

## Supplementary Information

### **Preparation and structural adjustment of vanadium oxide nanotube as cathode material for PIBs with improved performance**

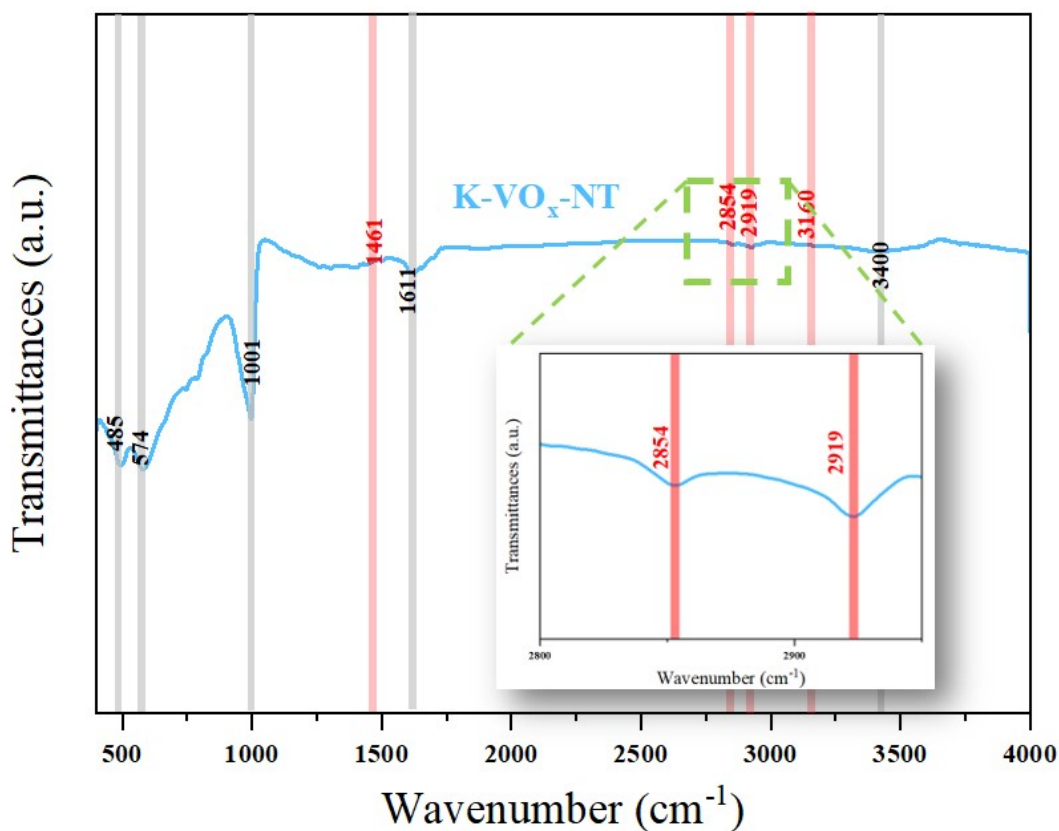
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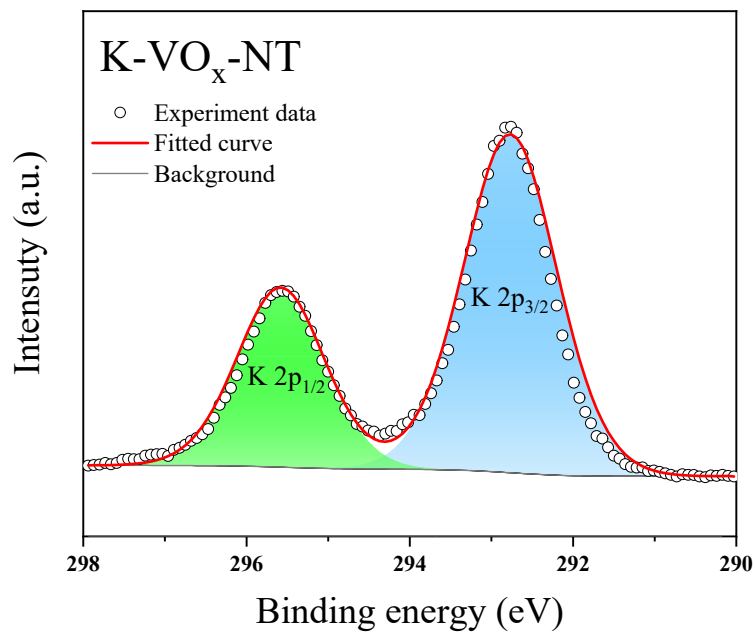
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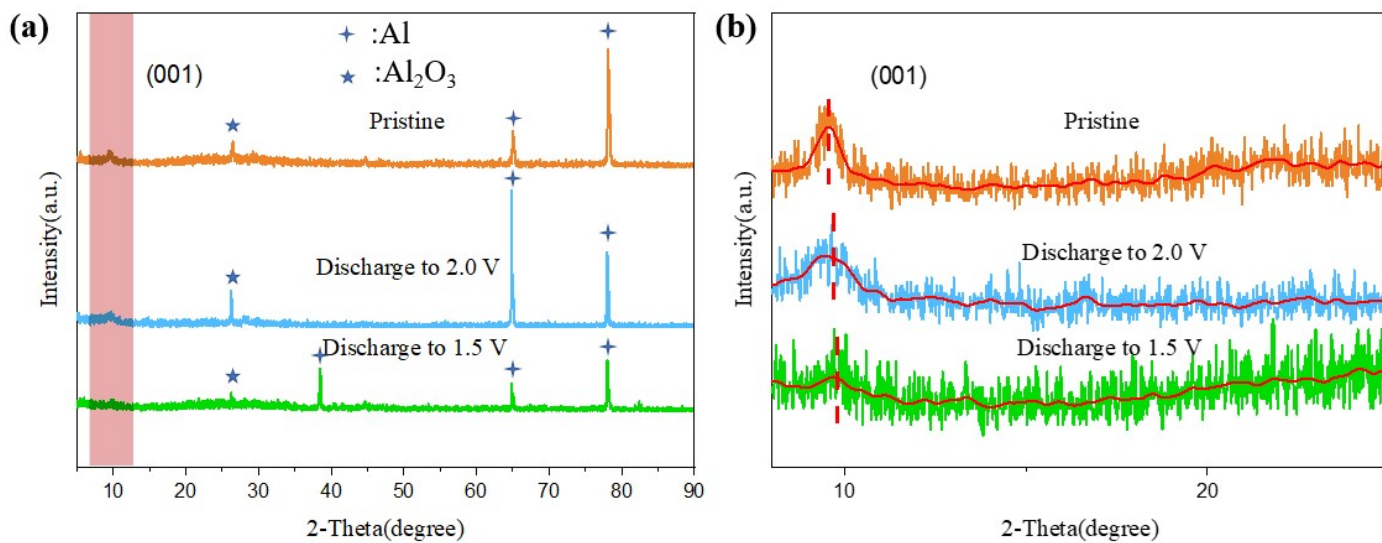


**Figure S1.** FTIR spectrum of K-VO<sub>x</sub>-NT with an enlarged inset highlighting specific vibrational modes.

Fig. S1 shows FTIR spectrum of K-VO<sub>x</sub>-NT sample. The inset provides a magnified view of two typical absorption peaks located at 2854 and 2919 cm<sup>-1</sup>, which correspond to different stretching and bending vibration modes of C-H bonds in dodecylamine. It is evident that although the K<sup>+</sup> exchange process can effectively remove the organic template (C<sub>12</sub>H<sub>25</sub>NH<sub>3</sub><sup>+</sup>) between VO<sub>x</sub> layers, a small amount of organic template is still preserved. This residual organic template is one of the reasons for clear detection of carbon signals in the subsequent EDX mappings of K-VO<sub>x</sub>-NT.

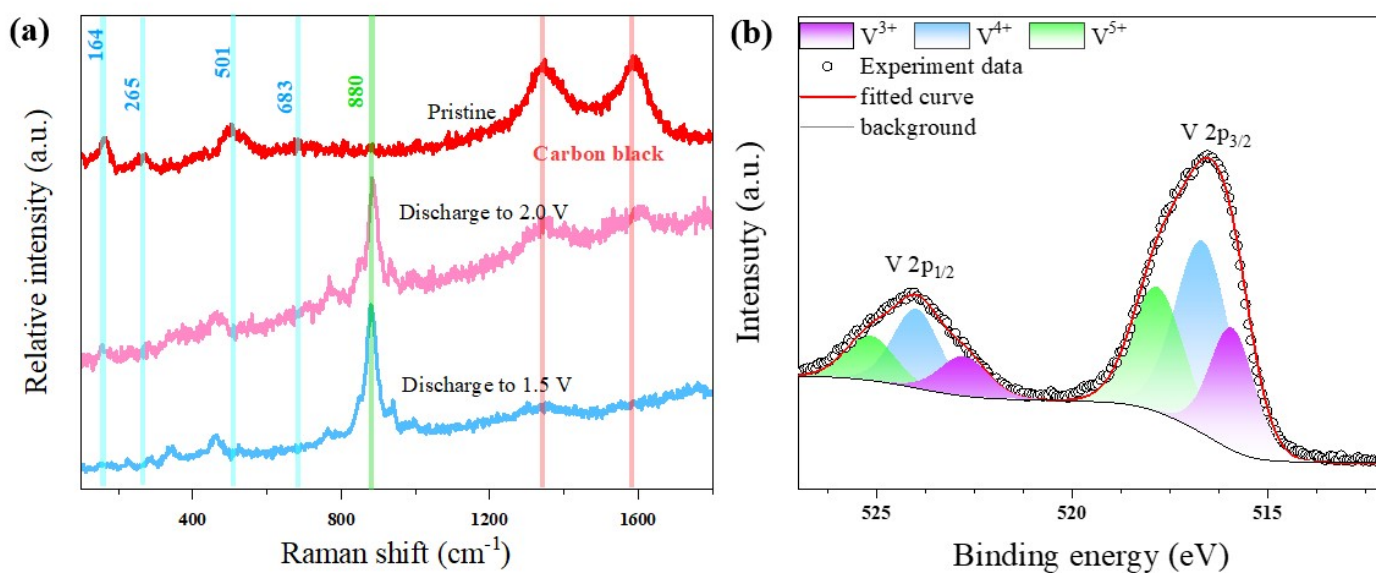


**Figure S2.** XPS fine survey spectrum of  $K_{2p}$  for  $K-VO_x-NT$ .



**Figure S3.** Ex-situ XRD patterns (a) of  $K-VO_x-NT$  at pristine, discharge to 2.0 V and 1.5 V states; Enlarged view (b) of (001) peaks in corresponding ex-situ XRD patterns.

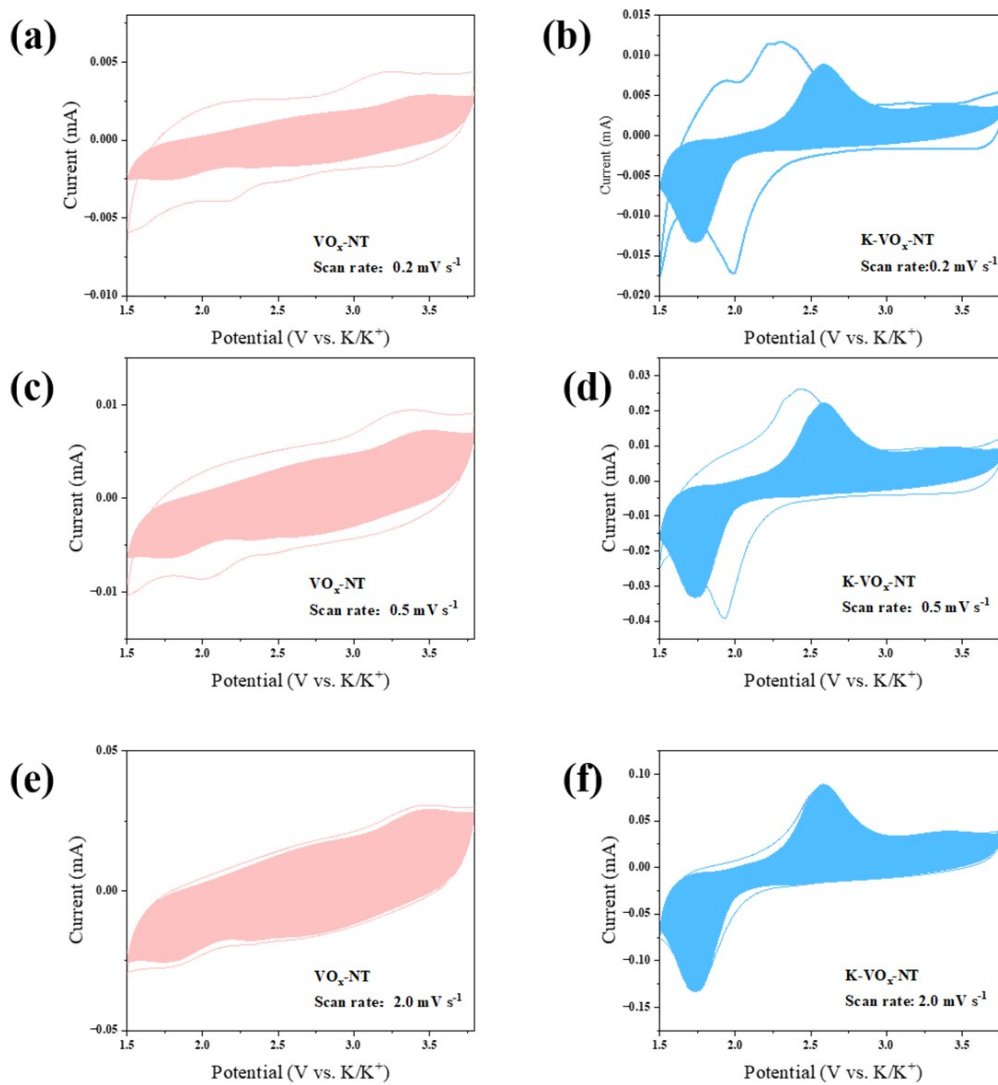
To further investigate  $K^+$  storage mechanism of  $K\text{-VO}_x\text{-NT}$  during the discharge process, ex-situ XRD tests at pristine, discharge to 2.0 V and 1.5 V states were performed as illustrated in Fig. S2 (a). The (001) peak for  $K\text{-VO}_x\text{-NT}$  exhibits a slight shift during the discharge process, moving from  $9.6^\circ$  in the pristine state to  $9.9^\circ$  at a discharge state of 1.5 V. The enlarged image for this shift was displayed in Fig. S2 (b). This shift corresponds to a minor contraction of interlayer spacing, which can be attributed to the enhanced interaction between intercalated  $K^+$  and negatively charged  $\text{VO}_x$  layers as well as the reduced electrostatic repulsion between  $\text{O}^{2-}$  in adjacent layers [1]. However, owing to the very small shift of only  $0.3^\circ$ , the change in (001) interlayer spacing is hard to be detected by TEM.



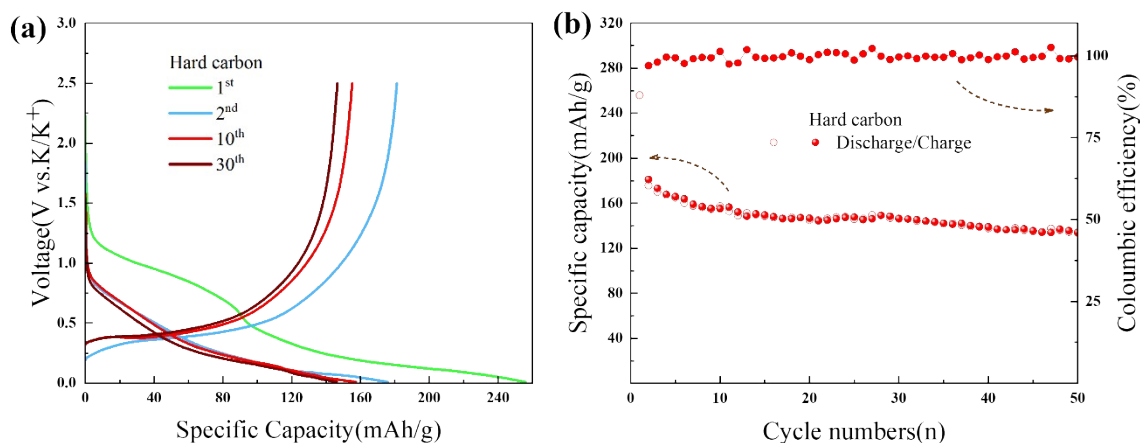
**Figure S4.** Ex-situ Raman spectra (a) of  $K\text{-VO}_x\text{-NT}$  at pristine, discharge to 2.0 V and 1.5 V states; (b) XPS fine survey spectra of  $\text{V}_{2p}$  for  $K\text{-VO}_x\text{-NT}$  under the fully

discharged (1.5 V) state.

Raman spectra were collected at pristine, discharge to 2.0 V and 1.5 V states for K-VO<sub>x</sub>-NT cathode in Fig. S3 (a). A distinct peak appears at 880 cm<sup>-1</sup> after discharging to 2.0 and 1.5 V, which can be attributed to the stretching vibration of V<sup>4+</sup>=O bond [2]. It indicates that V<sup>5+</sup> in pristine K-VO<sub>x</sub>-NT would gradually transform into V<sup>4+</sup>. Fig. S3(b) shows V2p fine spectrum of K-VO<sub>x</sub>-NT in the fully discharged state. In comparison to V2p fine spectrum in the fully charged state as shown in Fig. 5 (d), the appearance of V<sup>3+</sup> peak is observed, accounting for 24.1% of total V elements based on the peak area. Additionally, the V<sup>4+</sup>/V<sup>5+</sup> ratio increases significantly (from 66.2 % in the fully charged state to 159.0 % in the fully discharged state). This indicates that a portion of V<sup>4+</sup> and V<sup>5+</sup> is reduced to V<sup>3+</sup> and V<sup>4+</sup> upon K<sup>+</sup> intercalation, respectively. This is consistent with the ex-situ Raman results.

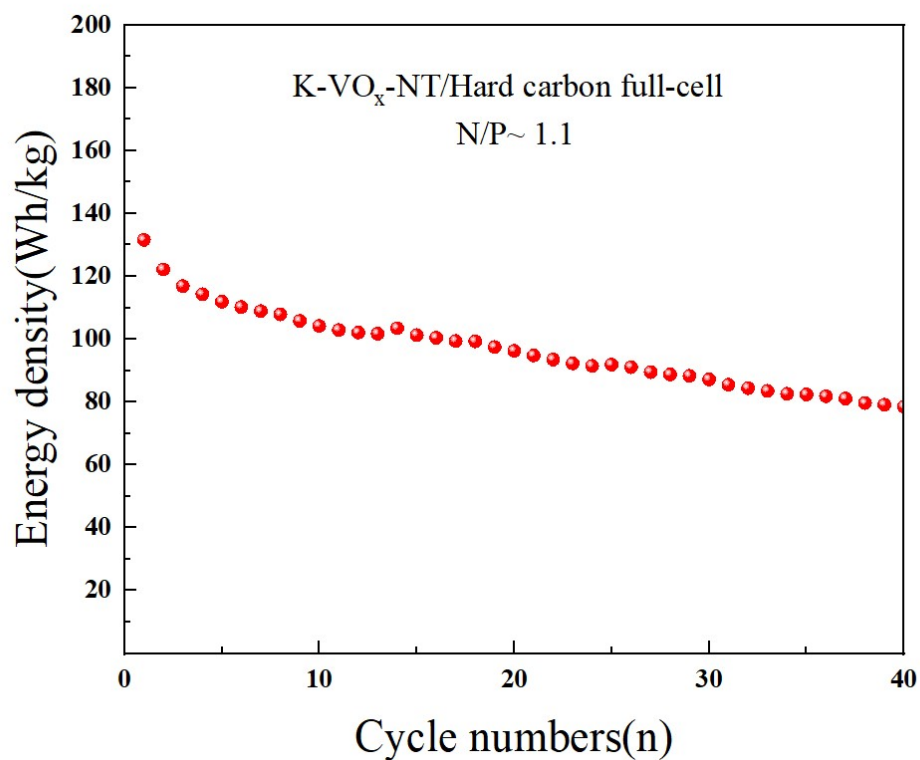


**Figure S5.** Pseudocapacitive contributions (shadow areas) from surface-induced process under representative CV test at 0.2, 0.5 and 2.0 mV s<sup>-1</sup> for VO<sub>x</sub>-NT (a, c, e) and K-VO<sub>x</sub>-NT (b, d, f).



**Figure S6.** The 1<sup>st</sup>, 2<sup>nd</sup>, 10<sup>th</sup> and 30<sup>th</sup> GCD profiles (a) and cycling performance (b) of standard hard carbon in half-cell configuration under 50 mA g<sup>-1</sup> between 0.01-2.5 V.

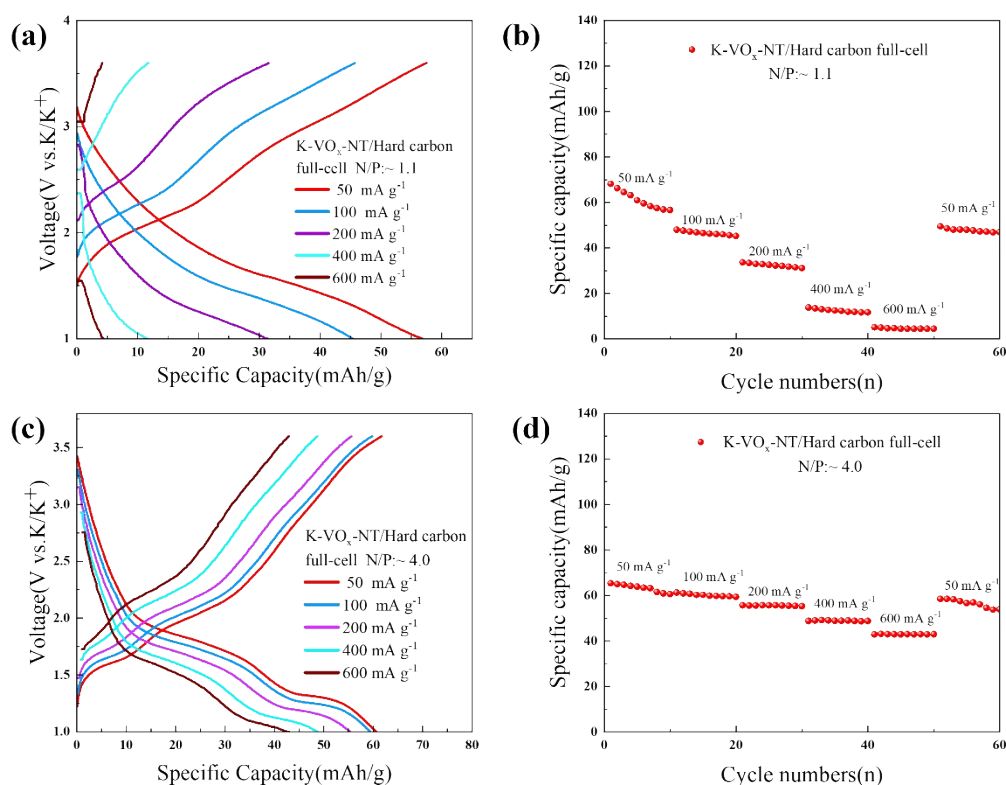
Fig. S6 (a) and (b) show the 1<sup>st</sup>, 2<sup>nd</sup>, 10<sup>th</sup> and 30<sup>th</sup> galvanostatic charge-discharge profiles and cycling performance of standard hard carbon in half-cell configuration. At a current density of 50 mA g<sup>-1</sup> within the voltage range of 0.01–2.5 V, this hard carbon delivered an initial discharge specific capacity of 256.0 mAh g<sup>-1</sup>, which decreased to 157.4, 146.2 and 133.7 mAh g<sup>-1</sup> at the 10<sup>th</sup>, 30<sup>th</sup> and 50<sup>th</sup> cycles, respectively.



**Figure S7.** The evolution of energy density for K-VO<sub>x</sub>-NT/Hard carbon full-cell in cycling process.

The energy density value is calculated based on the mass of the K-VO<sub>x</sub>-NT cathode, without considering the mass of electrolyte and standard anode. The energy densities at the 1<sup>st</sup>, 20<sup>th</sup>, and 40<sup>th</sup> cycles are 136.0, 96.3, and 78.5 Wh kg<sup>-1</sup>, respectively.





**Figure S8.** Typical GCD profiles (a) at 10<sup>th</sup> cycle for each rate tests and corresponding rate capability (b) of K-VO<sub>x</sub>-NT/Hard carbon full-cell (N/P ratio: ~ 1.1) under different current densities; Typical GCD profiles (c) at 10<sup>th</sup> cycle for each rate tests and corresponding rate capability (d) of K-VO<sub>x</sub>-NT/Hard carbon full-cell (N/P ratio: ~ 4) under different current densities.

The rate capabilities under different N/P ratios (~ 1.1 and 4.0) for K-VO<sub>x</sub>-NT in full-cell configuration were evaluated. At the condition of ~1.1 for N/P ratio, as depicted in Fig. S8(a) and (b), the full-cell exhibits the discharge capacities of 56.7, 45.3, 31.2, 11.7, 4.5, and 47.0 mAh g<sup>-1</sup> under continuously changing current densities of 50, 100, 200, 400, 600, and 50 mA g<sup>-1</sup> for the 10<sup>th</sup> cycle at each rate test. In comparison, with regard to the full-cell with an N/P ratio of ~4, as shown in Fig. S8(c) and (d), the

discharge capacities are 60.6, 59.4, 55.3, 48.8, 43.0, and 53.9 mAh g<sup>-1</sup> under the same current densities for the 10<sup>th</sup> cycle at each rate test, respectively.

This could be due to the fact that, under the condition of ~4 for N/P ratio, the hard carbon anode is very excessive, and only a small portion of hard carbon can fully participate in the storage of K<sup>+</sup> during cycling. As a result, the hard carbon anode predominantly stores K<sup>+</sup> via capacitive behavior, which does not obviously contribute to the polarization effect of full-cell. It allows the target K-VO<sub>x</sub>-NT cathode to demonstrate its optimal rate capability. In contrast, under the condition of ~1.1 for N/P ratio, almost all hard carbon participate in the storage of K<sup>+</sup>, which would fully utilize its both capacitance and diffusion-controlled contributions for K<sup>+</sup> storage [3, 4]. In this case, the hard carbon anode will lead to obvious polarization effect of full-cell configuration, thereby affecting the rate capability of target K-VO<sub>x</sub>-NT cathode.

### References:

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