoS<sub>2</sub> nanosheet changes remarkably. Combined with our data results, when the velocity of N<sub>2</sub> is too low ( $\leq$  100 sccm), the N<sub>2</sub> flow in the tubular furnace chamber is not enough to quickly bring the S steam from the low temperature region to the high temperature zone, resulting in the high concentration of MoO<sub>x</sub> near the substrate and finally deposited on the surface of the substrate[2]; Similarly, when the N<sub>2</sub> flow rate is too high ( $\geq$  200 sccm), the S vapor does not have time to react with MoO<sub>3</sub>, the vulcanization process is not complete and the intermediate products can be obtained. Only when the velocity of carrier gas is about 200 sccm, a good reaction can take place between the precursors and high quality MoS<sub>2</sub> can be formed and deposited on the substrate surface.

The chemical reaction equation for the conversion from  $MoO_3$  to  $MoS_2$  in  $N_2$  atmosphere is as follows :

$$2 \operatorname{MoO}_3 + 7 \operatorname{S} \rightarrow 2 \operatorname{MoS}_2 + 3 \operatorname{SO}_2 \tag{1}$$

$$MoO_3 + x/2 S \rightarrow MoO_{3-x} + x/2 SO_2$$
(1)

$$MoO_{3-x} + (7-x)/2 S \rightarrow MoS_2 + (3-x)/2 SO_2$$
 (2)

Fig. S1 (e-h) shows that the optical microscopy images of the MoS<sub>2</sub> growth state on the substrate surface when the MoO<sub>3</sub> is placed at different distances with the substrate, which is important influence on the growth of MoS<sub>2</sub>. When the distance is small (Fig S1(e, f)), the concentration of MoO<sub>3</sub> vapor is extremely high, and the kinds of materials deposited on the substrate surface are complex, and it is easier to grow multi-layer MoS<sub>2</sub>. Moreover, when the distance is too far (Fig.S1(h)), it will cause uneven MoO<sub>3</sub> vapor concentration, and it is difficult to control the growth of MoS<sub>2</sub>. In our experiments, we found that when the distance is 20 mm, it is easier to grow large area monolayer MoS<sub>2</sub>.

In our experiments, it is found that the optimum gas flow rates and placement distance are around 200 sccm and 20 mm, respectively. Certainly, it is worth noting that practical considerations will also cause the experimental results to differ. The fundamental reason is that the optimized gas flow rates and placement distance will increase the time of raw material reaction and enhance the size and quality of the MoS<sub>2</sub> nanosheet remarkably.



Fig S2. (a-f) The same substrate surface has different growth states at different positions.



Fig. S3. (a-f) The morphology of MoS2 grown on SiO2/Si shows many kinds, such as

dendritic, triangular, and butterfly shapes



Fig. S4. In the same environment, the thickness of MoS<sub>2</sub> nanosheets grown on four different substrate surfaces was measured. The mica surface showed a distinct step-like morphology, and height measurements confirmed the presence of three layers in the crystal domain. MoS<sub>2</sub> domains observed on SiO<sub>2</sub>/Si substrates appeared relatively smooth, while those on sapphire and Si substrates were observed to be monolayers with a thickness of approximately 0.7nm. The MoS<sub>2</sub> grown on Si exhibited more pronounced sawtooth patterns, while the boundaries of the MoS<sub>2</sub> grown on sapphire showed small particle aggregation, suggesting the presence of unreacted MoO<sub>3</sub>. <sub>X</sub> or MoO<sub>x</sub>S<sub>2-y</sub> nanoparticles.



Fig. S5. Pre-treatment of the sapphire substrate with a NaCl solution at a concentration of 25 mg/mL resulted in the morphology of the grown MoS<sub>2</sub> nanosheets shown in (a-c). Clearly demonstrating the vapor-liquid-solid (VLS) reaction [3] model on the substrate surface, with monodomain sizes of approximately 1 mm. (d) Shows a comparison of the substrate before and after the growth of MoS<sub>2</sub> on a sapphire substrate, where MoS<sub>2</sub> appears pale yellow to the naked eye, indicating growth throughout the entire substrate surface.



Fig. S6. Pre-treatment of the Si substrate with the same concentration (25 mg/mL) of NaCl solution for 5 cycles achieved a high concentration pre-treatment target, followed by CVD growth. The growth conditions at different positions on the substrate surface are shown in (a-c). A high concentration pre-treatment implies the introduction of more ion suspension bonds on the substrate surface, resulting in decreased controllability of the growth state and an increased probability of contamination of the grown MoS<sub>2</sub> nanosheets by impurities.



Fig. S7. (a) SEM image of MoS<sub>2</sub> grown on a sapphire substrate surface after NaCl solution pre-treatment, and (b) EDS elemental distribution map at the red box position in (a).

Chemical element	Wt%	At%			
С	4.32	9.61			
0	21.63	36.12			
Na	8.10	9.41			
Al	19.79	19.60			
S	21.30	17.76			
K	1.41	0.96			
Мо	23.45	6.53			
Total	100.00	100.00			



Fig S8. (a) High-resolution TEM image of a single MoS<sub>2</sub> nanosheet. The red circles in the outline the six-fold-symmetric diffraction spots. The labelled plane in a is indexed to be the (002) plane, with 0.311 nm lattice spacing. (b) EDS image at the cross-section position, with the positions of elements such as O, S, and Mo labeled. (c) XRD characterization of the same

batch of samples.

(a)	Thin Film Standardless Standardless Quantitative Analysis									
	Element C K O K Al K S K	(keV) 0.277 0.525 1.486 2.307	Mass% 6.59 1.82 1.46 0.58	Counts 10087.46 6569.42 9024.75 3291.63	Sigma 0.07 0.04 0.04 0.05	Atom% 45.23 9.37 4.45 1.50	Compound	Mass%	Cation	K 1.1341 0.4807 0.2803 0.3087
	Ga K Mo K	9.241	3.14	1956.50	0.05	2.70				2.7844
752	W M Au M (Ref.) Total	1.774 2.120	1.04 84.70 100.00	4 2102.08 0 147115.55 0	0.08 0.44	0.46 35.47 100.00				0.8556 1.0000
( <b>b</b> )	Thin Film St	andardle	ss Star	ndardless Qu	antitat	ive Anal	ysis			
(0)	Fitting Coef	ficient	: 0.5449	9						
the second se	Element	(keV)	Mass	& Counts	Sigma	Atom%	Compound	Mass%	Cation	K
	C K	0.277	6.00	0 8907.20	0.07	41.96				1.1341
	O K	0.525	1.89	9 6629.22	0.04	9.94				0.4807
	AI K	1.486	1.43	3 8588.80	0.04	4.45				0.2803
and the second	SK	2.307	1.00	U 54/4./2	0.05	2.63				0.3087
<u> </u>	Ga K	17 441	4 92	2 2076 20	0.05	4 31				2 7844
753	W M	1.774	0.92	8 1919.48	0.08	0.45				0.8556
	Au M (Ref.)	2,120	83.08	8 139860.41	0.44	35.43				1.0000
	Total		100.00	0		100.00				
$(\alpha)$	Thin Film Stan	dardless	Stand	ardless Qua	ntitativ	e Analys	sis			
(C)	Fitting Coeffi	cient :	0.5407							
``	Element	(keV)	Mass*	Counts	Sigma	Atom%	Compound	Mass∜	Cation	1 1 2 4 1
per l'anne de la facture de la service de	C K	0.277	1 51	10000.37	0.07	9 02				1.1341
	A) K	1 486	1 09	7189 61	0.04	3 42				0.2803
	SK	2.307	0.52	3089.02	0.04	1.36				0.3087
	Ga K	9,241	0.74	2016.80	0.05	0.90				0.6780
1. A	Mo K	17.441	3.18	2110.86	0.13	2.81				2.7844
	WM	1.774	1.10	2371.59	0.08	0.51				0.8556
/55	Au M (Ref.)	2.120	85.32	157826.08	0.43	36.74				1.0000
	Total		100.00			100.00				

Fig S9. (a-c) EDS data summary and comparison of Mo and S element contents marked at three different positions.

Due to the hardness of sapphire, the sample was sent for FIB sectioning before TEM imaging. To clearly locate the target sample during the sectioning process, gold deposition was performed on the sample surface in advance, with a thickness of approximately 6-10 nm. This also posed some difficulties in observing the MoS<sub>2</sub> sample. Therefore, in the observed crosssection, only a thin layer (~1 nm) close to the sapphire substrate is the MoS<sub>2</sub> sample.

## **REFERENCES:**

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