

Supporting information for:

## **Understanding the evolution of ternary alloyed nanoparticles during reversible exsolution from double perovskite oxides**

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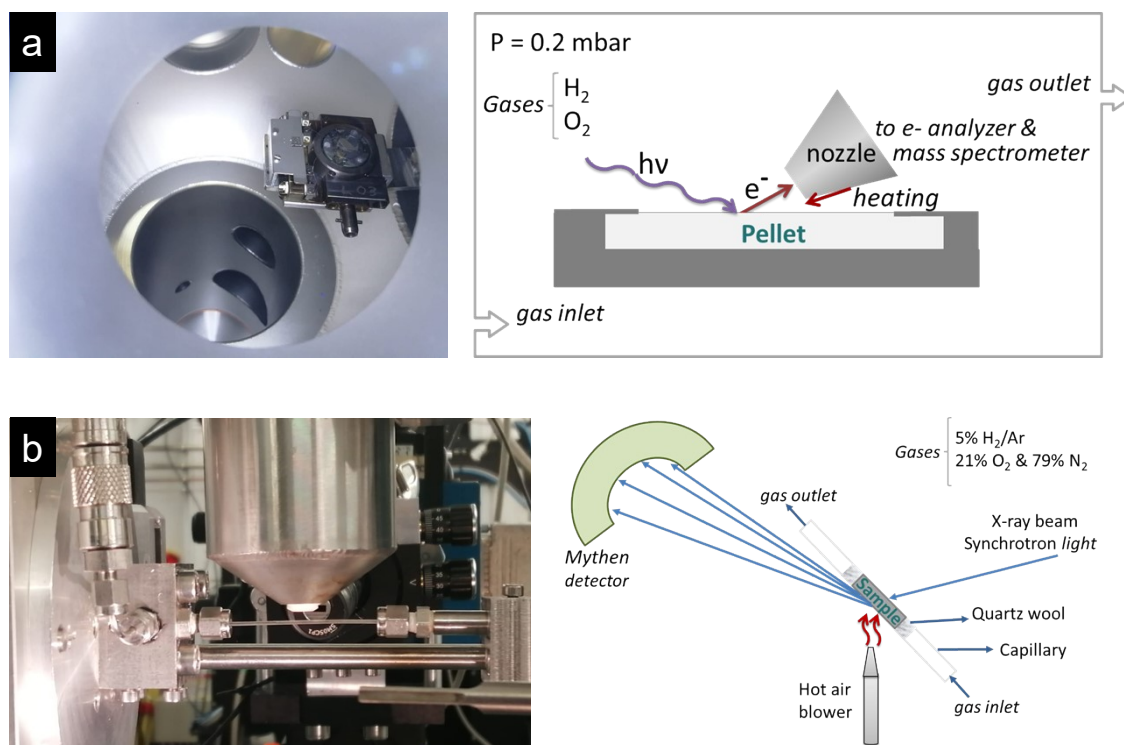
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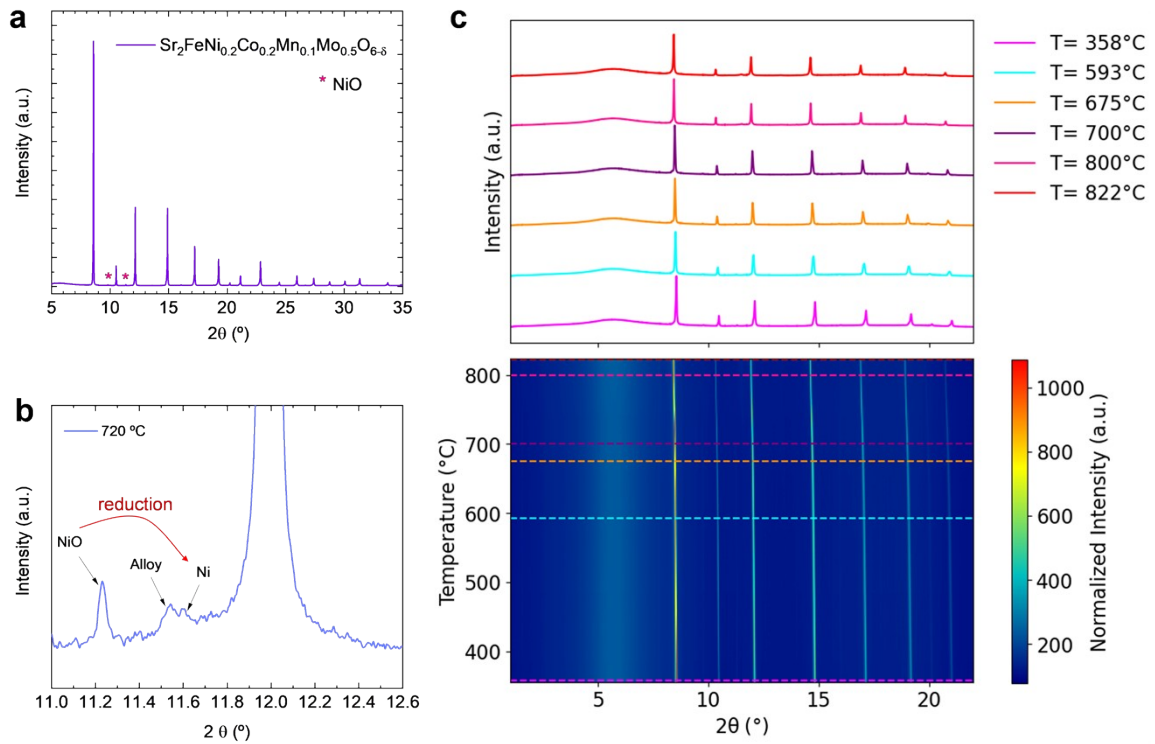
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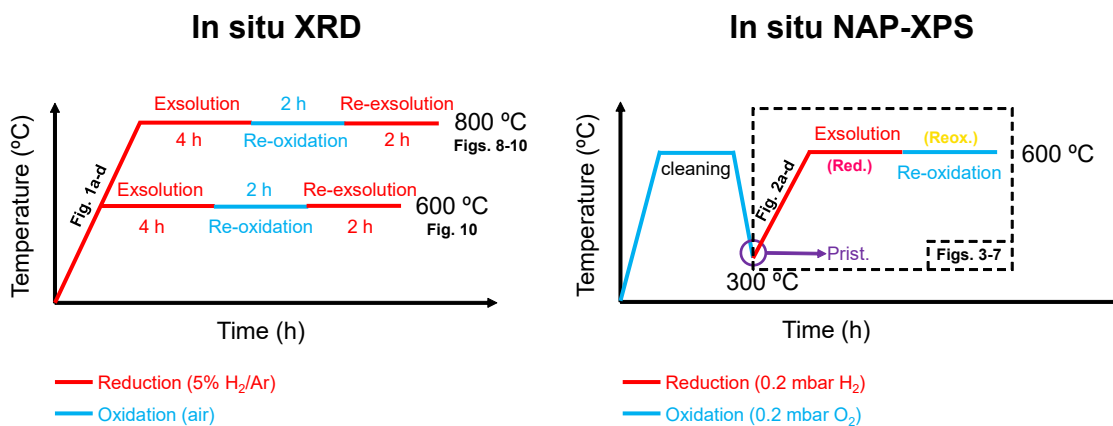
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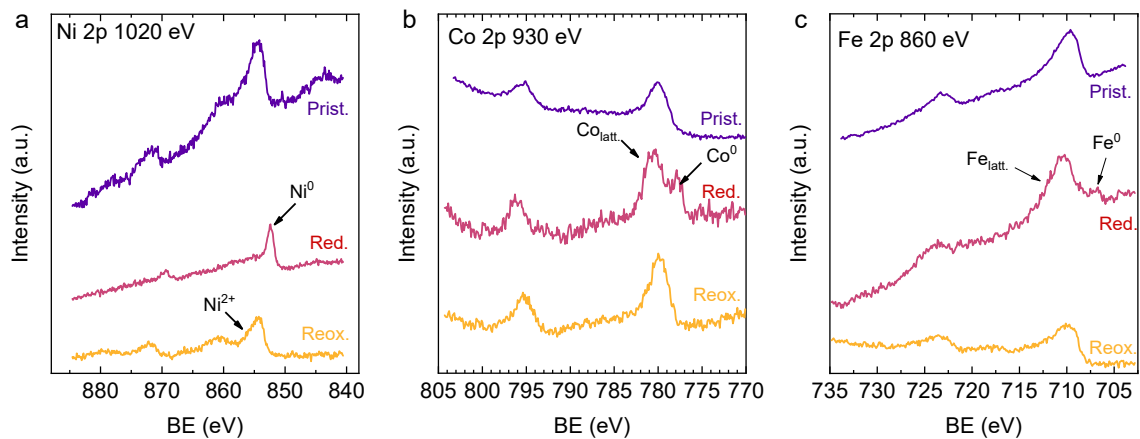
**Figure S1. (a)** NAP-XPS chamber picture and detailed schematic with the sample holder and the perovskite oxide pellet. **(b)** Setup picture and detailed schematic for measuring in-situ XRD showing the quartz capillaries and the hot air gun.



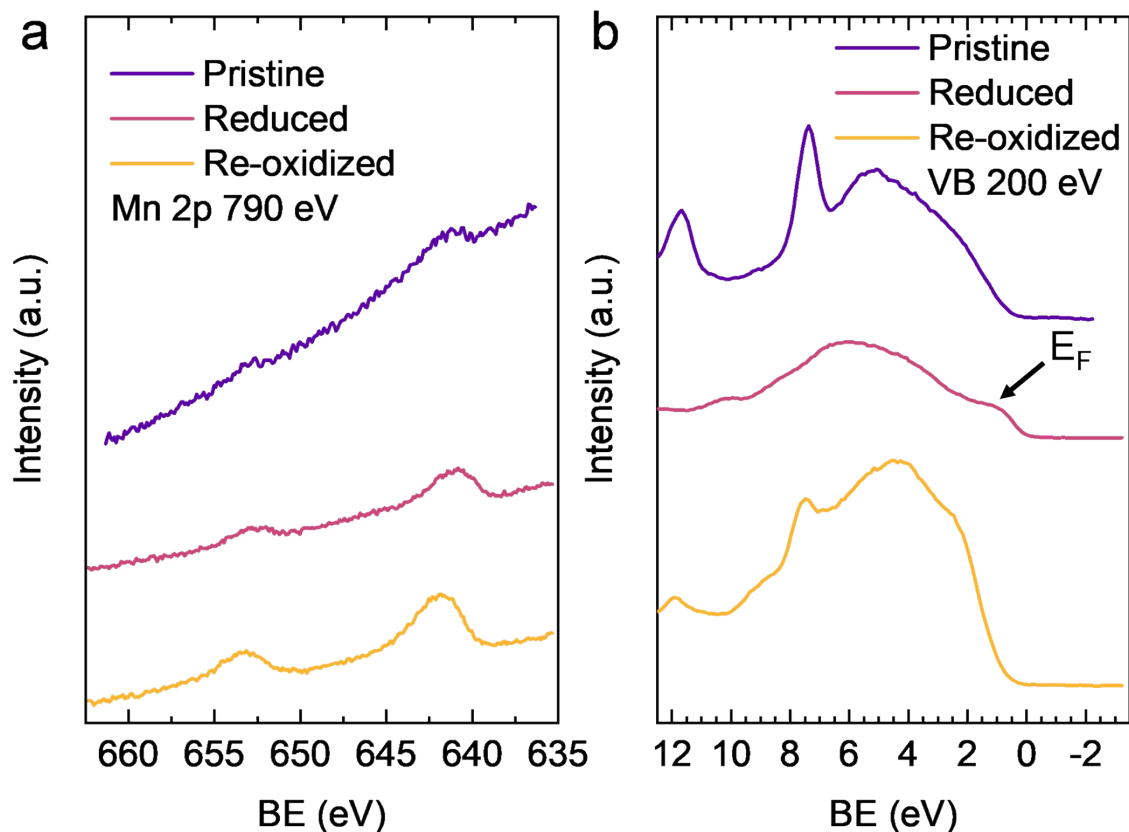
**Figure S2.** (a) X-ray diffractogram of pristine  $\text{Sr}_2\text{FeCo}_{0.2}\text{Ni}_{0.2}\text{Mn}_{0.1}\text{Mo}_{0.5}\text{O}_{6-\delta}$ , showing the minor presence of NiO. (b) XRD diffractogram collected at  $720^\circ\text{C}$  under 5%  $\text{H}_2/\text{Ar}$  flow, exhibiting a double metallic signal (“Alloy” & “Ni”). Diffractograms taken at different temperatures showing the whole  $2\theta$  range and a contour plot depicting the evolution of the observed reflections (c).



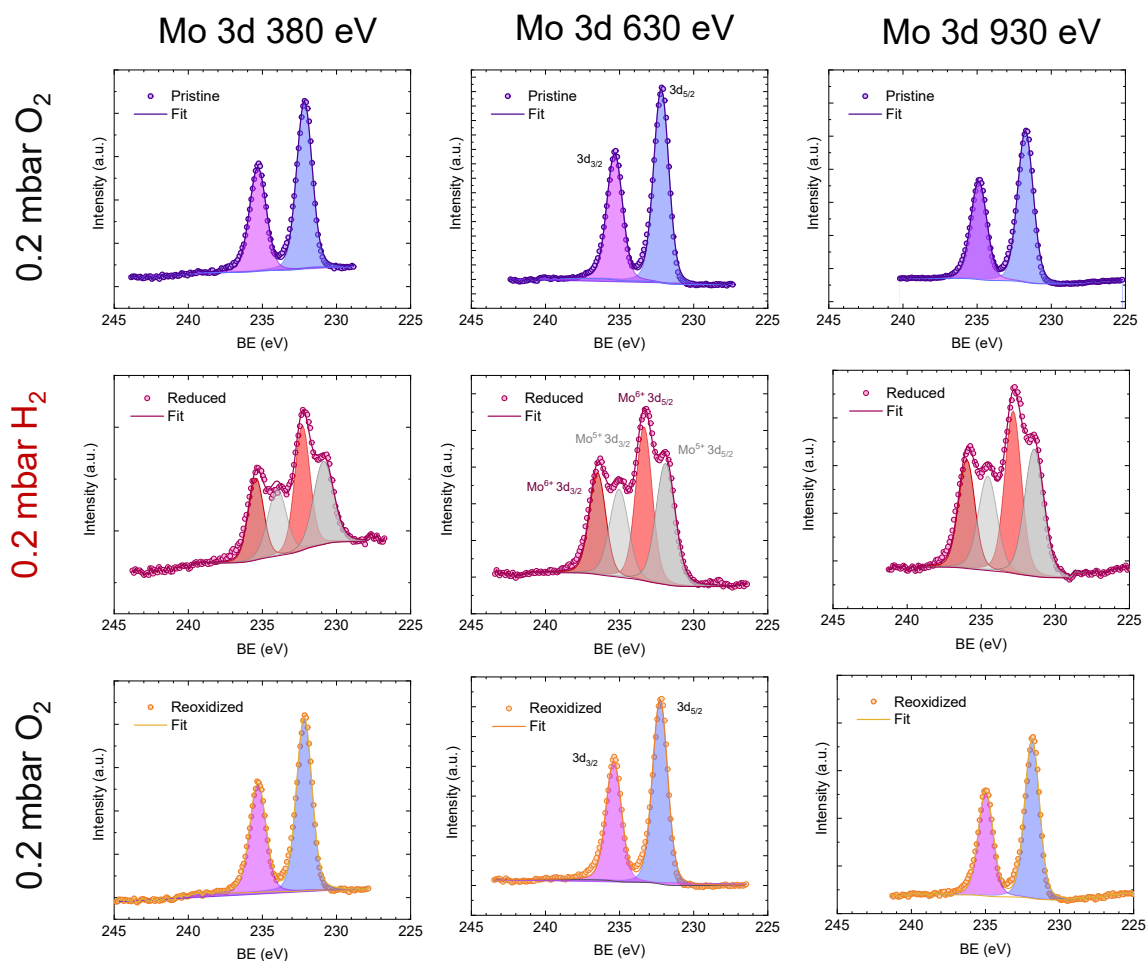
**Figure S3.** Temperature-time measuring protocols for the XRD (left) and NAP-XPS (right) measurements employed for in-situ analyzing the formation of exsolved ternary alloyed NPs.



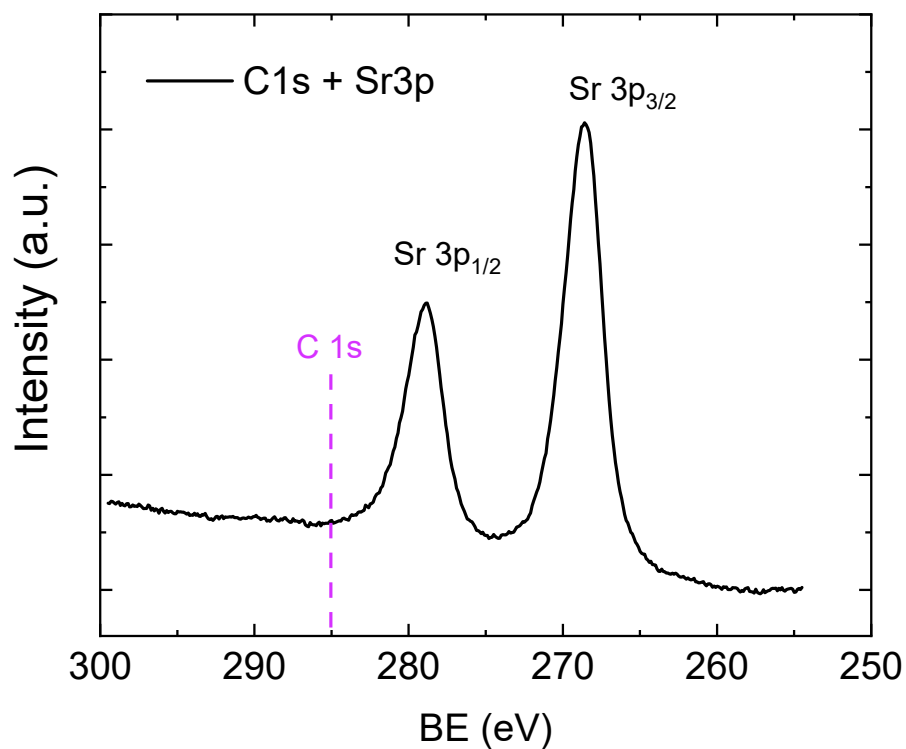
**Figure S4.** NAP-XPS spectra of **(a)** Ni 2p **(b)** Co 2p, and **(c)** Fe 2p core levels acquired *in-situ* at 600 °C in pristine (“Prist.”, 0.2 mbar O<sub>2</sub>), reduced (“Red.”, 0.2 mbar H<sub>2</sub>) and reoxidized (“Reox.”, 0.2 mbar O<sub>2</sub>) conditions. The used photon energies (to result in a similar electron kinetic energy of 150 eV) are indicated in the panels. A constant offset for better clarity is added to the spectra.



**Figure S5.** NAP-XPS spectra of (a) Mn 2p core level and (b) the valence band (VB) region acquired in-situ at 600 °C in pristine (0.2 mbar O<sub>2</sub>), reduced (0.2 mbar H<sub>2</sub>) and re-oxidized (0.2 mbar O<sub>2</sub>) conditions.  $E_F$  indicates the position of the Fermi level ( $E_F$ ) in panel (b). Used photon energies are indicated in the panels. Constant offset added for clarity.

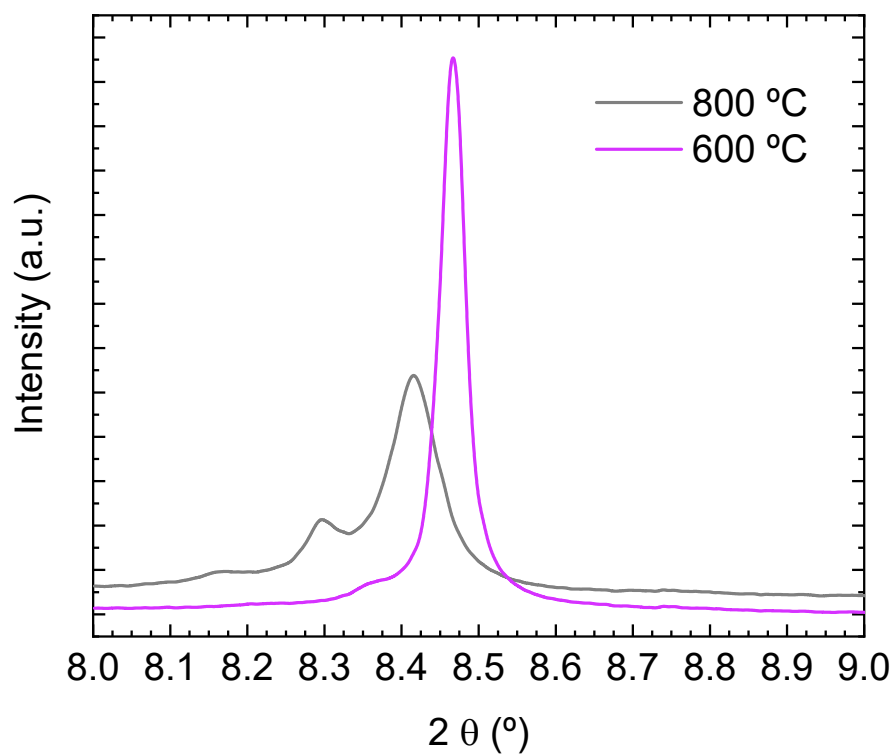


**Figure S6.** NAP-XPS fitted spectra of Mo 3d at the pristine (0.2 mbar O<sub>2</sub>), reduced (0.2 mbar H<sub>2</sub>), and re-oxidized (0.2 mbar O<sub>2</sub>) sample states using three photon energies: 380, 630 and 930 eV. The spectra of the reduced sample show the presence of both, Mo<sup>5+</sup> and Mo<sup>6+</sup>, whereas in the pristine and oxidized sample states only Mo<sup>6+</sup> is present.

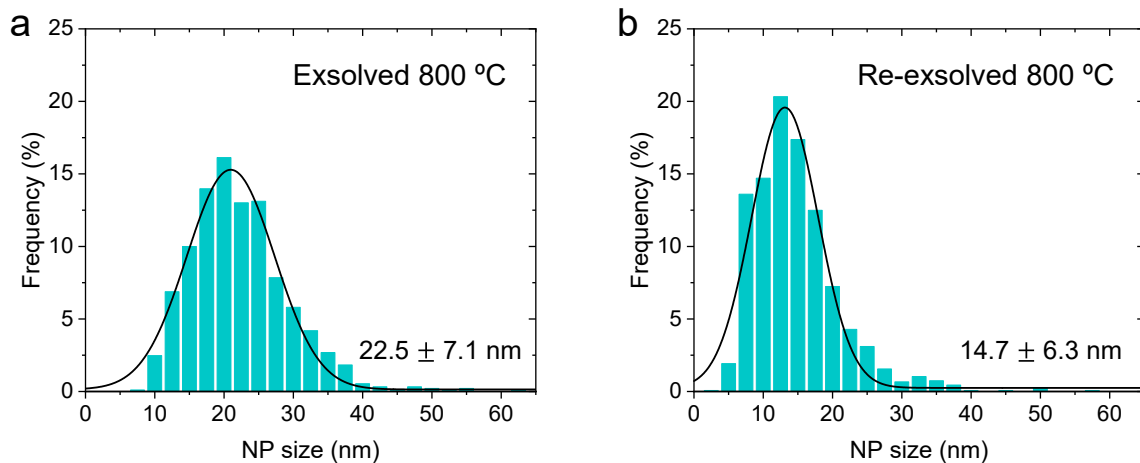


**Figure S7.** NAP-XPS spectra of the (overlapping) C 1s and Sr 3p core level regions of  $\text{Sr}_2\text{FeCo}_{0.2}\text{Ni}_{0.2}\text{Mn}_{0.1}\text{Mo}_{0.5}\text{O}_{6-\delta}$  sample in the pristine sample state. Data acquired with a photon energy of 1030 eV. The visible spin-orbit doublet corresponds to the Sr 3p core level, whereas the C 1s line around 284.8 eV is not observed, indicating successful carbon-based species removal during the initial heating cleaning process.

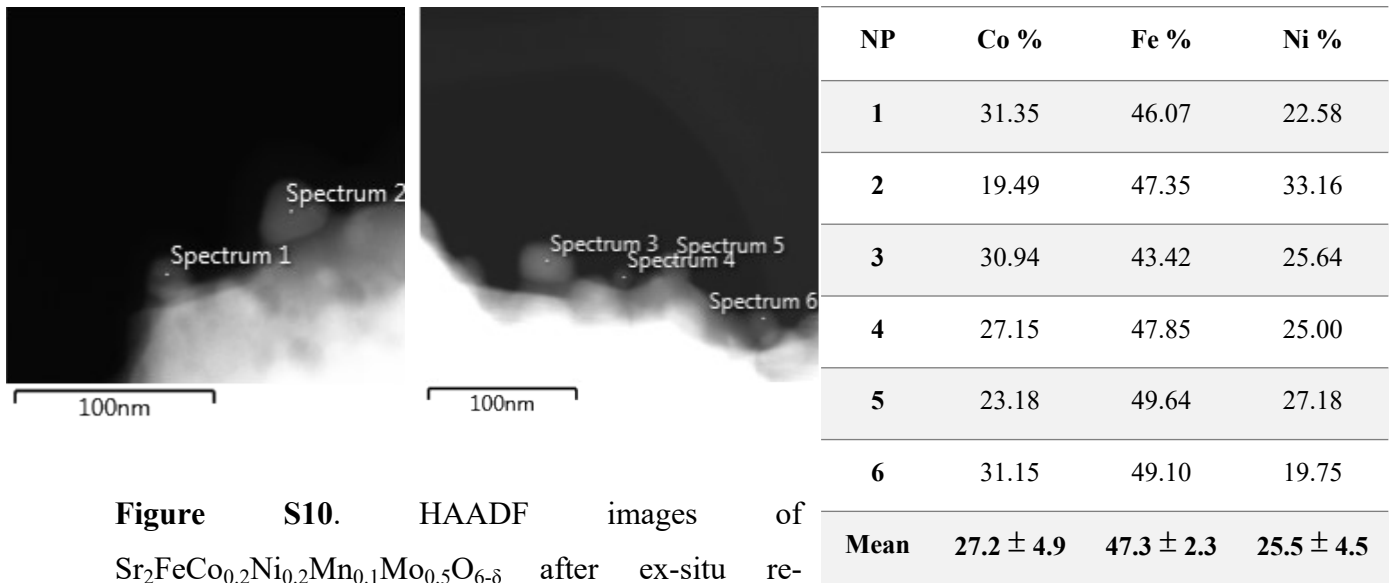




**Figure S8.** Comparison of the XRD data at  $2\theta$  (°) : 600 and 800 °C, after the second reduction treatment, consisting of 2 h under 5 %  $\text{H}_2/\text{Ar}$  for  $\text{Sr}_2\text{FeCo}_{0.2}\text{Ni}_{0.2}\text{Mn}_{0.1}\text{Mo}_{0.5}\text{O}_{6-\delta}$  main reflection (2 0 0) region.



**Figure S9.** Histograms for the size distribution of the exsolved NPs for  $\text{Sr}_2\text{FeCo}_{0.2}\text{Ni}_{0.2}\text{Mn}_{0.1}\text{Mo}_{0.5}\text{O}_{6-\delta}$ , after ex-situ first exsolution (**a**) and re-exsolution of the re-oxidized material (**b**). Both reduction treatments were performed at 800 °C, 4 h and under 5%  $\text{H}_2/\text{Ar}$  flow.



**Figure S10.** HAADF images of  $\text{Sr}_2\text{FeCo}_{0.2}\text{Ni}_{0.2}\text{Mn}_{0.1}\text{Mo}_{0.5}\text{O}_{6-\delta}$  after ex-situ re-exsolution of the re-oxidized material. Reduction treatment was performed at 800 °C, 4 h and under 5%  $\text{H}_2/\text{Ar}$  flow, after 24 h re-oxidation under air flow and 800 °C. The table shows the compositional fractions of the re-exsolved NPs, where Fe becomes the main component of the ternary alloy instead of Ni.