Robust Nanoporous NiMn oxide Electrocatalysts for Oxygen Evolution Reaction through Defect Engineering

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Figure S1: Scanning electron microscope (SEM) images of (a) NM_{DL} , and (b) NM_{QVD} .

Figure S2: BET analysis for NM_{OVD} and NM_{DL} samples.

Figure S3: Electron energy loss spectroscopy (EELS) analysis- counts vs binding energy (a) for pure Nickel; (b) for $NiMnO₃$ Phase

Figure S4: (a) XPS survey scan for as-synthesized pristine dealloyed (NM_{DL}) and deformed dealloyed (NM_{OVD}) specimen, (b) Comparison of oxygen vacancy peak for NM_{OVD} and NM_{DL} specimen

Figure S5: Linear sweep voltammetry curves for NM_{OVD} specimen, dealloyed for different time intervals.

Figure S6: Cyclic voltammetry curves in the non-faradaic region for (a) NM_{OVD} , (b) NM_{DL} , (c) double layer capacitance (*Cdl*) from cyclic voltammetry response, (d) electrochemical active surface area (ECSA) normalized LSV curves for NM_{OVD} and NM_{DL} .

Figure S7: (a) XPS survey scan for NM_{OVD} after chronopotentiometry at 500 mA/cm² for 24 hours; (b)-(d) high resolution XPS spectra of Ni2p, Mn2p and O1s for NM_{OVD} specimen after 24 hour chronopotentiometry test at 500 mA/cm². Inset of (d) shows the comparison of O1s peak before and after OER testing.

Computational details: Our calculations are based on first-principles density functional theory (DFT) and spin-polarized plane-wave pseudopotential method as implemented in Vienna Ab Initio Simulation Package (VASP)^{1, 2}. The ionic core−valence electron interactions are represented with Projector augmented wave (PAW) potentials³ and electronic exchange−correlation energy is approximated with a generalized gradient approximation (GGA) as parameterized by Perdew−Burke−Ernzerhof⁴ . As transition-metal oxides are strongly-correlated systems, the short-range interactions between electrons and electronlocalization effects are accounted for by adding Hubbard *U* terms. We use rotationally invariant approach to DFT+U, introduced by Dudarev et al.⁵ with $U = 1.0$ and 6.0 eV for Mn and Ni in NiMnO₃ and U = 2.5 and 3.5 eV for Mn²⁺ and Mn³⁺ species in Mn₃O₄. The parameters are determined self-consistently to match the electronic band gap and magnetic moments of elements as determined experimentally in the earlier literature. Kohn−Sham wave functions are expanded in a plane-wave basis set truncated with an energy cut-off of 450 eV. Integrations over Brillouin Zone (BZ) are sampled with a uniform $(4 4 4)$ and $(2 2 1)$ k-grid for Mn₃O₄ and $NiMnO₃$ bulk structures respectively, and the atomic structures are relaxed using Broyden, Fetcher, Goldfarb, Shanno (BFGS)-based algorithm until the Hellman−Feynman forces on each atom are lower than 0.01 eV/Angstrom. The transition states (TSs) are obtained using the climbing-image nudged elastic band (NEB) method⁶. Four intermediate images are used in all the NEB calculations. All intermediate images are relaxed until the Hellman−Feynman forces on each atom are within 0.03 eV/Angstrom.

Nature of OER on NiMnO³ in high-coverage regime

To further determine how the LOM mechanism proceeds in the presence of surrounding OH ions in the medium (high-coverage regime), we added OH group in the neighborhood of O*L*-O bond (in Figure 6(d)). On energy minimization, the optimized structure shows the formation of *OOH, *OH, H₂O species, along with the O_2 formation (Figure 6(i,j)). Evidently, the chemical interactions of the OH ions and the surface can result in cleaved Ni-O bonds(Figure 6(i)) or dipolar interactions allowing electronic charge transfer between adsorbed species over the $NiMnO₃$ surface (Figure 6(j)). The latter stems primarily from the high electronegativities of O_L of Mn- O_L -Ni and Mn-O^L sites, which bond strongly with H ion of the OH groups, hence releasing O-O molecules⁷. The broken Ni-O bond re-forms itself in this process (Figure 6(j)), highlighting the cyclic and self-propagating nature of the OER process on $NiMnO₃$ surface. This reinforces the prevalence of multi-site OER mechanisms, like LOM, in $NiMnO₃$ over the AEM mechanism prevalent in most transition metal-oxides.

Figure S8: Electronic structure of bulk NiMnO3, (a) showing a semi-metallic nature with a band gap of ~2.0 eV, with prevalence of Mn-3d and O-2p states at the valence band maximum crossing Fermi level, (b) higher density of occupied Mn-e_g orbitals than Mn-t_{2g}, suggesting crucial role of Mn in catalytic performance of $NiMNO₃$

Figure S9: Electronic structures of distinct Ni atoms on the (001) surface; (a) Ni atoms with dangling Ni-O bonds (dark blue), and (b) Ni with Ni-Mn-Ni coordinated O atoms (cyan). The former shows an absence of band gap, with high density of O-2p states, signifying electronic charge transfer from O- to Ni. This Ni-O bond cleaves readily and lead to direct evolution of $O₂$ through LOM mechanism.

Figure S10. Transition state for the reaction intermediate $*OH \rightarrow *O$ on pristine NiMnO₃ **surface:** (a) starting configuration, $DE = 0$ eV, (b) transition state, showing formation of H bonds between two OH groups, $DE = 0.37$ eV, and (c) release of water molecule and resulting deprotonation of OH group, $DE = -1.1$ eV.

Figure S11: OER activity as a function of pH

Figure S12: Electronic structure of two different vacancy configurations; (a) very low formation-energy configuration involving cleavage of Ni-O bond, and (b) structure involving structural re-construction and formation of Ni-O and Mn-O distorted polyhedra. The e-DoS of latter shows a sharp increase in Mn-3d states in conduction bands signifying enhanced oxidation kinetics of adsorbates.

Figure S13. Transition state for the reaction intermediate $*OH \rightarrow *O$ on $NIMnO_3$ surface **containing O vacancy:** (a) starting configuration, $DE = 0$ eV, (b) transition state, showing long-ranged interaction of OH group with adsorbed OH, resulting in the elongation of Ni-O bond from 1.92 Å to 2.06 Å, signifying cleavage of Ni-O bond and activation of LOM mechanism, $DE = 1.32$ eV, (c) formation of O_L -O group and $H₂O$ molecule, showing formation of hydrogen bonds, $DE = -1.36$ eV, and (d) release of water molecule, $DE = -2.2$ eV.

Figure S14: Electronic density of Mn-3d, Ni-3d and O-2p states at the Ni/NiMnO3 junction, showing increase in the density of Mn-3d states in conduction bands. The inset shows the e-Dos of under-coordinated Mn at the interface, showing a sharp increase in density of states at the Fermi-level and in conduction band, strongly indicating its role in OER activity of Ni/NiMnO3 interfaces, (b) direct O_2 evolution from OH group adsorbed in the neighbourhood of Mn-O-O* site, due to H adsorption on electronegative O atoms and resulting cleavage of O-H groups.

Table S1: Comparison of electro-catalytic OER performance of the NM_{OVD} and other selfsupported electrocatalysts in 1 M KOH electrolyte. The η_x corresponds to the overpotential at the current density of $x \text{ mA cm}^{-2}$.

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